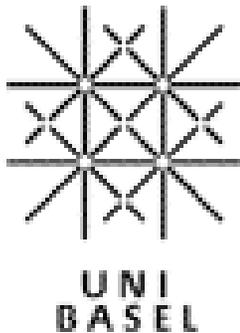


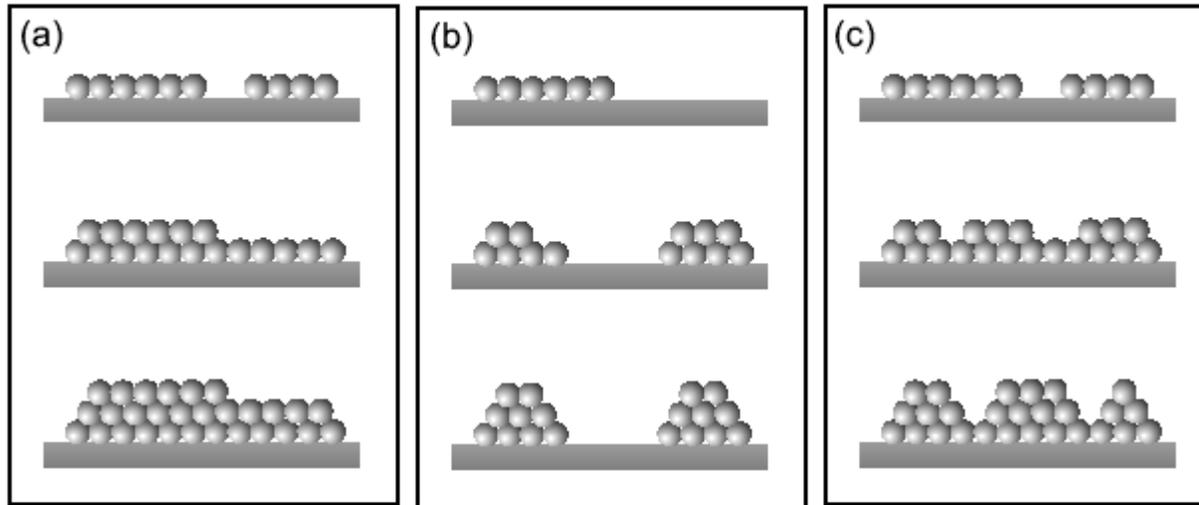
Surface Physics 2010

Growth of Thin Films



Lecturer: Dr. Enrico Gnecco
NCCR Nanoscale Science

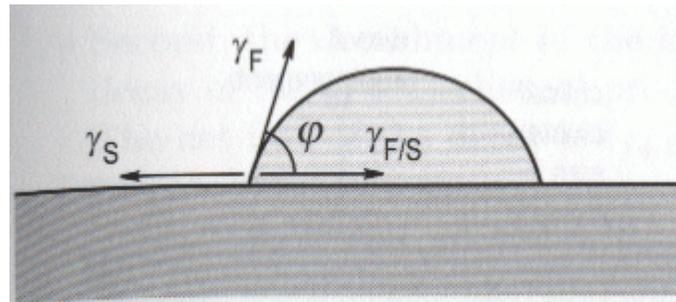
Growth Modes



- (a) **Layer-by-layer** or **Frank-van der Merve** mode \rightarrow 2D islands
- (b) **Island** or **Vollmer-Weber** mode \rightarrow 3D islands
- (c) **Layer plus island** or **Stranski-Krastanov** mode \rightarrow 2D layer + 3D islands

Growth Modes

- **Surface tension** (γ) = work required to build a surface of unit area
(\equiv force per unit length)

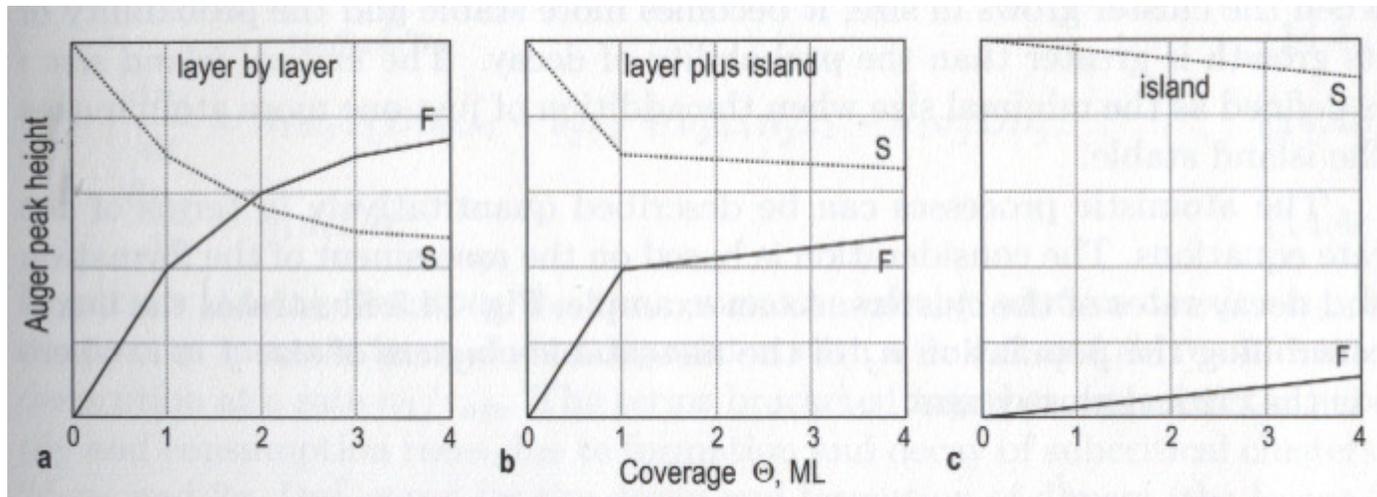


$$\gamma_S = \gamma_{SF} + \gamma_F \cos \varphi$$

- Island growth: $\varphi > 0 \rightarrow \gamma_S < \gamma_{SF} + \gamma_F$
- Layer-by-layer growth: $\varphi = 0 \rightarrow \gamma_S \geq \gamma_{SF} + \gamma_F$

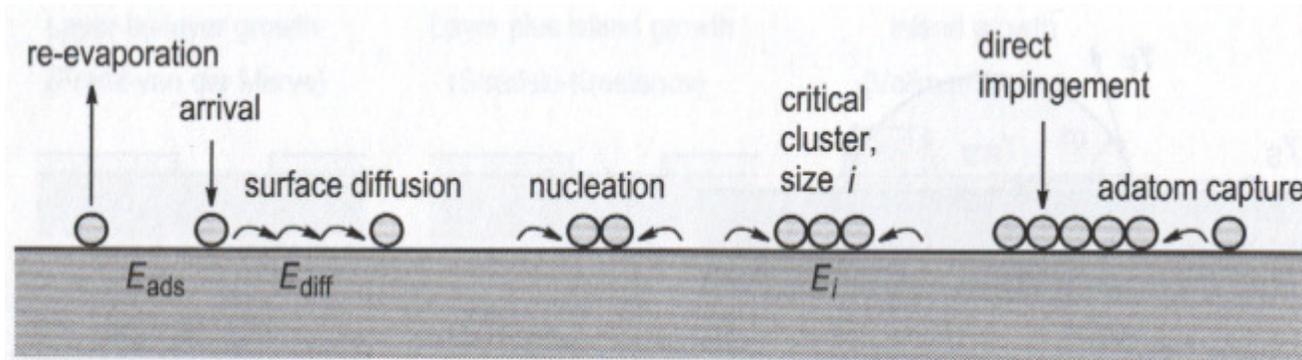
Growth Modes

- Exp: Monitor Auger signals from film and substrate while depositing...



