

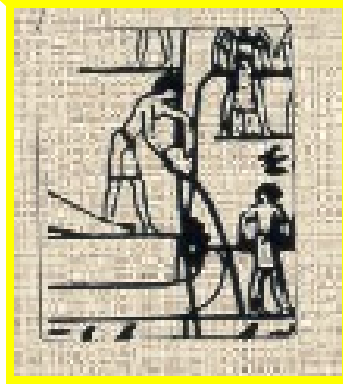
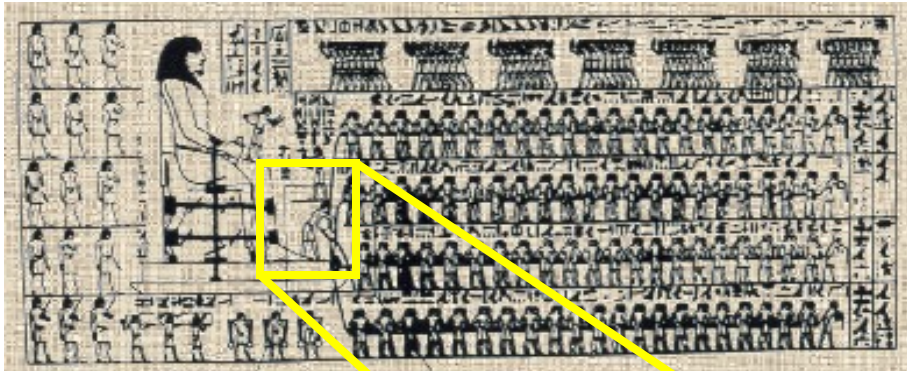
Oberflächenphysik

Scanning Probe Microscopy

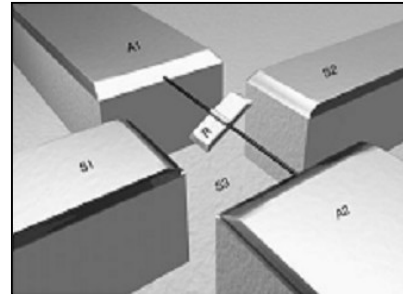
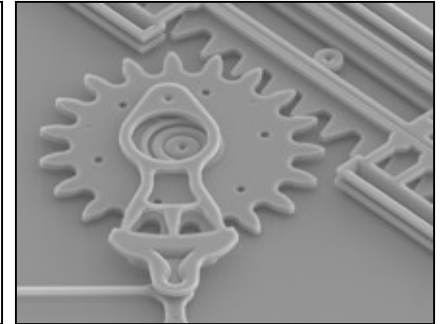
- **Friction Force Microscopy**
 - Force Calibration
 - Atomic Stick Slip
 - Tomlinson Model
 - Nano-manipulation
- **Atomic Force Microscopy**
 - Short- and Long-Range Forces
 - Kelvin Probe Force Microscopy
 - Measurements on Semiconducting Devices
 - Molecules on Insulating Surfaces

Importance of Friction

Long time ago...

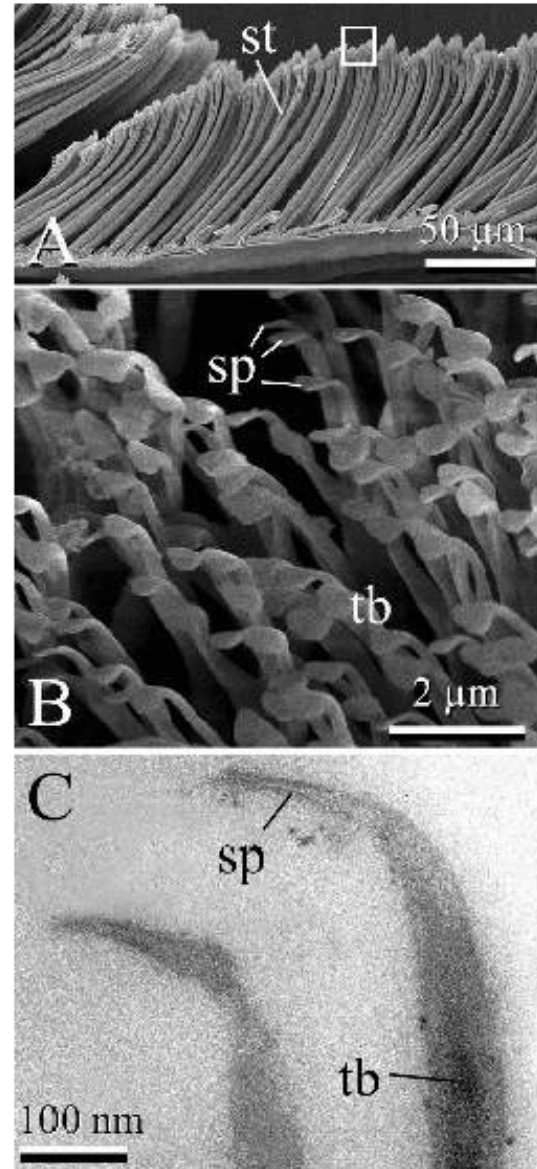


Nowadays...



In all cases: It is highly desirable to reduce and control friction

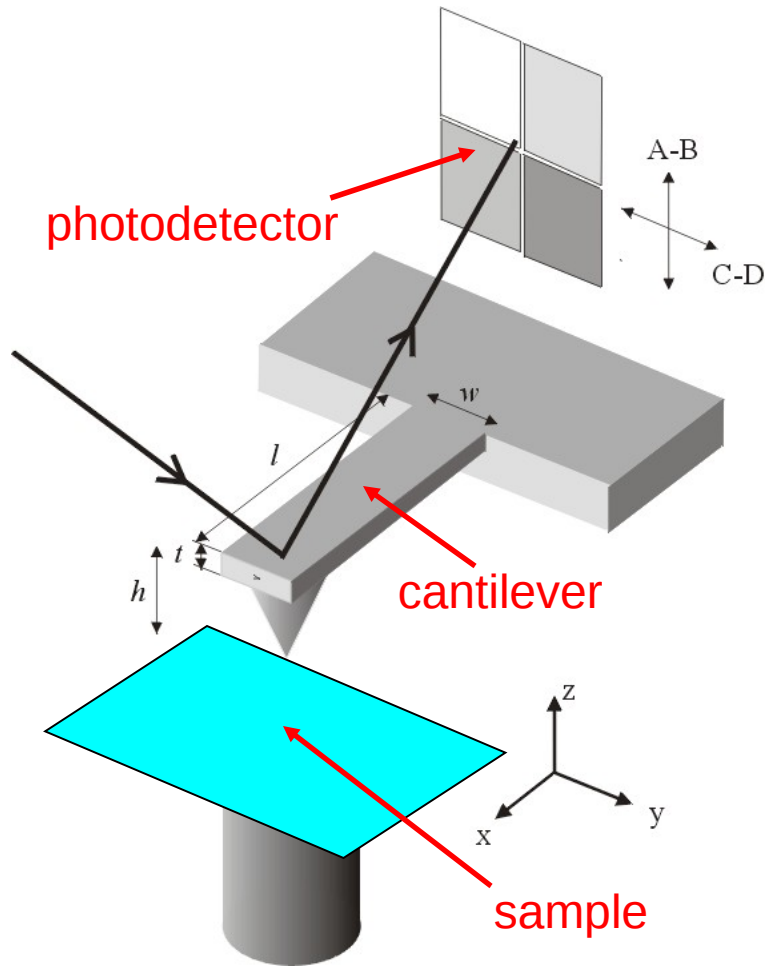
Gecko uses nanometer-sized contacts to climb walls



Gecko is able to control the contact area on all length scales

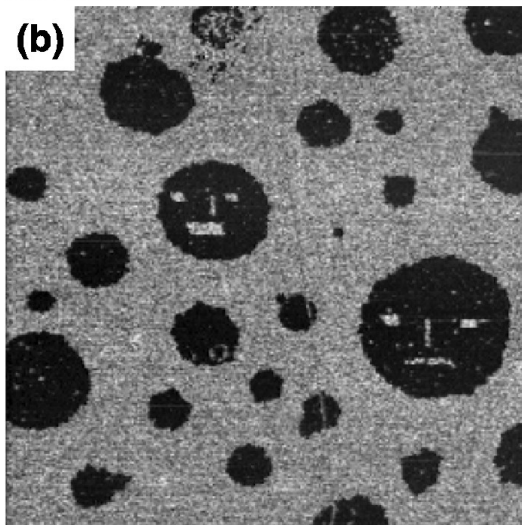
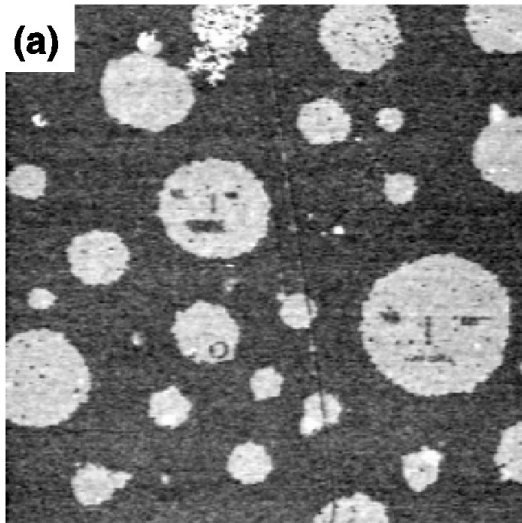
From B. Persson and S. Gorb
JCP, 119, 11437 (2003)

Atomic Force Microscopy (beam deflection)



- The normal and lateral forces on a sharp tip sliding on a surface are sensed using a laser beam
- Forces < 1 nN can be measured

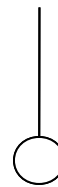
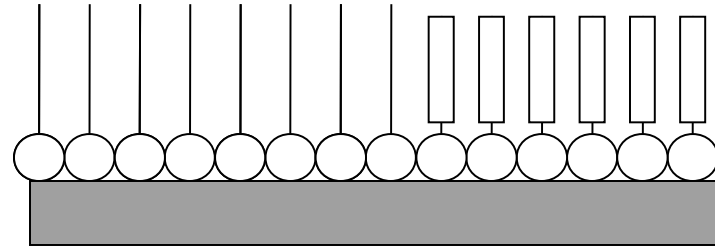
FFM on Langmuir-Blodgett films



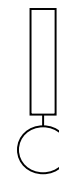
2.8 μm

- **Material contrast** on mixed Langmuir-Blodgett films:

Topo



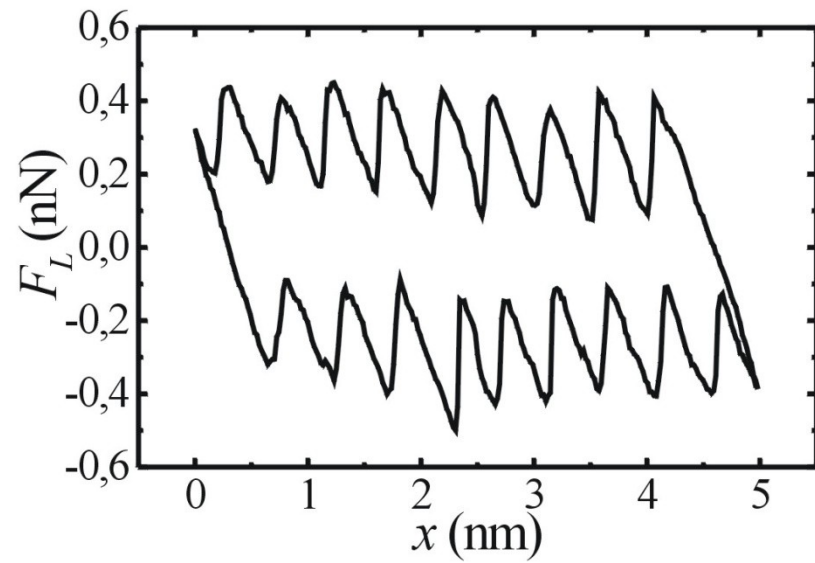
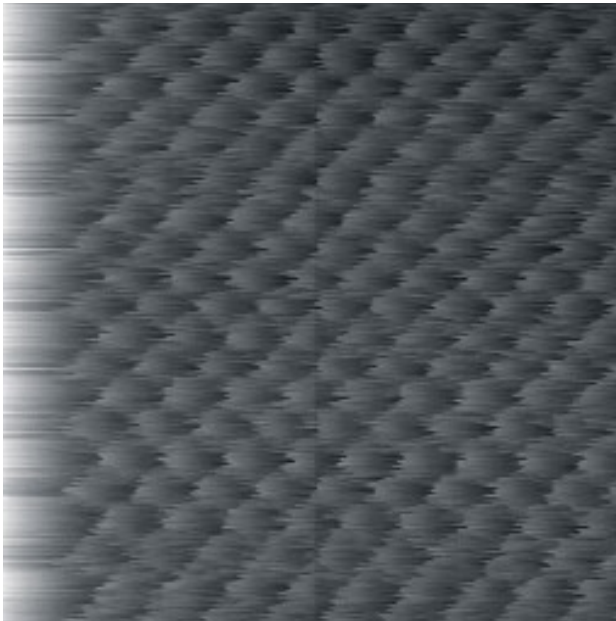
C-terminated



F-terminated

Lat. Force

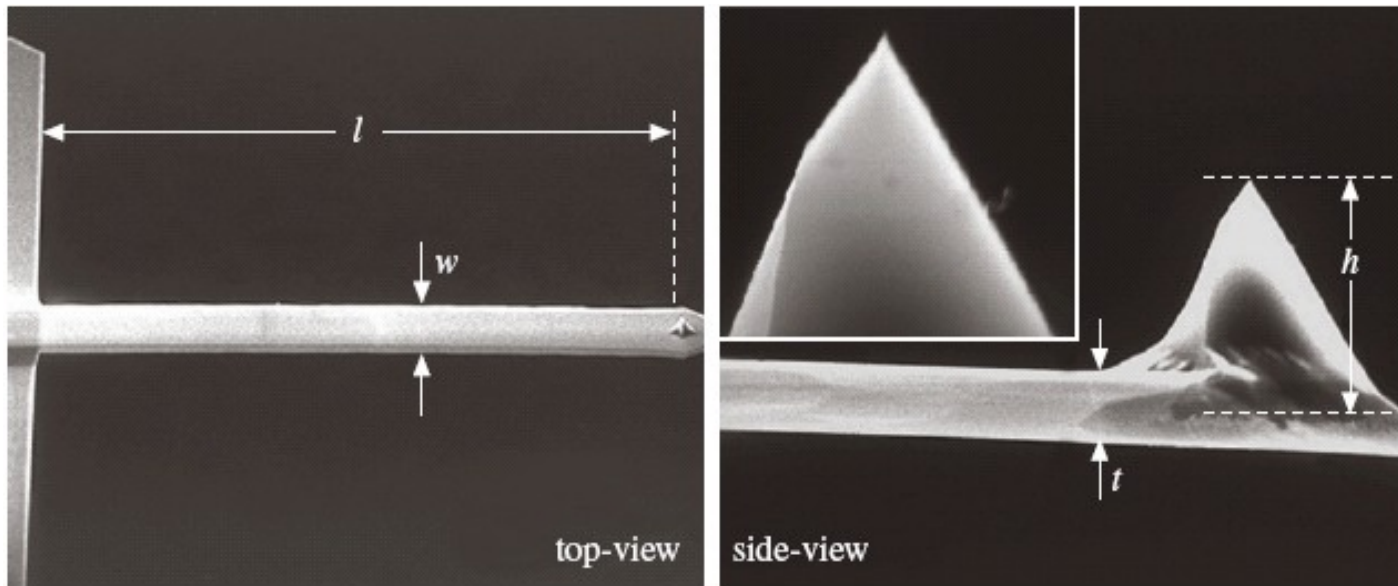
Atomic stick-slip



(friction map and friction loop on NaCl(100) in UHV)

Force Calibration

- Simple if **rectangular** cantilevers are used
- Cantilever width, thickness and length, tip height: from **SEM pictures**



Force Calibration

- Cantilever **thickness** also from the **resonance frequency**:

$$t = \frac{2\sqrt{12}\pi}{1.875^2} \sqrt{\frac{\rho}{E}} f_0 l^2$$

- ρ , E : density and Young modulus
(Nonnenmacher et al., JVSTB 1991)
- For pure silicon:

$$\rho = 2.33 \cdot 10^3 \text{ kg/m}^3$$

$$E = 1.69 \cdot 10^{11} \text{ N/m}^2$$

Force Calibration

- **Normal and lateral spring constants** of cantilever:

$$c_N = \frac{Ewt^3}{4l^3} \qquad c_L = \frac{Gwt^3}{3h^2l}$$

- G: shear modulus
- For pure silicon:

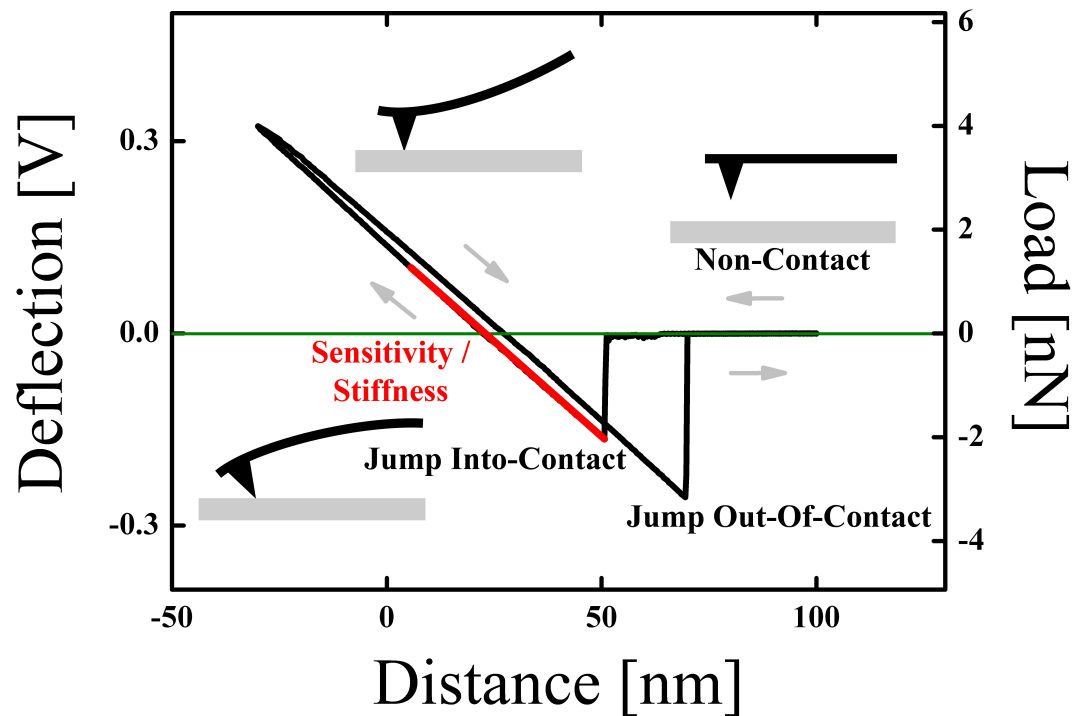
$$\rho = 2.33 \cdot 10^3 \text{ kg/m}^3$$

$$E = 1.69 \cdot 10^{11} \text{ N/m}^2$$

$$G = 0.5 \cdot 10^{11} \text{ N/m}^2$$

Force Calibration

- Next step: **sensitivity of photodetector**
- Force-distance curves on hard surfaces (e.g. Al_2O_3):



- Scanner movement = cantilever deflection
- Slope \rightarrow sensitivity

Force Calibration

- **Normal and lateral forces:**

$$F_N = c_N S_z V_N \qquad F_L = \frac{3}{2} c_L \frac{h}{l} S_z V_L$$

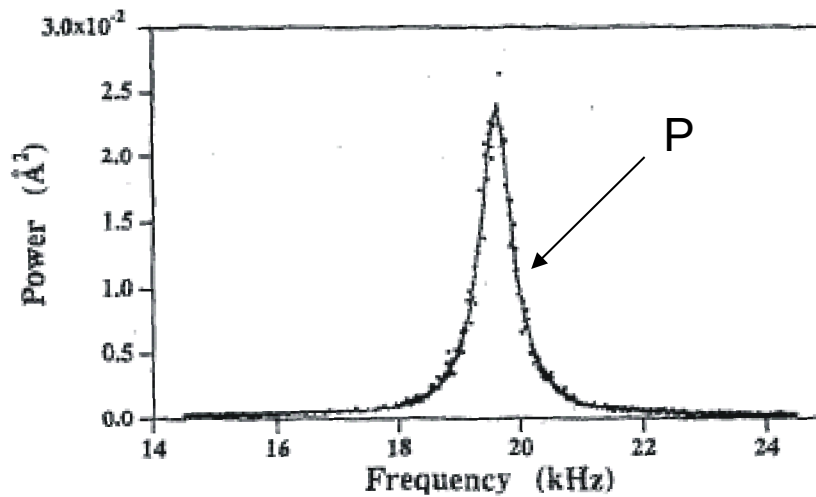
(if the laser beam is above the probing tip!)

- V_N , V_L : normal and lateral signals

Force Calibration

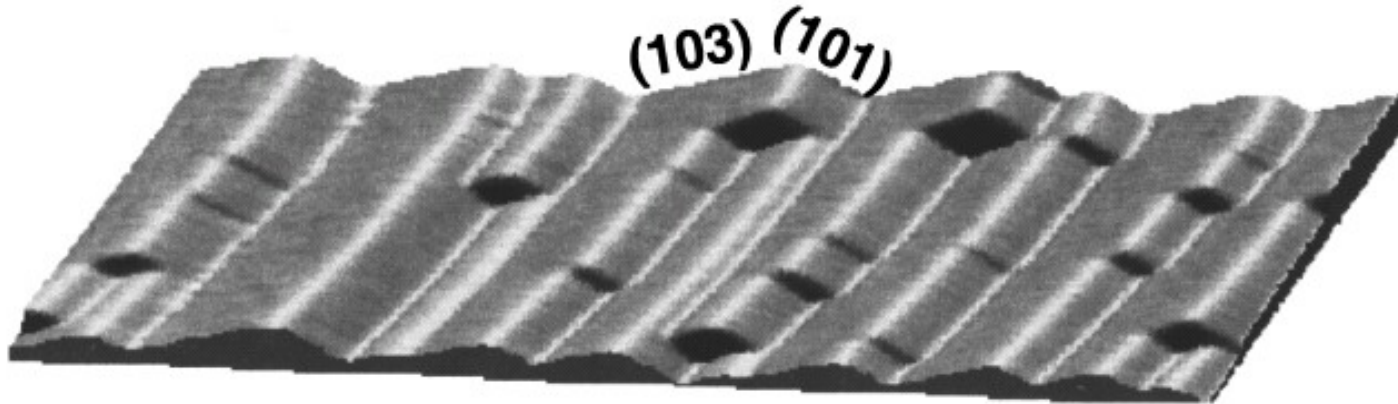
- Alternative method: Spring constant **from thermal power spectrum** (Hutter et al., RSI 1993)
- Correct relation (Butt et al., Nanotech. 1995):

$$c_N = \frac{4k_B T}{3P}$$

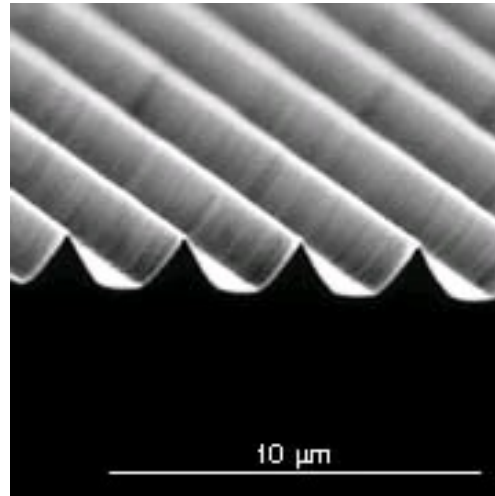


Force Calibration

- Alternative method: Scanning over profiles with **well-defined slope** (Ogletree et al., RSI 1996)



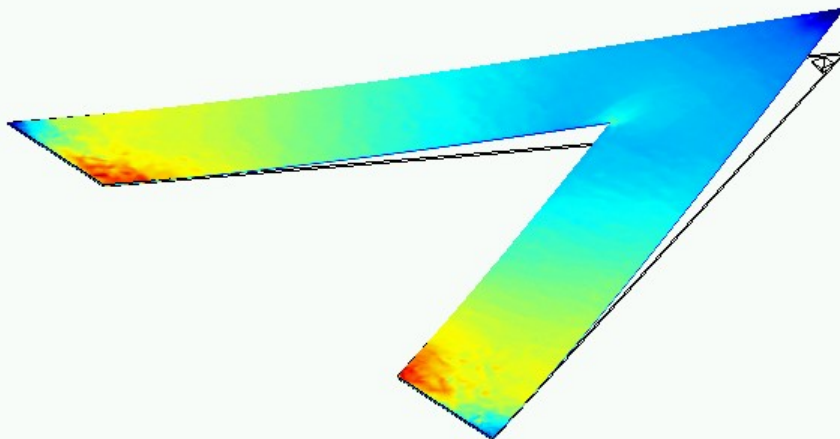
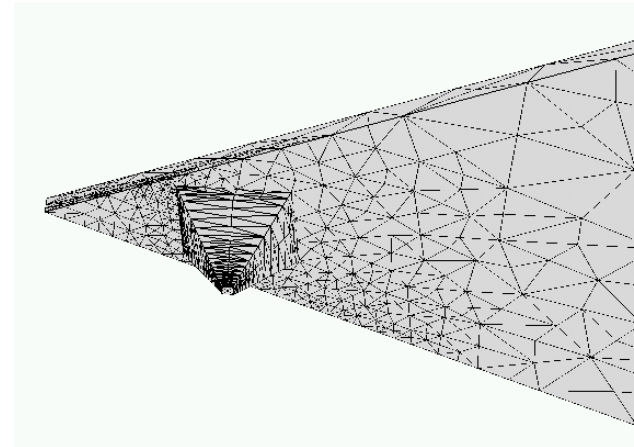
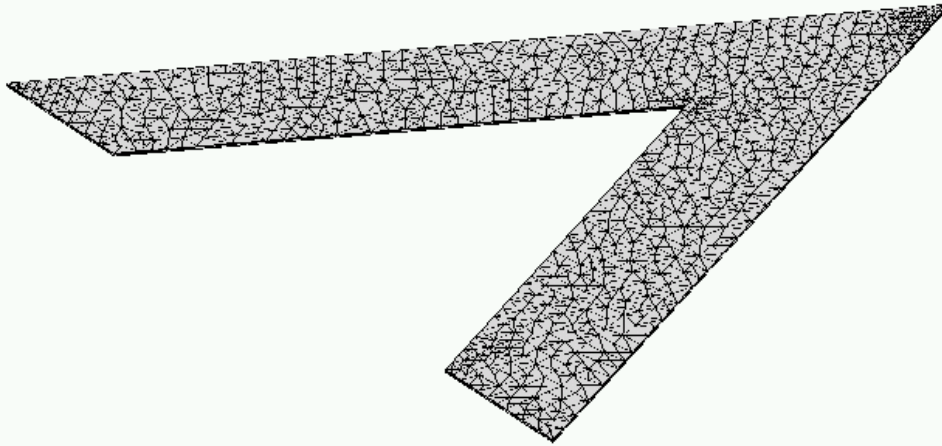
Commercially available grating:



(TGG01, NT-MDT,
Moscow)

Force Calibration

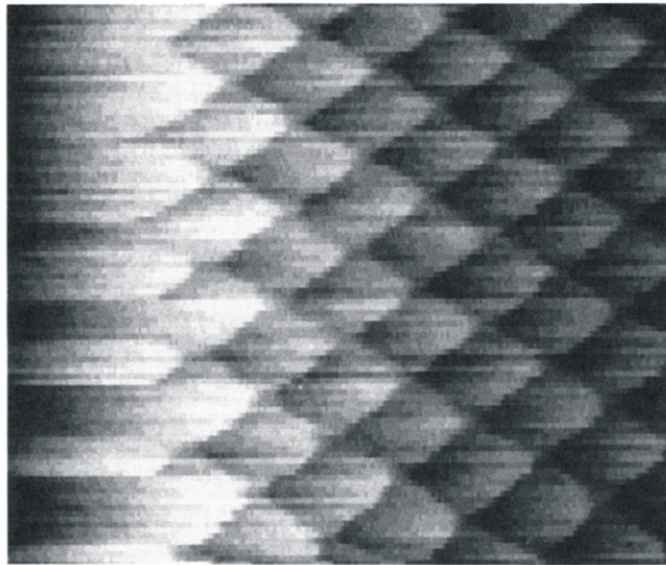
- Different shapes → **Finite elements analysis**



(Femlab™ 3.0)

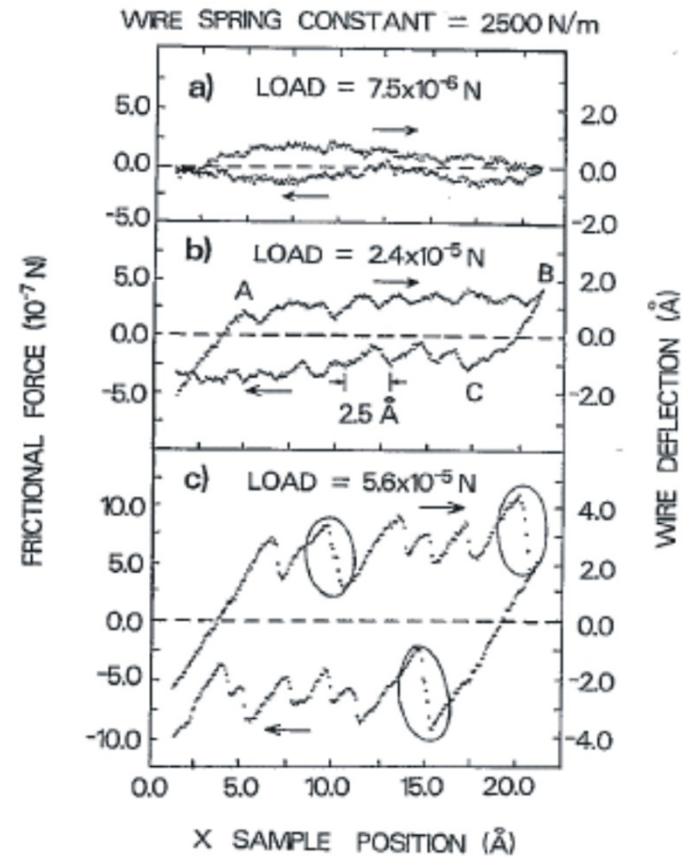
Atomic-Scale Measurements

- Atomic friction on **graphite**:



