

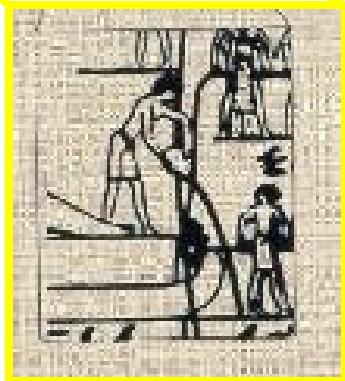
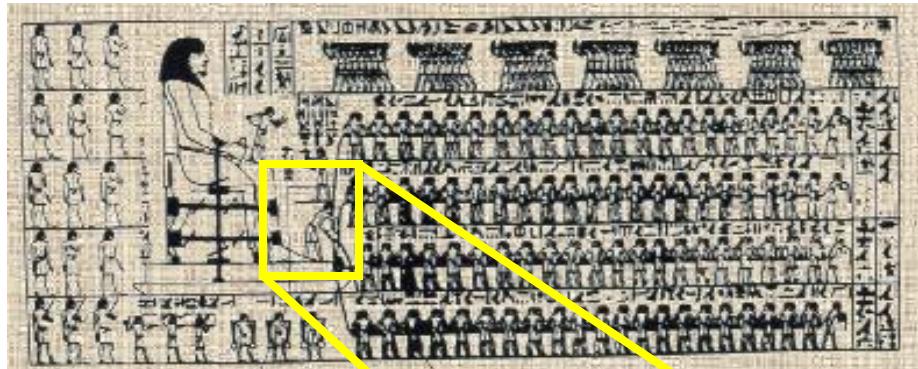
Oberflächenphysik

Scanning Probe Microscopy

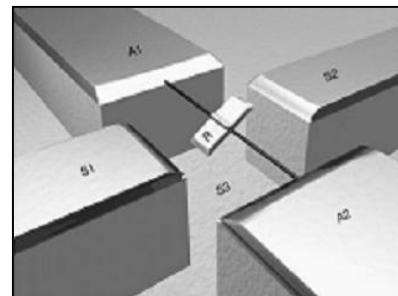
- **Friction Force Microscopy**
- Force Calibration
- Atomic Stick Slip
- Tomlinson Model
- Nano-manipulation
- **Atomic Force Microscopy**
- Short- and Long-Range Forces
- Kelvin Probe Force Microscopy
- Measurements on Semiconducting Devices
- Molecules on Insulating Surfaces

Importance of Friction

Long time ago...

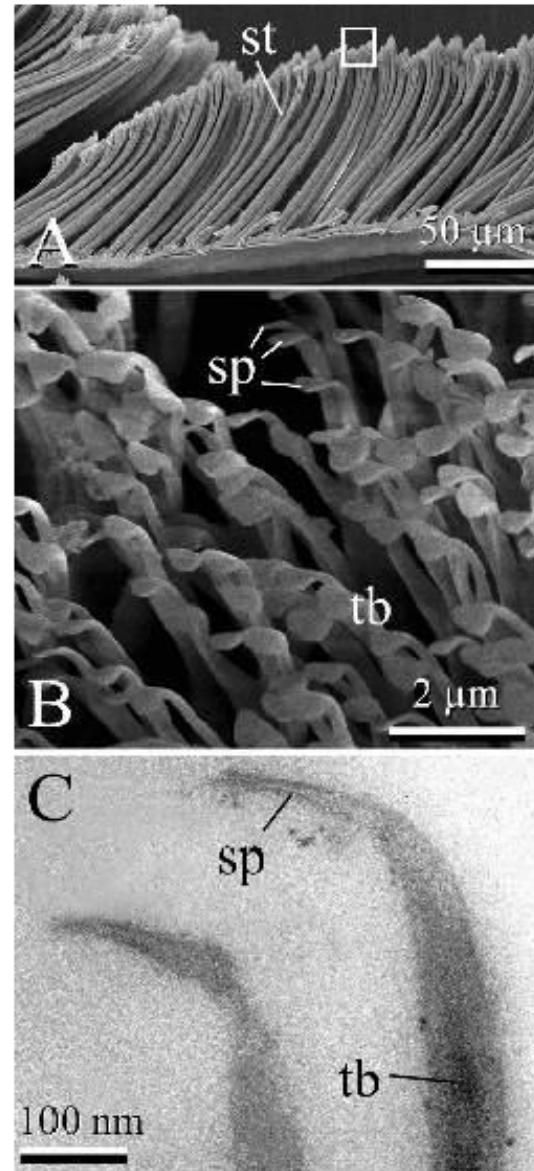


Nowadays...



In all cases: It is highly desirable to reduce and control friction

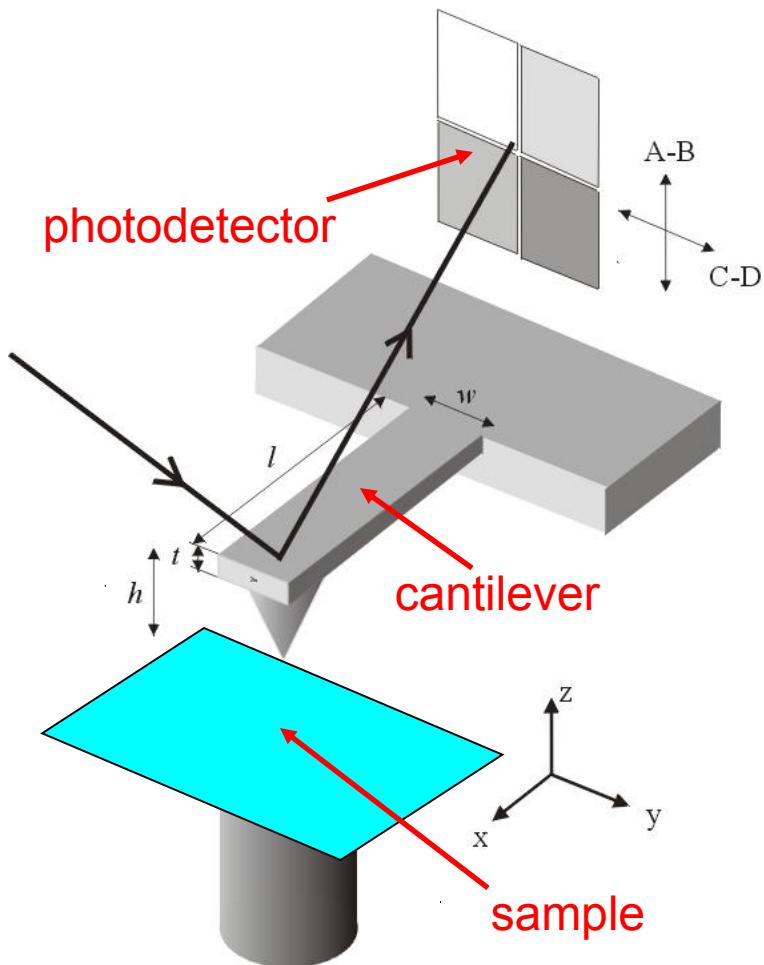
Gecko uses nanometer-sized contacts to climb walls



Gecko is able to control the contact area on all length scales

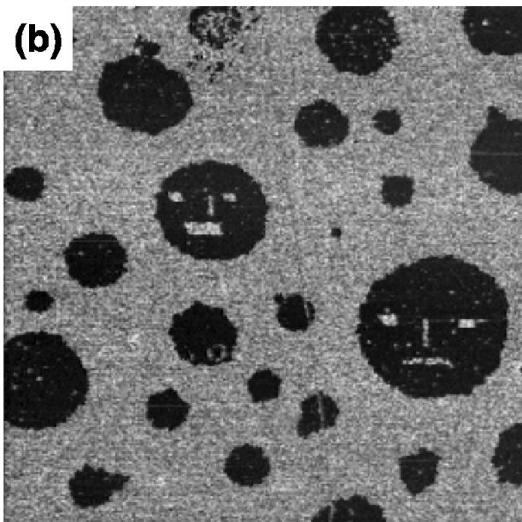
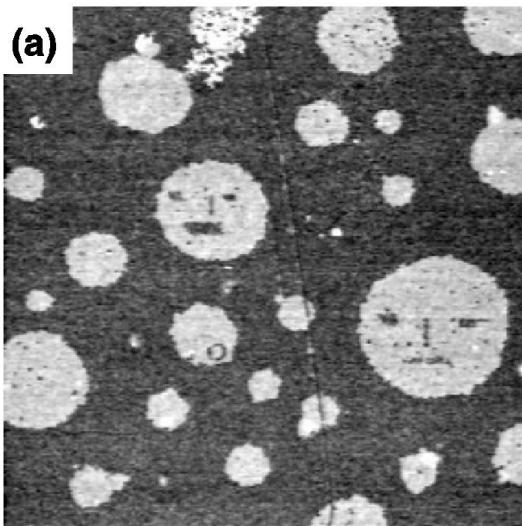
From B. Persson and S. Gorb
JCP, 119, 11437 (2003)

Atomic Force Microscopy (beam deflection)