Island Number Density

- Nucleation and growth on surfaces:



- Diffusion coefficient:

$$D = \frac{v}{4n_0} \exp\left(-\frac{E_{diff}}{k_B T}\right)$$

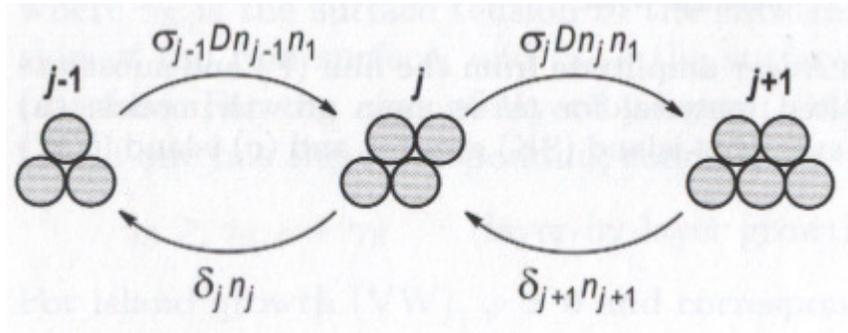
- Residence time:

$$\tau_{ads} = \frac{1}{v} \exp\left(\frac{E_{ads}}{k_B T}\right)$$

- **Critical island size i** : minimal size when the addition of one atom makes the island stable

Island Number Density

- Capture and decay processes → cluster size



- Rate equations:

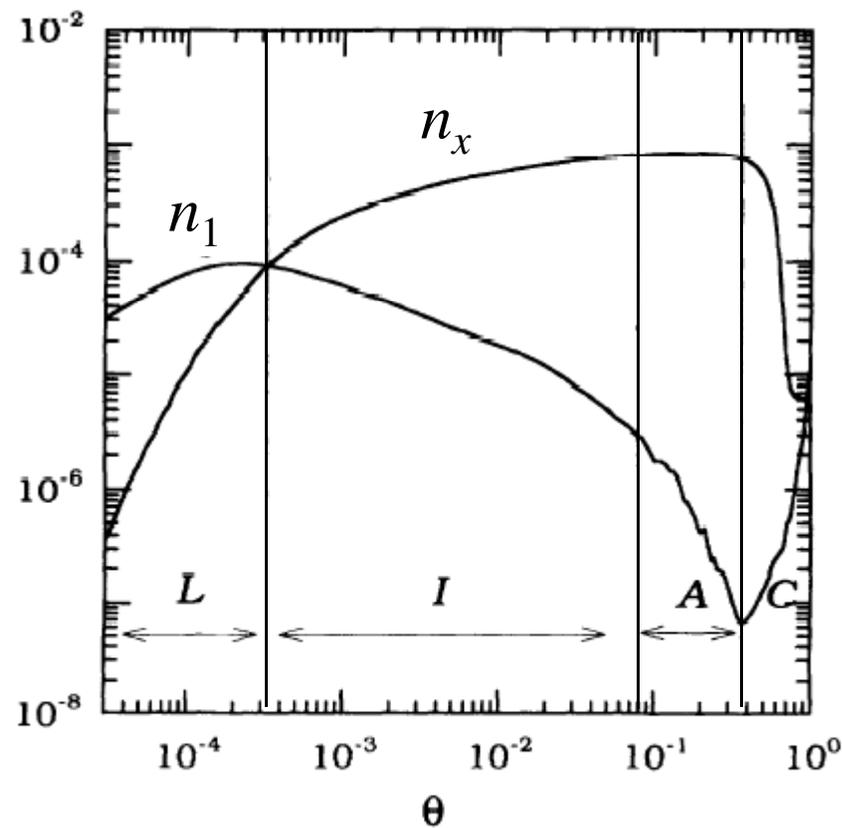
$$\frac{dn_1}{dt} = \underbrace{R}_{\text{adatom density}} - \frac{n_1}{\tau_{ads}} + \underbrace{\left(2\delta_2 n_2 + \sum_{j=3}^i \delta_j n_j - 2\sigma_1 D n_1^2 - n_1 \sum_{j=2}^i \sigma_j D n_j \right)}_{\text{subcritical clusters}} - \underbrace{n_1 \sigma_x D n_x}_{\text{stable clusters}}$$

$$\frac{dn_j}{dt} = n_1 \sigma_{j-1} D n_{j-1} - \delta_j n_j + \delta_{j+1} n_{j+1} - n_1 \sigma_j D n_j \longrightarrow \text{metastable clusters}$$

$$\frac{dn_x}{dt} = n_1 \sigma_i D n_i \longrightarrow \text{stable clusters}$$

Island Number Density

- Numerical solution for $i = 1$ without re-evaporation (Amar, Family and Lam, PRB 1994):



- Four coverage regimes are found

Island Number Density

(1) Low-coverage nucleation regime: $n_1 \gg n_x$

- In such case:

$$n_1 \propto \Theta \quad n_x \propto \Theta^3$$

When $n_x \sim n_1 \rightarrow$ **(2) Intermediate-coverage regime**

- In such case:

$$n_1 \propto \Theta^{-1/3} \quad n_x \propto \Theta^{1/3}$$

When mean island separation \sim mean free path of adatoms

\rightarrow **(3) Aggregation regime** ($\Theta \sim 0.1-0.4$ ML)

When the island join together

\rightarrow **(4) Coalescence and percolation regime**

Island Number Density

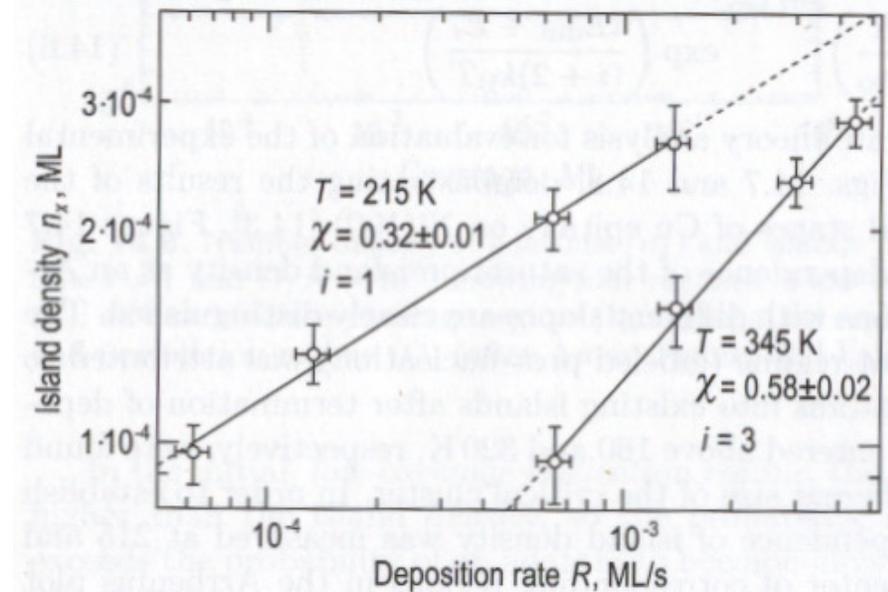
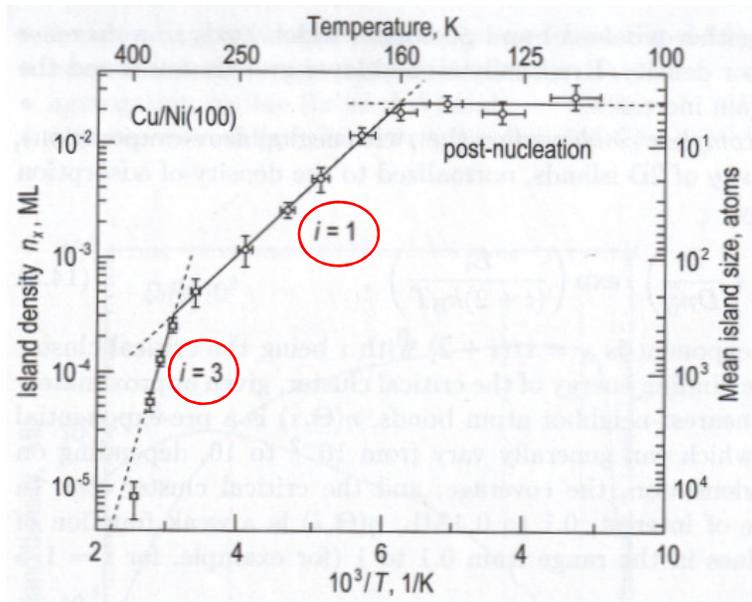
- Saturation density (Venables et al., 1984):