2 nm

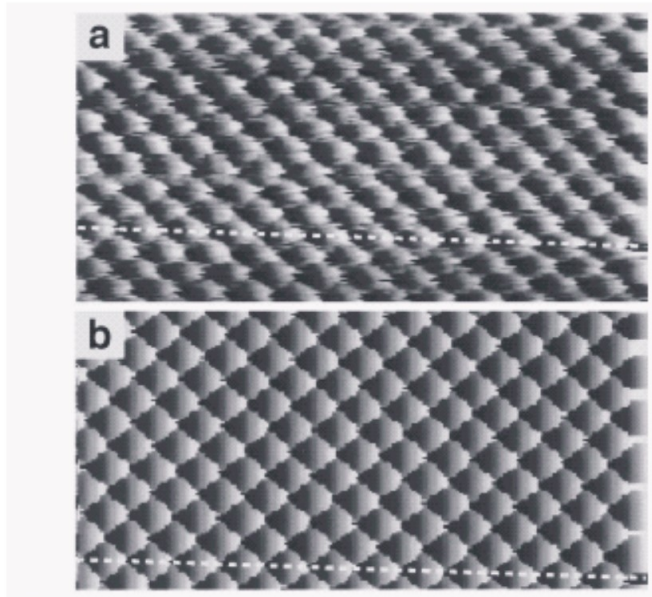
(Mate at al., PRL 1987)



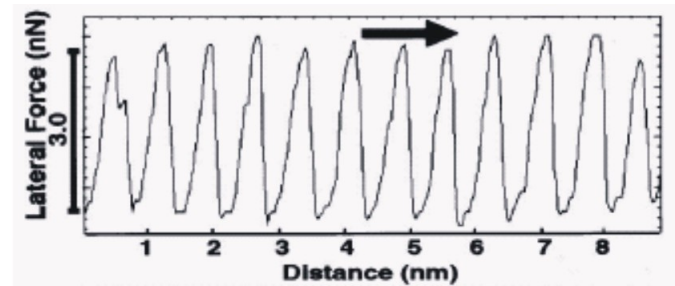
Atomic-Scale Measurements

- Friction on **insulating surfaces** (Lüthi et al., JVSTB 1996):

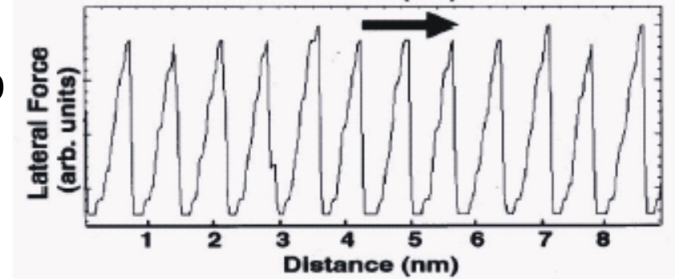
KBr



Exp



Theo

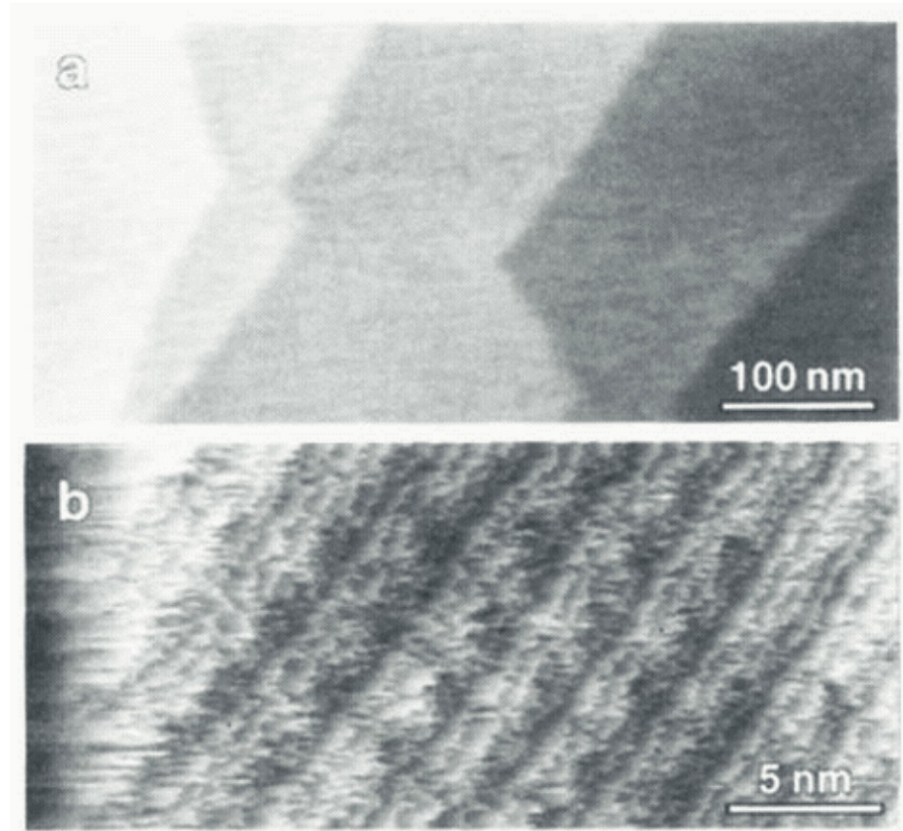


- No individual defects are observed

Atomic-Scale Measurements

- Friction on **semiconductors** (Howald et al., PRB 1995):

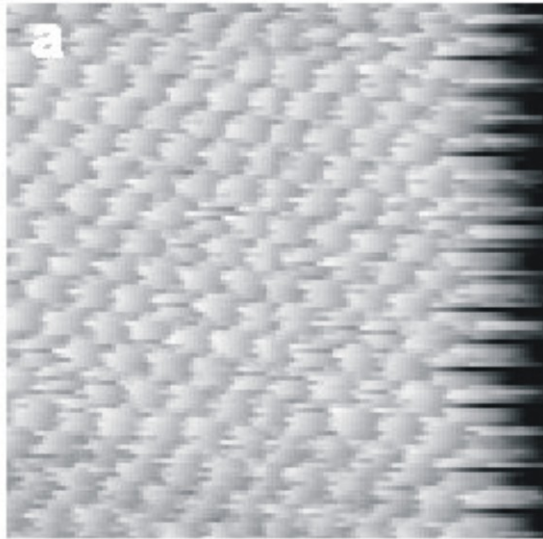
Si(111)7x7



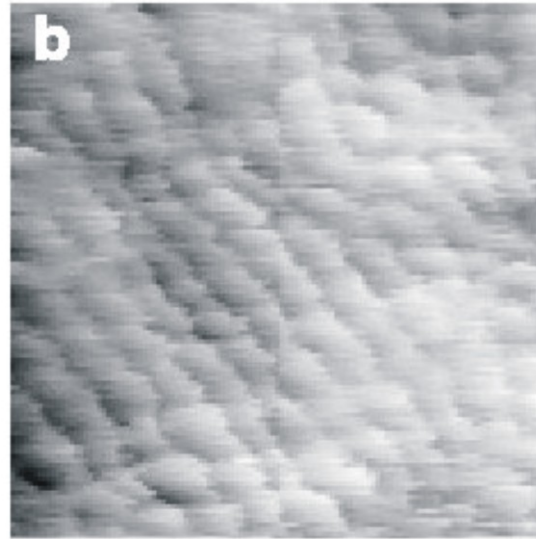
(tip coated with PTFE)

Atomic-Scale Measurements

- Friction on **metal surfaces** (Bennewitz et al., Trib. Lett. 2001):



Cu(111)

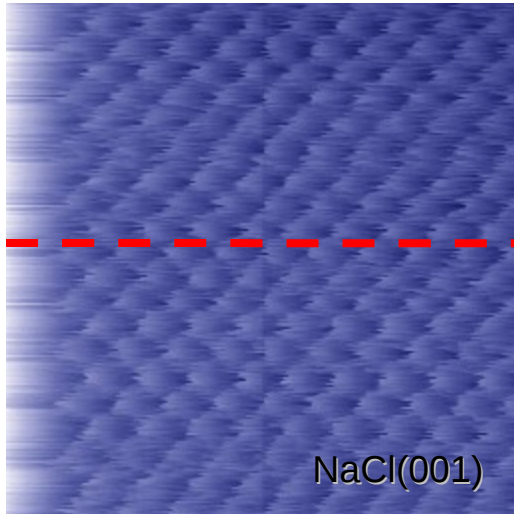


Cu(100)

Irregular features on the (100) surface (less packed!)

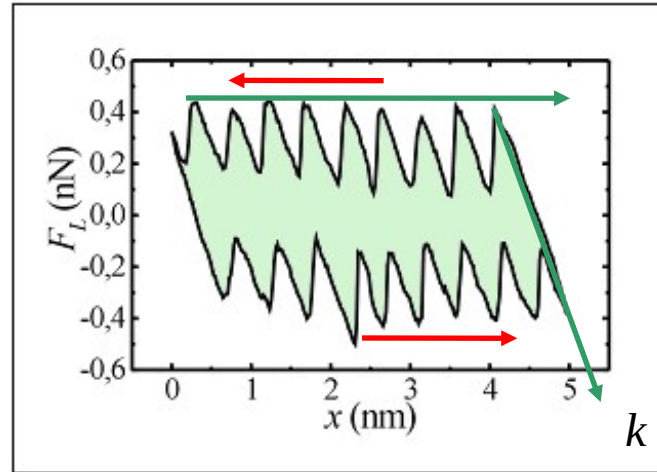
Atomic friction on crystal surfaces

Our model systems: alkali halide surfaces (easy preparation, simple structure)

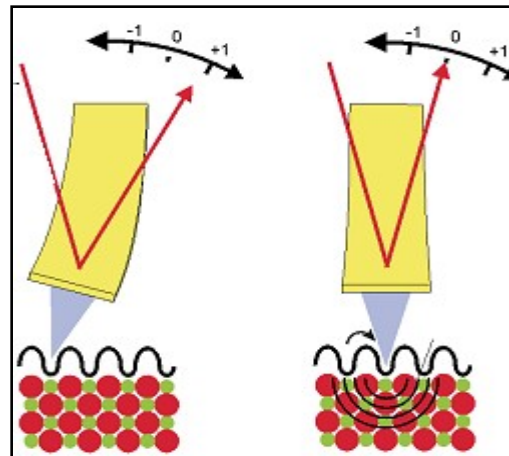


5 nm

Tomlinson model:
(Phyl. Mag. 1929)



$$F_L^{\max} = \frac{2\pi V_0}{a}$$

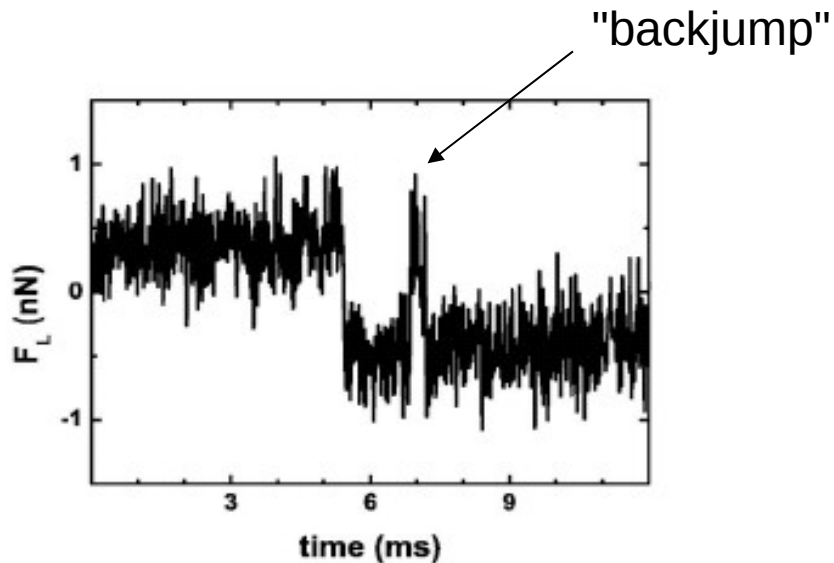
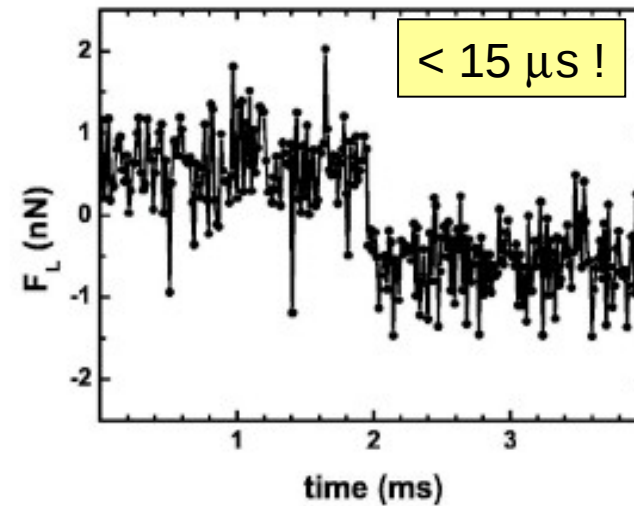
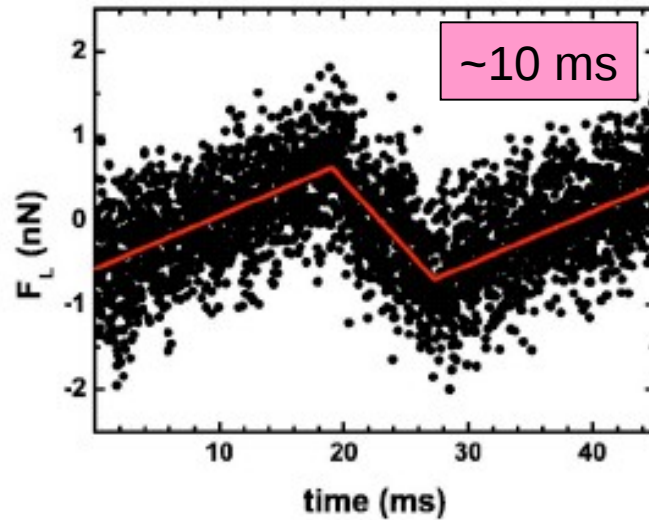


$$V_0 \sim 1 \text{ eV}$$

$$k \sim 1 \text{ N/m}$$

Atomic-Scale Measurements

- Wide distribution of slip durations:



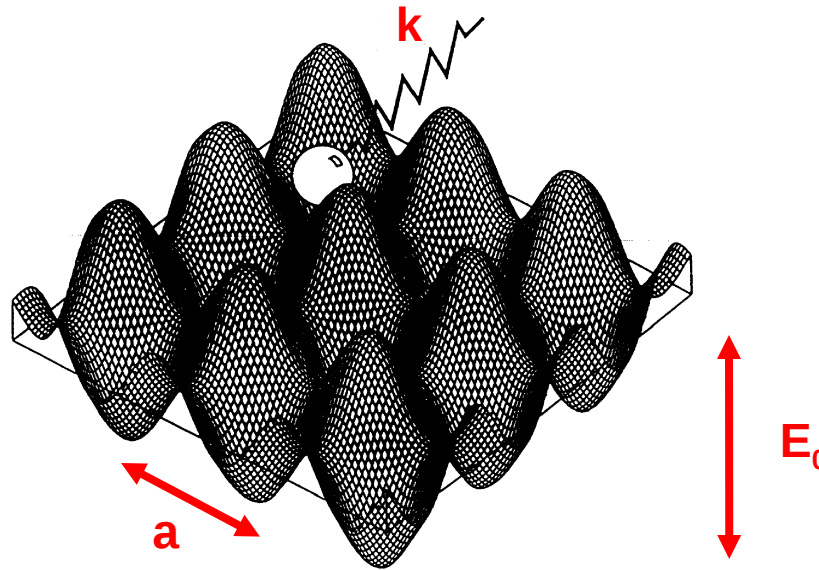
Why?

Modelling Atomic Friction

- The tip is subject to

1) periodic interaction with the underlying surface

2) elastic deformation of the cantilever



- In 1D the corresponding potential energies are represented by

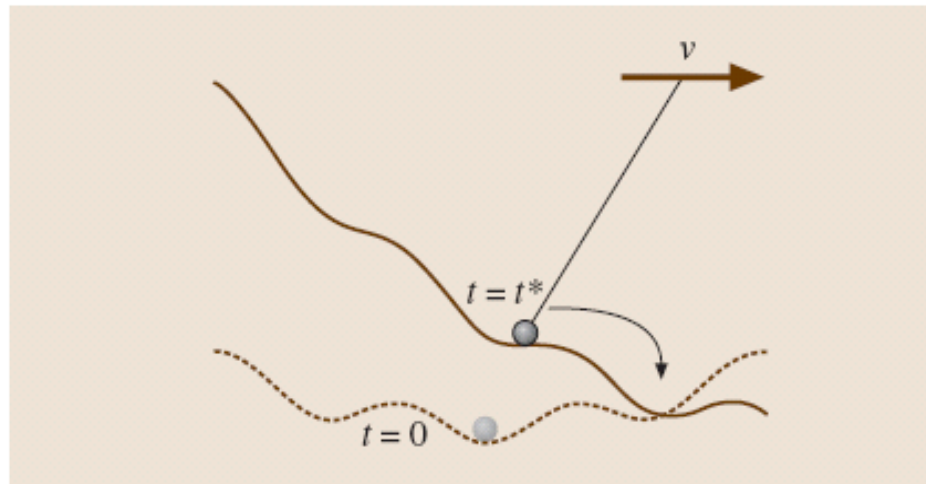
1) a sinusoid

2) a parabola

Modelling Atomic Friction

- Total energy of the system:

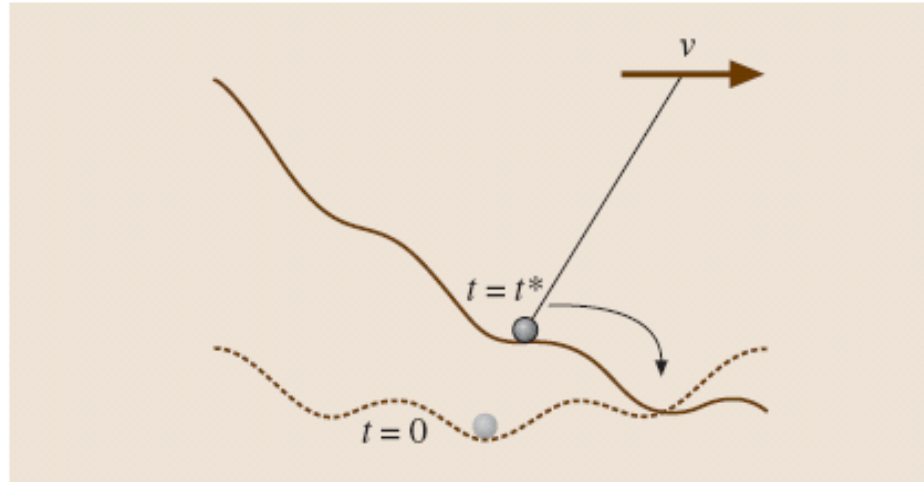
$$U_{\text{tot}}(x, t) = -\frac{E_0}{2} \cos \frac{2\pi x}{a} + \frac{1}{2} k_{\text{eff}} (vt - x)^2$$



- The tip can "stick" to the minima of the potential profile

Modelling Atomic Friction

- Tip position at a given time t :
$$\frac{\partial U_{\text{tot}}}{\partial x} = \frac{\pi E_0}{a} \sin \frac{2\pi x}{a} - k_{\text{eff}}(vt - x) = 0$$

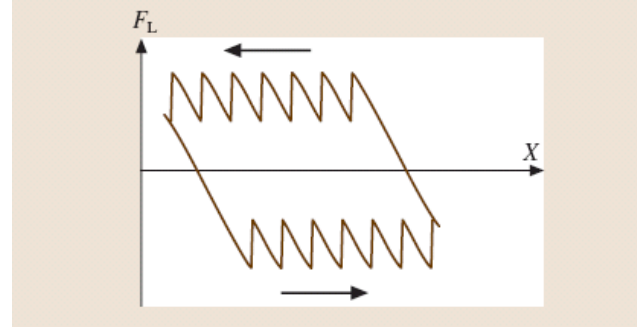
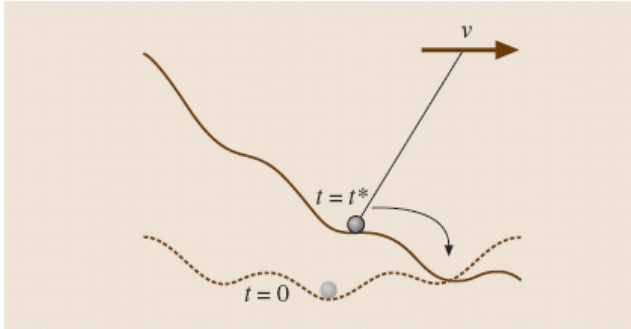


- Critical position (reached at $t = t^*$):

$$x^* = \frac{a}{4} \arccos\left(-\frac{1}{\eta}\right) \quad \eta = \frac{2\pi^2 E_0}{k_{\text{eff}} a^2}$$

- **Frictional parameter η** \rightarrow tip-surface interaction vs. lateral stiffness

Modelling Atomic Friction

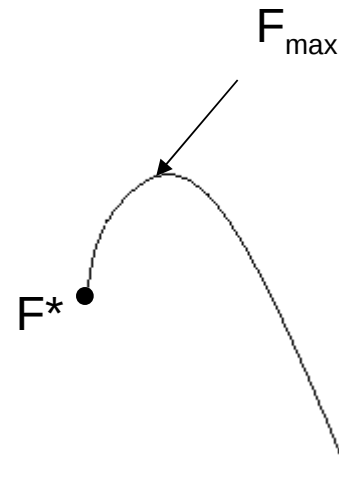


- Critical lateral force (at $t = t^*$):

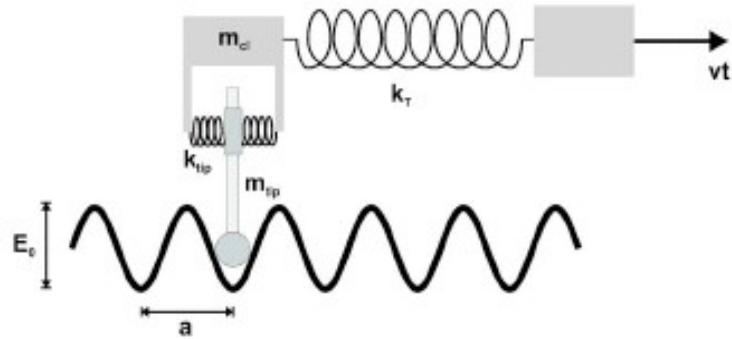
$$F^* = \frac{\pi E_0}{a} \sqrt{1 - \frac{1}{\eta^2}}$$

- Note that $F^* < F_{\max}$!

$$F_{\max} = \frac{\pi E_0}{a}$$

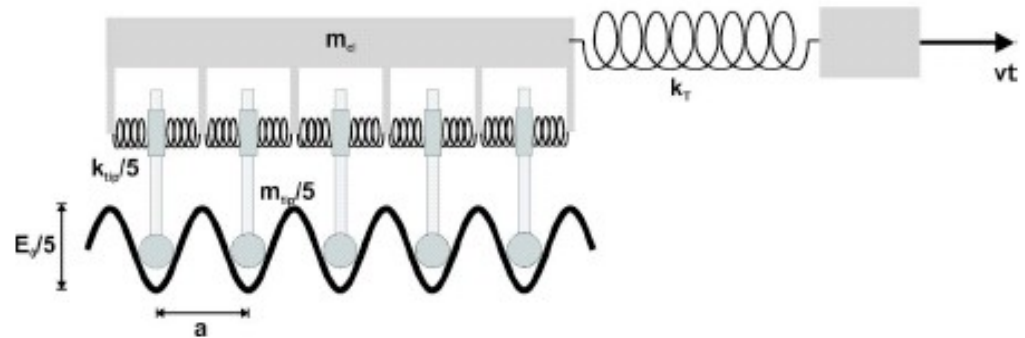


Modelling Atomic Friction



single contact

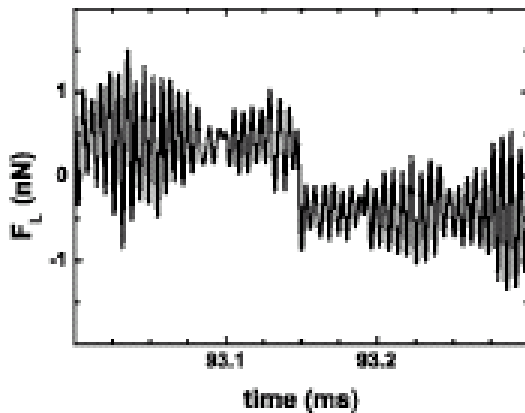
multiple contact



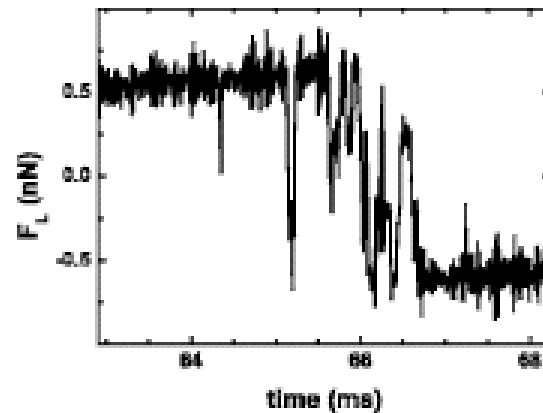
Modelling Atomic Friction

- Tip \rightarrow Langevin equation (including thermal noise)
- Cantilever \rightarrow Newton equation (without thermal noise)

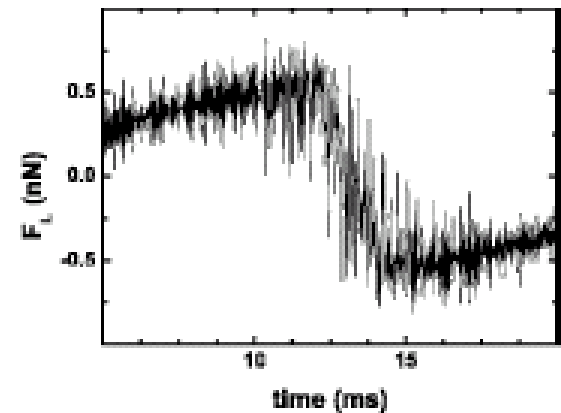
One tip



Three tips



Five tips

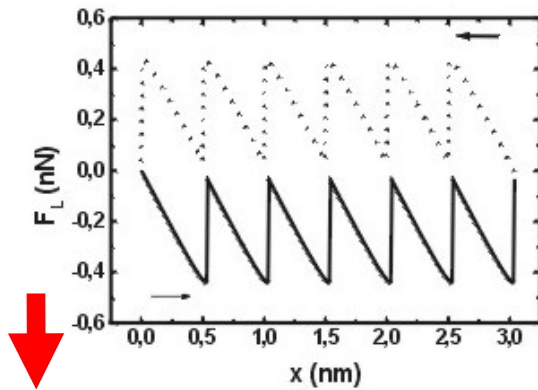


- Long slip times are found with multiple tips only
(Maier et al., PRB 2005)

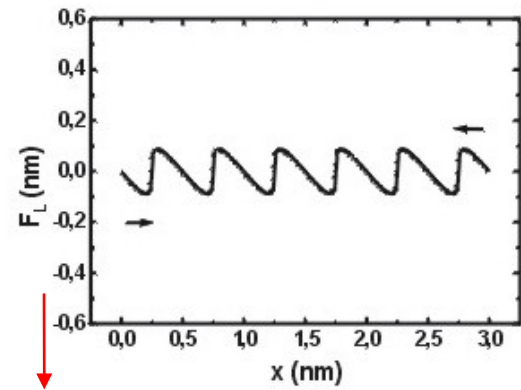
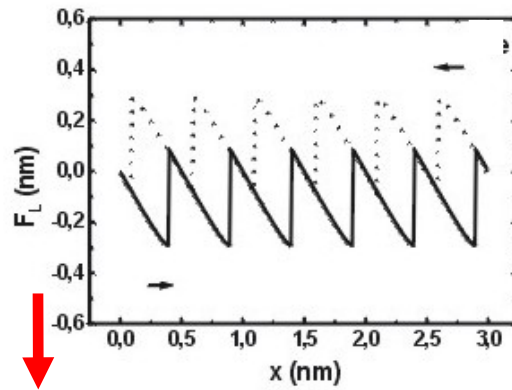
Superlubricity

- From the Tomlinson model (without thermal activation):

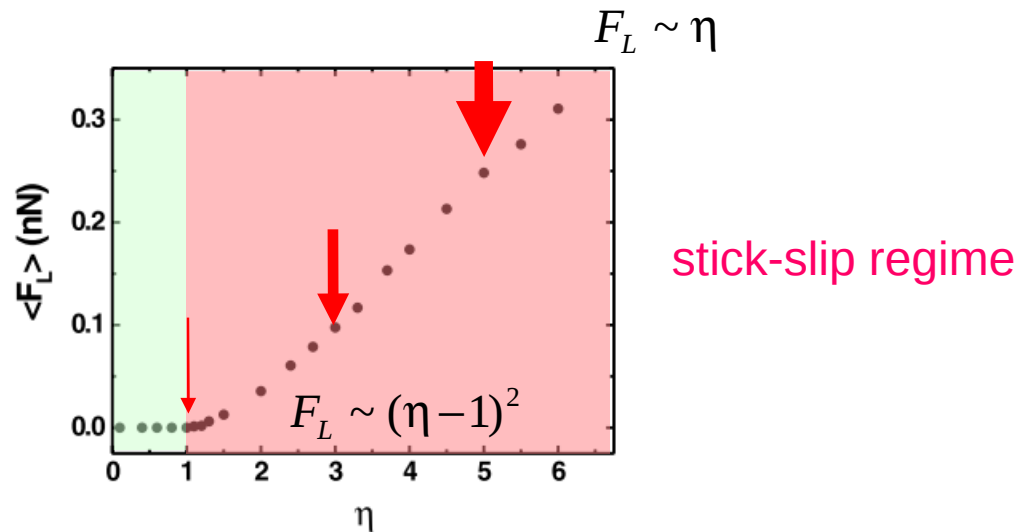
$$\eta > 1$$



$$\eta < 1$$

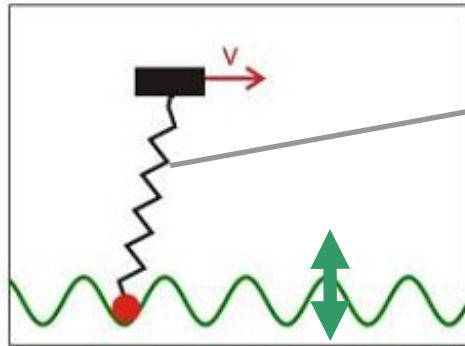


superlubric regime



“Dynamic superlubricity”

A third way to reduce friction: Tomlinson model with TIME modulation



$$V_{\text{elas}} = \frac{ka^2}{2}$$

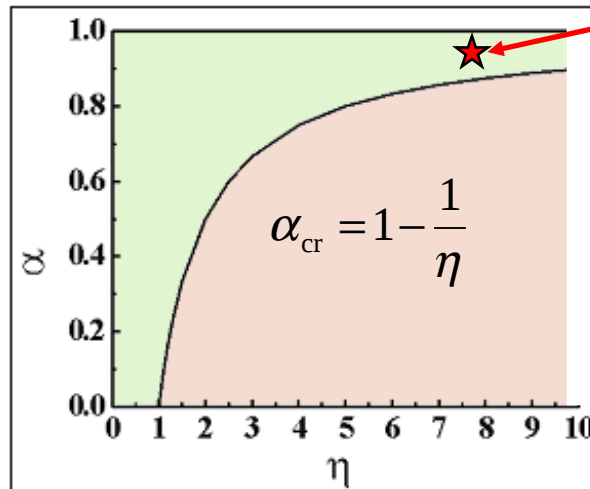
$$V_{\text{int}} = V_0 \cos \frac{2\pi x}{a}$$

$$\eta = \frac{(2\pi)^2 V_0}{ka^2}$$

Phase-diagram in the η - α plane:

$$V_0 \rightarrow V_0(1 + \alpha \cos \omega t)$$

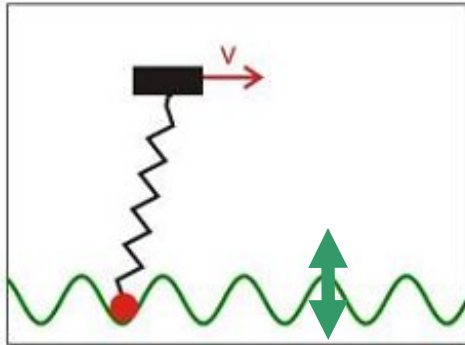
$$\eta_{\text{eff}} = \eta(1 - \alpha)$$



High loads
can be applied!