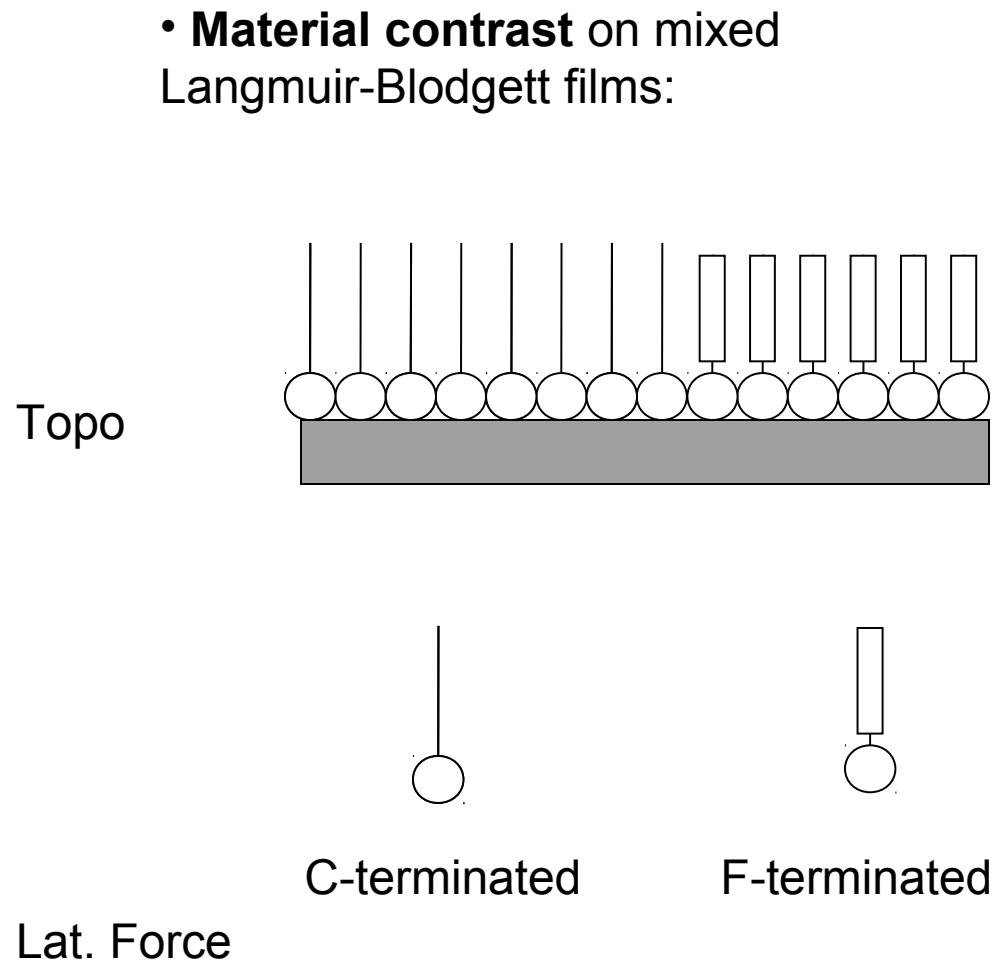
- The normal and lateral forces on a sharp tip sliding on a surface are sensed using a laser beam
- Forces $< 1 \text{ nN}$ can be measured

FFM on Langmuir-Blodgett films

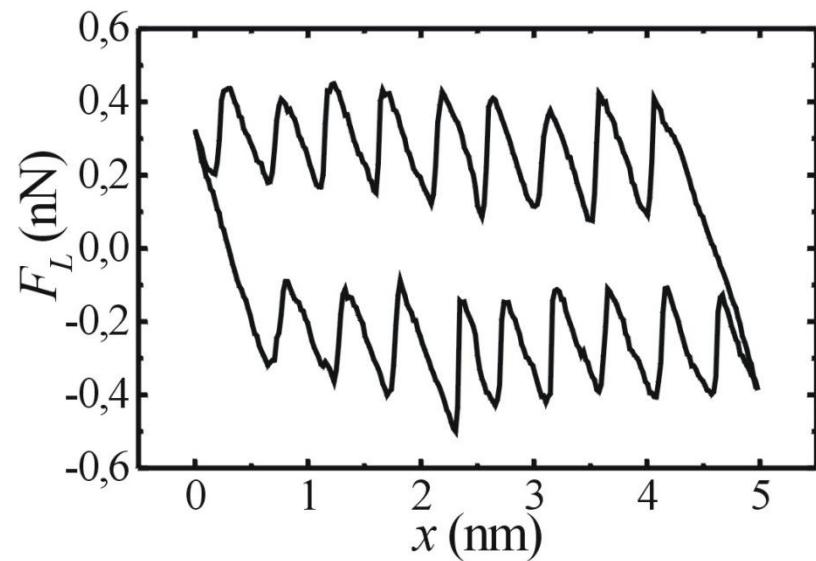
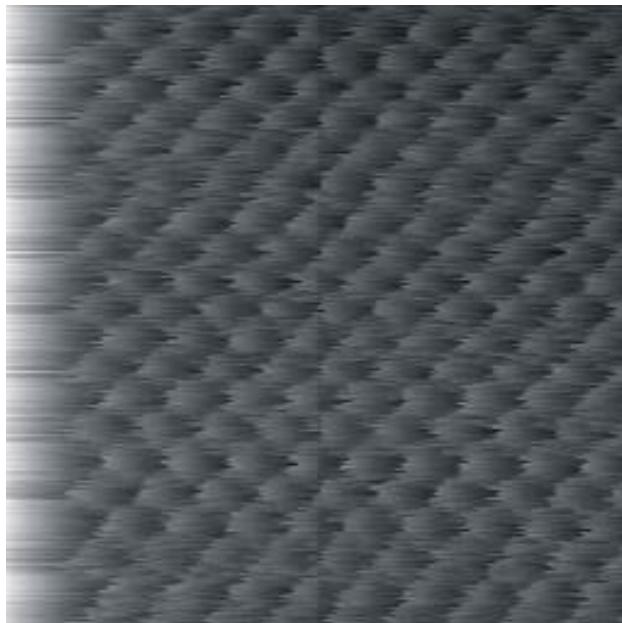


Lat. Force

2.8 μm



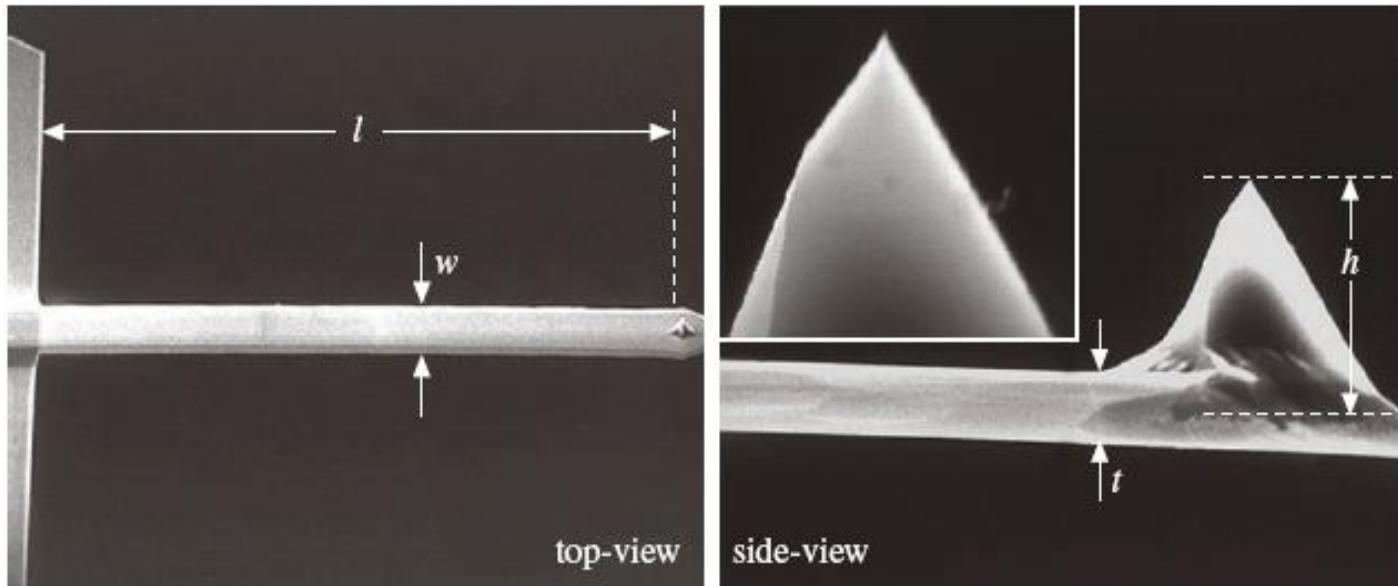
Atomic stick-slip



(friction map and friction loop on NaCl(100) in UHV)

Force Calibration

- Simple if **rectangular** cantilevers are used
- Cantilever width, thickness and length, tip height: from **SEM pictures**



Force Calibration

- Cantilever **thickness** also from the **resonance frequency**:

$$t = \frac{2\sqrt{12}\pi}{1.875^2} \sqrt{\frac{\rho}{E}} f_0 l^2$$

- ρ , E : density and Young modulus
(Nonnenmacher et al., JVSTB 1991)
- For pure silicon:

$$\rho = 2.33 \cdot 10^3 \text{ kg/m}^3$$

$$E = 1.69 \cdot 10^{11} \text{ N/m}^2$$

Force Calibration

- **Normal and lateral spring constants** of cantilever:

$$c_N = \frac{Ewt^3}{4l^3} \quad c_L = \frac{Gwt^3}{3h^2l}$$

- G: shear modulus
- For pure silicon:

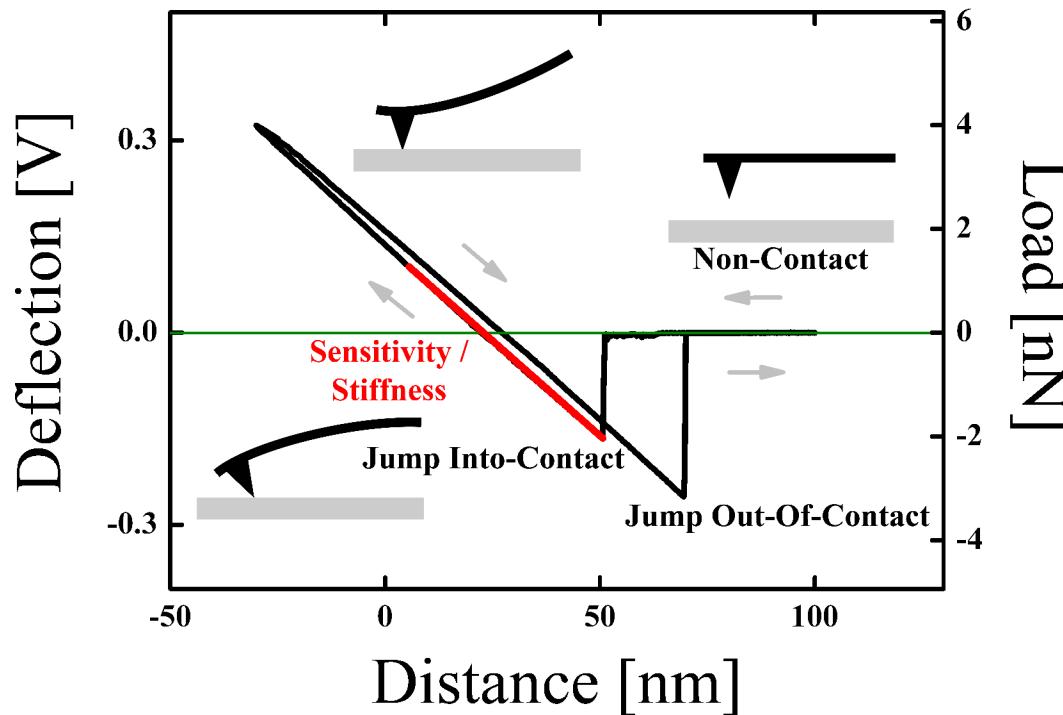
$$\rho = 2.33 \cdot 10^3 \text{ kg/m}^3$$

$$E = 1.69 \cdot 10^{11} \text{ N/m}^2$$

$$G = 0.5 \cdot 10^{11} \text{ N/m}^2$$

Force Calibration

- Next step: **sensitivity of photodetector**
- Force-distance curves on hard surfaces (e.g. Al_2O_3):



- Scanner movement = cantilever deflection
- Slope → sensitivity

Force Calibration

- **Normal and lateral forces:**

$$F_N = c_N S_z V_N \quad F_L = \frac{3}{2} c_L \frac{h}{l} S_z V_L$$

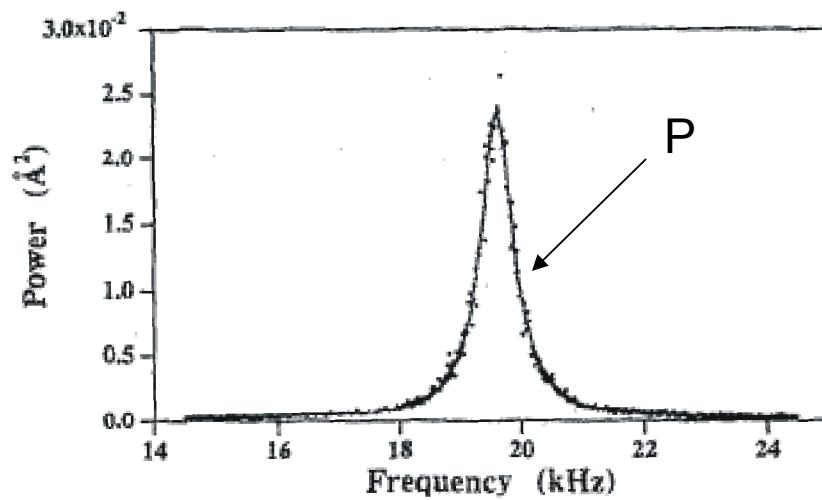
(if the laser beam is above the probing tip!)

- V_N, V_L : normal and lateral signals

Force Calibration

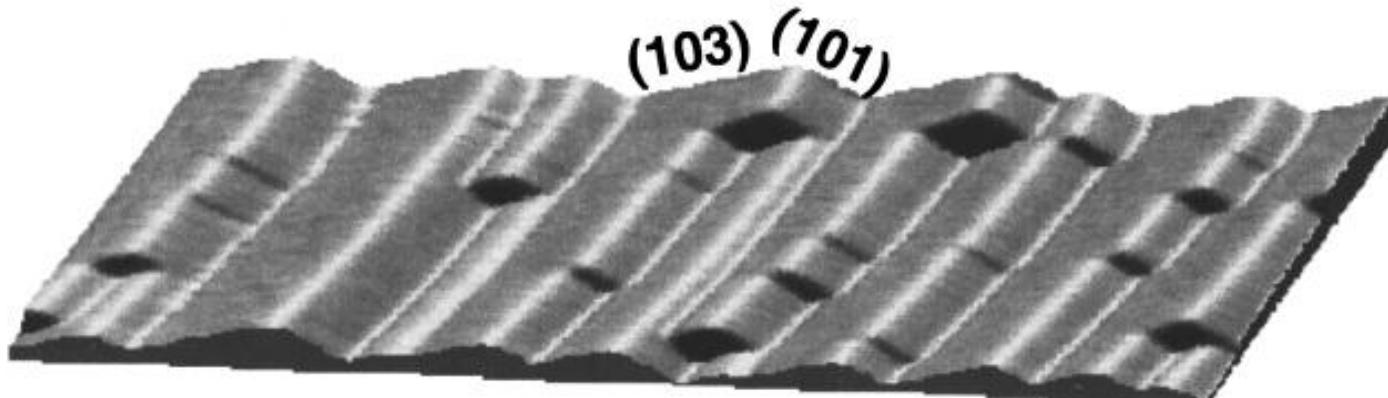
- Alternative method: Spring constant **from thermal power spectrum** (Hutter et al., RSI 1993)
- Correct relation (Butt et al., Nanotech. 1995):

$$c_N = \frac{4k_B T}{3P}$$

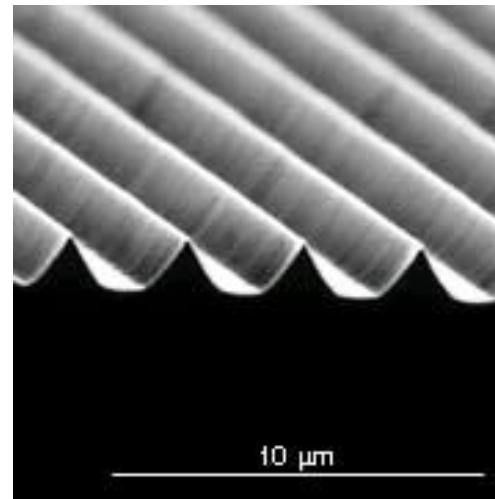


Force Calibration

- Alternative method: Scanning over profiles with **well-defined slope**
(Ogletree et al., RSI 1996)



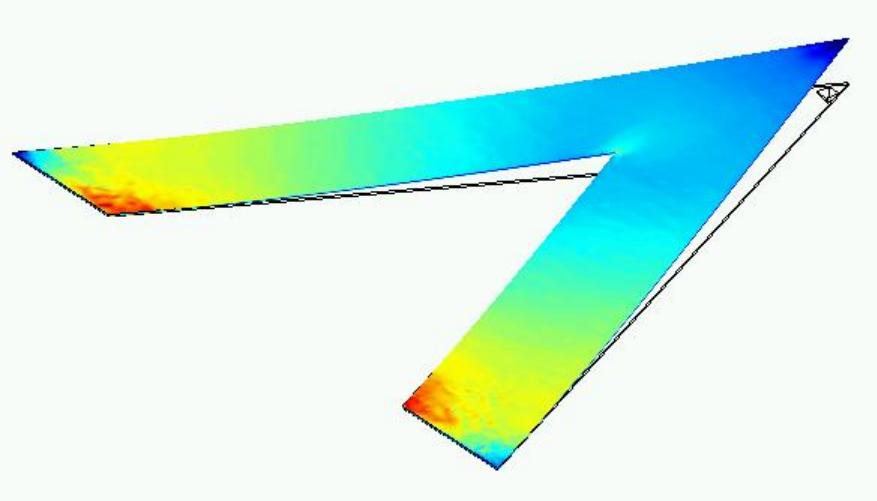
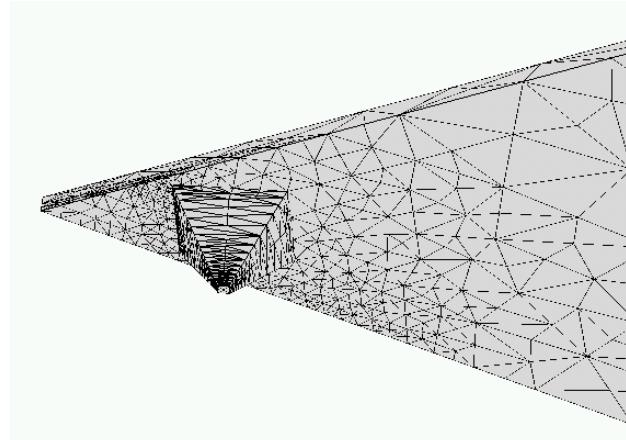
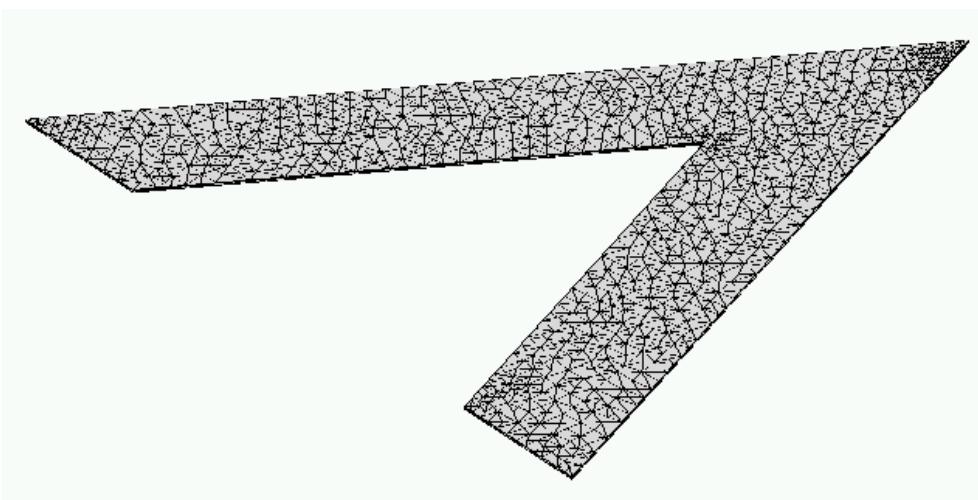
Commercially available grating:



(TGG01, NT-MDT,
Moscow)

Force Calibration

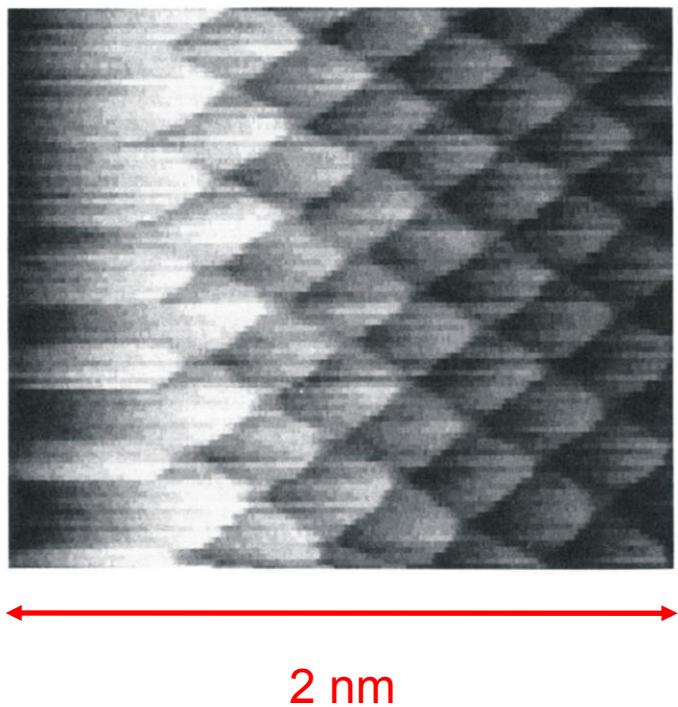
- Different shapes → Finite elements analysis



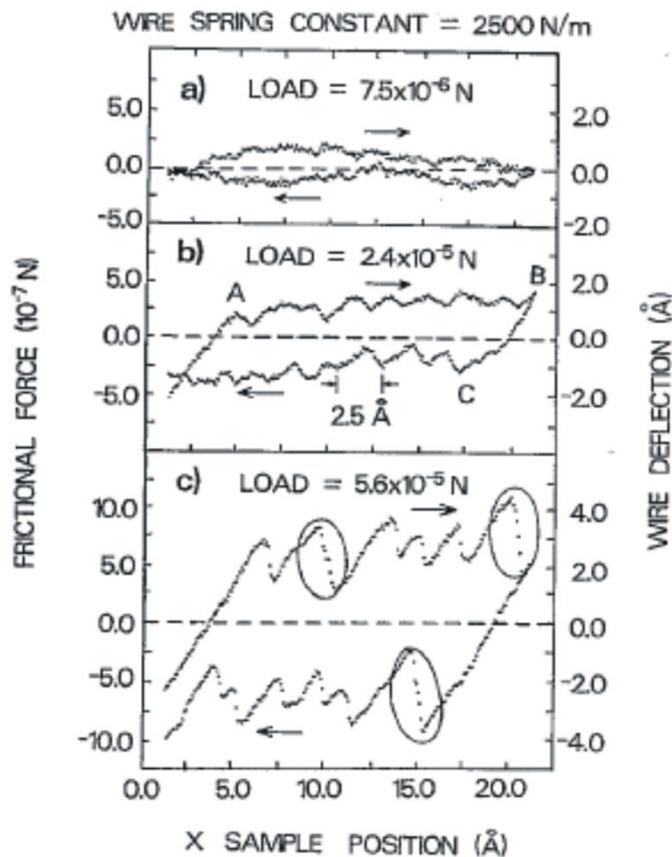
(Femlab™ 3.0)

Atomic-Scale Measurements

- Atomic friction on graphite:



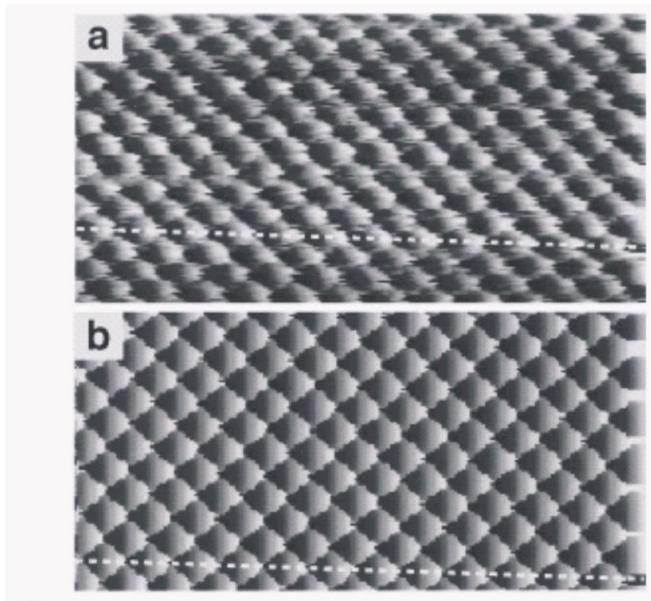
(Mate et al., PRL 1987)



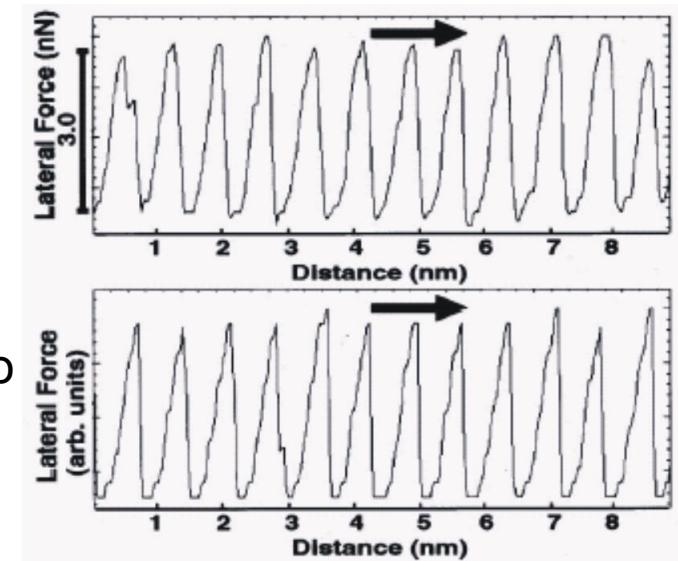
Atomic-Scale Measurements

- Friction on **insulating surfaces** (Lüthi et al., JVSTB 1996):