$$n_x = n_0 \eta(\Theta, i) \left(\frac{4R}{v_0 n_0} \right)^{\frac{i}{i+2}} \exp\left(\frac{iE_{diff} + E_i}{(i+2)k_B T} \right)$$

binding energy
of critical cluster

$$\eta(\Theta, i) \sim 0.1-1$$

- Example: Cu on Ni(100) (Müller et al., PRB 1996)



Island Shape

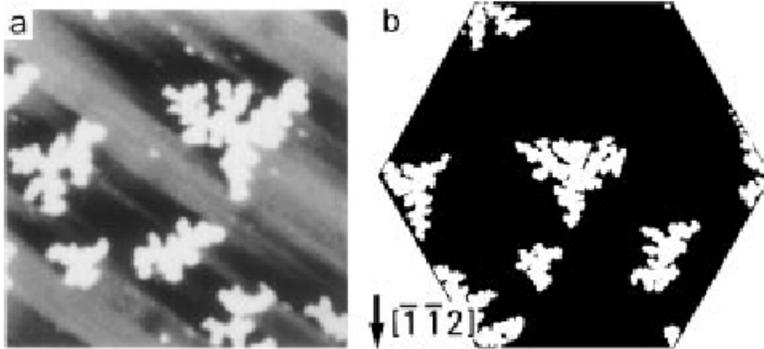
- At low T (slow edge diffusion): **ramified islands**
- **Diffusion-limited-aggregation (DLA) model** (Witten & Sander, PRL 1981):



- Adatoms stick at islands
 - Fractal shape
 - Branch thickness ~ 1 atom
 - No influence of lattice geometry
-
- In real growth (STM experiments):
 - Fractal shape
 - Branch thickness > 1 atom
 - Influence of lattice geometry

Island Shape

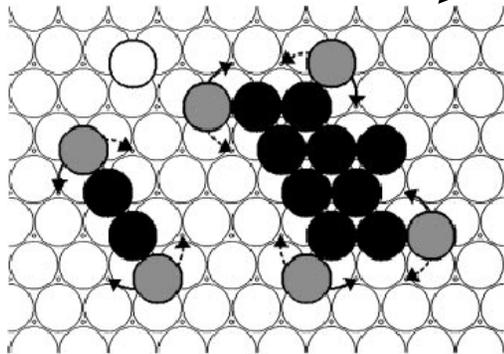
- Example: Pt/Pt(111) (Hohage et al., PRL 1996)



- Branch thickness ~ 4 atoms
- Trigonal symmetry

experiment

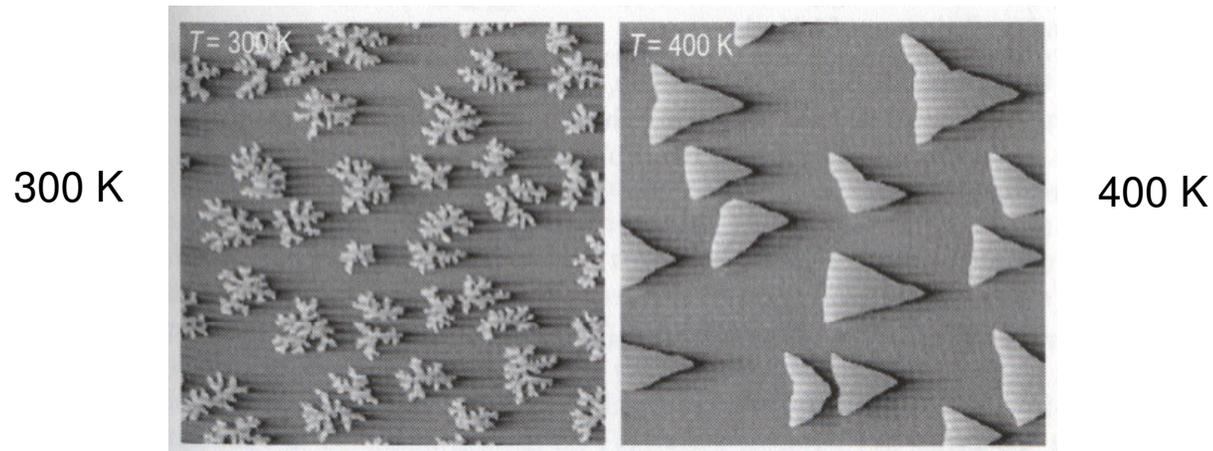
simulation



- Higher coordination is preferred

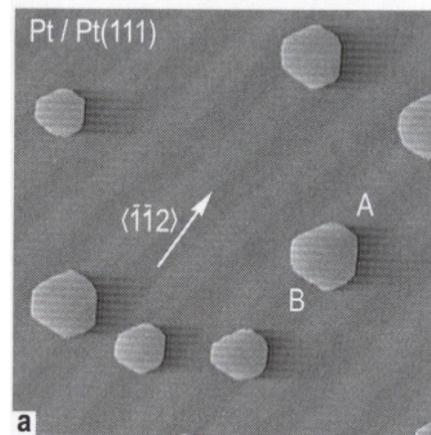
Island Shape

- At higher T: **compact islands**
- Example: Pt/Pt(111) (Bott et al., PRL 1992)



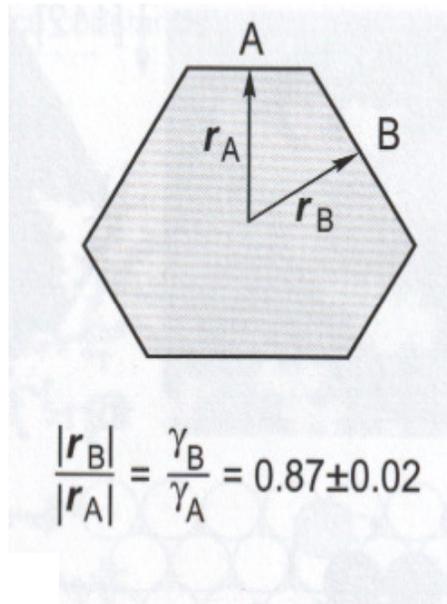
- The equilibrium shape is hexagonal:

(deposition at 425 K + annealing at 700 K)



Island Shape

- **2D Wulff theorem:** In a 2D crystal at equilibrium, the distances of the borders from the crystal center are proportional to their free energy per unit length



- For Pt(111): B/A ~ 0.87

Island Size Distribution

- **Island size distribution** depends on:

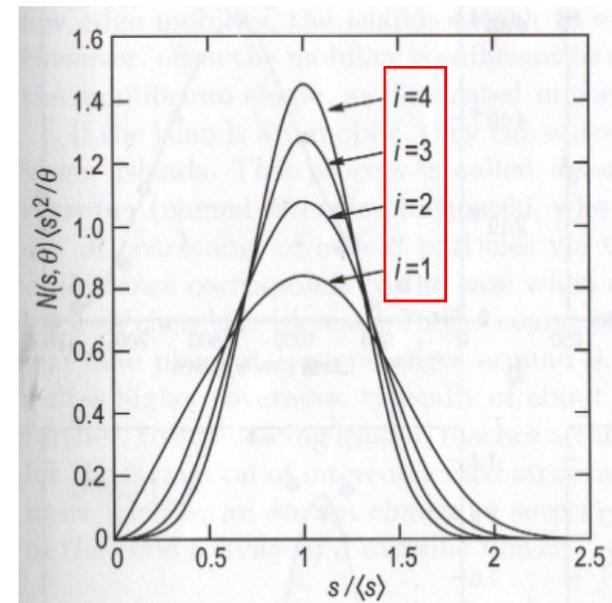
- Critical island size
- Coverage
- Substrate structure
- “Coarsening” (at high Θ)

- From **scaling theory** (Amar et al., PRB 1994):

$$N_s = \frac{\Theta}{\langle s \rangle^2} f_i \left(\frac{s}{\langle s \rangle} \right)$$

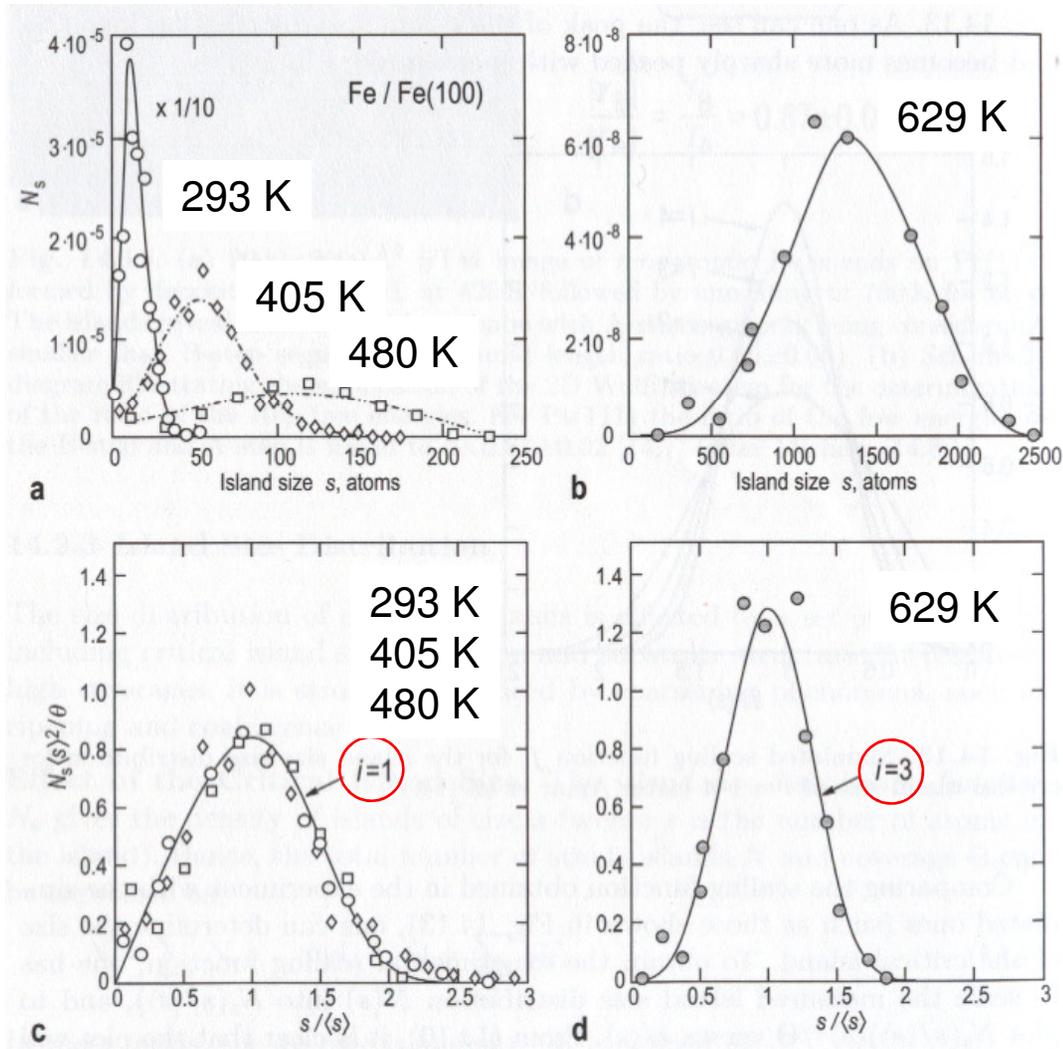
- Comparing with experimental results

→ critical island size



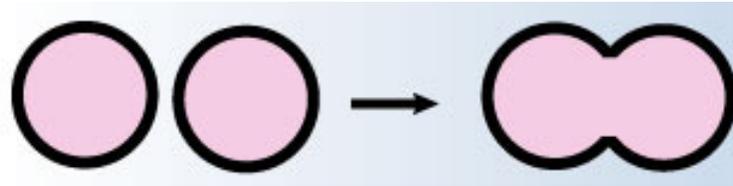
Island Size Distribution

- Example: Fe on Fe(100) (Stroscio et al., PRL1993, Amar et al., PRB 1994)



Coarsening Phenomena

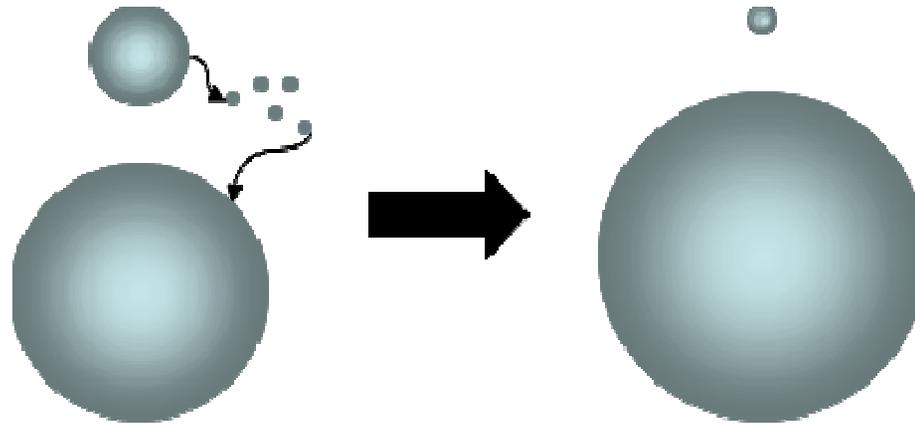
1) Coalescence:



- At 0.1 ML coverage: **dynamic coalescence (Smoluchowski ripening)**
- At 0.4-0.5 ML: **static coalescence**
- Higher coverages → **percolation** growth (→ change of physical properties!)