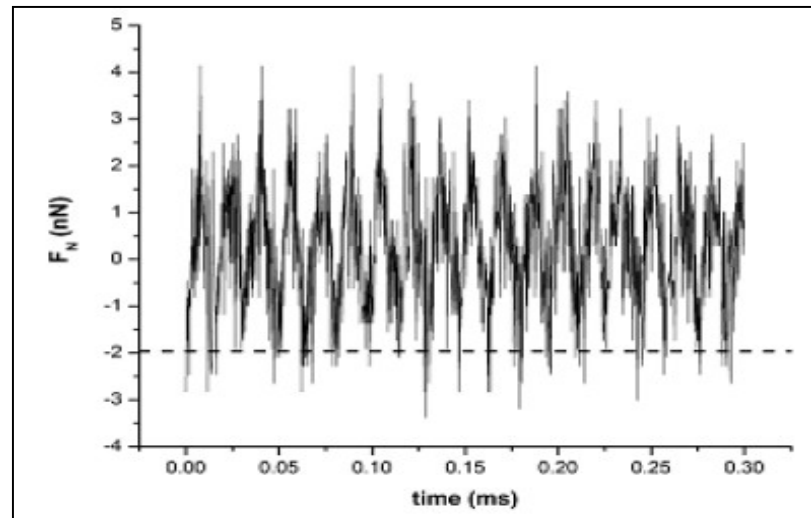
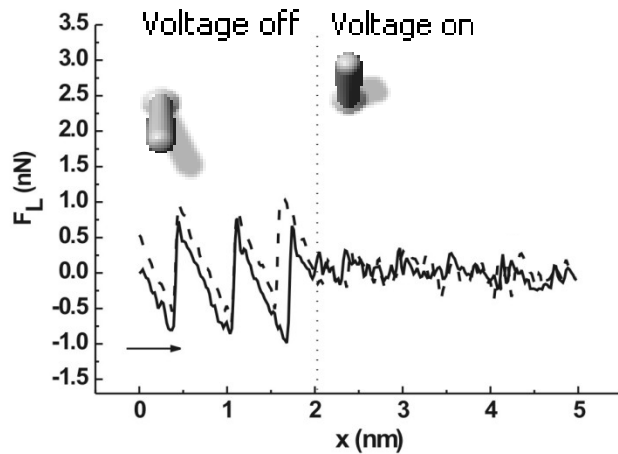
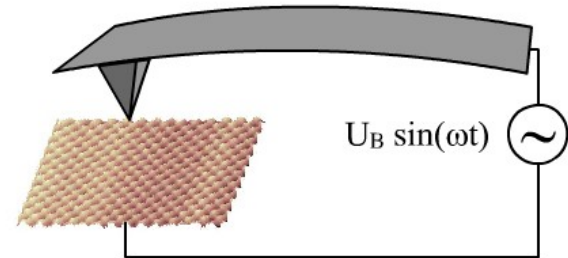
“Dynamic superlubricity”

A third way to reduce friction: Tomlinson model with TIME modulation



NaCl(001)

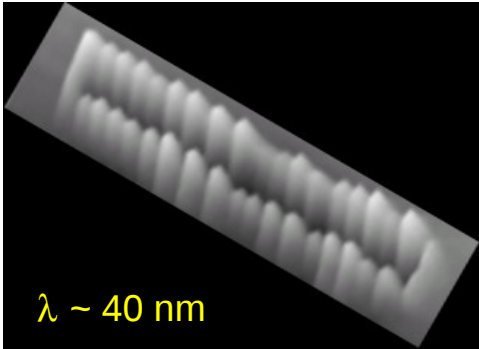
AC actuation
of the nanocontact:



Abrasion wear at the nanoscale

Ripples induced by localized abrasion:

Scratching single lines:

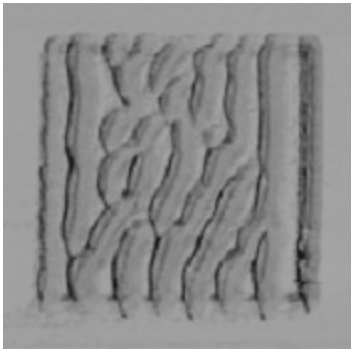


Analogies to waterjet cutting:

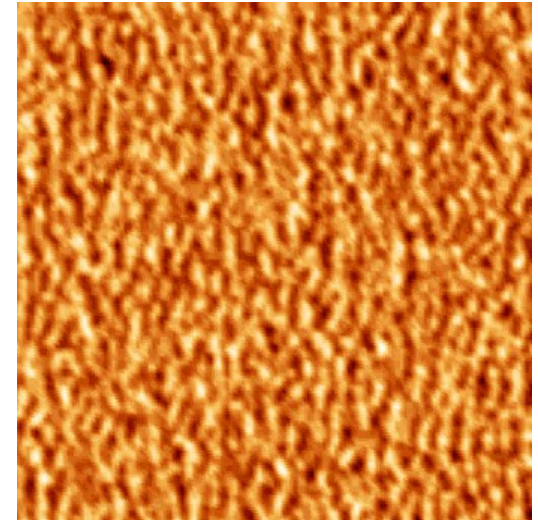
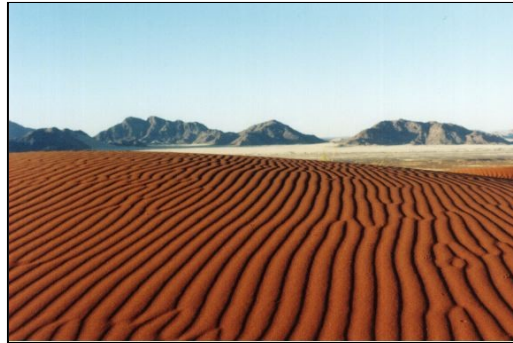


- Combined erosion and relaxation
- Thermal activation of atomic-scale wear
- Numerical analysis in progress

Scratching square areas:



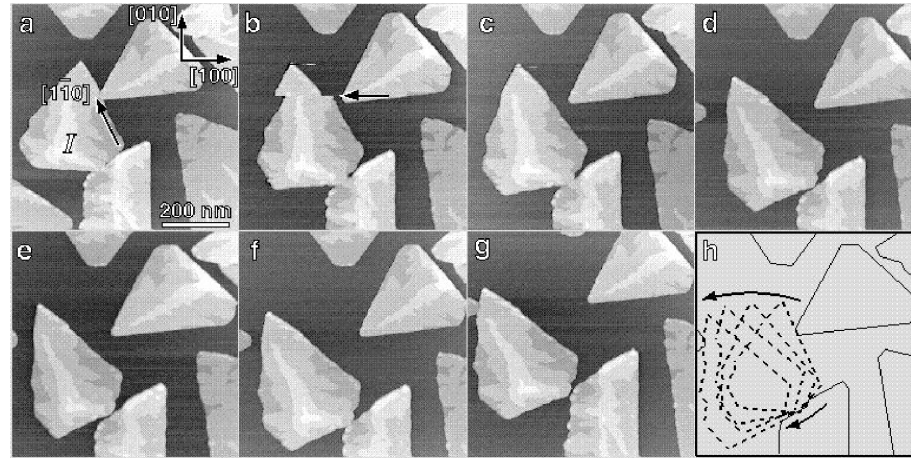
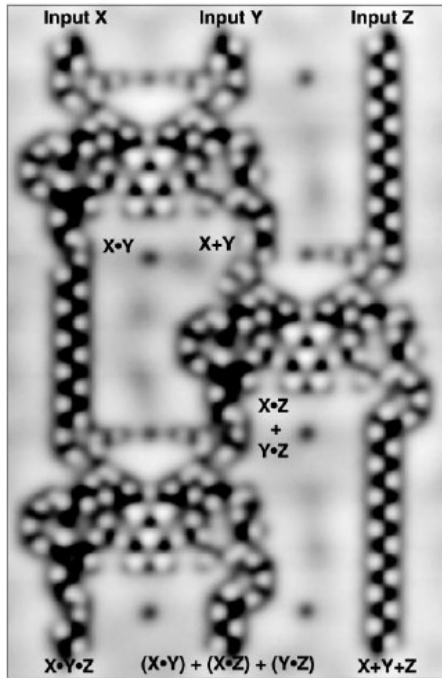
Analogies to sand ripples:



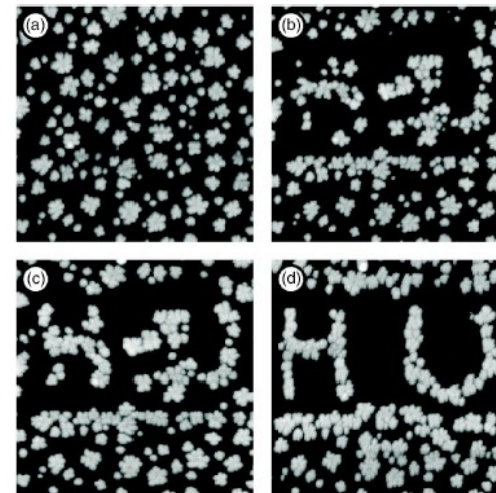
Nanomanipulation Techniques

By STM (Heinrich et al. 2002):

By contact AFM (Lüthi et al. 1994):



By dynamic AFM
(Ritter et al. 1995):

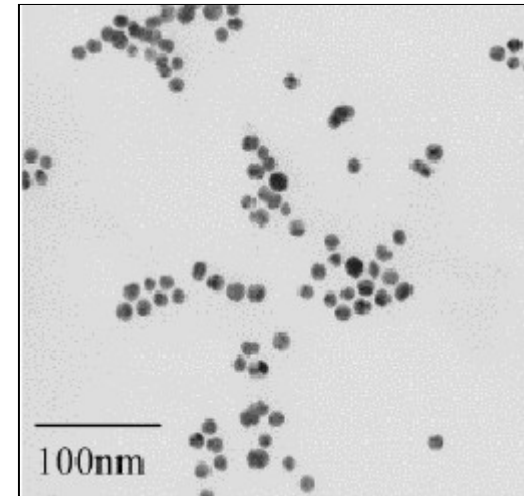


Manipulation of Nanoparticles



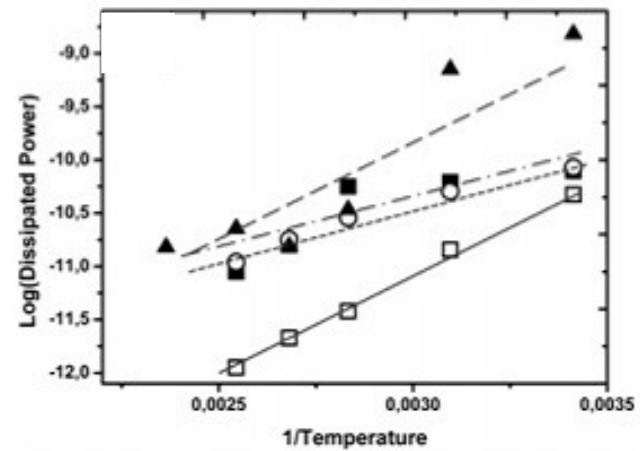
Power dissipation at onset of motion:

$$P_{\text{tip}} = \frac{1}{2} \frac{k\omega_0}{Q_{\text{cant}}} (Q_{\text{cant}} A_d A \sin \varphi - A^2)$$



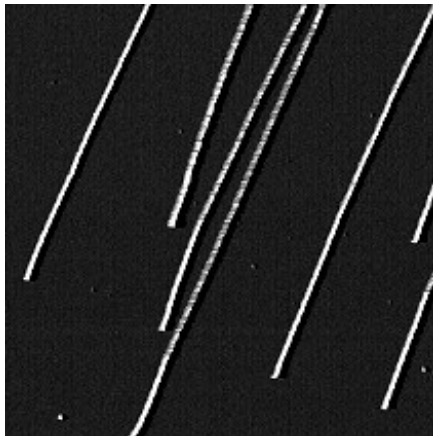
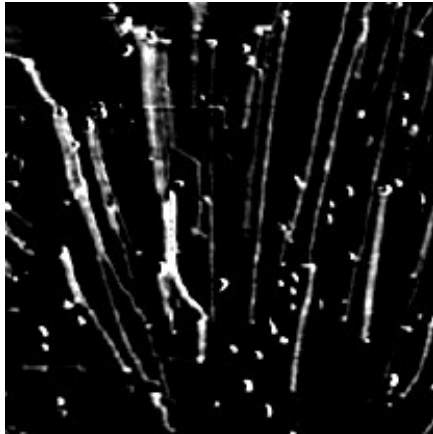
Thermal activation of the particle motion:

$$\log P_{\text{tip}} \sim \frac{1}{T}$$

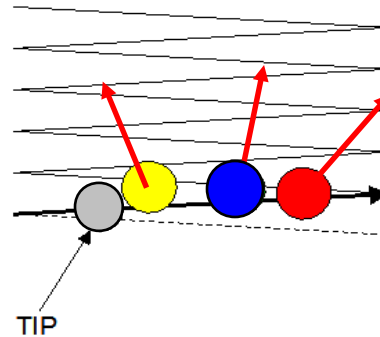


Manipulation of Nanoparticles

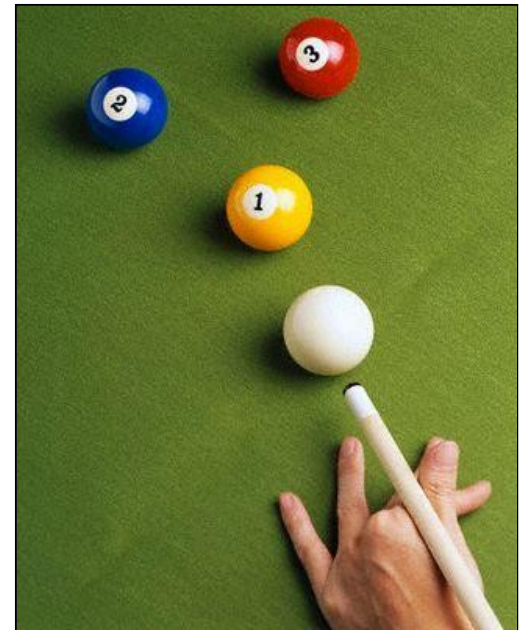
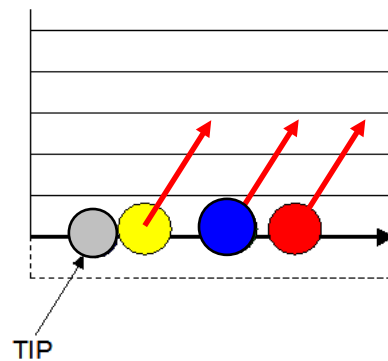
Nanoparticles can be “scattered” by the AFM tip:



Zigzag path:

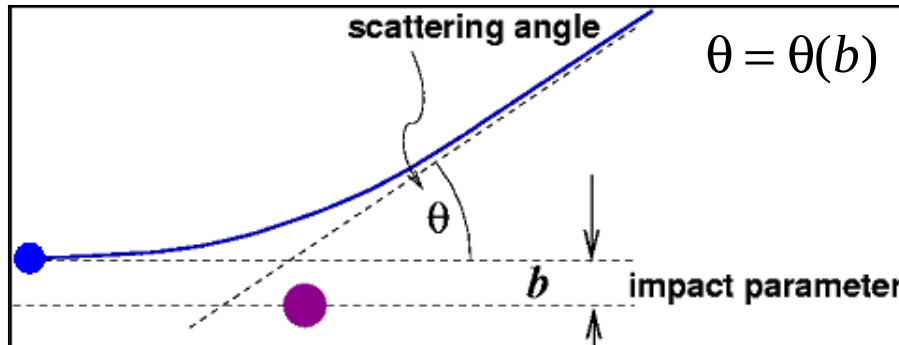


Raster path:

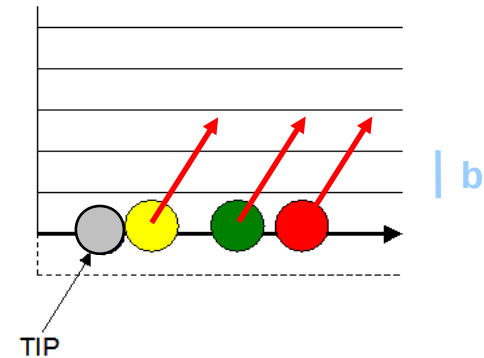


Manipulation of Nanoparticles

Analogy to classical scattering:



Raster path:



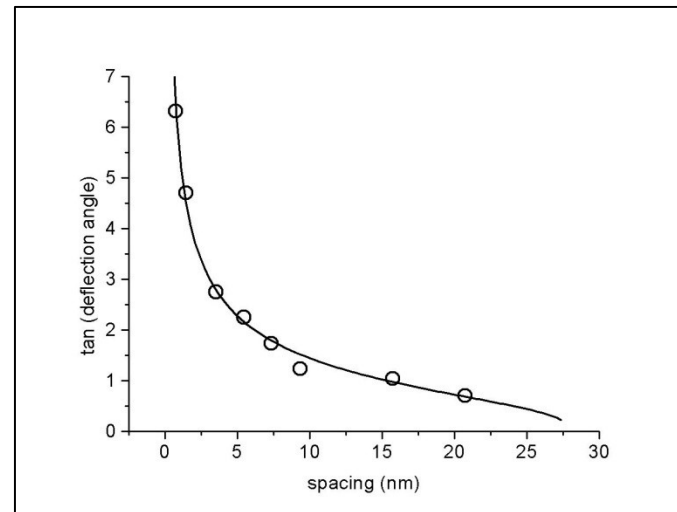
In tapping AFM:

$$\tan \theta = -\frac{b}{2R \left(\cos \alpha_0 + \log \tan \frac{\alpha_0}{2} \right)}$$

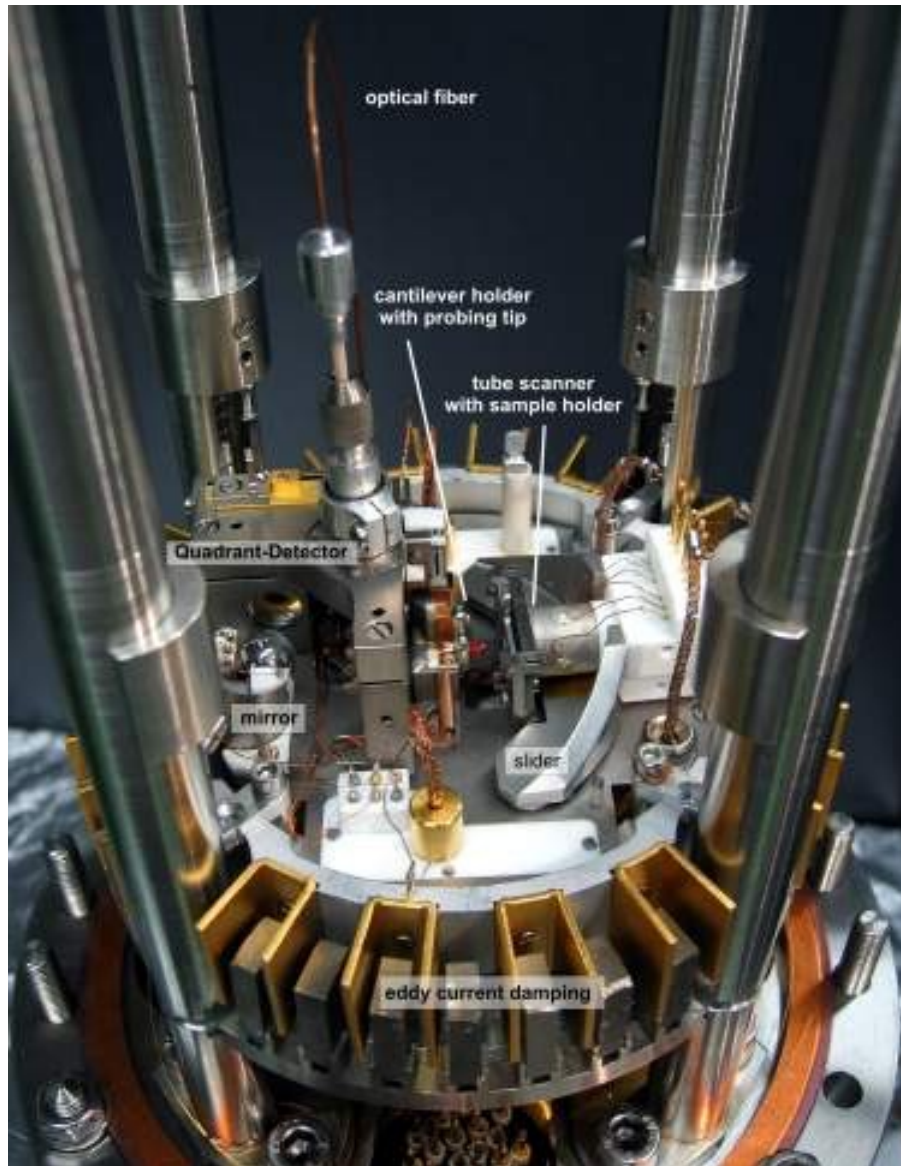
$$\alpha_0 = \arcsin \left(1 - \frac{b}{R} \right)$$

R ← tip and particle radii

b : spacing between consecutive scan lines

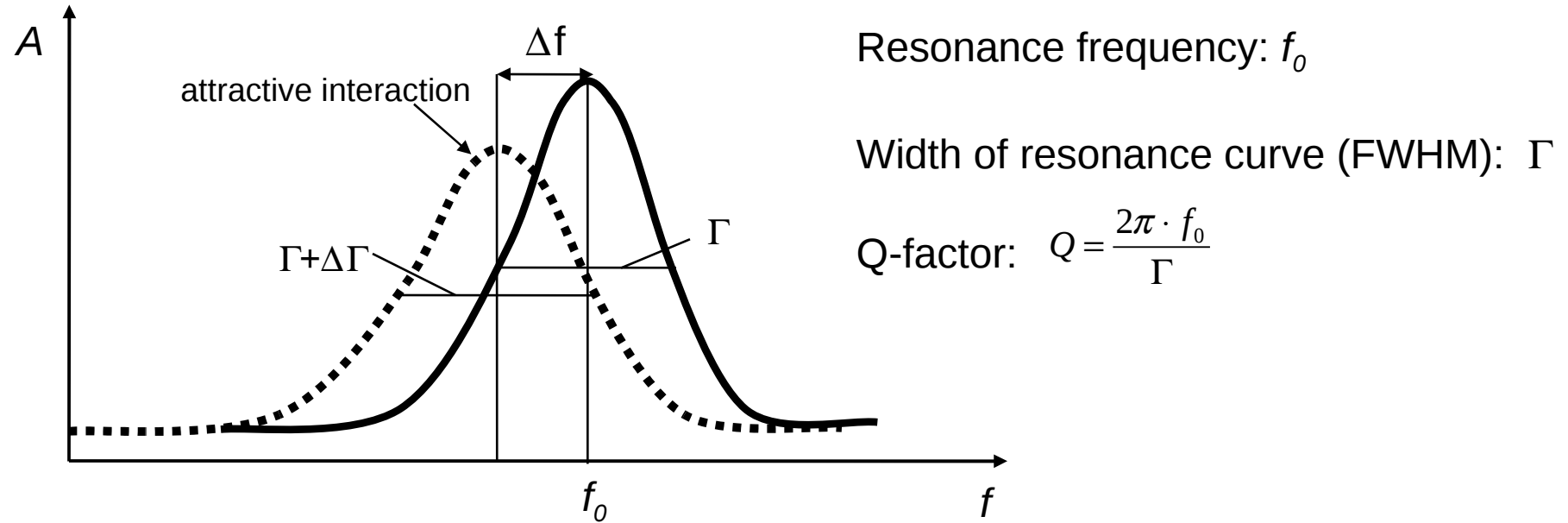


Noncontact-AFM (nc-AFM)



- UHV: Base pressure below 1×10^{-10} mbar
- Operation at room temperature
- Mixed mode: AFM/STM
- Beam deflection method
- Bandwidth of the photodetector: 3MHz
- Evaporation of molecules from a k-cell kept at 165°C or 170°C

Quantitative understanding of nc-AFM



Conservative forces \Rightarrow shift of resonance curve Δf
Dissipative forces \Rightarrow broadening of curve $\Delta\Gamma$

Forces in nc-AFM

Frequency modulation: $f_0 = \frac{1}{2\pi} \sqrt{\frac{k}{m^*}}$ $\Delta f = -\frac{f_0}{2k} \frac{\partial F_{tot}}{\partial z}$