KBr



Exp

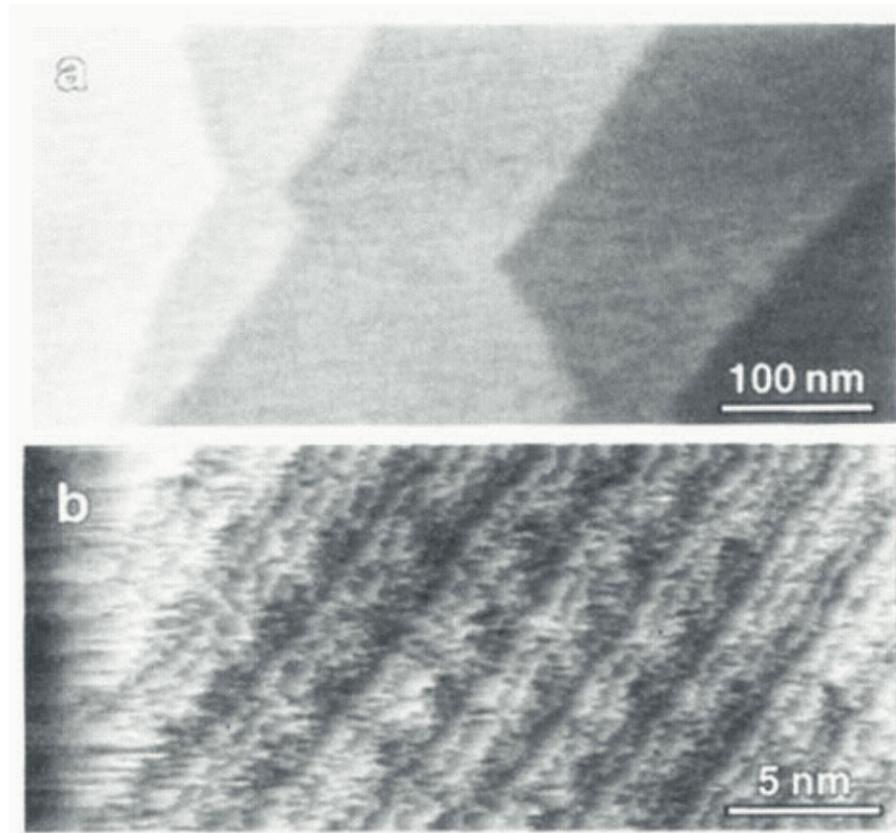


- No individual defects are observed

Atomic-Scale Measurements

- Friction on **semiconductors** (Howald et al., PRB 1995):

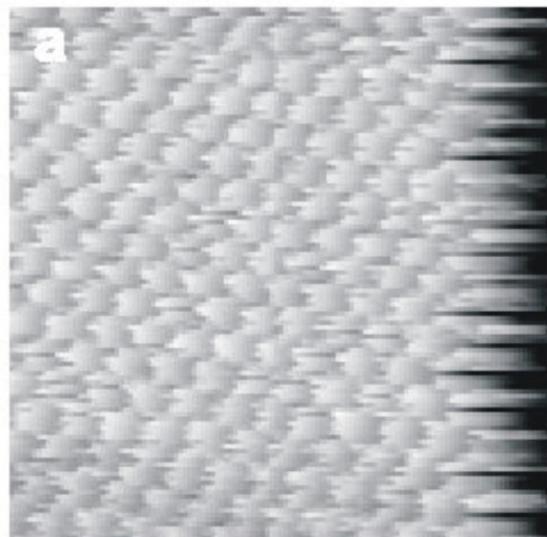
Si(111)7x7



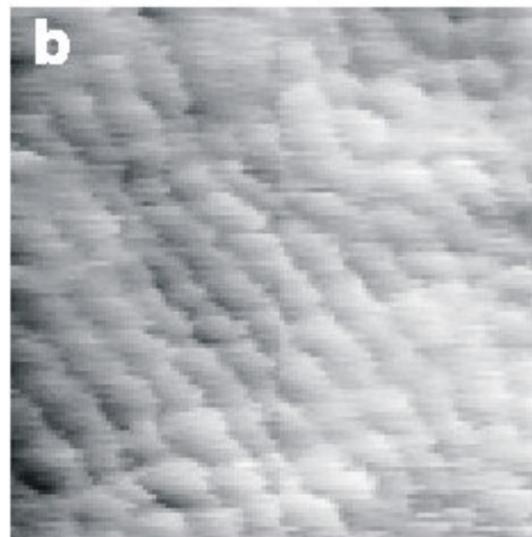
(tip coated with PTFE)

Atomic-Scale Measurements

- Friction on **metal surfaces** (Bennewitz et al., Trib. Lett. 2001):



Cu(111)

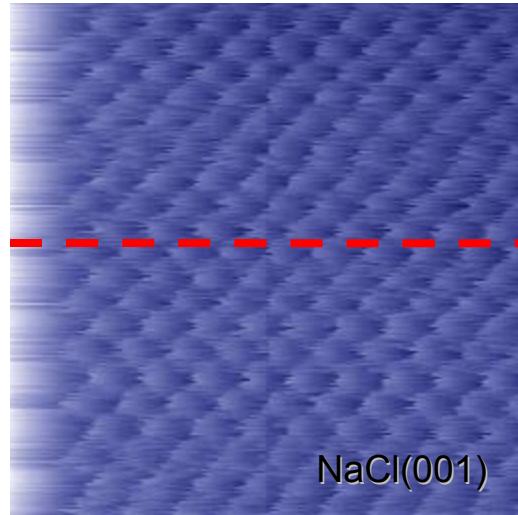


Cu(100)

Irregular features on the (100) surface (less packed!)

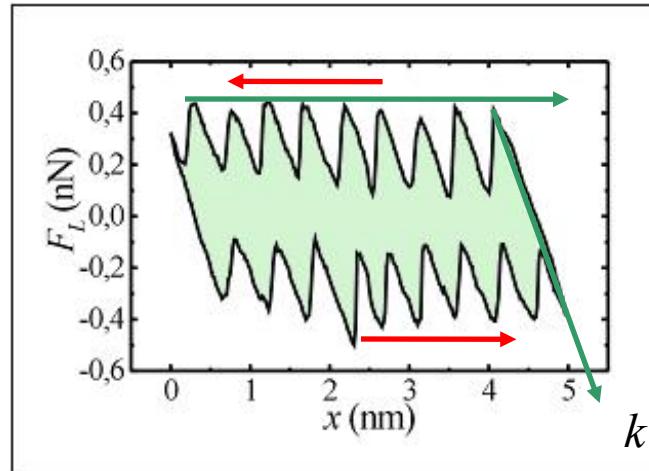
Atomic friction on crystal surfaces

Our model systems: alkali halide surfaces (easy preparation, simple structure)

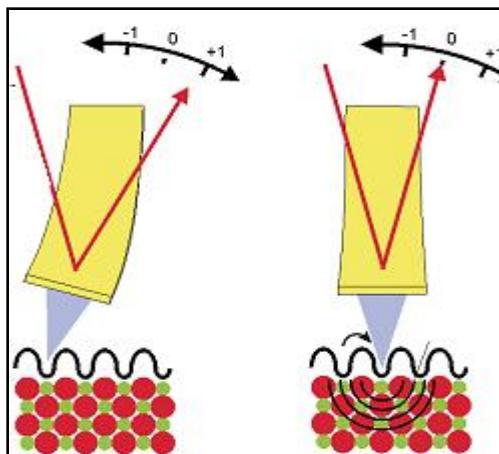


5 nm

Tomlinson model:
(Phyl. Mag. 1929)

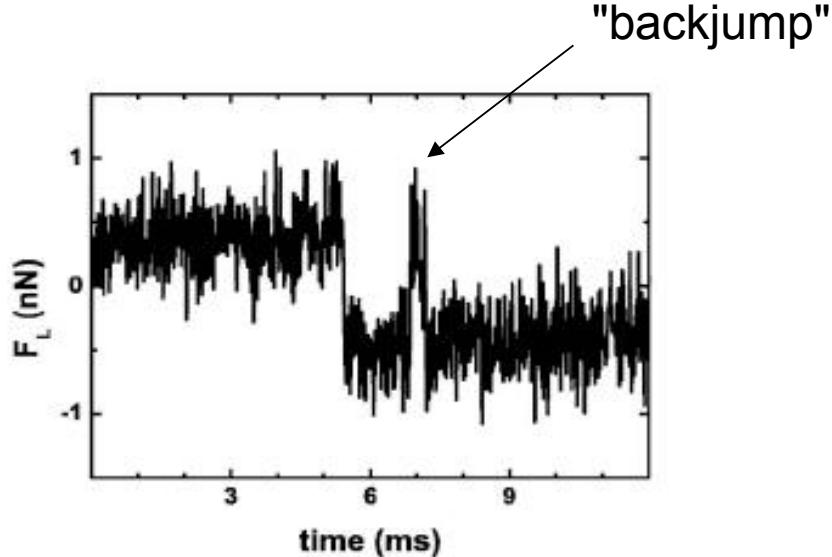
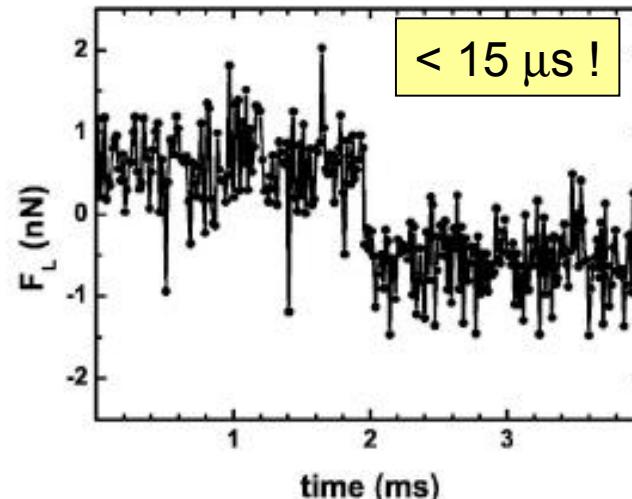
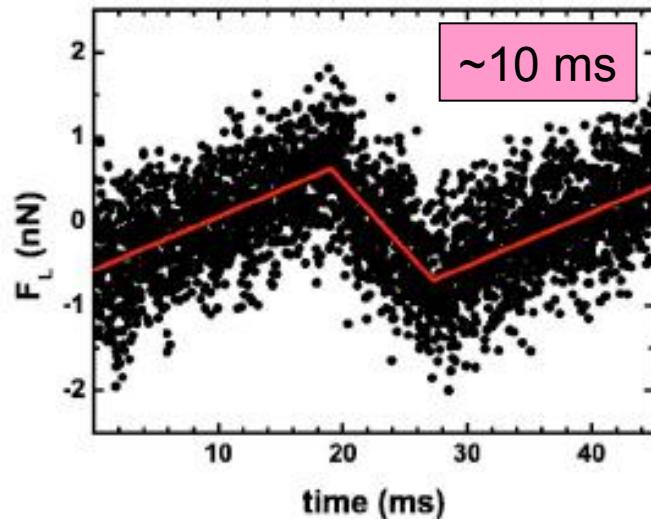