Coarsening Phenomena

2) (Ostwald) Ripening:



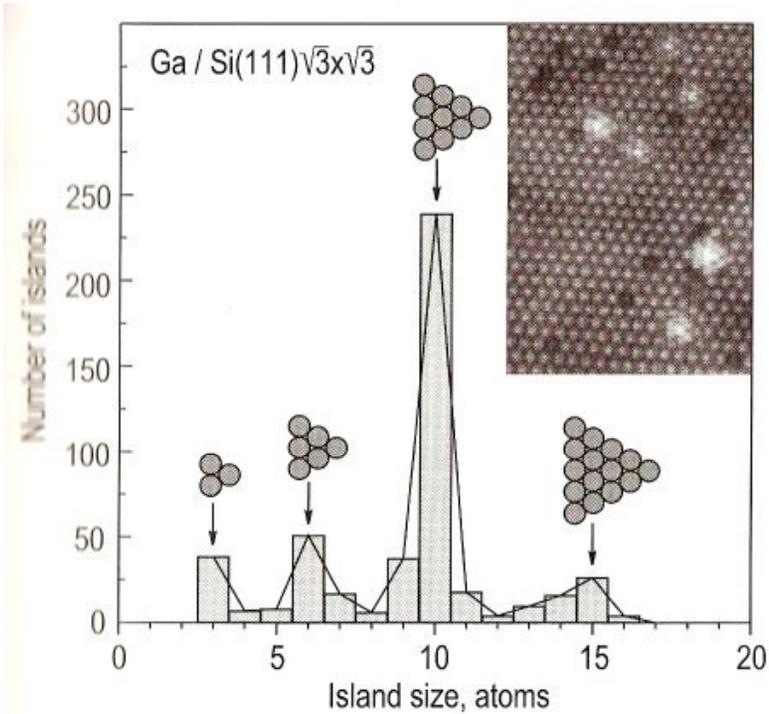
- Chemical potential of a circular island:

$$\mu(r) \propto \frac{\gamma}{r}$$

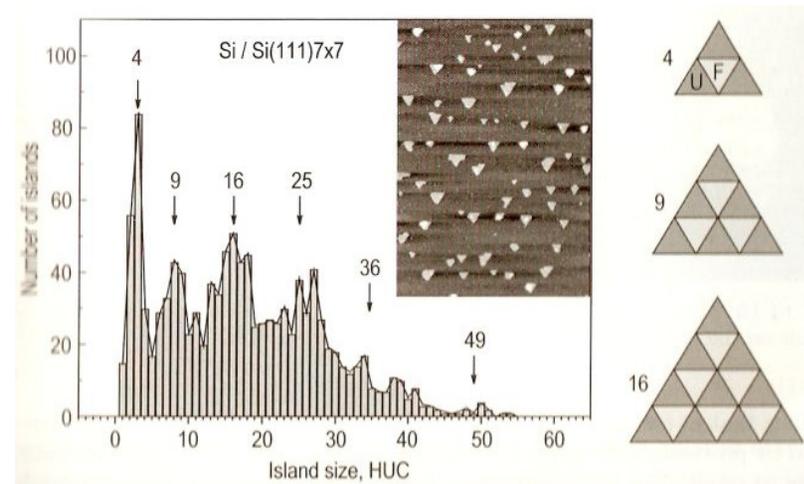
Reducing free energy \rightarrow net flow from smaller to larger islands!

Magic Islands

- Ga on Si(111) $\sqrt{3}\times\sqrt{3}$ (Lai & Wang, PRL 1998):



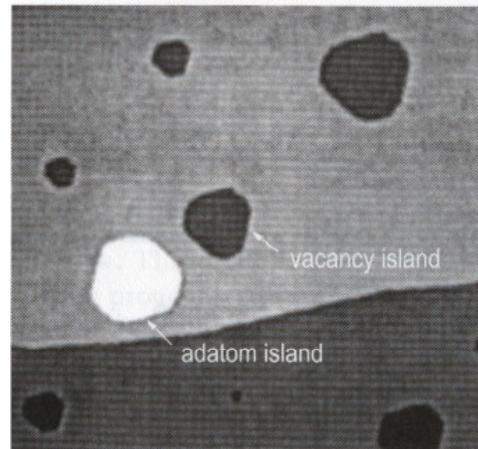
- Si on Si(111)7x7 (Voigtländer et al., PRL 1998):



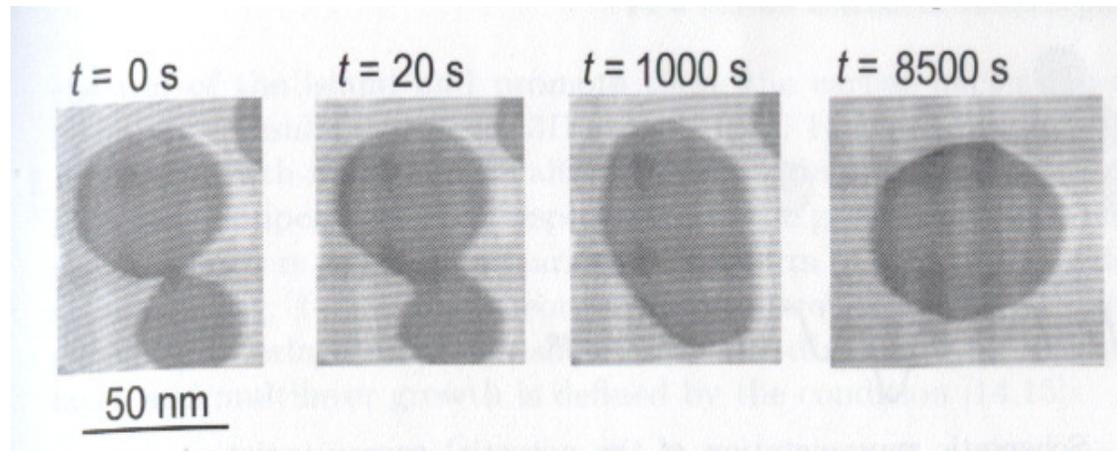
Formation of a new row \rightarrow high energy cost!

Vacancy Islands

- Ion bombardment → formation of **vacancy islands**

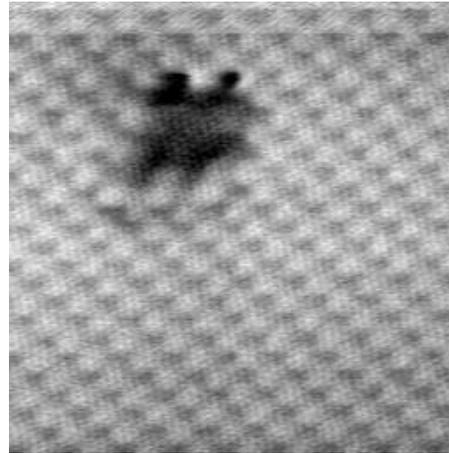
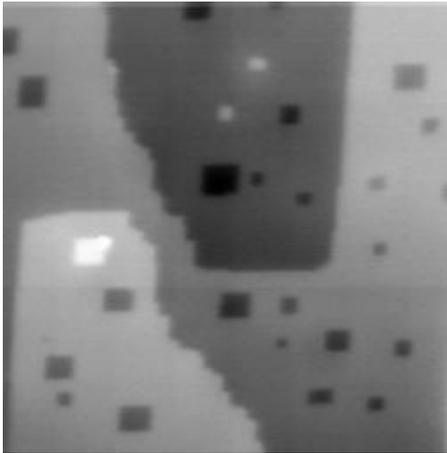