⇒ measured topography = surface of constant $\frac{\partial F}{\partial z}$

$$F_{tot} = F_{chem} + F_{mag} + F_{el} + F_{vdW}$$

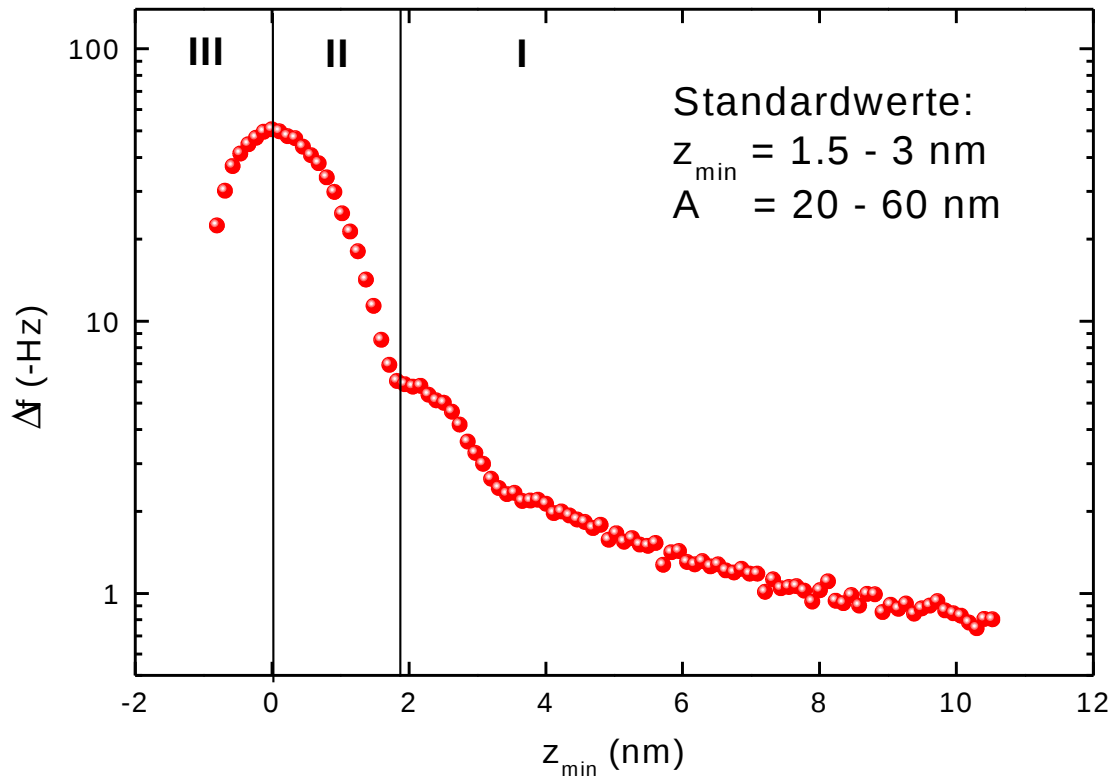
bonding between
tip and sample
atoms
(only for $d < 5 \text{ \AA}$)

only for
magnetically
sensitive tips

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V^2$$

$$F_{vdW} = -\frac{HR}{6d^2}$$

Dynamic Mode, non-contact

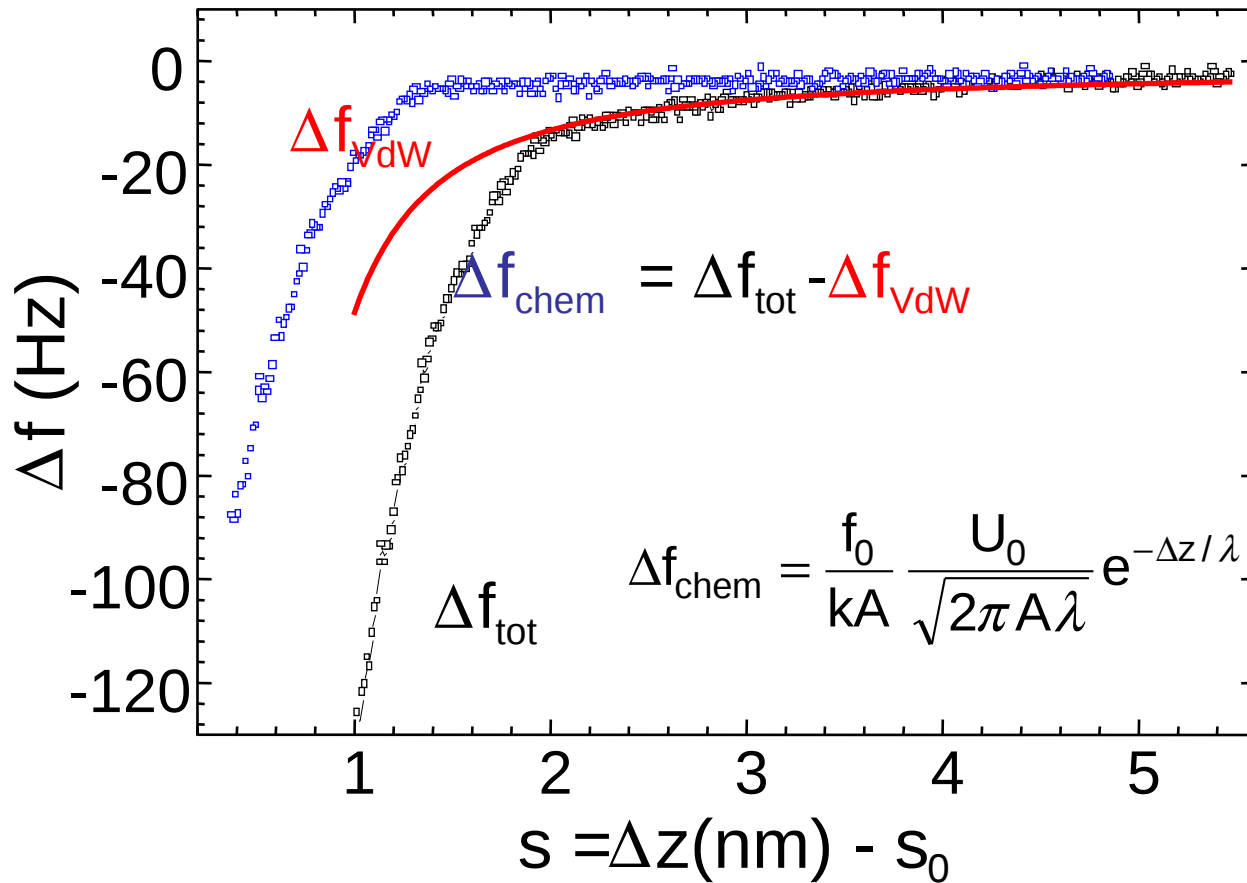


region I:
attractive forces
non-contact mode

region II:
attractive forces
atomic resolution

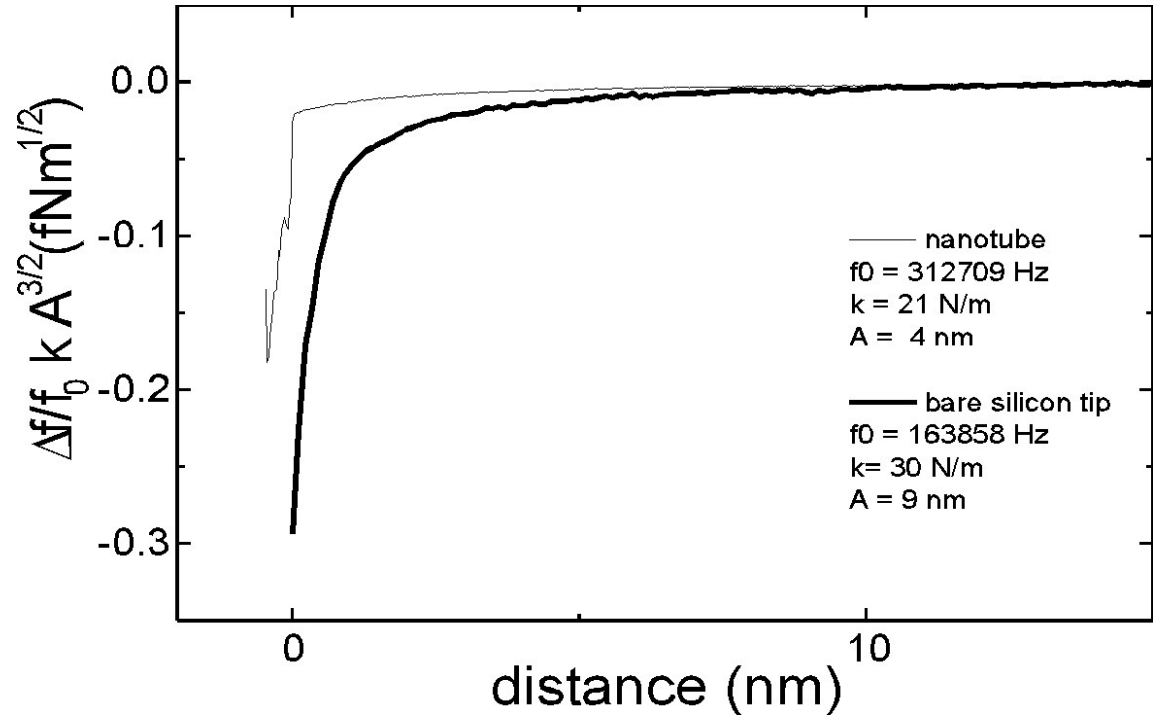
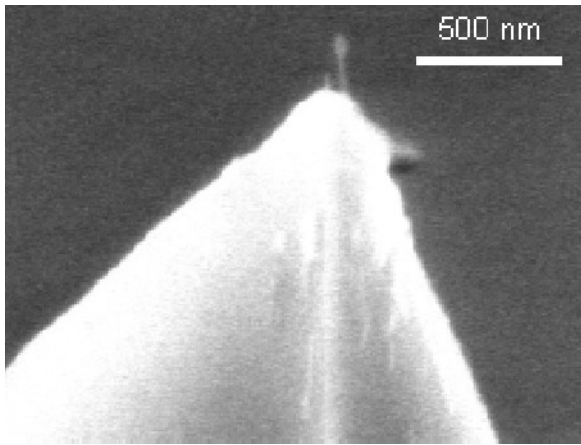
region III:
repulsive forces
tapping mode

Short range interaction



$\lambda = 0.35 \text{ nm}$
 $U_0 = -4.7 \text{ eV}$
 $s_0 = 0.45 \text{ nm}$

Carbon nanotubes as probing tips for nc-AFM

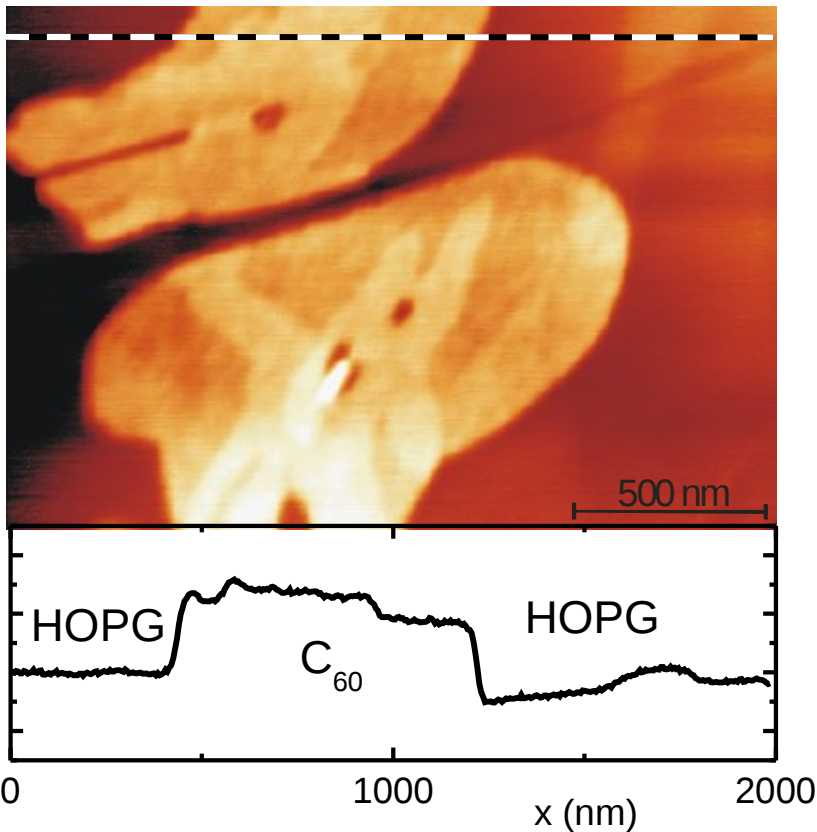


⇒ Long-range forces are reduced

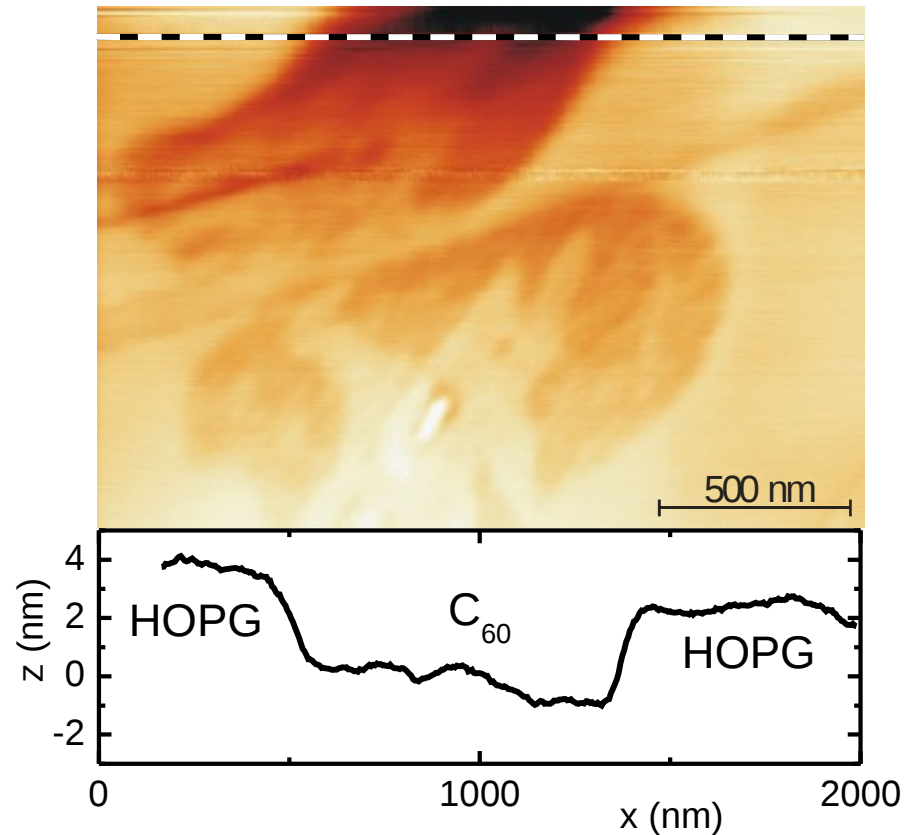
inhomogeneous sample: HOPG + $\frac{1}{2}$ monolayer C₆₀

Topography

$V_{\text{bias}} = 0 \text{ V}$



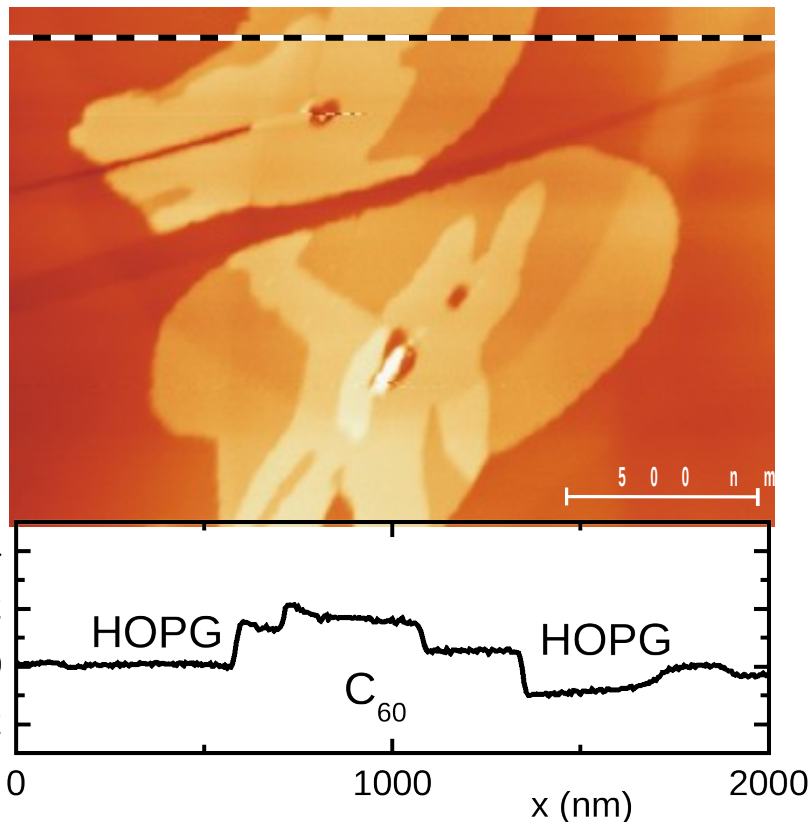
$V_{\text{bias}} = 1.34 \text{ V}$



→ contrast inversal: HOPG ↔ C₆₀

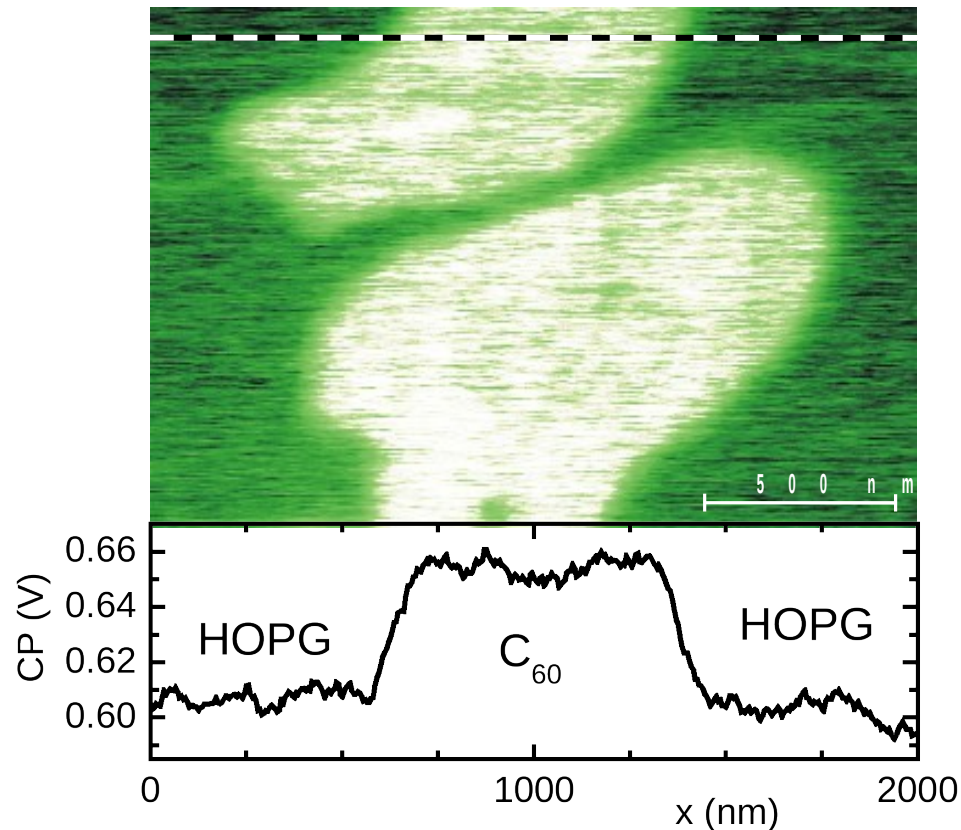
inhomogeneous sample: HOPG + $\frac{1}{2}$ monolayer C60

topography



HOPG: $V_{CP} \cong 0.61$ V

contact potential

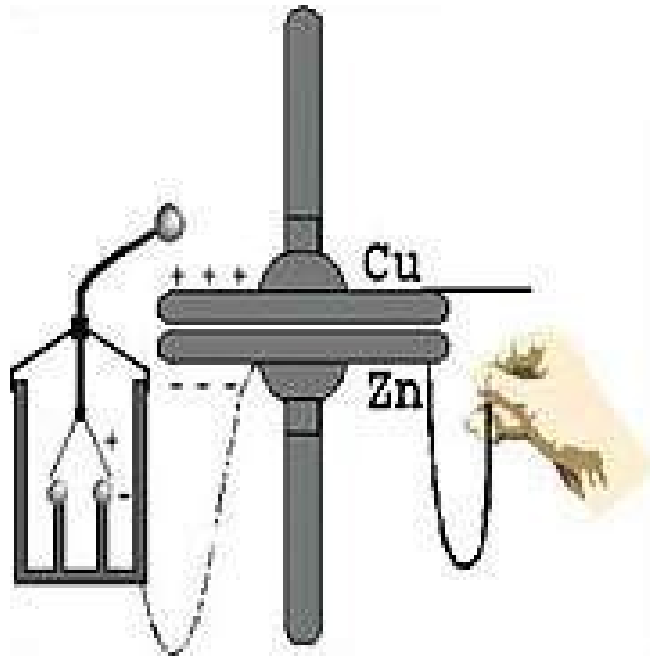


C₆₀: $V_{CP} \cong 0.66$ V

\Rightarrow NC-AFM: residual electrostatic force for fixed V_{bias}

Makroskopische Kelvin-Sonde

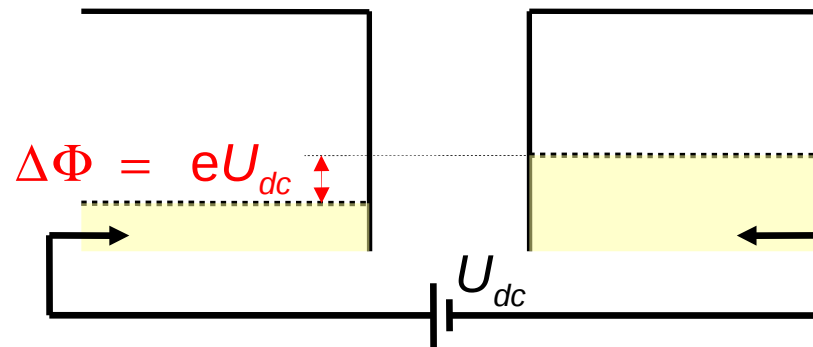
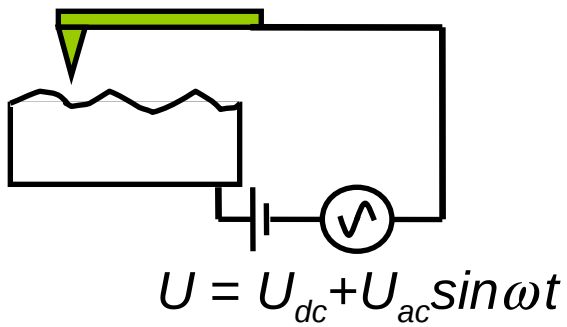
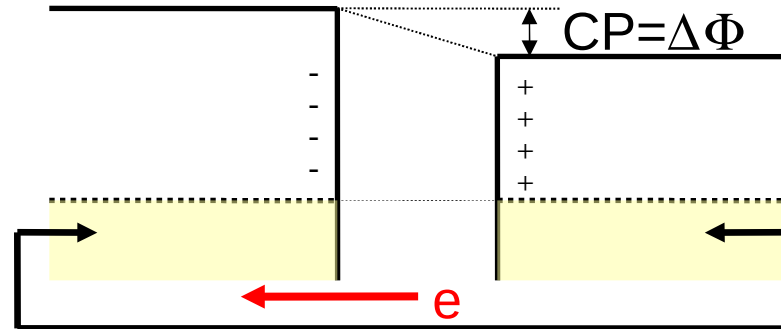
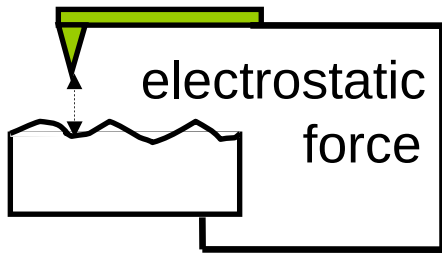
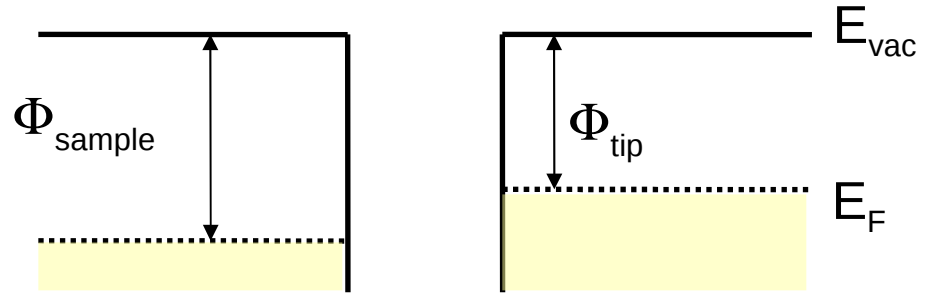
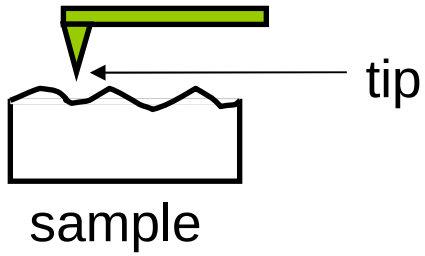
Lord Kelvin 1861



Verschiebestrom

$$I(t) = (U_{dc} - U_{CPD}) f \Delta C \cos \omega t.$$

Kelvin Principle



Electrostatic Forces in nc-AFM

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V_{eff}^2 \quad \Rightarrow \quad F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} (V_{bias} - V_{CP})^2$$

$$V_{CP} = 1/e \cdot (\Phi_{tip} - \Phi_{sample})$$

contact potential

Φ - work function

apply bias:

$$V_{bias} = V_{dc} + V_{ac} \cdot \sin(\omega t)$$

Kelvin Probe Force Microscopy

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V_{eff}^2 = F_{dc} + F_{\omega} + F_{2\omega}$$

$$F_{dc} = -\frac{\partial C}{\partial z} \left[\frac{1}{2} (V_{dc} - V_{CP})^2 + \frac{V_{ac}^2}{4} \right]$$

$$F_{\omega} = -\frac{\partial C}{\partial z} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$

$$F_{2\omega} = \frac{\partial C}{\partial z} \frac{V_{ac}^2}{4} \cos(2\omega t)$$

AM-KPFM

Amplitude Modulation

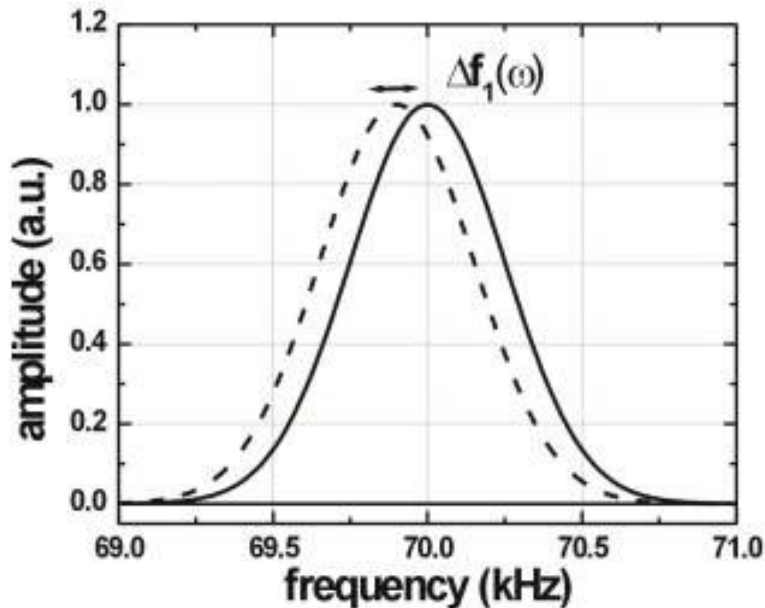
FM-KPFM

Frequency Modulation

FM – KPFM

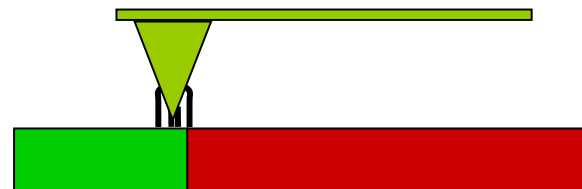
Frequency Modulation Detection

$$\Delta f(\omega) \propto \frac{\partial F_{el}}{\partial z} \propto \frac{\partial^2 C}{\partial z^2} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$



- frequency ω of V_{ac} between 1-3 kHz
- detection of the oscillation of $A(\Delta f_1)$ with a lock-in
- limiting factor: bandwidth of the FM-demodulator / PLL

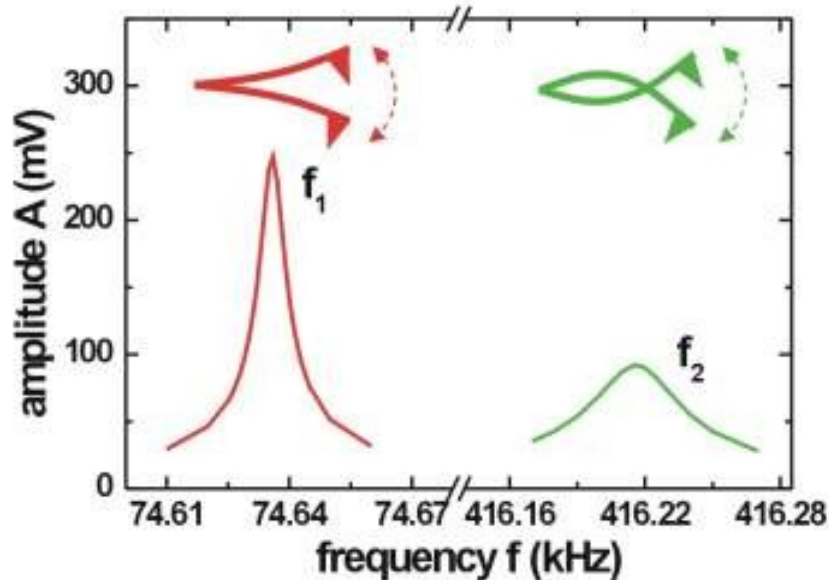
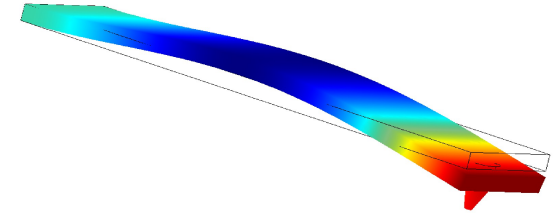
$$A(\Delta f_1) \propto \partial F_{el} / \partial z$$



AM – KPFM

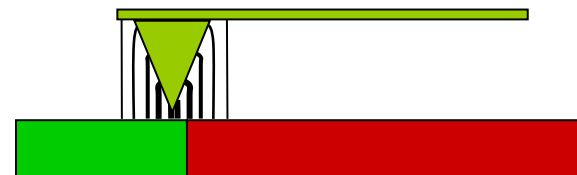
Amplitude Modulation Detection

$$F_{\omega} = -\frac{\partial C}{\partial z} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$