$$F_L^{\max} = \frac{2\pi V_0}{a}$$



Atomic-Scale Measurements

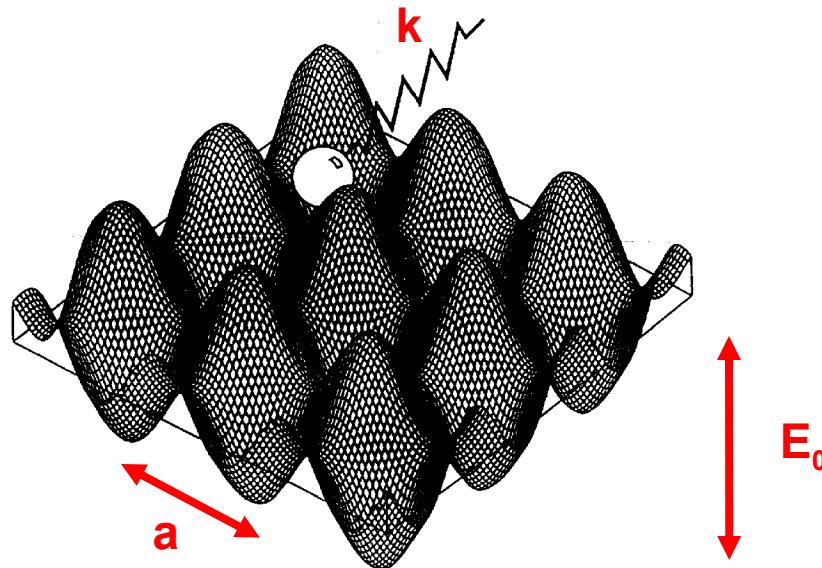
- Wide distribution of slip durations:



Why?

Modelling Atomic Friction

- The tip is subject to
 - 1) periodic interaction with the underlying surface
 - 2) elastic deformation of the cantilever

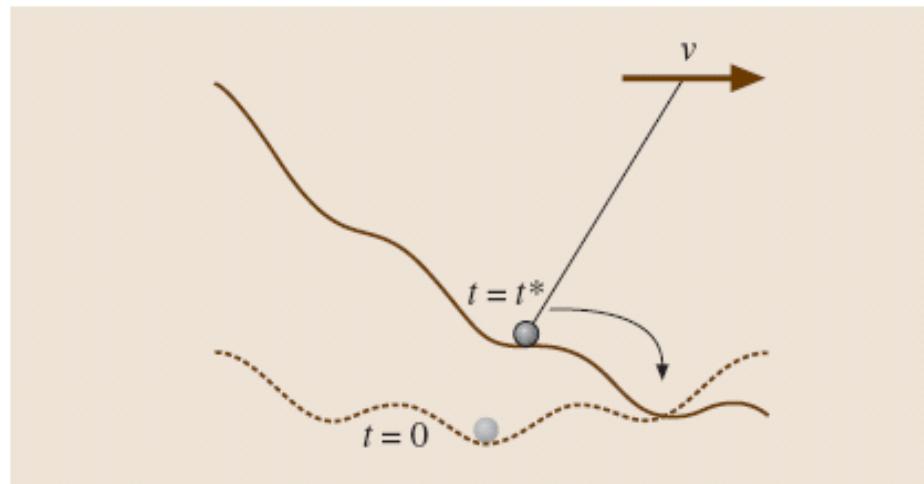


- In 1D the corresponding potential energies are represented by
 - 1) a sinusoid
 - 2) a parabola

Modelling Atomic Friction

- Total energy of the system:

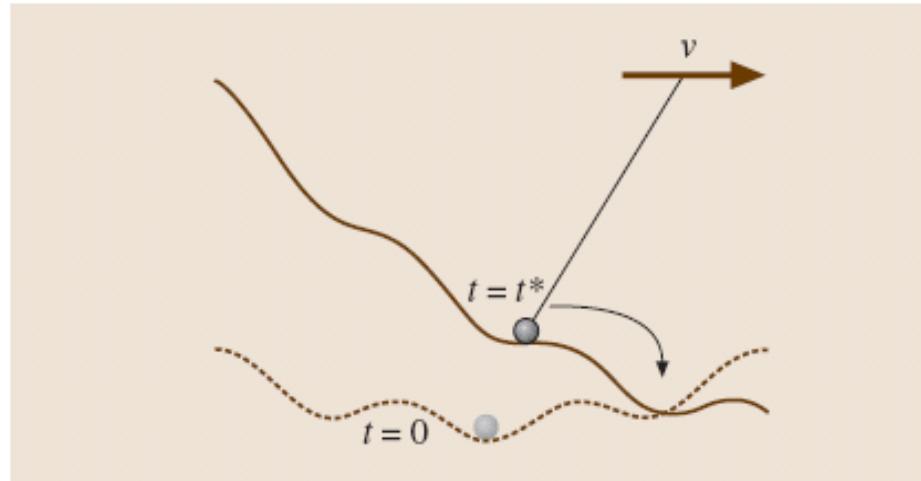
$$U_{\text{tot}}(x, t) = -\frac{E_0}{2} \cos \frac{2\pi x}{a} + \frac{1}{2} k_{\text{eff}} (vt - x)^2$$



- The tip can "stick" to the minima of the potential profile

Modelling Atomic Friction

- Tip position at a given time t:
$$\frac{\partial U_{\text{tot}}}{\partial x} = \frac{\pi E_0}{a} \sin \frac{2\pi x}{a} - k_{\text{eff}}(vt - x) = 0$$



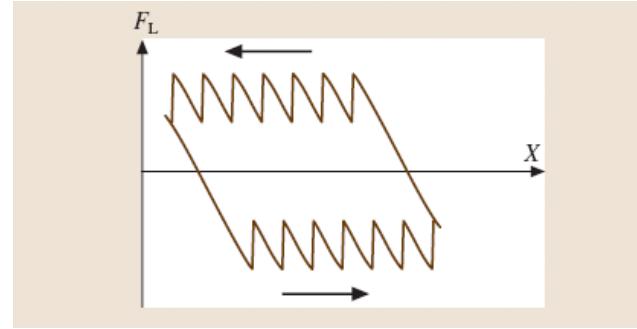
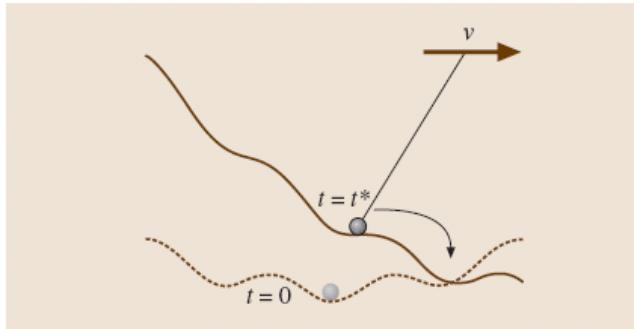
- Critical position (reached at $t = t^*$):

$$x^* = \frac{a}{4} \arccos \left(-\frac{1}{\eta} \right)$$

$$\eta = \frac{2\pi^2 E_0}{k_{\text{eff}} a^2}$$

- **Frictional parameter η** → tip-surface interaction vs. lateral stiffness

Modelling Atomic Friction

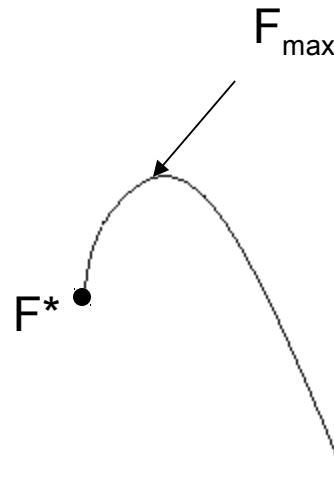


- Critical lateral force (at $t = t^*$):

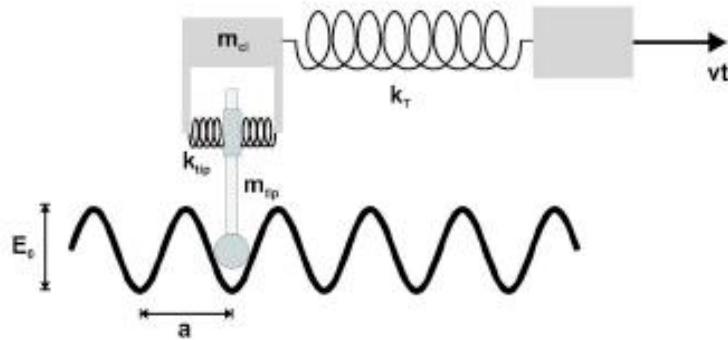
$$F^* = \frac{\pi E_0}{a} \sqrt{1 - \frac{1}{\eta^2}}$$

- Note that $F^* < F_{\max}$!