- Analogies with adatom islands



Vacancy Islands

- Electron bombardment on insulating surfaces (Bennewitz et al., SS 2001):

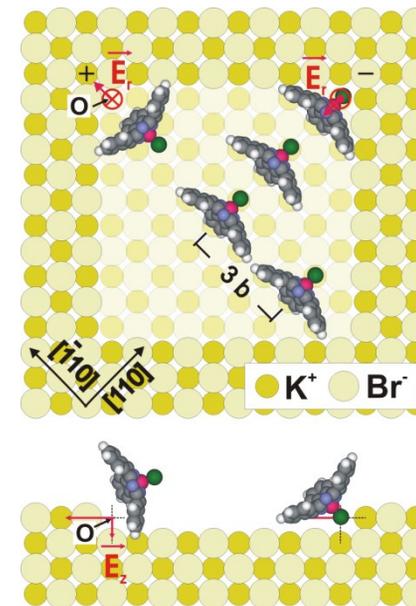
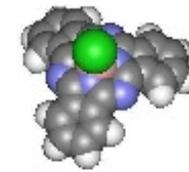
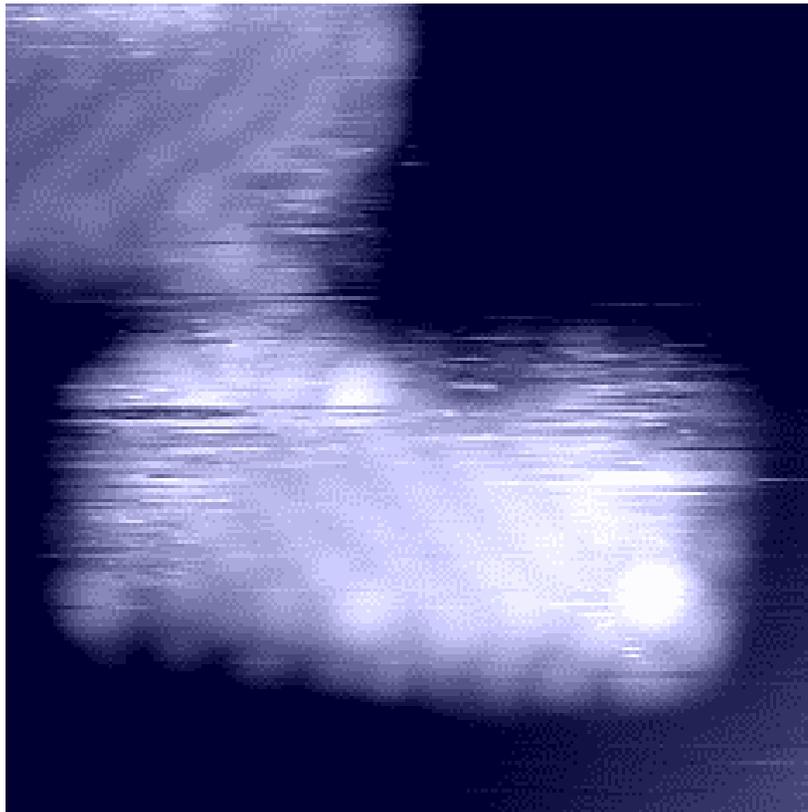


KBr(100)

- Irradiation with 1 keV electrons at 130 °C
- Rectangular pits with area of 1x1 nm² up to 30x30 nm²
- 1 ML deep (0.33 nm)

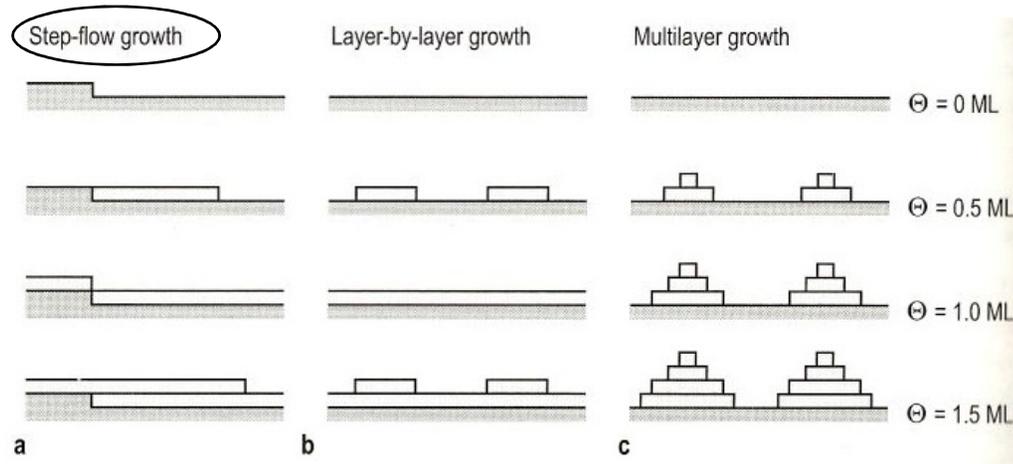
Vacancy Islands

- Vacancy islands can be used as molecular traps (Nony et al., NL 2004)

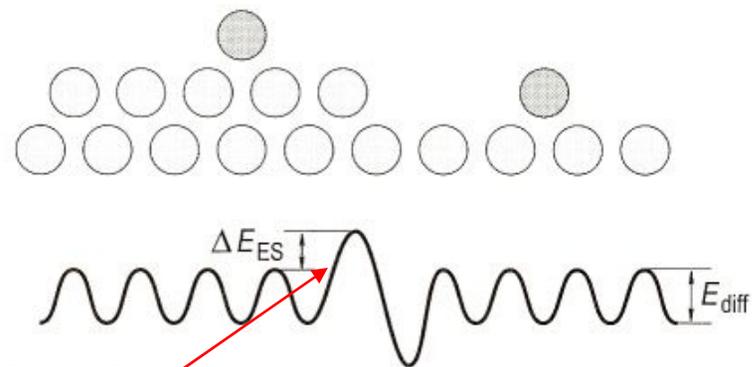


Kinetic Effects in Homoepitaxy

- Thermodynamics \rightarrow Layer-by-layer growth but...
- Kinetic processes \rightarrow Other growth modes are possible!



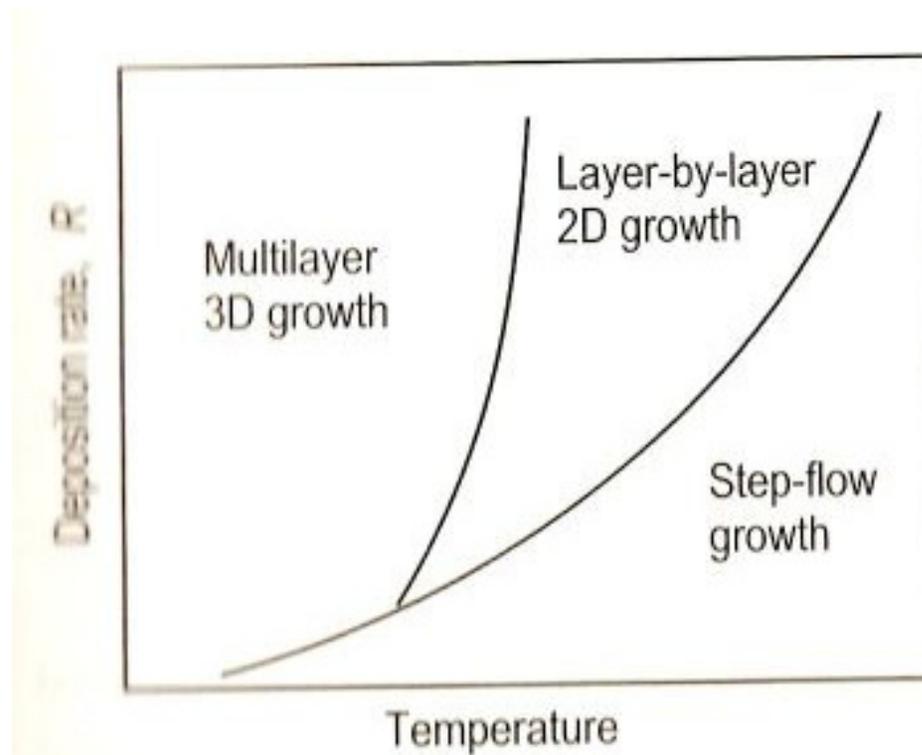
- **Interlayer** mass transport:



Ehrlich-Schwöbel barrier

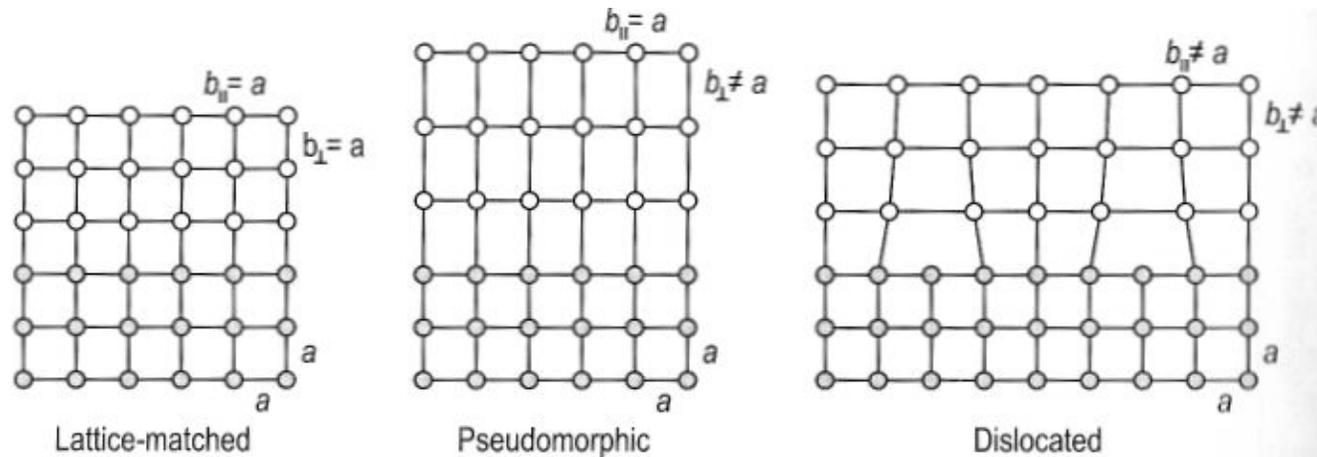
Kinetic Effects in Homoepitaxy

- Deposition rate and Temperature are important!
- Growth mode diagram (Rosenfeld et al., 1997):



Strain Effects in Heteroepitaxy

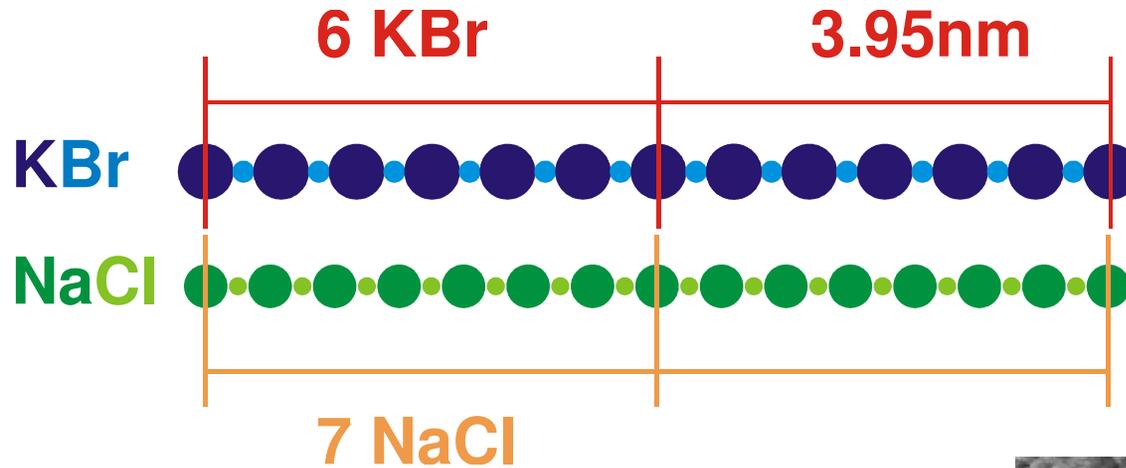
- Heteroepitaxy growth modes:



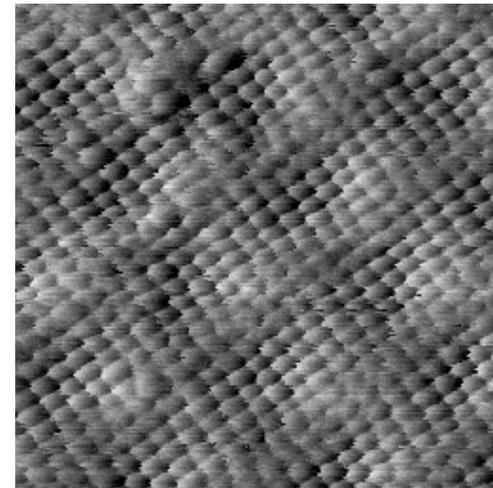
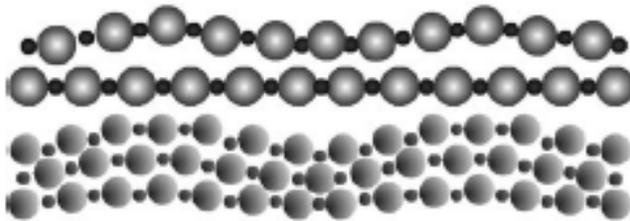
- Lattice **misfit** → elastic strain and dislocations
- Pseudomorphic growth below critical misfit and film thickness