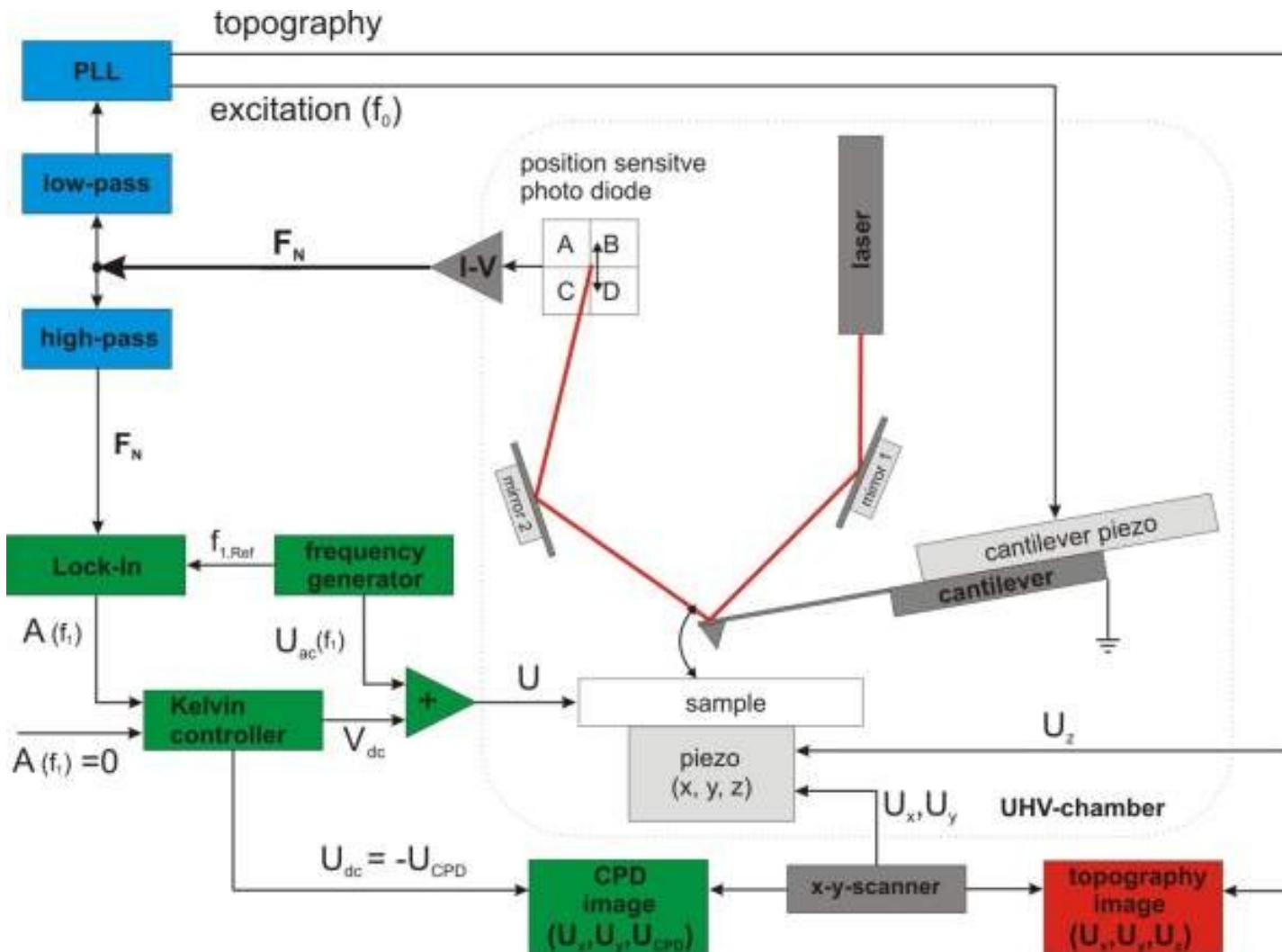


- tune ω to the second resonance f_2
- detection of the oscillation amplitude A_{ω} with a lock-in
- limiting factor: bandwidth of the photodiode

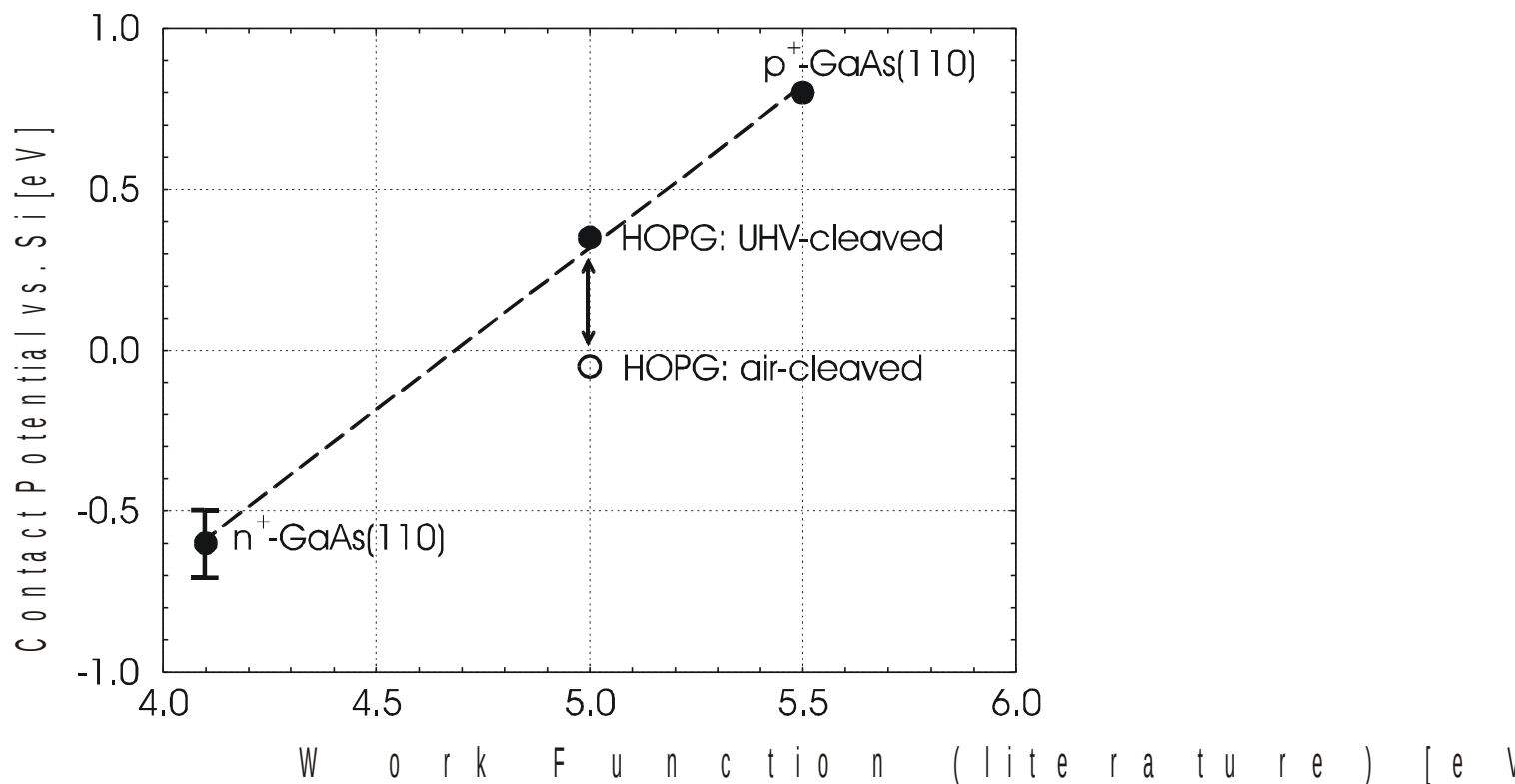
$$A_{\omega} \propto F_{\omega}$$



Experimental Setup nc-AFM & AM-KPFM



KPFM calibration and absolute work function



Φ -Si-Cantilever = 4.70 (± 0.1) eV

$U_{ac} = 100$ mV

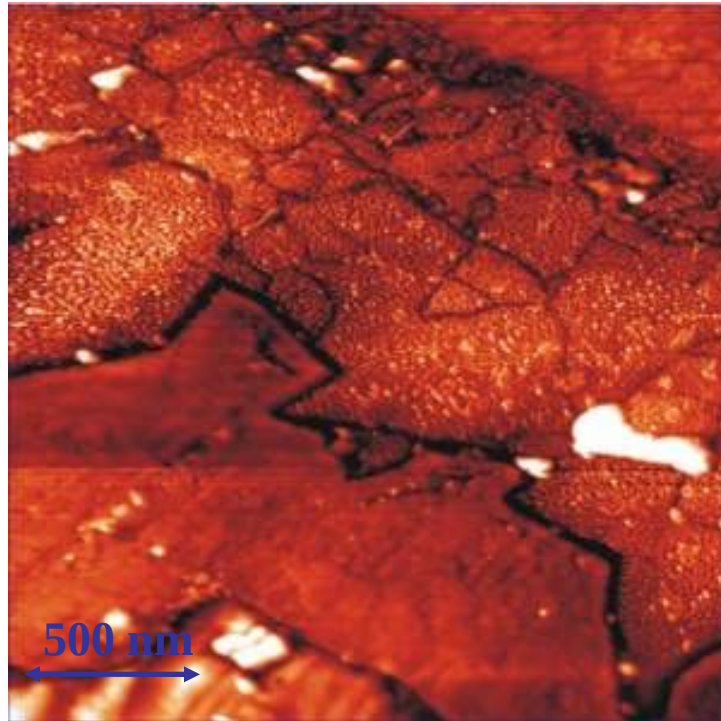
→ absolute and quantitative work function determination

Polished Cross Section of a CuGaSe₂ Solar Cell

CuGaSe₂ solar cell device: $V_{oc} = 820$ mV, $\eta = 4.6$ %

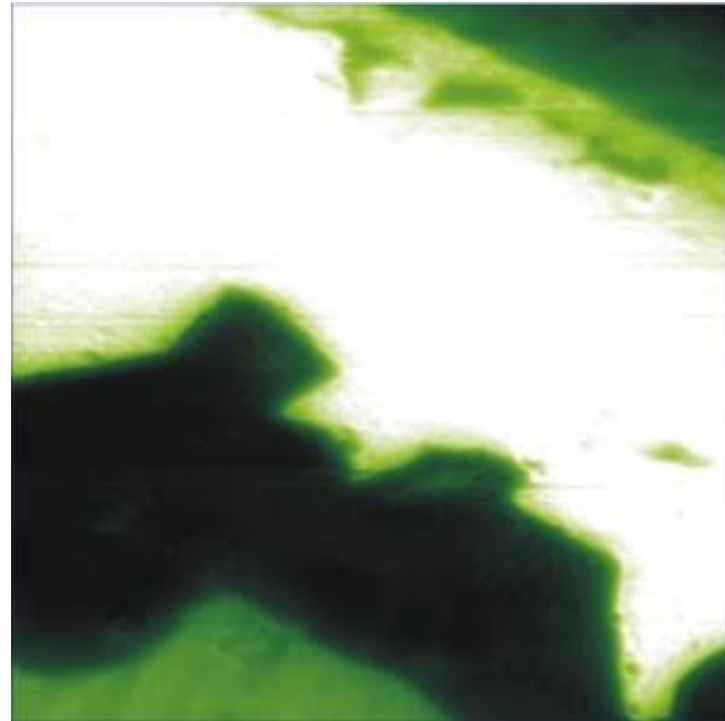
polished and Ar-ion sputtered cross section

topography



$\Delta z = 65$ nm

work function



Mo,
 $\Phi = 4.20$ eV

MoSe₂,
 $\Phi = 4.40$ eV

CuGaSe₂,
 $\Phi = 4.80$ eV

n-ZnO:Ga,
 $\Phi = 4.0$ eV

AM-KPFM measurement on GaP pn-junction

n-type GaP wafer with p-type GaP layer, $\sim 10^{18} \text{ cm}^{-3}$, UHV cleavage along (110) surface

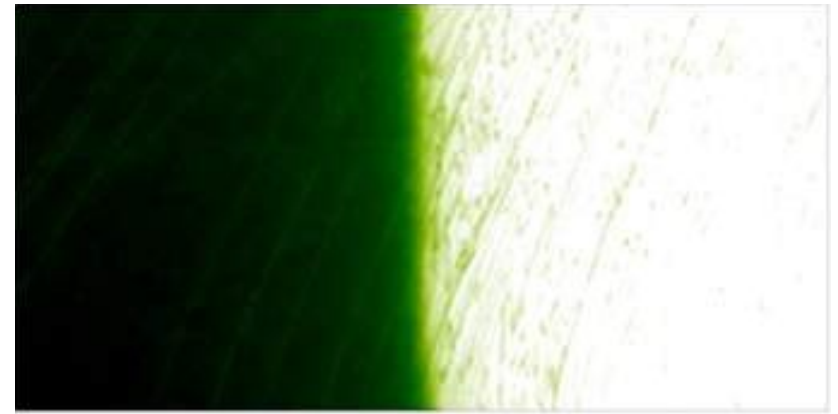
topography



$\Delta z = 4.5 \text{ nm}$

$2.5 \times 5 \mu\text{m}^2$

work function



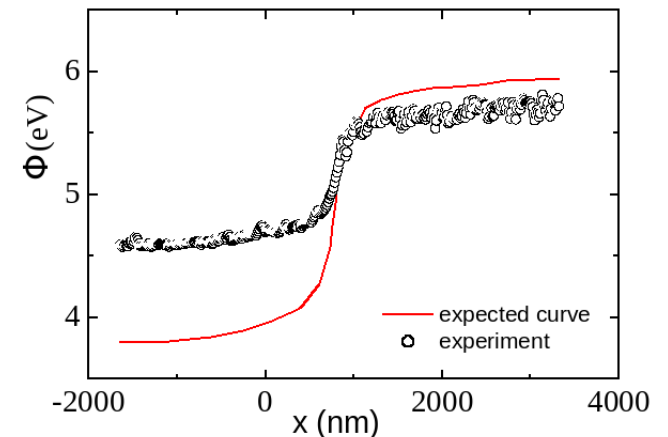
$\Phi = 4.56 - 5.77 \text{ eV}$

most III-V semiconductors:

no surface states on the (110) surface

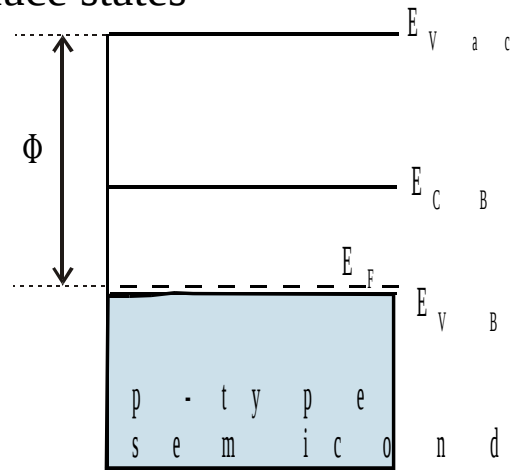
☞ GaP does show surface states

⇒ discrepancy of Φ_{exp} to Φ_{theo} due to surface states!

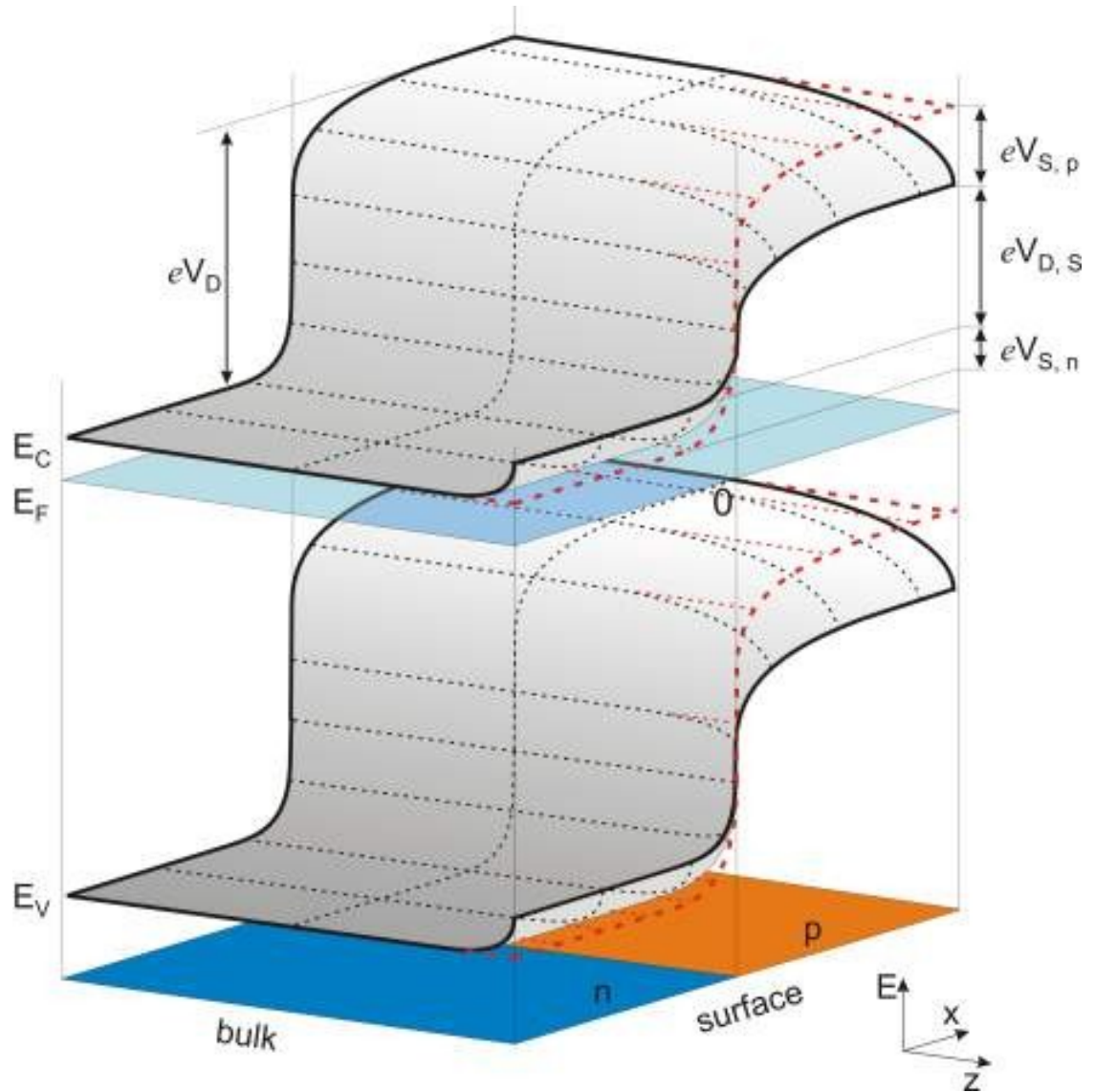
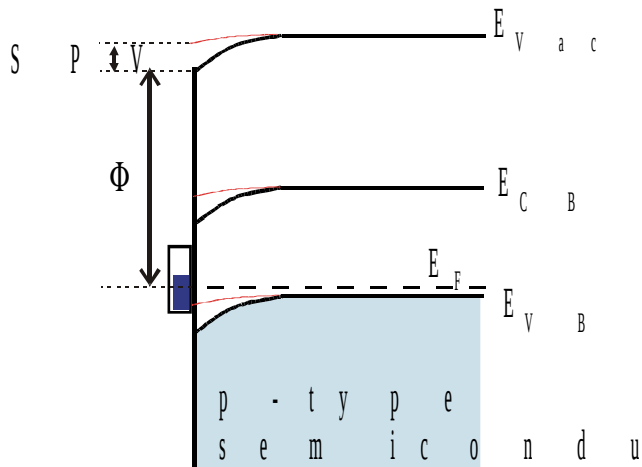


Surface Effects

no surface states



surface states



Surface Photovoltage GaP pn-Interface

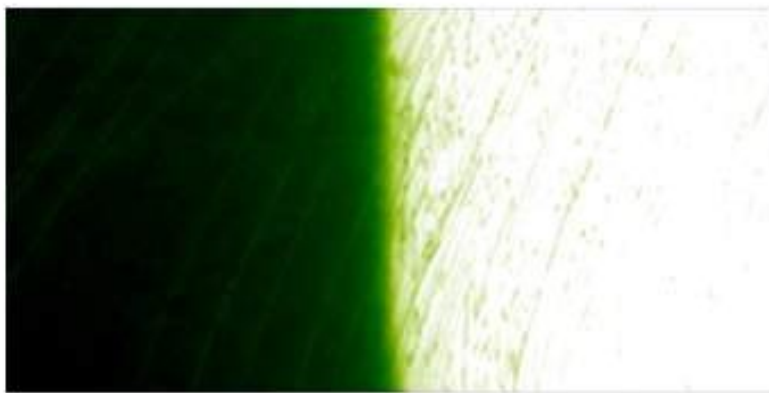
n- GaP

p- GaP



$\Delta z = 4.5 \text{ nm}$

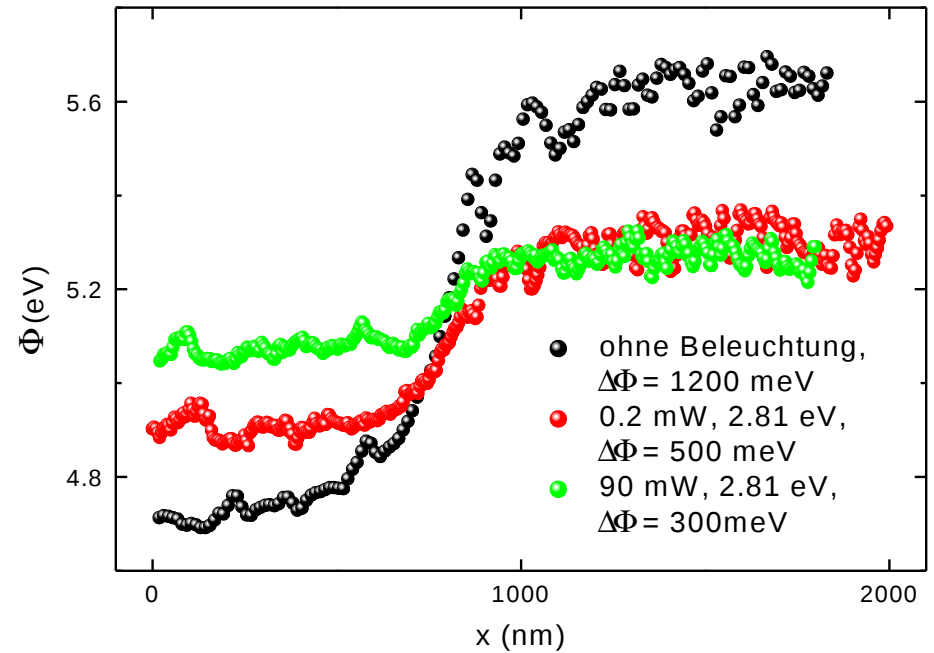
(5000x2500 nm)



$\Phi = (4.56 - 5.77) \text{ eV}$

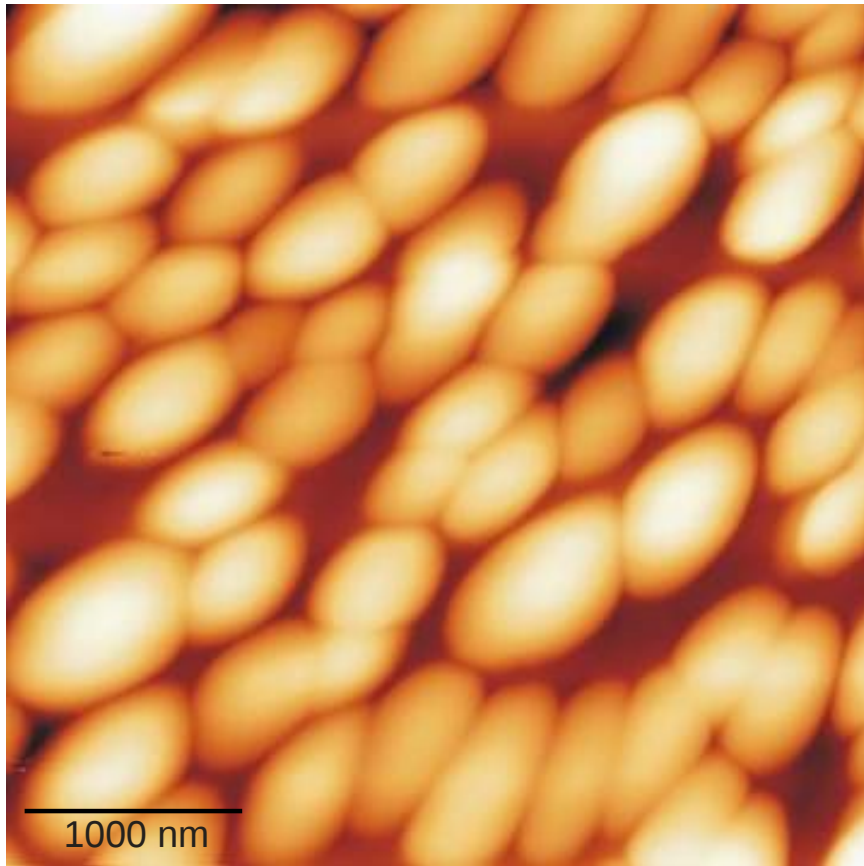
n- GaP

p- GaP

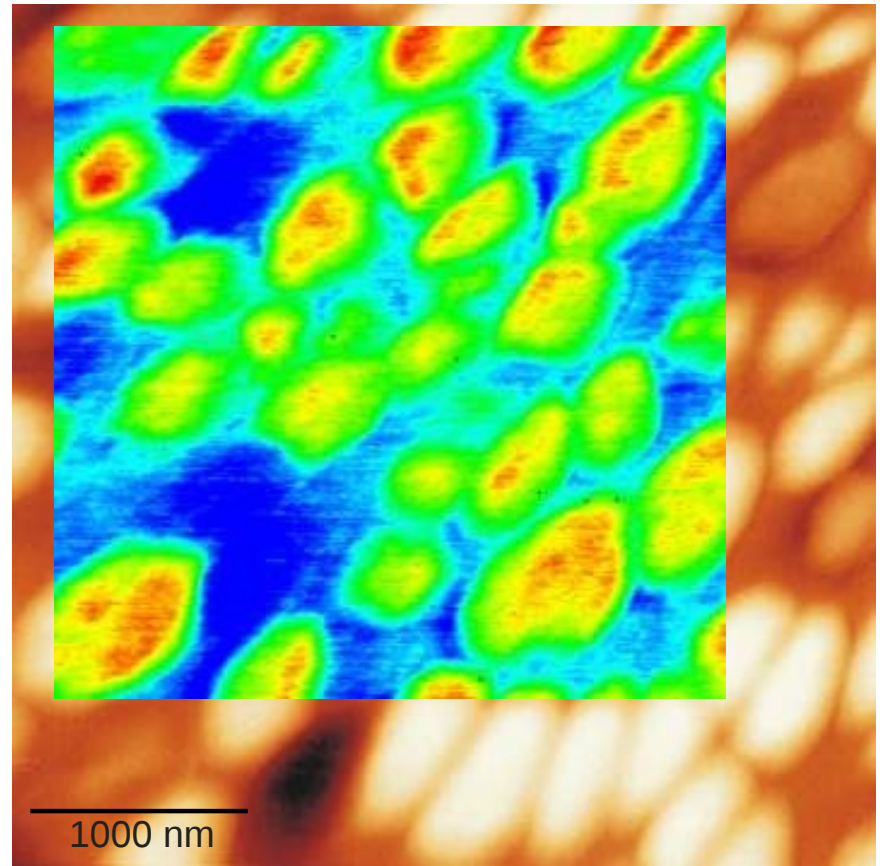


Surface Photovoltage

MDMO-PPV/PCBM – 675nm



0 nm  105.6 nm

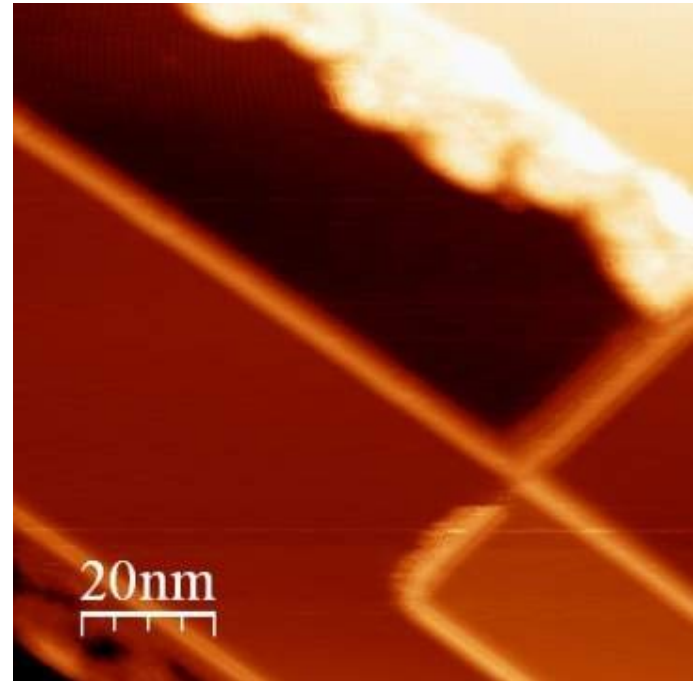
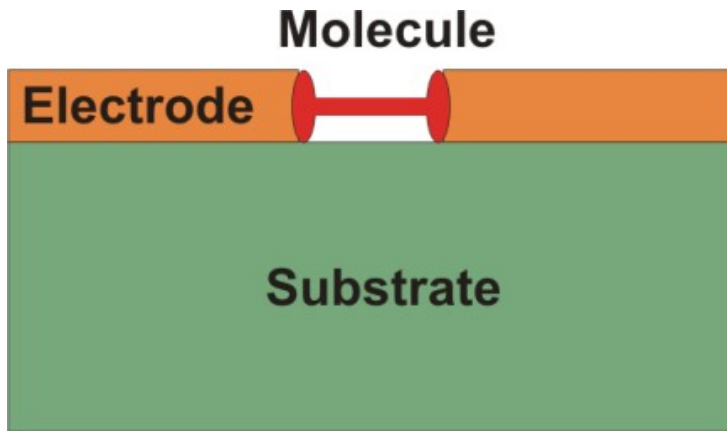


4.19 eV  4.62 eV

-50 mV  220mV

Motivation

Molecular electronics

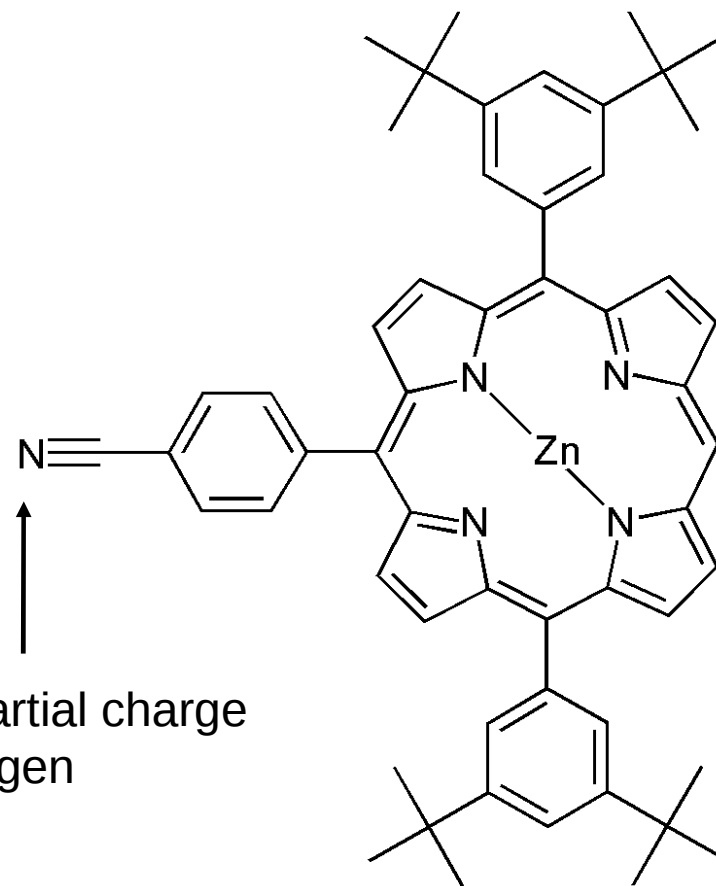
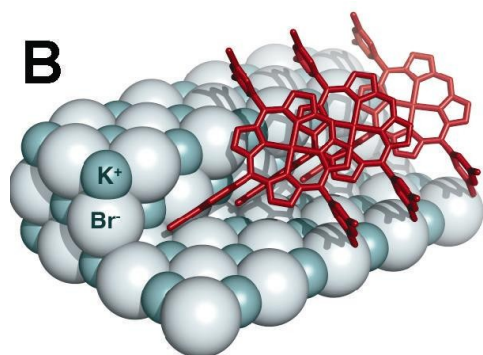
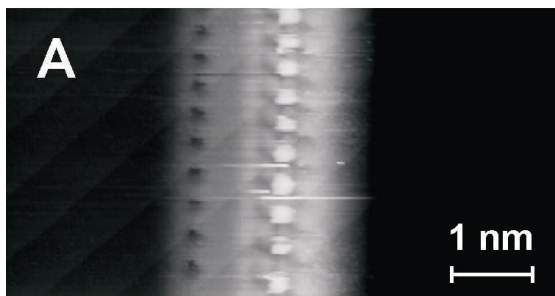