$$F_{\max} = \frac{\pi E_0}{a}$$

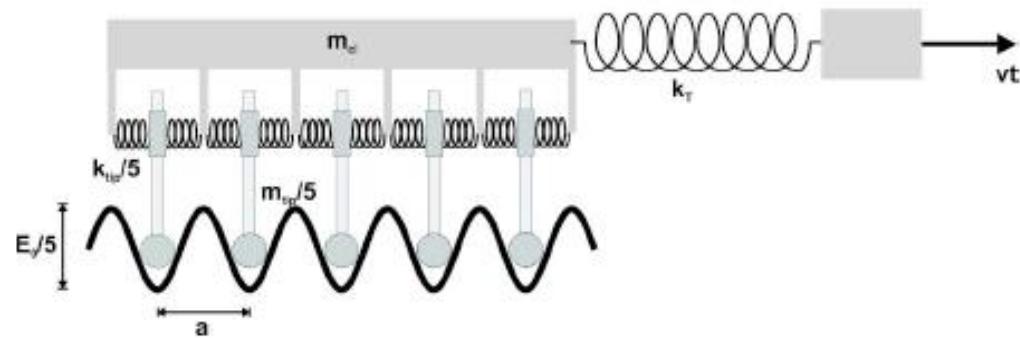


Modelling Atomic Friction



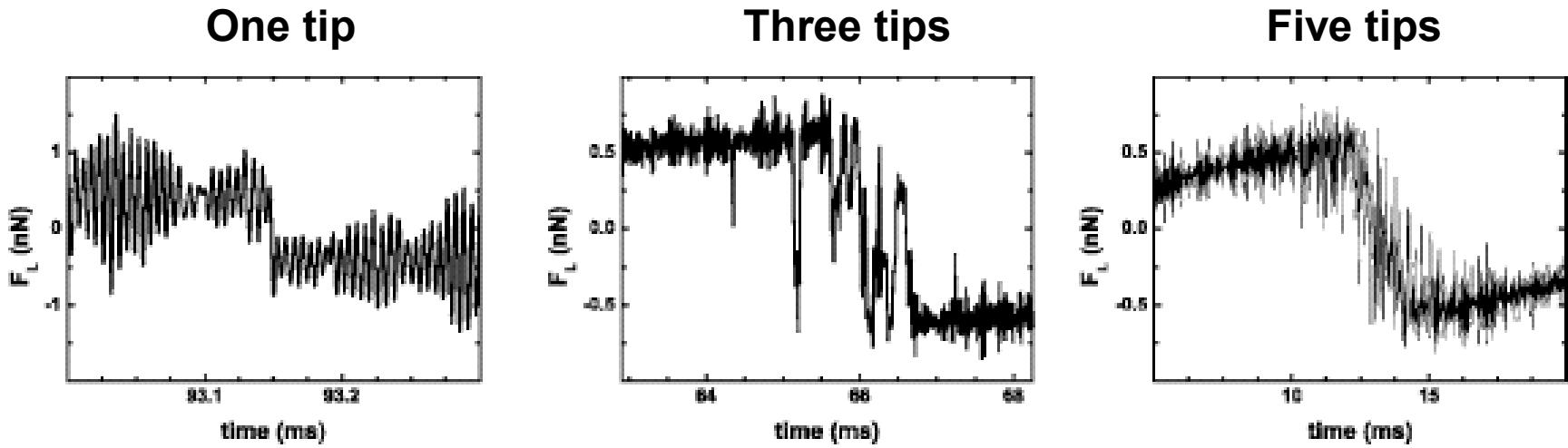
single contact

multiple contact



Modelling Atomic Friction

- Tip → Langevin equation (including thermal noise)
- Cantilever → Newton equation (without thermal noise)

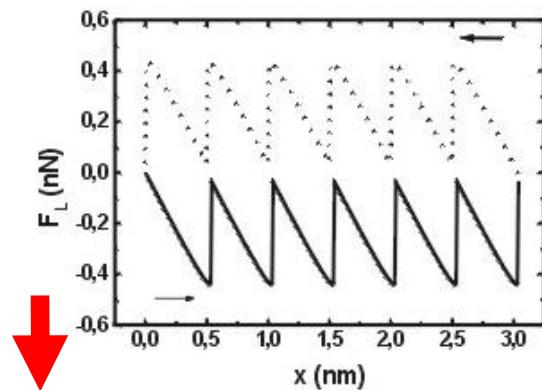


- Long slip times are found with multiple tips only
(Maier et al., PRB 2005)

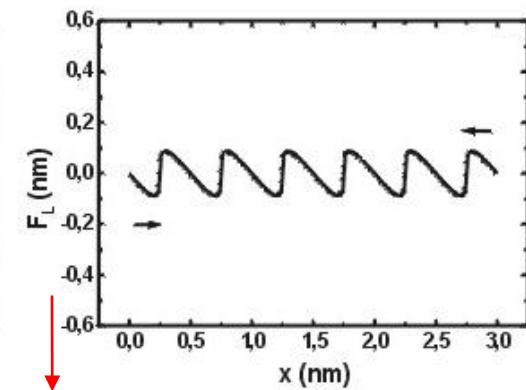
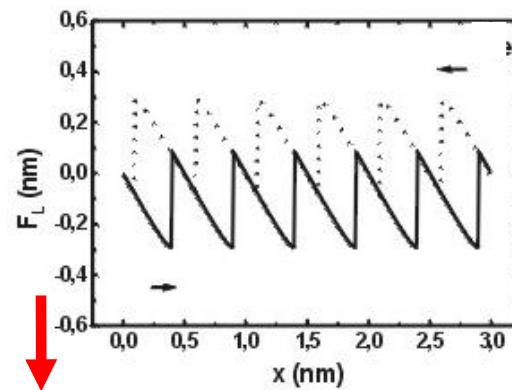
Superlubricity

- From the Tomlinson model (without thermal activation):

$$\eta > 1$$

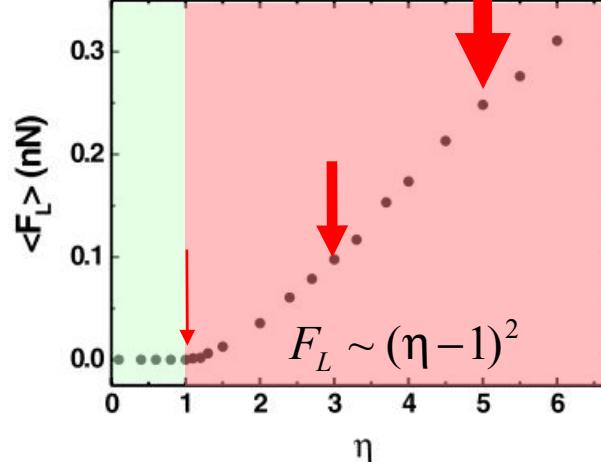


$$\eta < 1$$



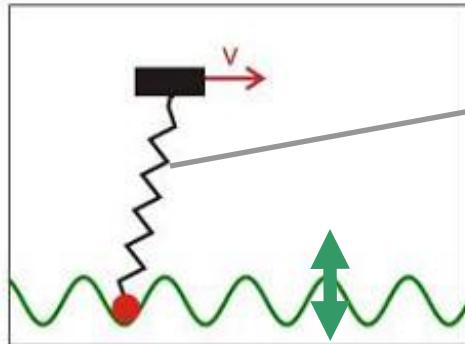
superlubric regime

$$F_L \sim \eta$$



“Dynamic superlubricity”

A third way to reduce friction: Tomlinson model with TIME modulation



$$V_{\text{elas}} = \frac{ka^2}{2}$$

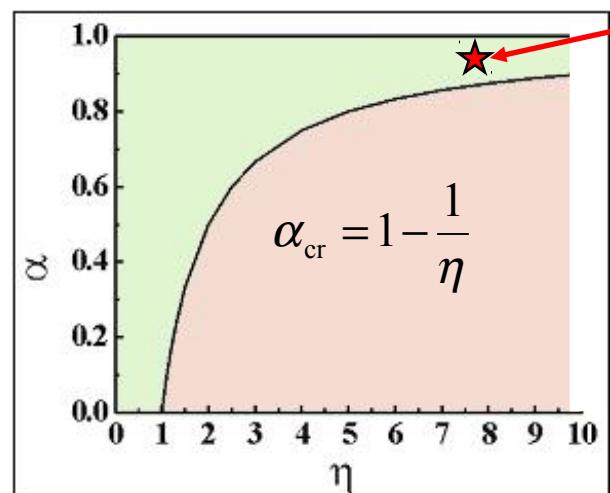
$$V_{\text{int}} = V_0 \cos \frac{2\pi x}{a}$$

$$\eta = \frac{(2\pi)^2 V_0}{ka^2}$$

$$V_0 \rightarrow V_0(1 + \alpha \cos \omega t)$$

$$\eta_{\text{eff}} = \eta(1 - \alpha)$$

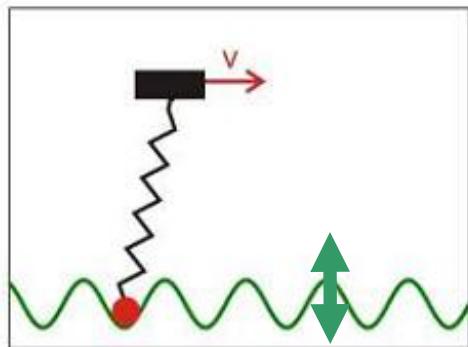
Phase-diagram in the η - α plane:



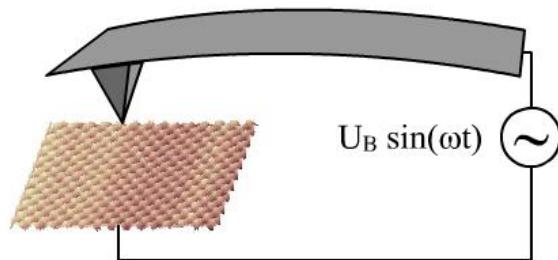
High loads
can be applied!