Strain Effects in Heteroepitaxy

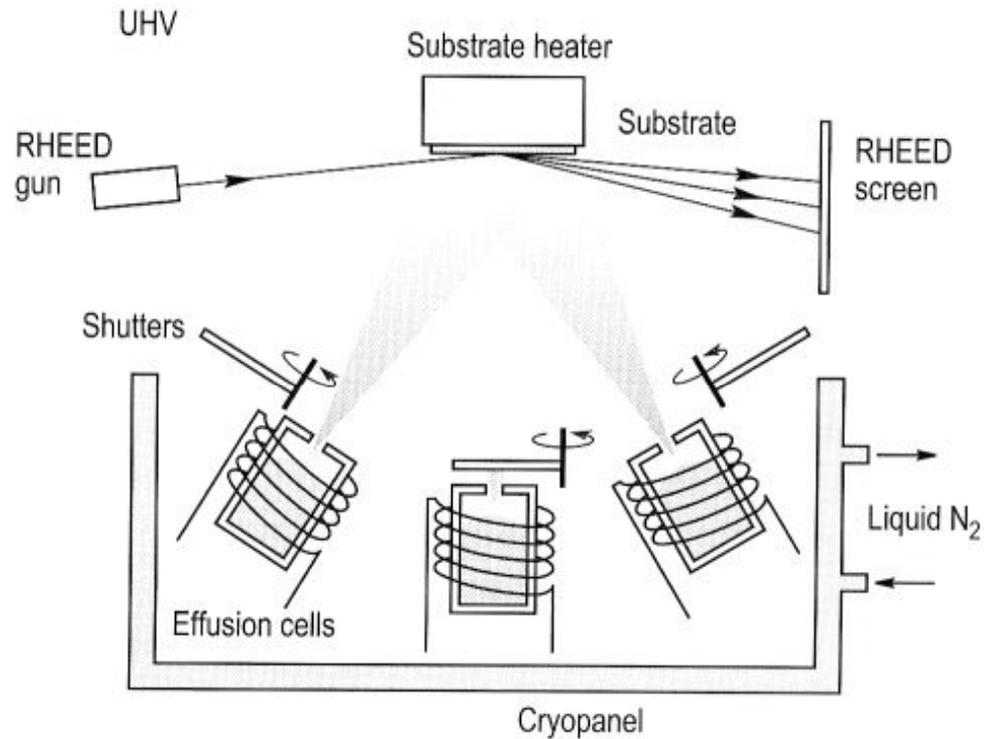
- Heteroepitaxy on insulating surfaces (Maier et al., PRB 2007):



- A “Moiré pattern” appears (also in the substrate?)



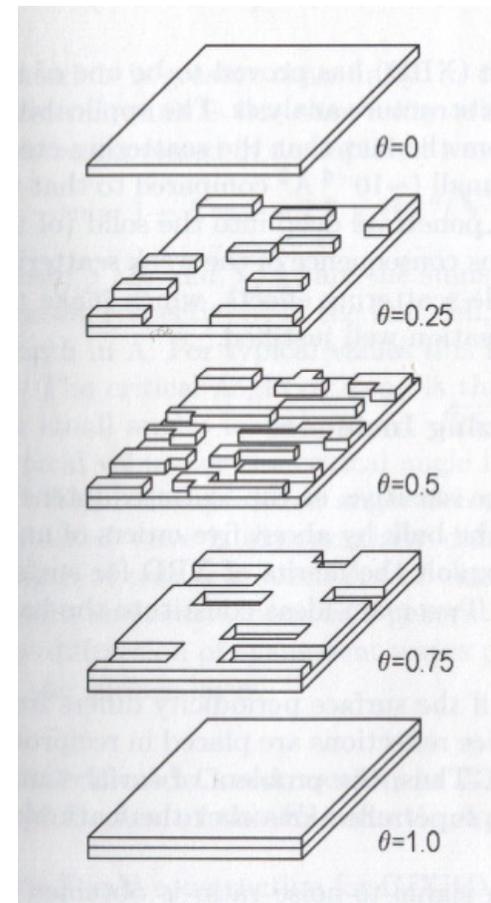
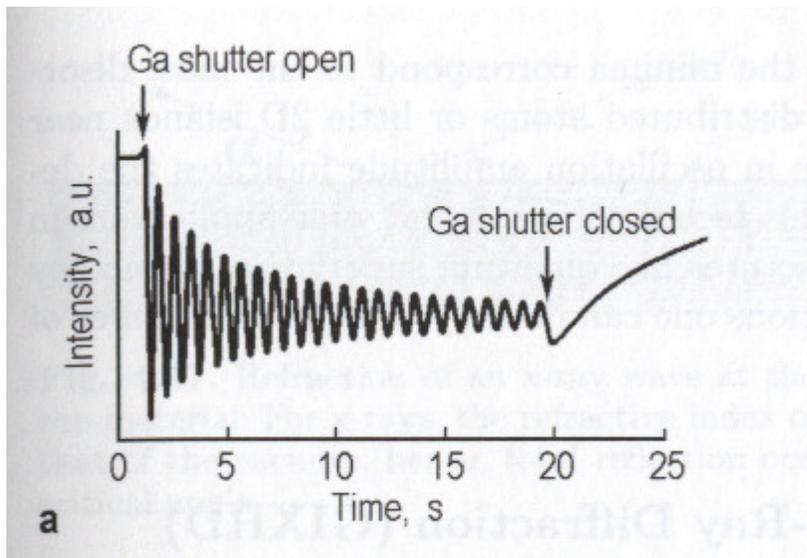
Molecular Beam Epitaxy



- Used both in research and in semiconductor device fabrications

Molecular Beam Epitaxy

- The growth process can be monitored by RHEED:



Solid Phase Epitaxy

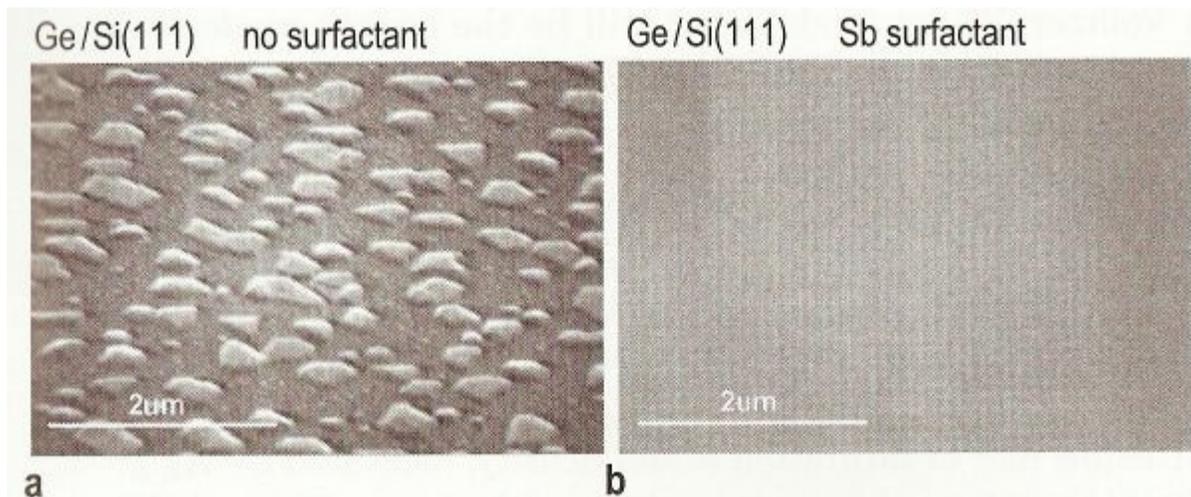
- Amorphous film deposited at low T and then crystallized upon heating at high T
- Lower crystallinity than MBE
- Used in semiconductor industry

Chemical Beam Epitaxy

- Growth by surface chemical reactions
- High temperature required
- High grow rate and cristallinity

Surfactant-Mediated Growth

- Impurity (**surfactant**) → different growth mode!
- Surfactant can be either segregated or trapped
- Example: Ge on Si(111) (Zahl et al., APA 1999)
 - Bare surface → Stranski-Krastanov mode
 - With surfactant (Sb) → Layer-by-layer mode



Further Reading

- K. Oura et al., Surface Science, Springer 2003, chapter 14
- J.A. Venables et al., Rep. Prog. Phys. 47 (1984) 399
- H. Brune, Surf. Sci. Rep. 31 (1998) 121