Molecules on Insulators:

- No STM possible – nc-AFM mandatory
- Low diffusion barrier but high intermolecular interaction
- Low temperatures – easier to “fix” molecules but not so easy to find applications

Asymmetric Cyano-Porphyrins

Natural light harvesting complexes

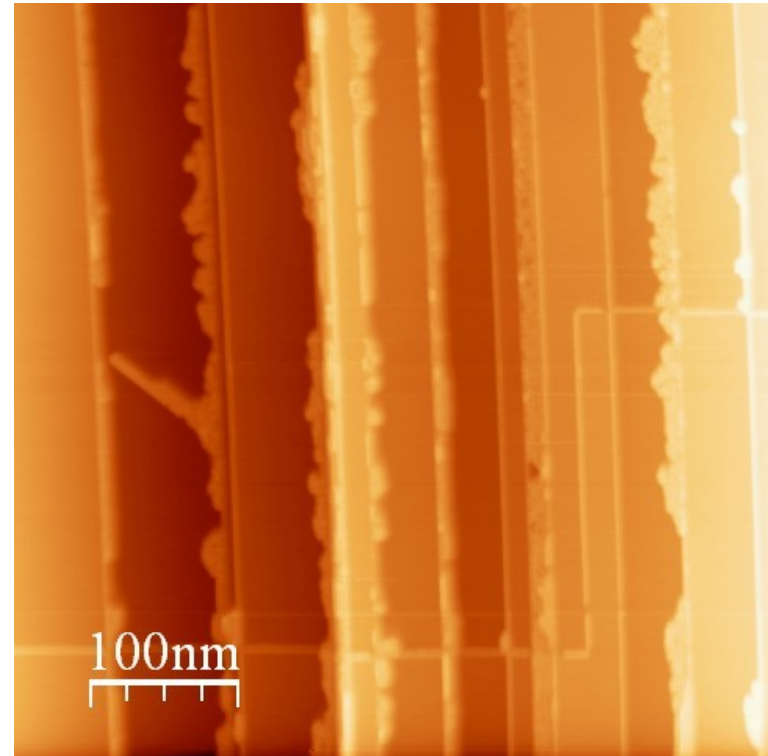


S. Meier et al., *Small*, 2008, 4, 1115

Wire Formation

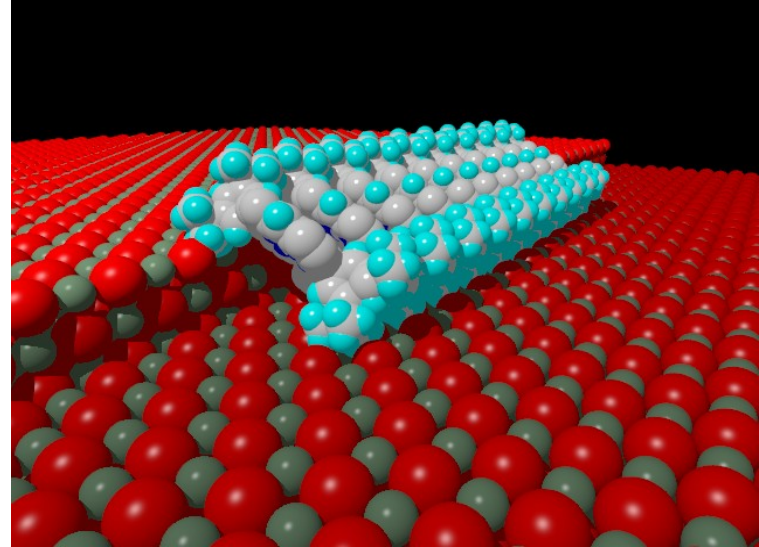
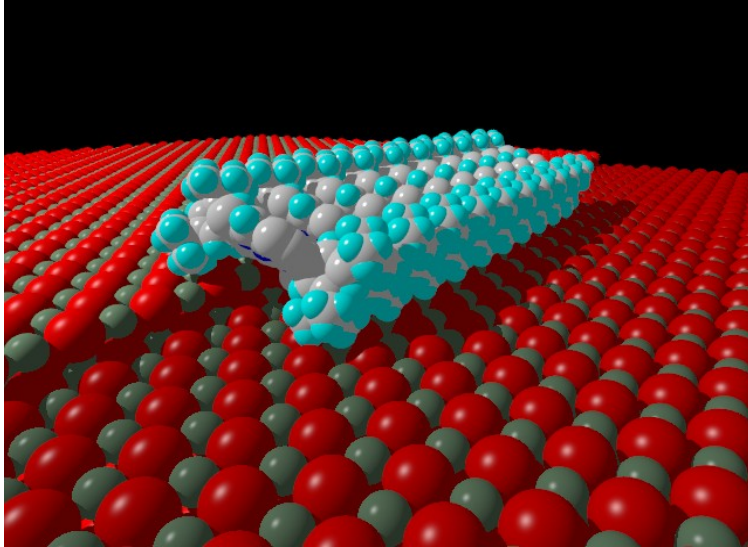
Decoration of step edges on KBr(100)

- In situ cleaved KBr with 0.5 ML of molecules
- Steps (< 1nm) are decorated with monowires
- Higher steps act as nucleation sites for structure growth across terraces

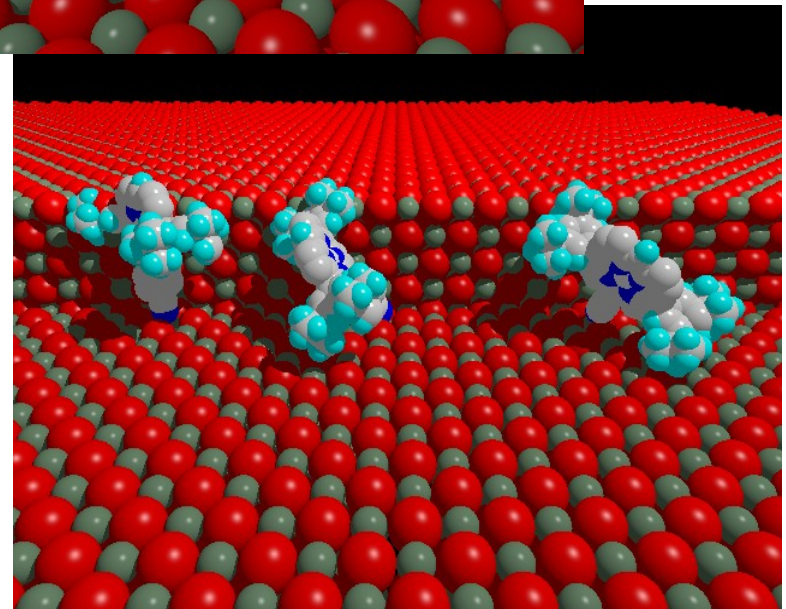


Wire Formation

Structural model

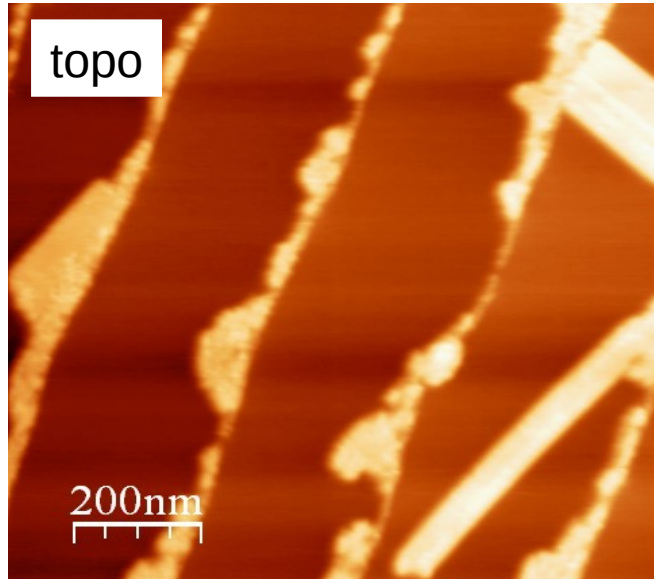


- The tilt angle of the molecules is determined by the side groups, the π - π stacking and the step height.
- Steps higher than 3 ML prevent a π - π stacking.

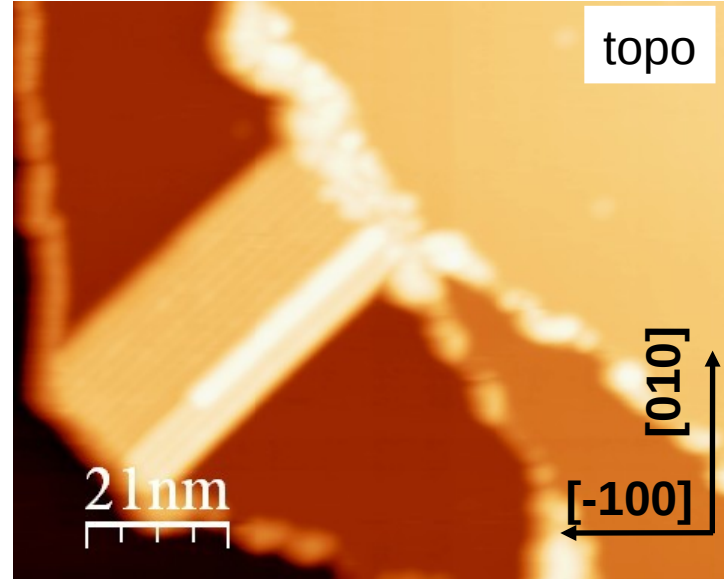


Molecular Assemblies

Multiwires on KBr



$f_0 \approx 174054\text{Hz}$, $\Delta f = -8\text{Hz}$, $Q = 15k$, $A = 5\text{nm}$

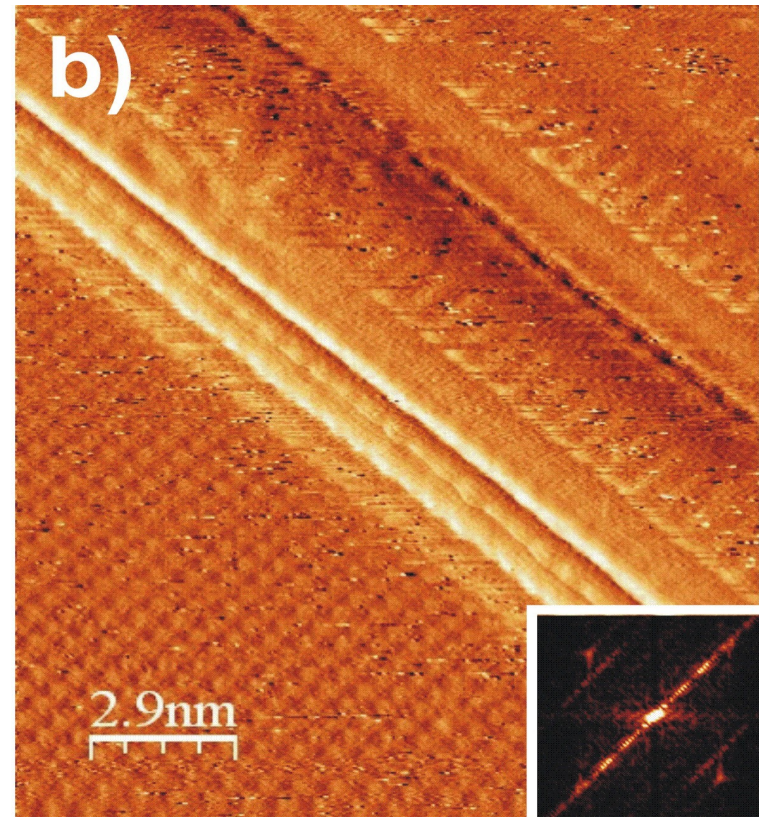
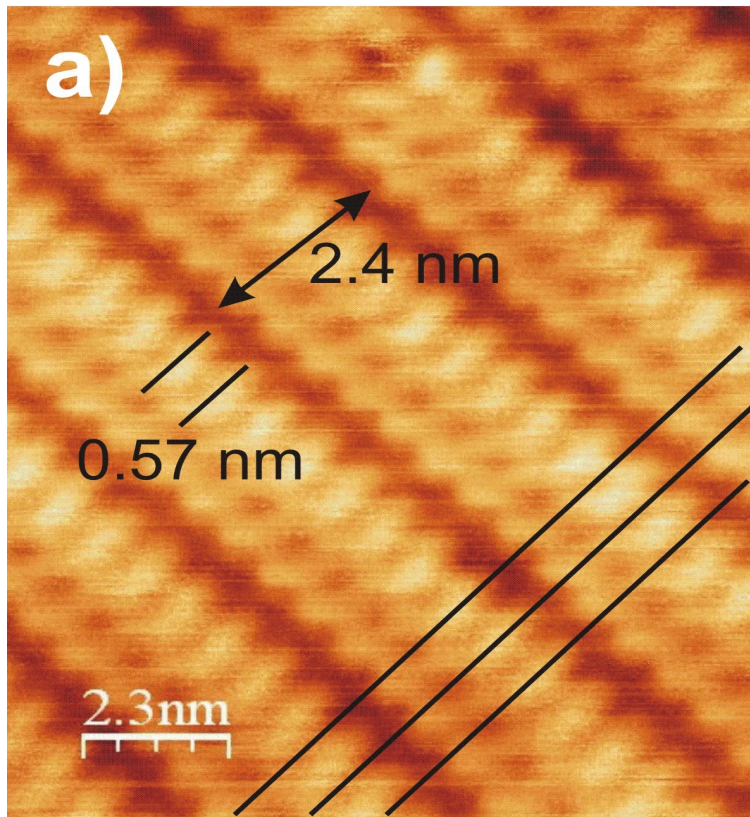


$f_0 \approx 173886\text{Hz}$, $\Delta f = -52\text{Hz}$, $Q = 15k$, $A = 5\text{nm}$

- Multiwire growth across terraces
- The $\langle 110 \rangle$ directions are clearly preferred
- Different heights are visible

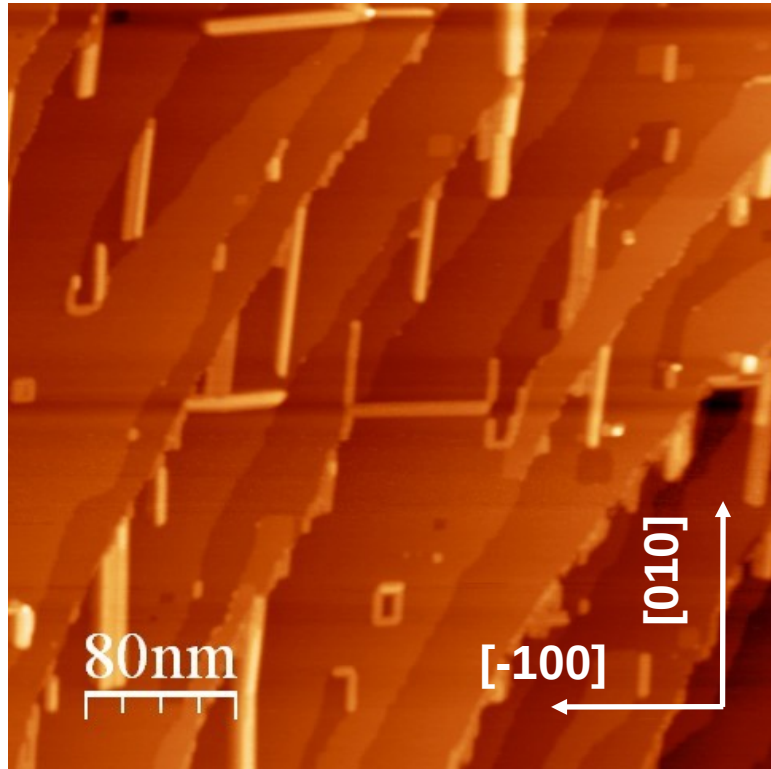
Molecular Assemblies

High resolution imaging

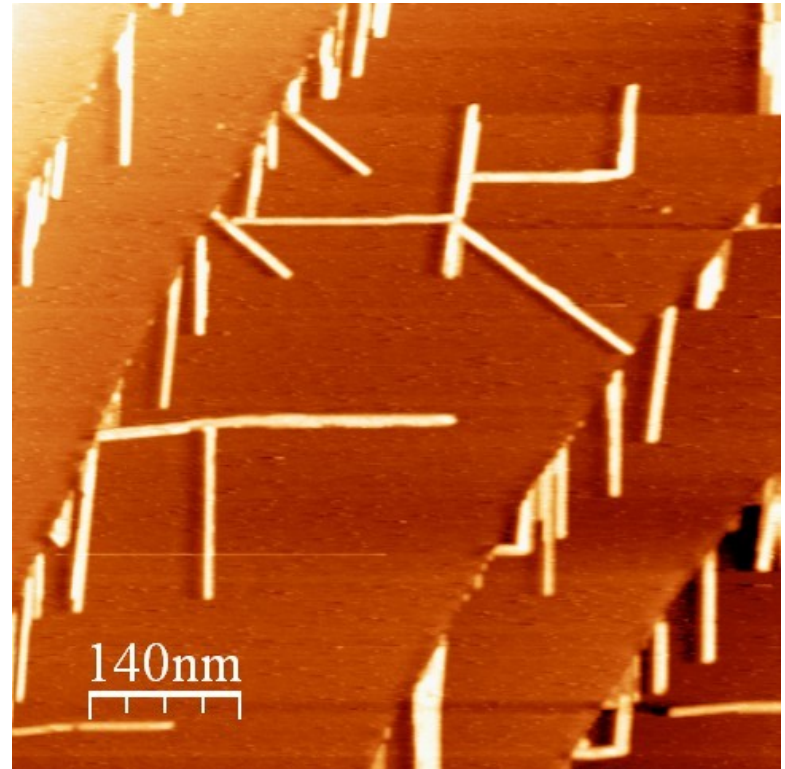


Molecular Assemblies

Molecular wires on NaCl



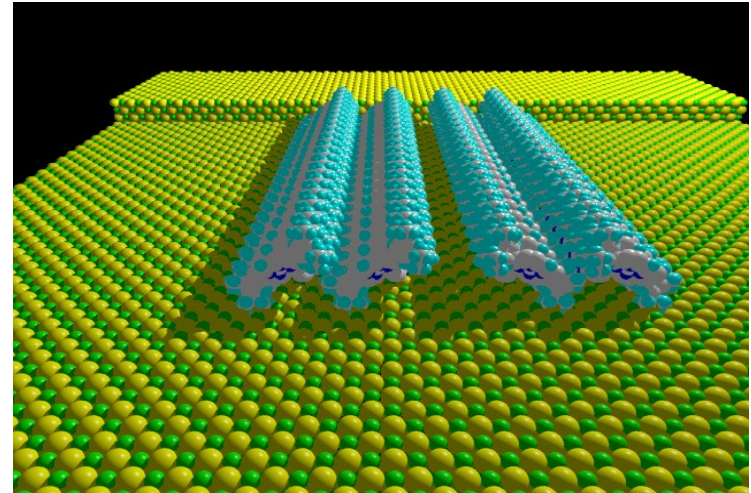
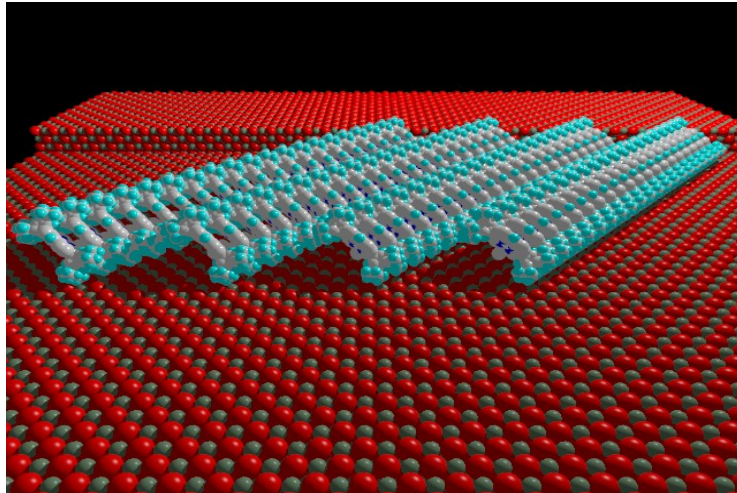
$f_0 \approx 170992\text{Hz}$, $\Delta f = -9.5\text{Hz}$, $Q = 15k$, $A = 40\text{nm}$



$f_0 \approx 170992\text{Hz}$, $\Delta f = -11\text{Hz}$, $Q = 15k$, $A = 40\text{nm}$

Molecular Assemblies

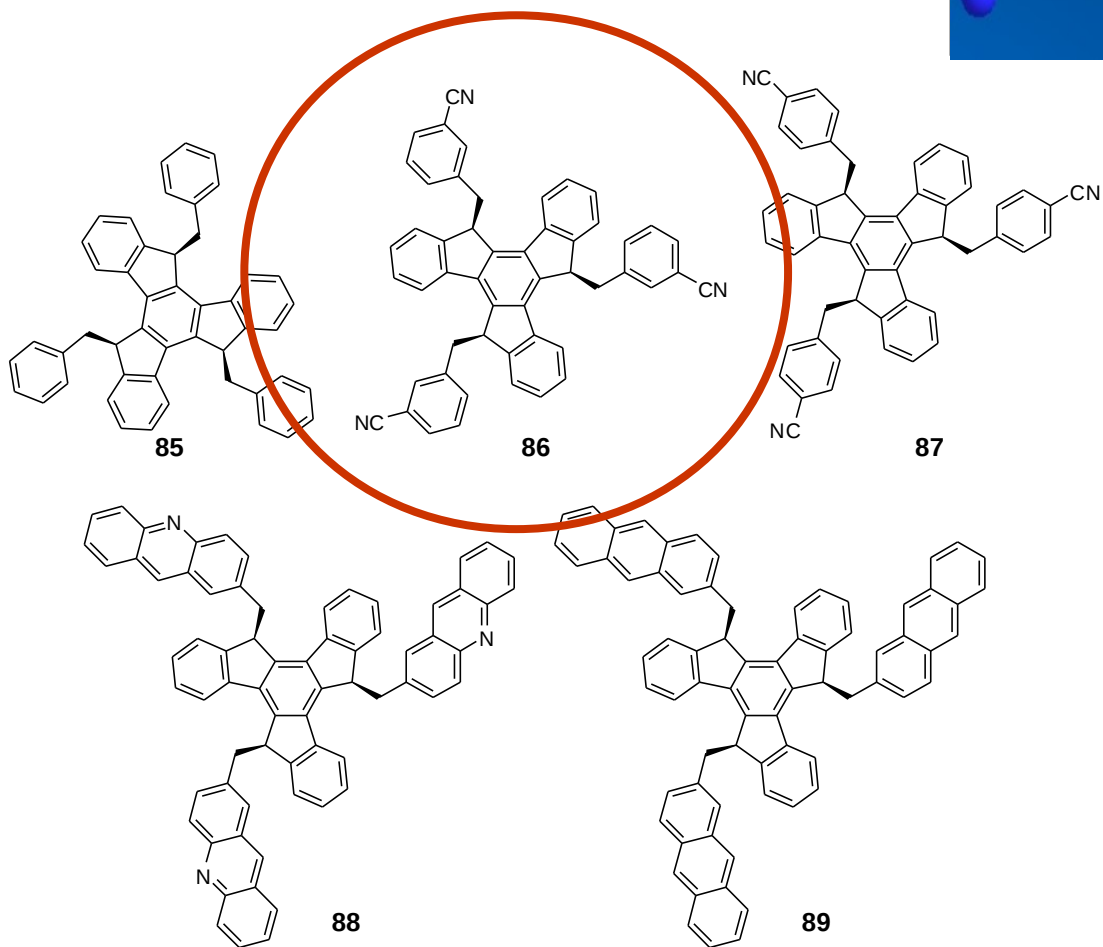
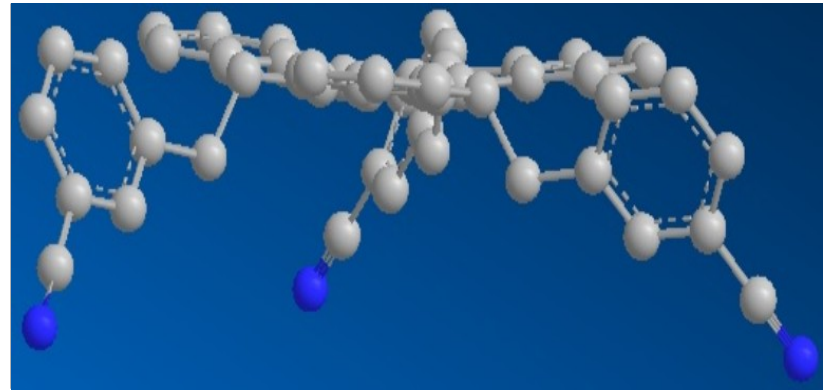
Structural model




- Intermolecular equilibrium separation $\sim 5.7 \text{ \AA}$
- Directed growth by the substrate

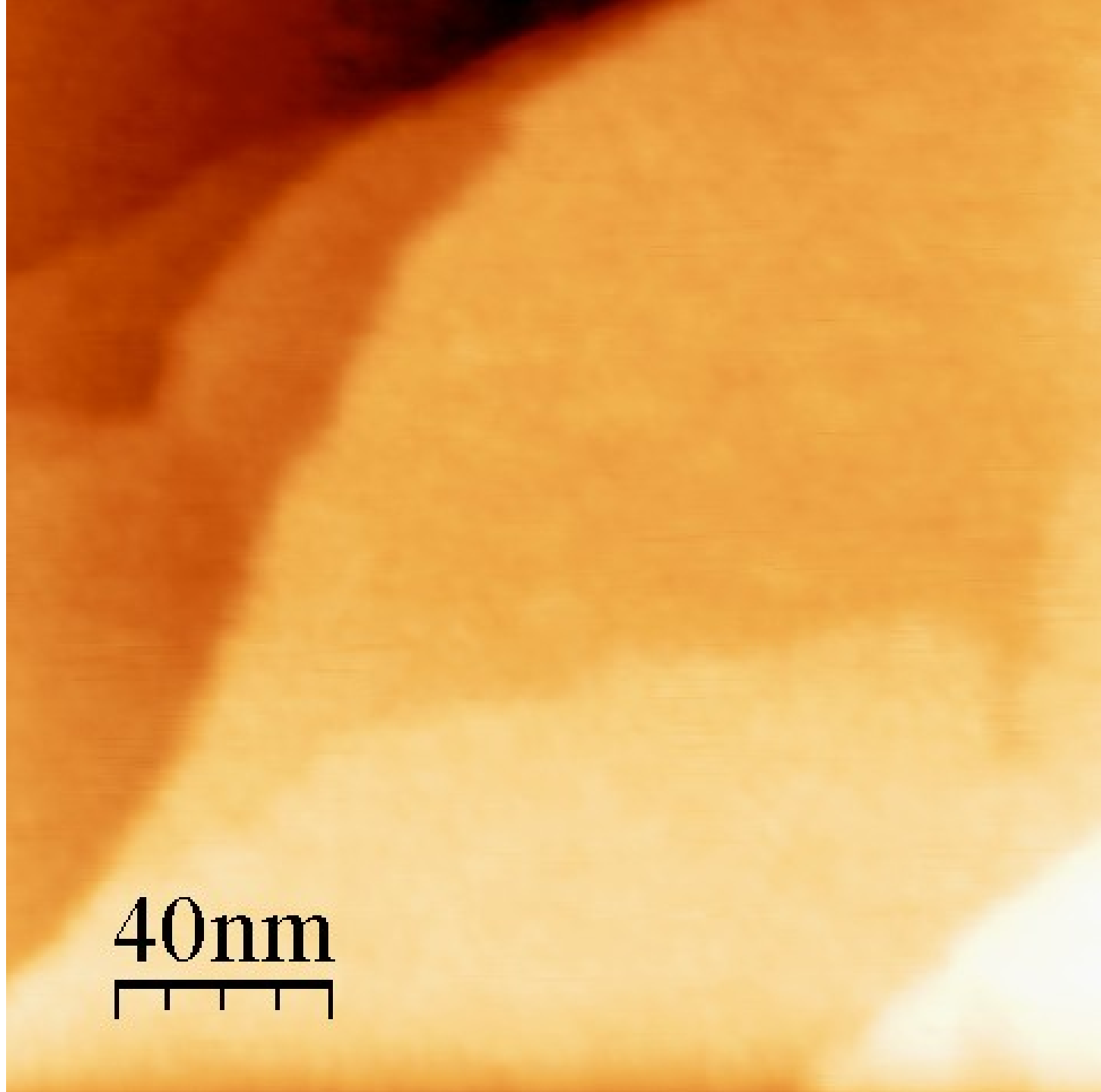
Functionalized Truxenes

Structure

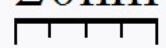


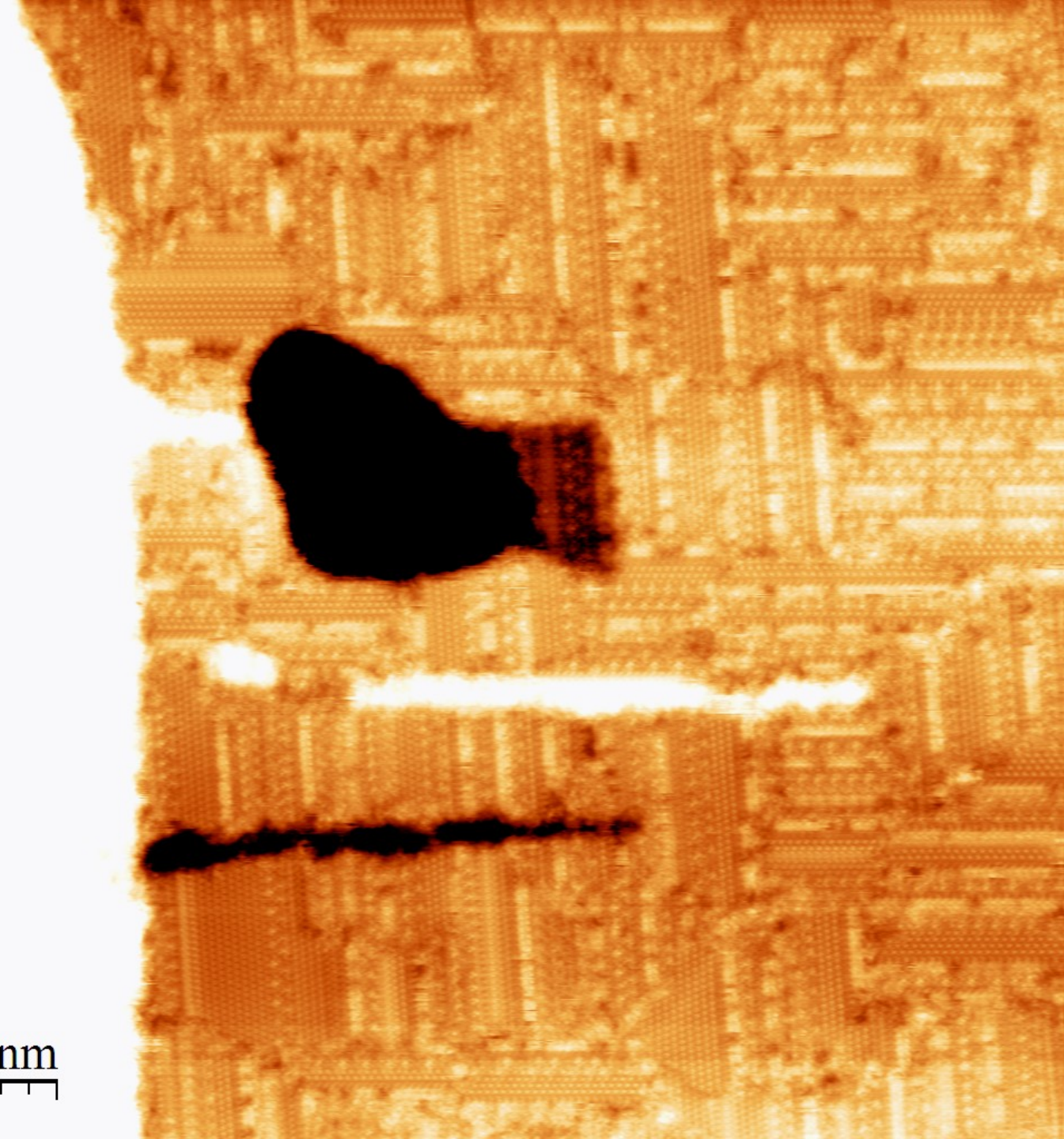
evaporation onto
the sample at RT

40nm




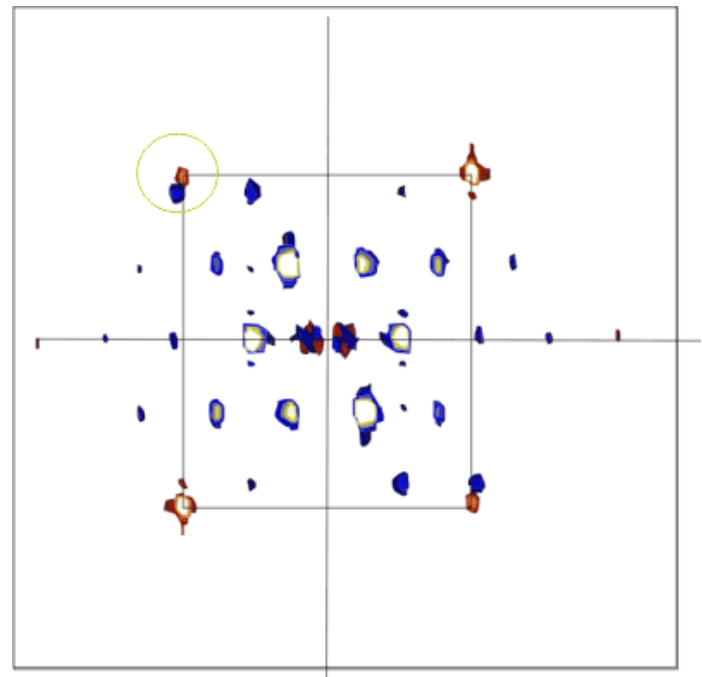
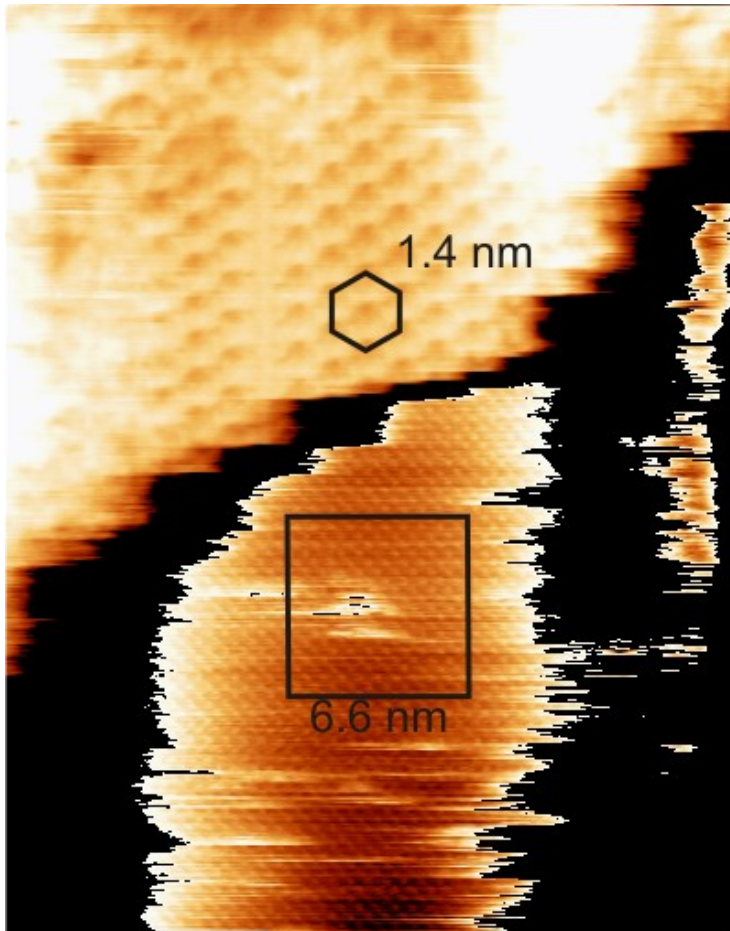
and the result of
post annealing at
155 C for 15 mins

20nm




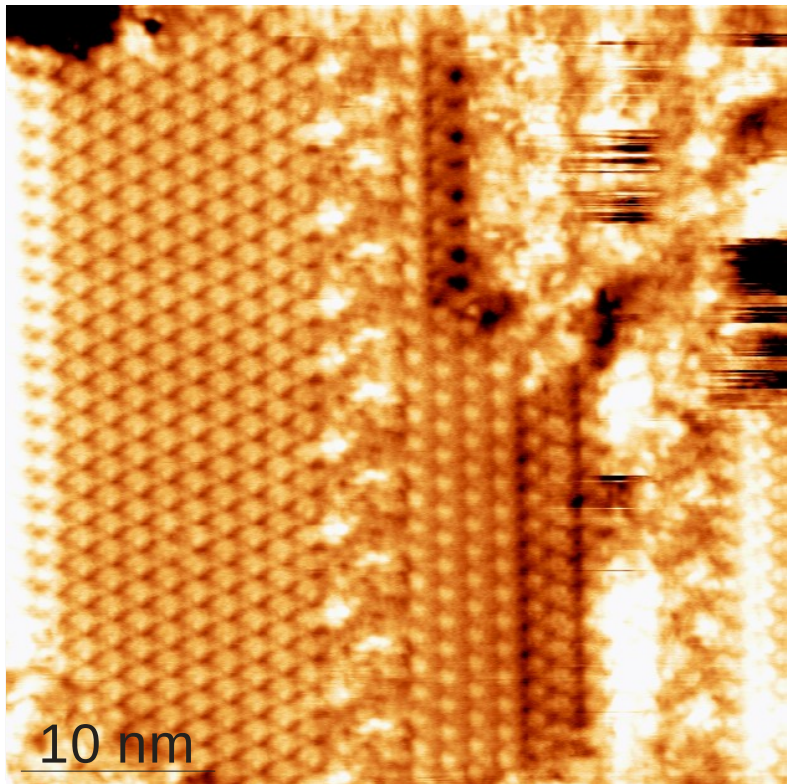
Truxene Self Assemblies

Hexagonal coordination on KBr

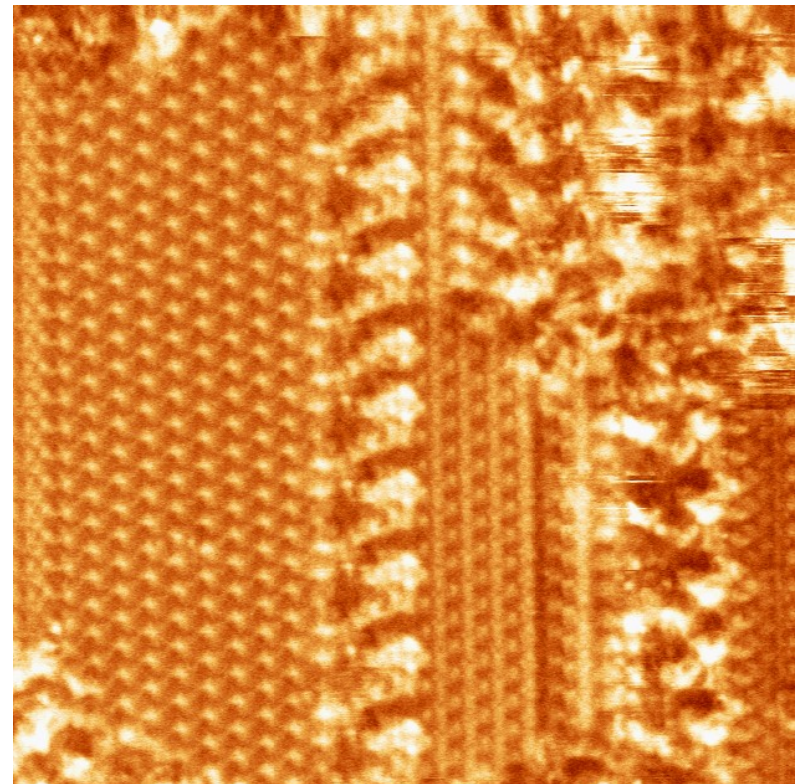


Truxene Self Assemblies

Various distinct structures



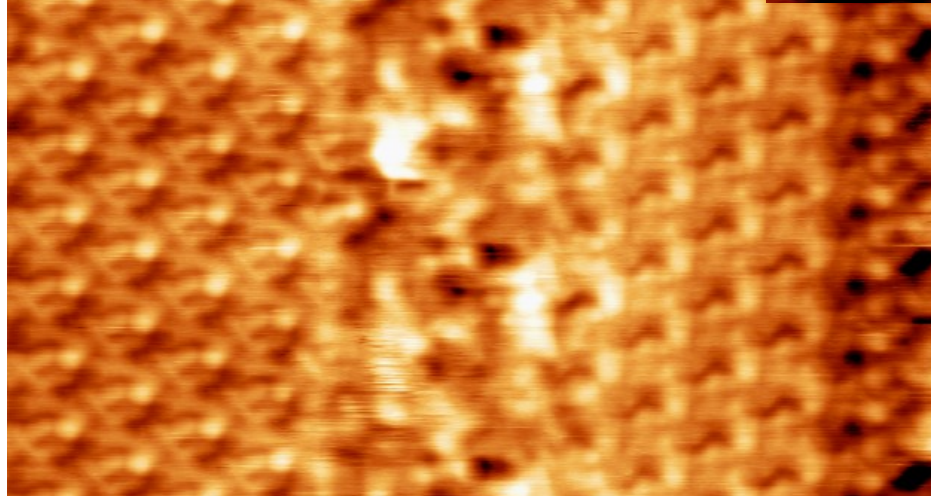
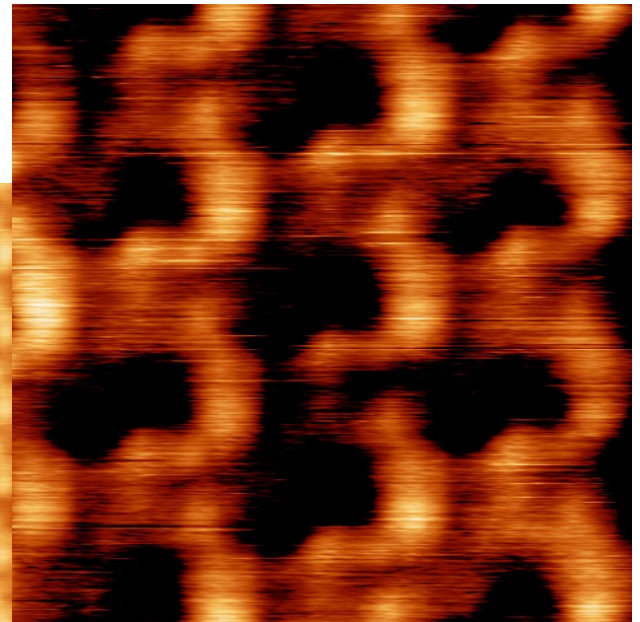
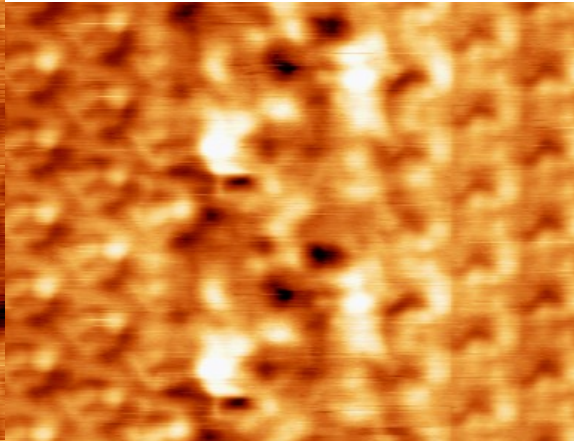
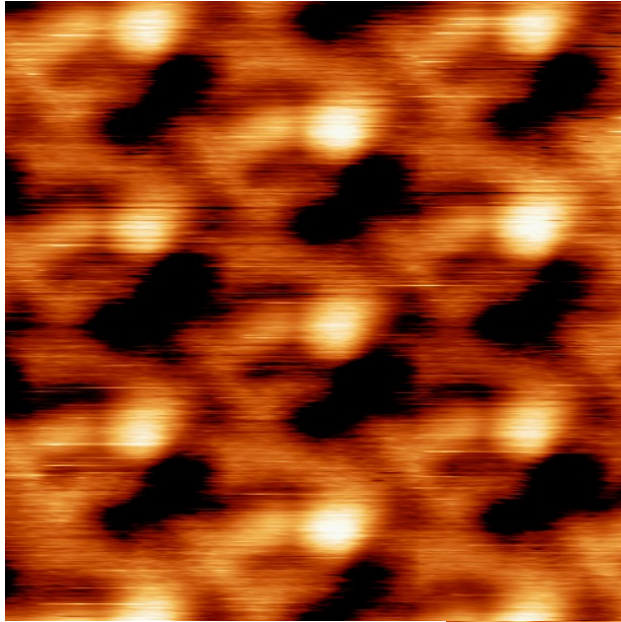
topography



dissipation

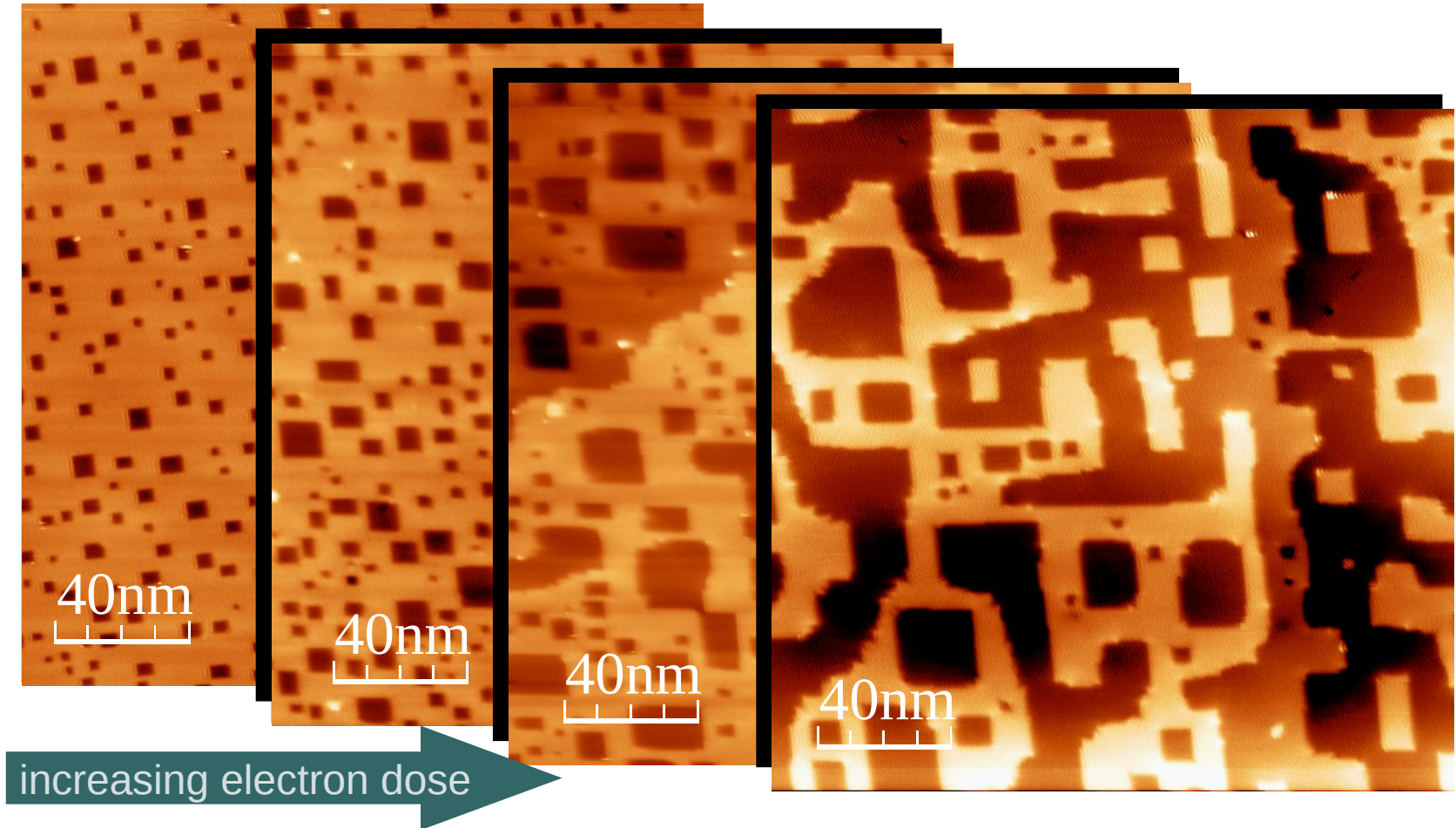
Truxene Self Assemblies

High resolution measurements



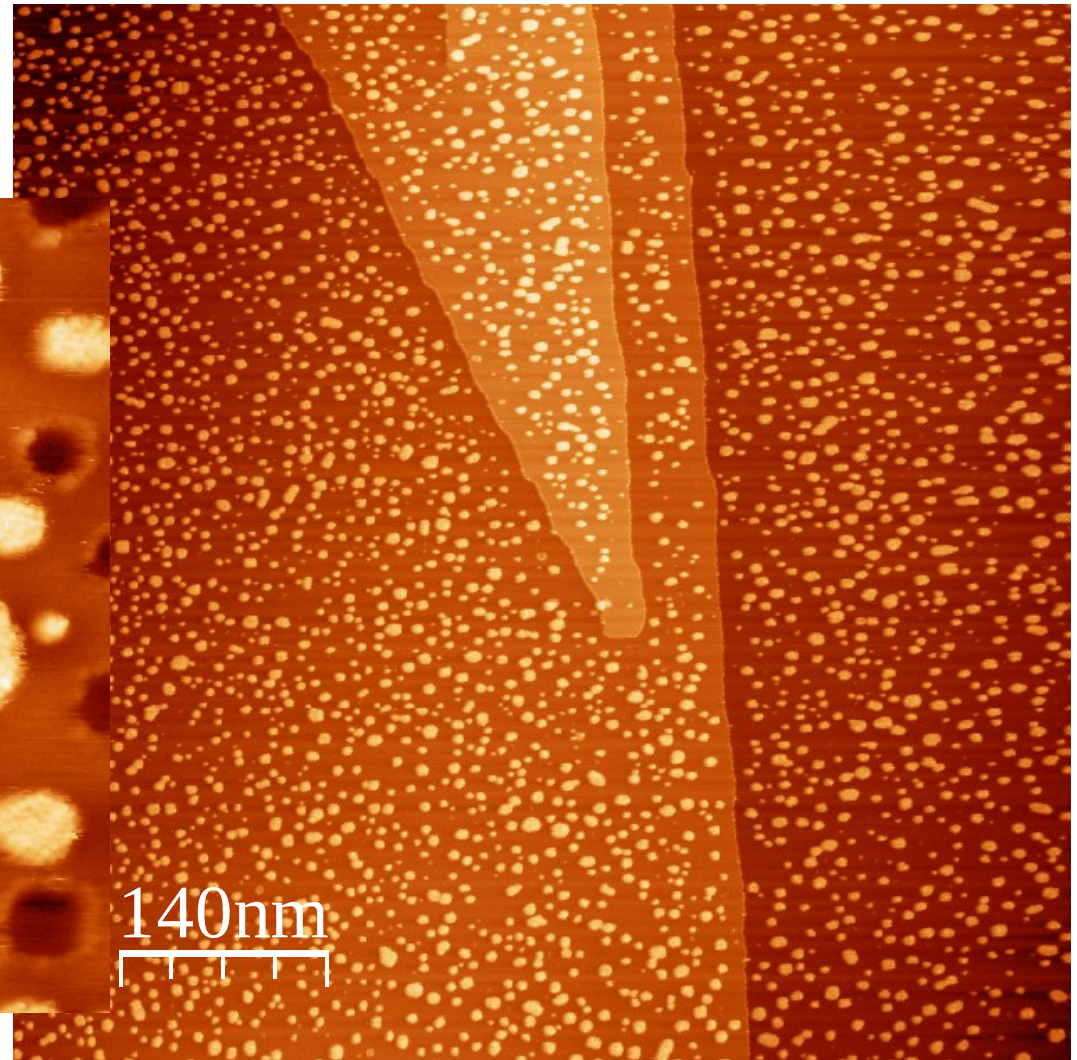
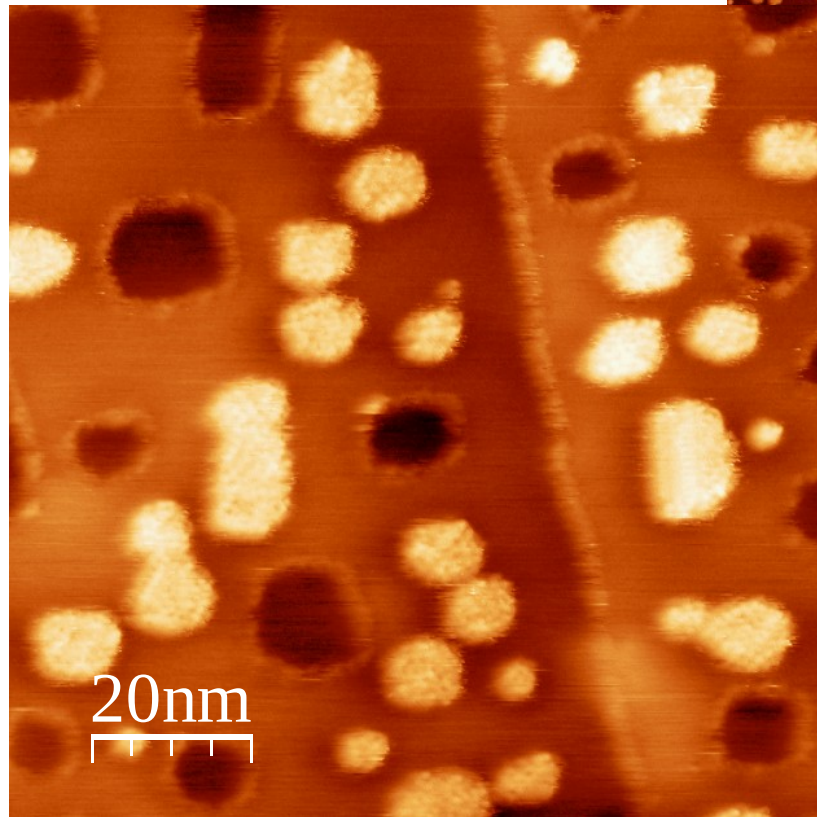
Single Crystal KBr

Substrate patterning by electron irradiation



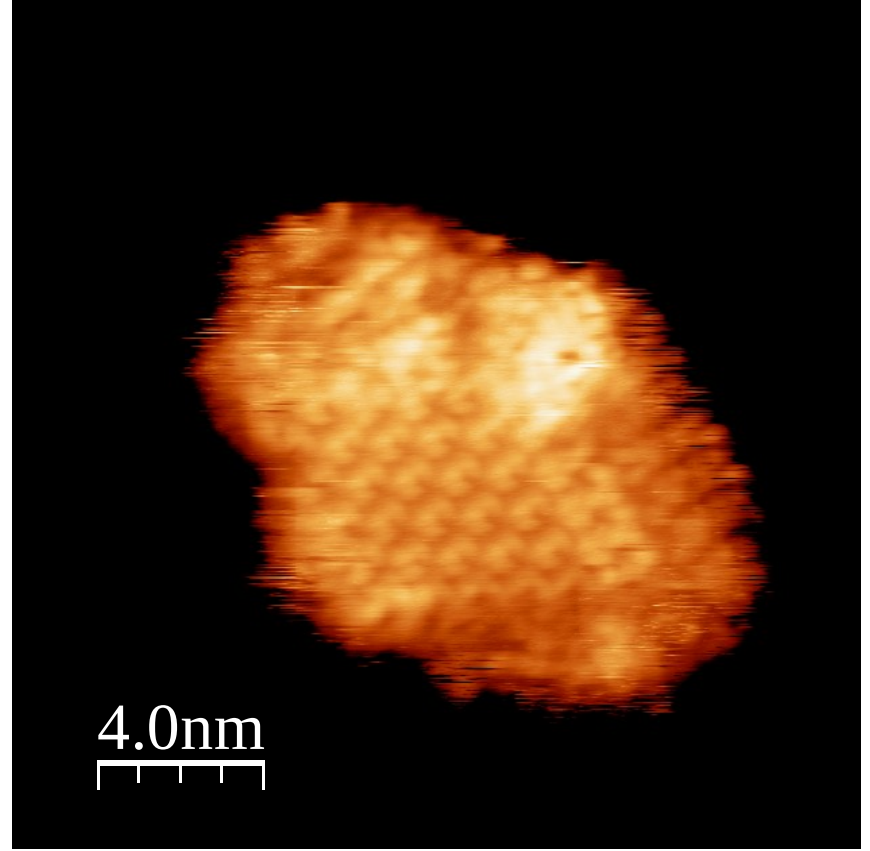
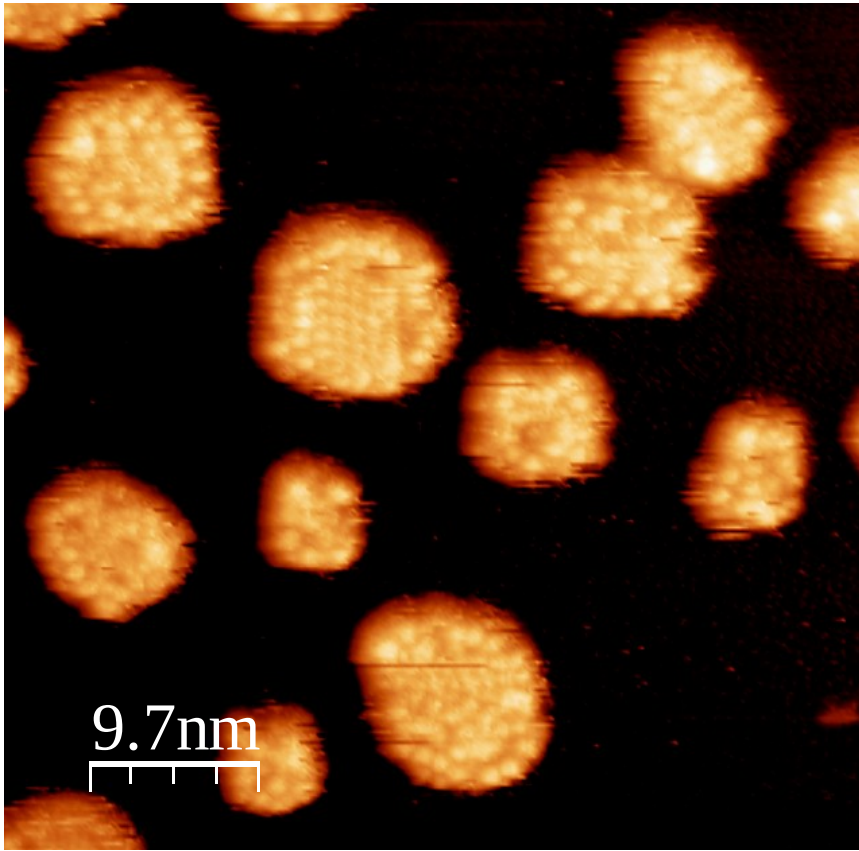
Truxenes on patterned surface

Filled and unfilled pits



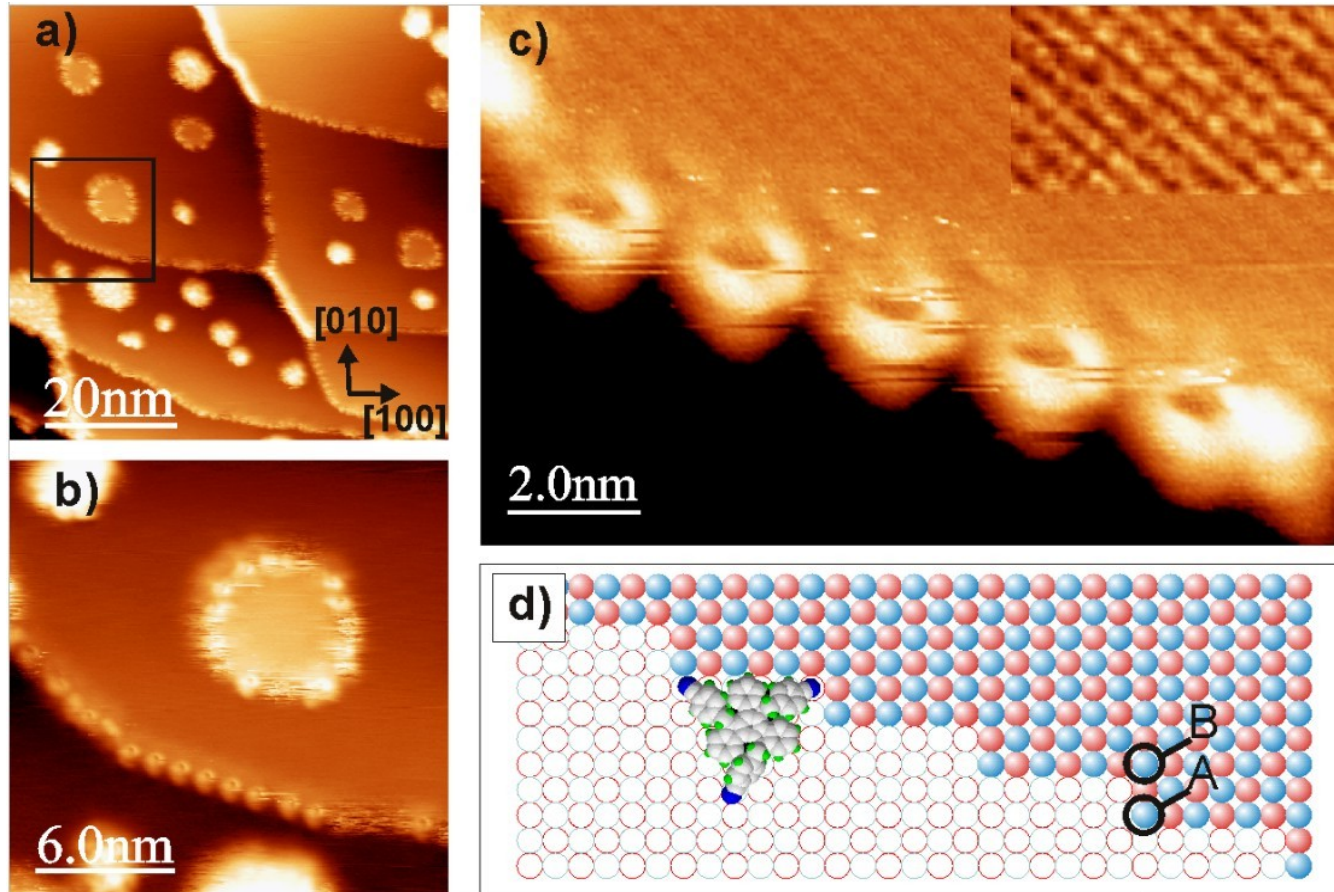
Truxenes on patterned surface

Organization within the pits



Imaging a Single Molecule

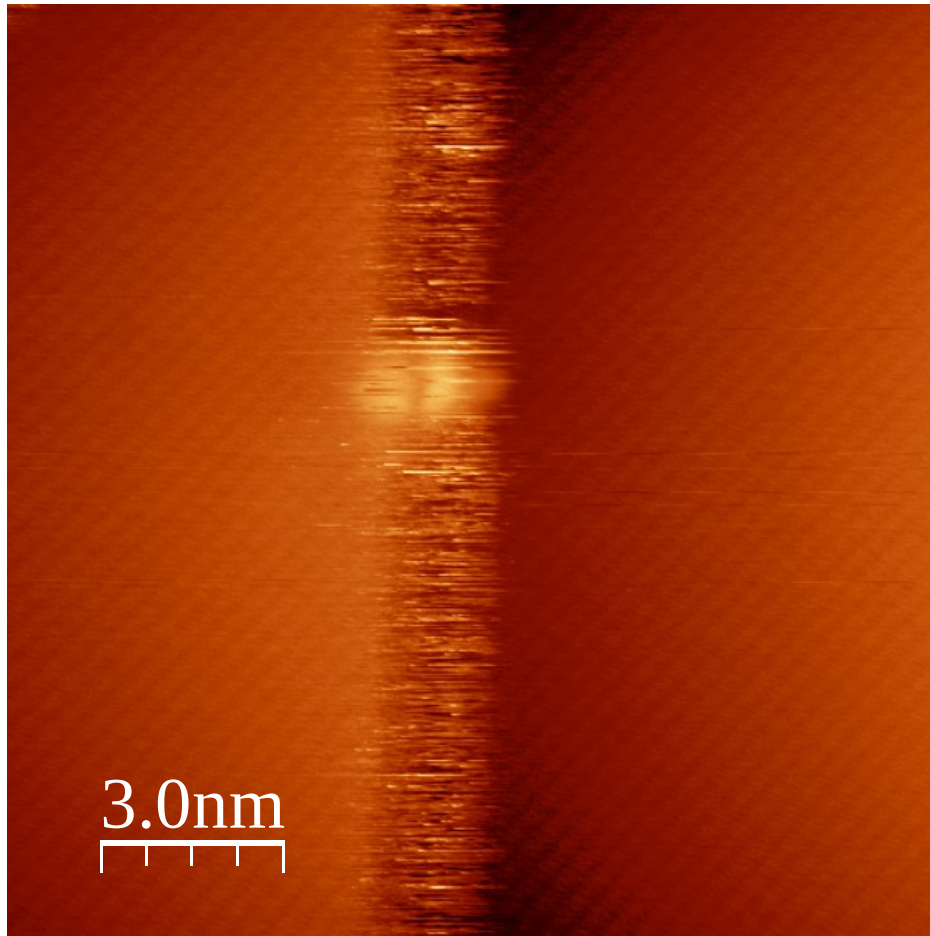
Measurements at RT



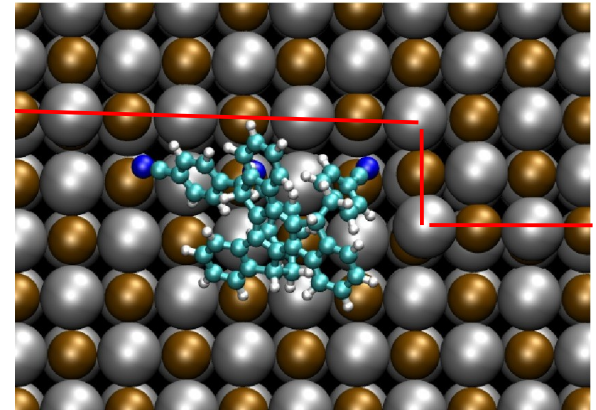
- Re-arrangement of the substrate, edges are running in the $[-3\ 1\ 0]$ direction
- no chemical interaction with the surface
- adsorbed on K or Br terminated double atomic kink

Imaging a Single Molecule

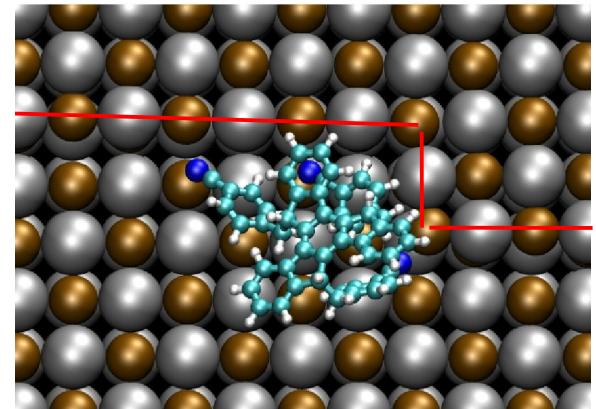
Measurements at RT



Br terminated, $E_b = 1.33\text{eV}$

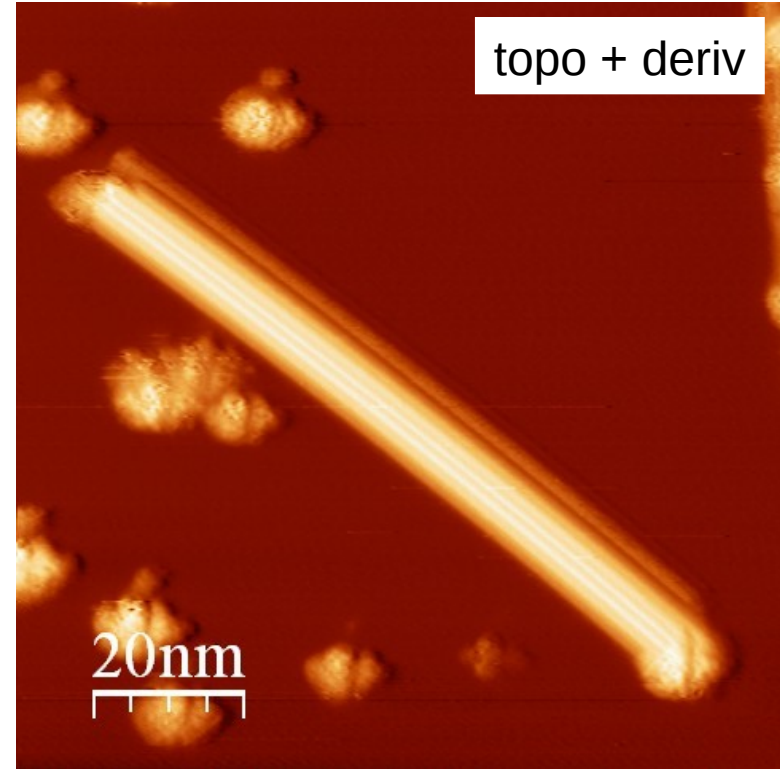
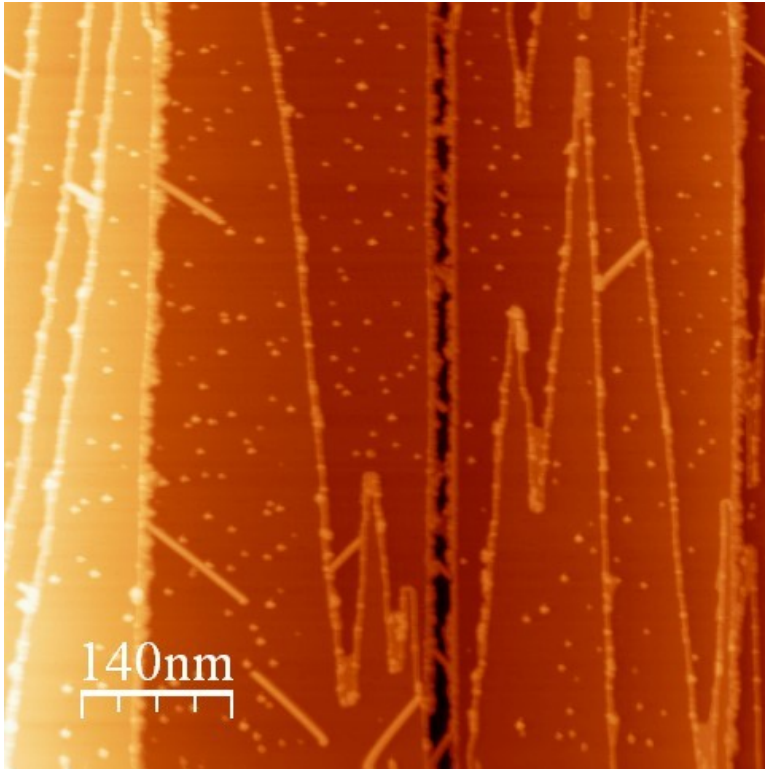
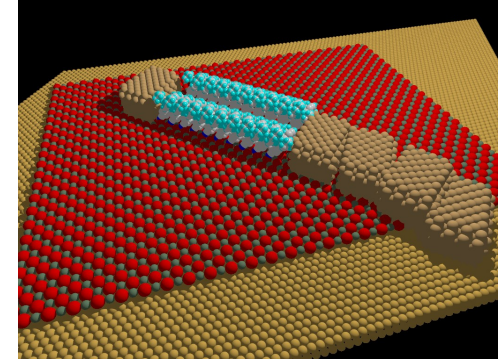


K terminated, $E_b = 1.17\text{eV}$



Contacting Molecular Assemblies

Au-Molecules-Au

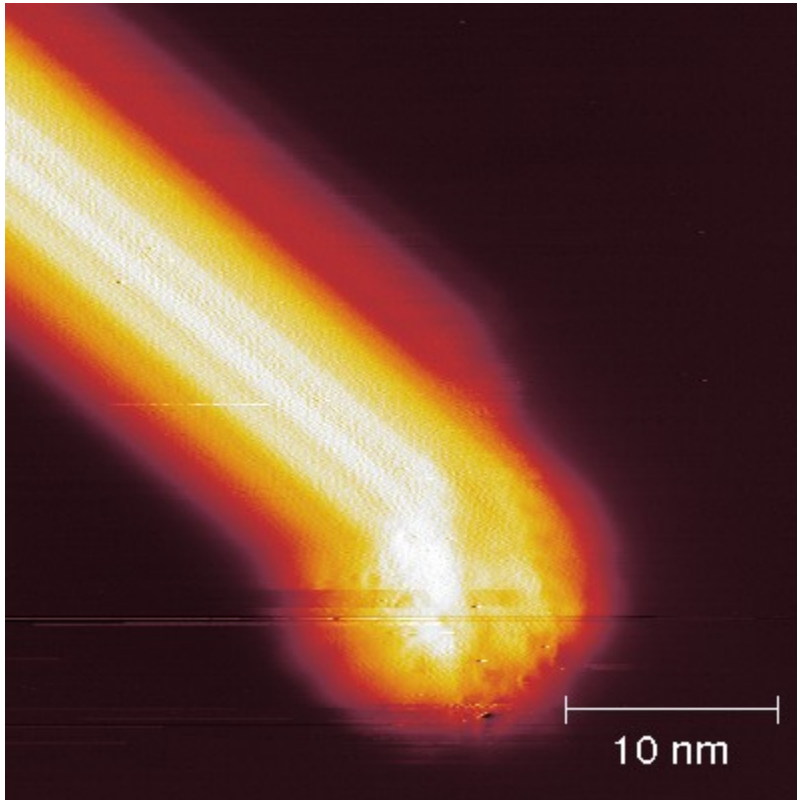


- Molecules arrange at steps and across terraces
- The growth is started/stopped at gold clusters.

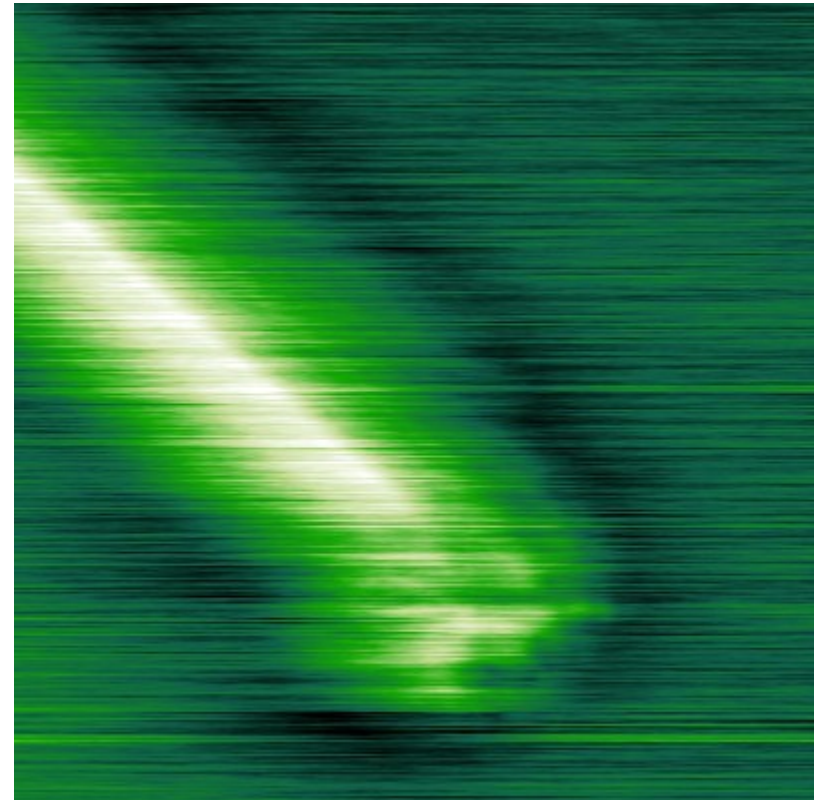
Contacting Molecular Assemblies

KPFM

Topography



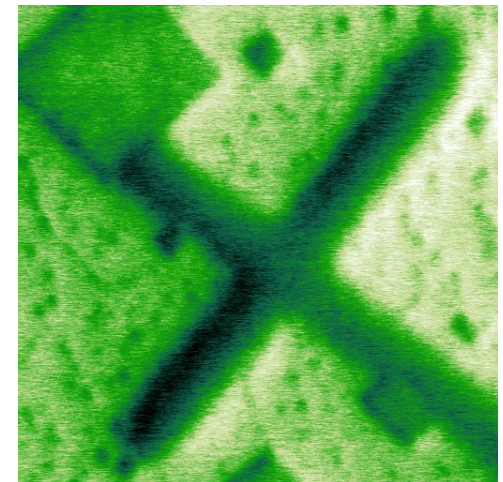
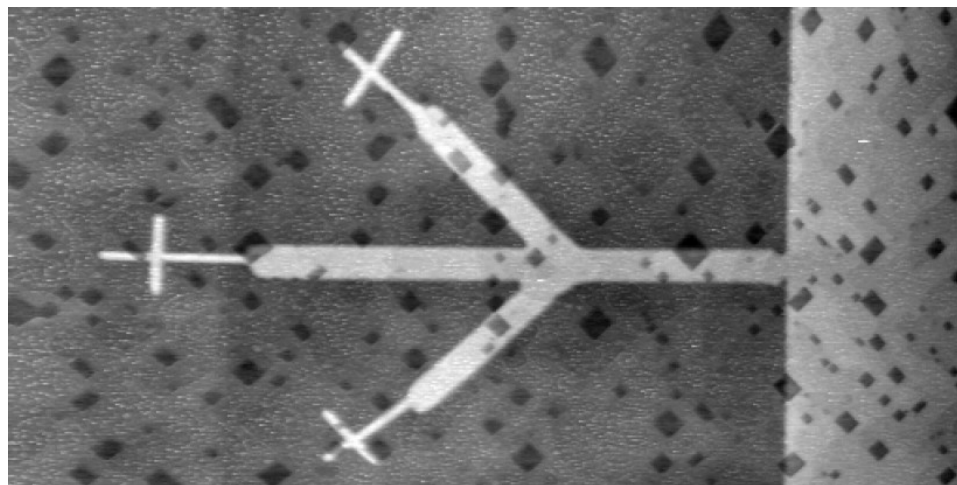
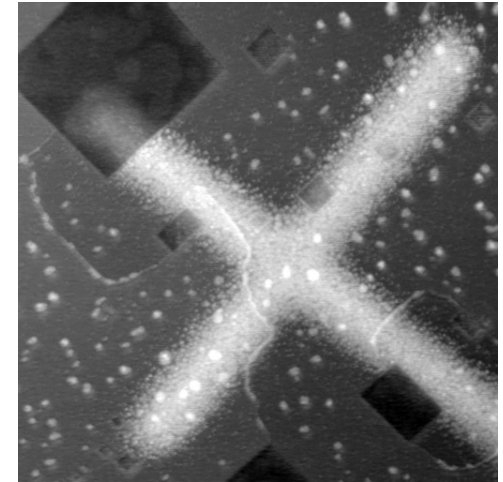
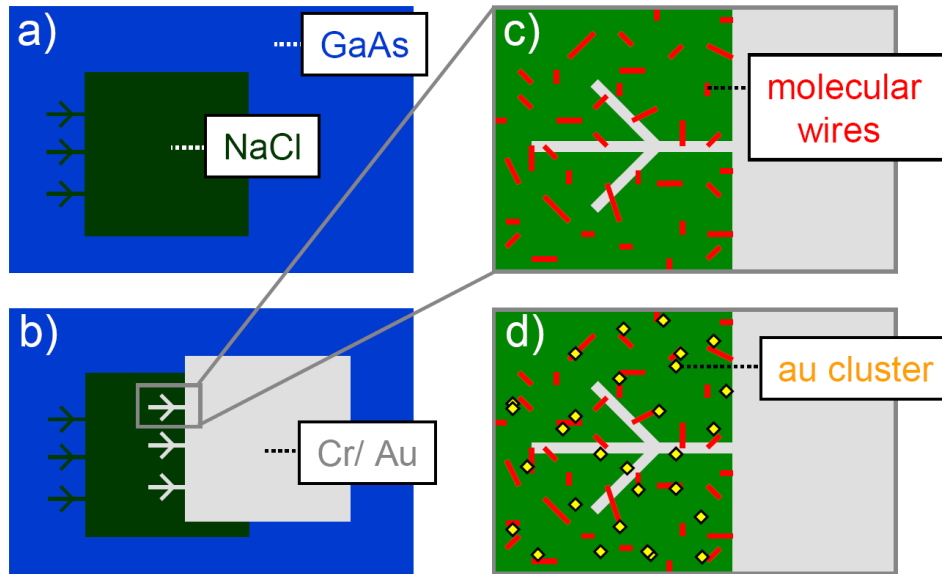
LCPD



LCPD = -0.5...0.2 eV

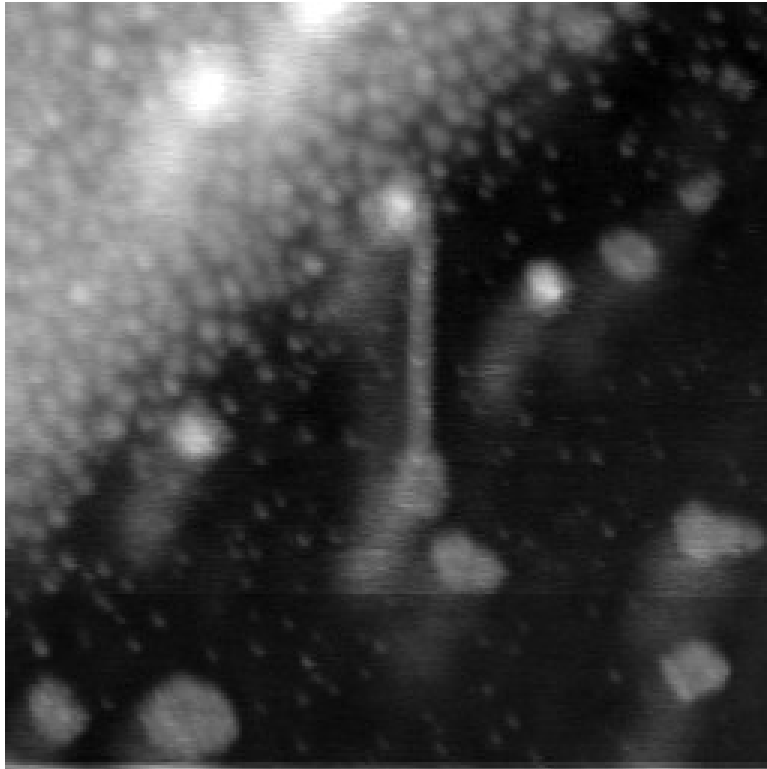
Contacting Molecular Assemblies

Nanostencil (IBM Rüschnikon)

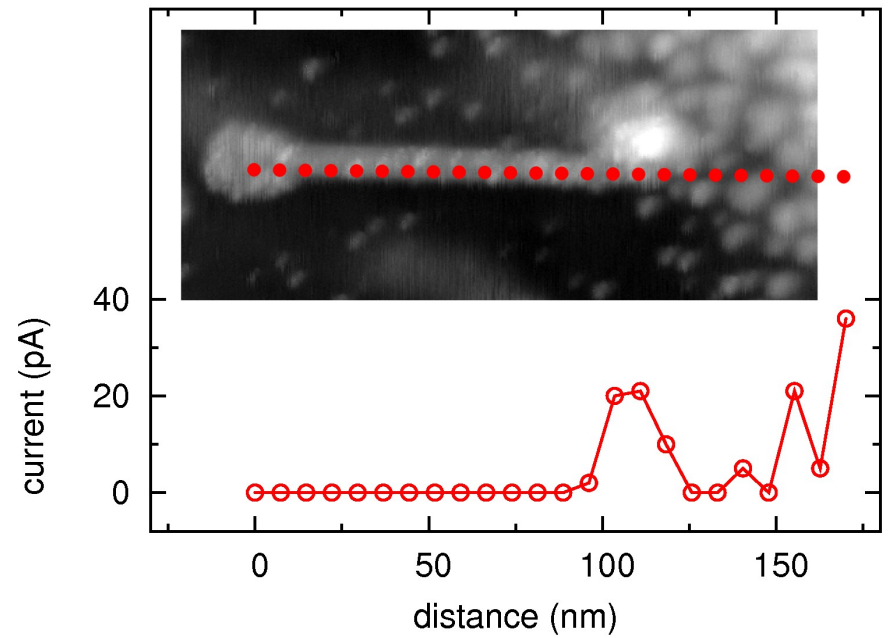


Contacting Molecular Assemblies

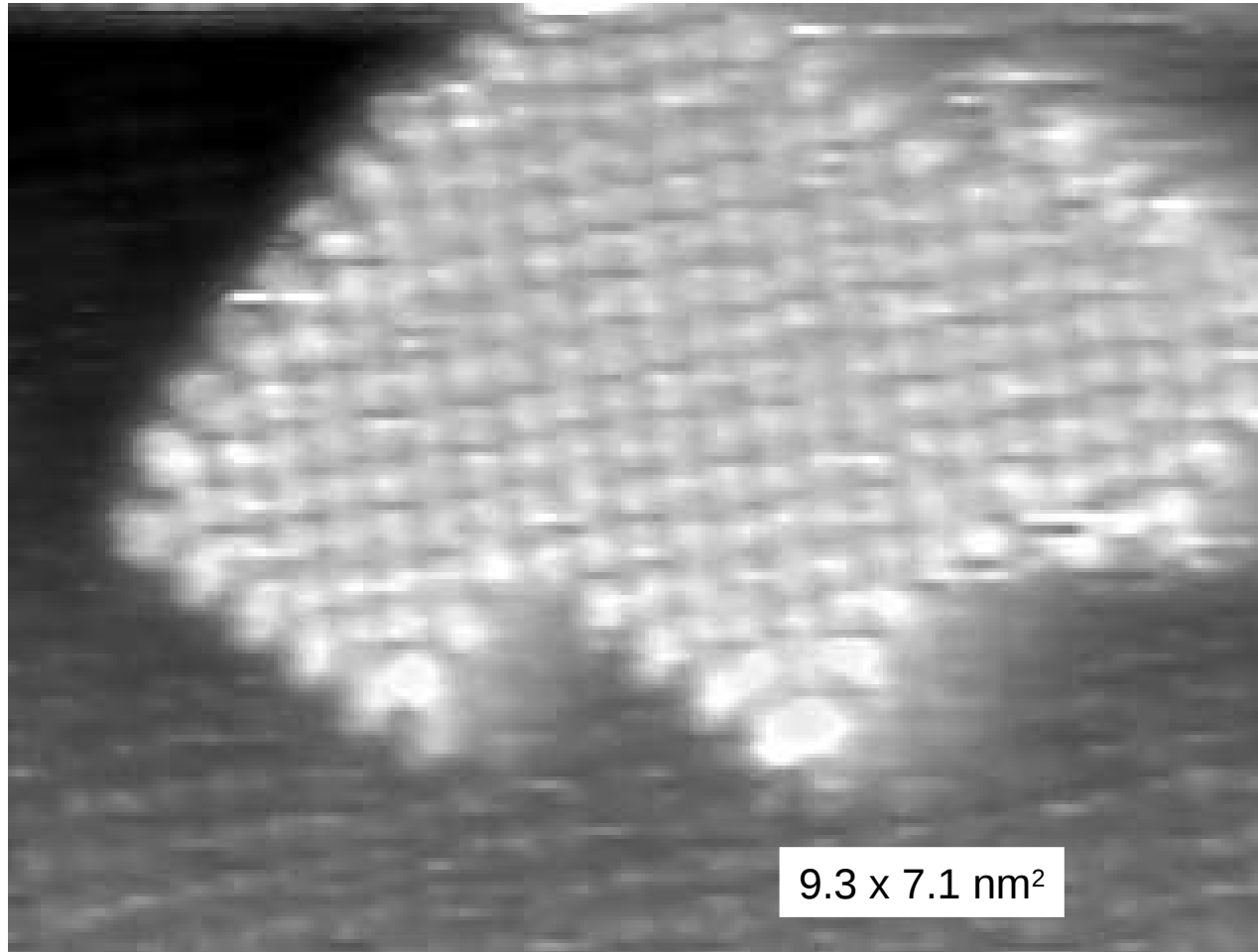
Nanostencil (IBM Rüslikon)



300x300nm²



Die “Nano-Schweiz”



NaCl-Insel mit AFM abgebildet