“Dynamic superlubricity”

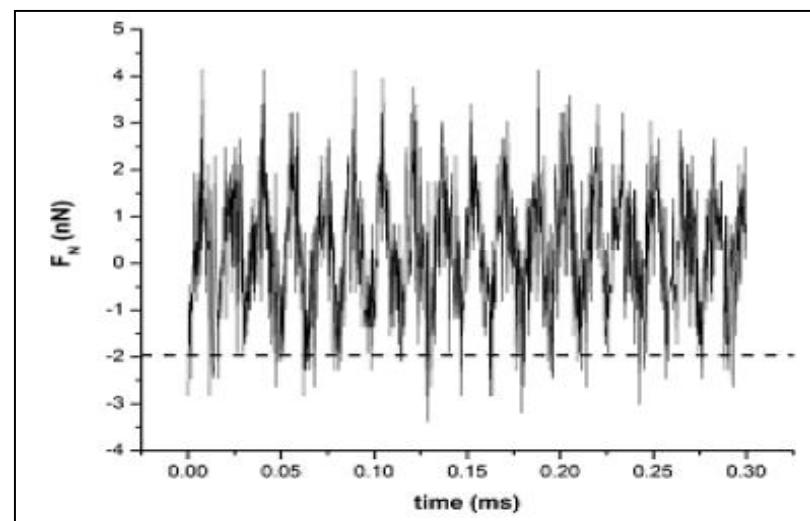
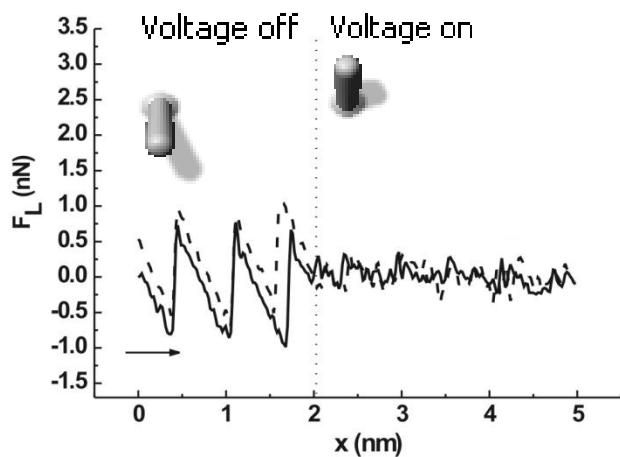
A third way to reduce friction: Tomlinson model with TIME modulation



AC actuation
of the nanocontact:



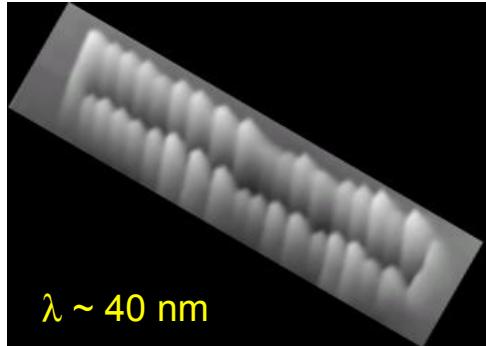
NaCl(001)



Abrasion wear at the nanoscale

Ripples induced by localized abrasion:

Scratching single lines:

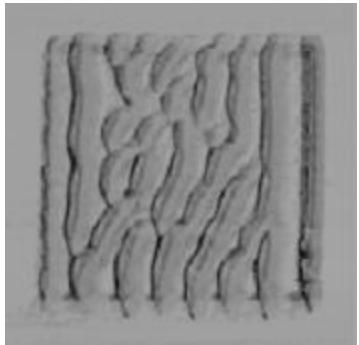


Analogies to waterjet cutting:

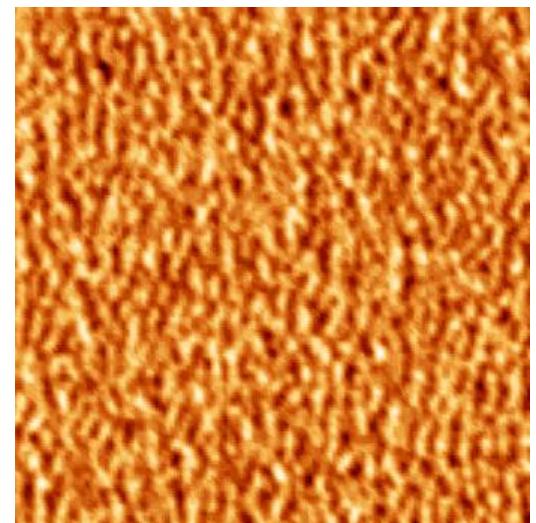


- Combined erosion and relaxation
- Thermal activation of atomic-scale wear
- Numerical analysis in progress

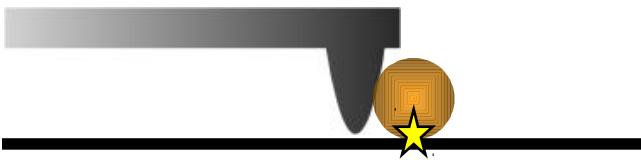
Scratching square areas:



Analogies to sand ripples:

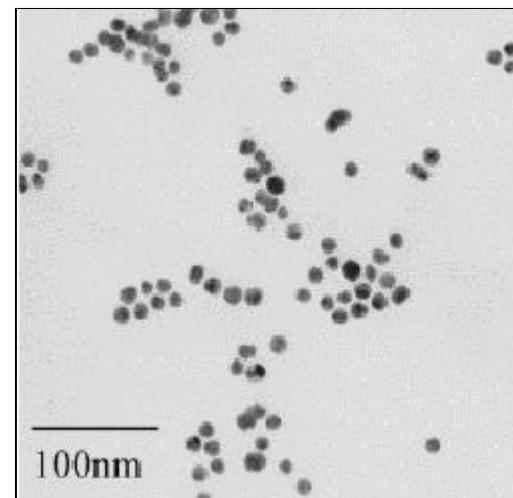


Manipulation of Nanoparticles



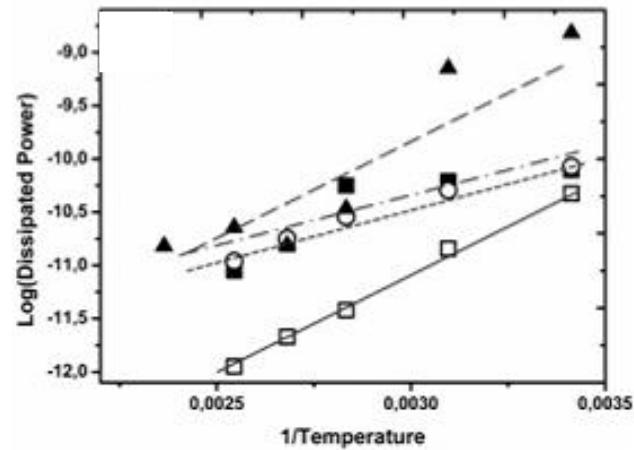
Power dissipation at onset of motion:

$$P_{\text{tip}} = \frac{1}{2} \frac{k\omega_0}{Q_{\text{cant}}} (Q_{\text{cant}} A_d A \sin \varphi - A^2)$$



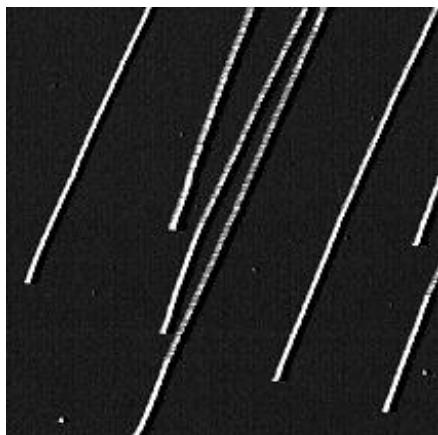
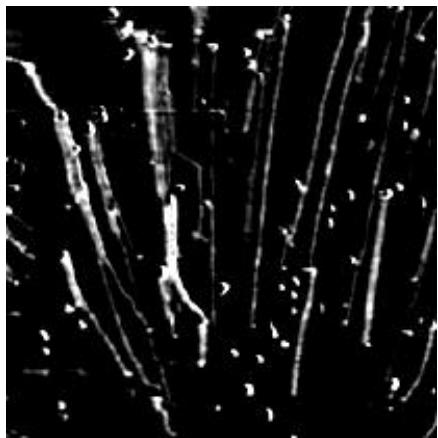
Thermal activation of the particle motion:

$$\log P_{\text{tip}} \sim \frac{1}{T}$$

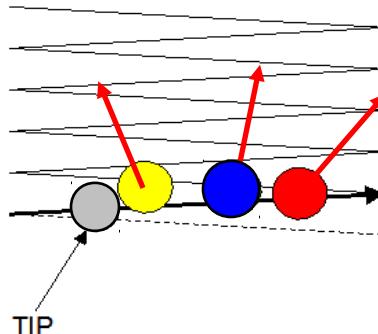


Manipulation of Nanoparticles

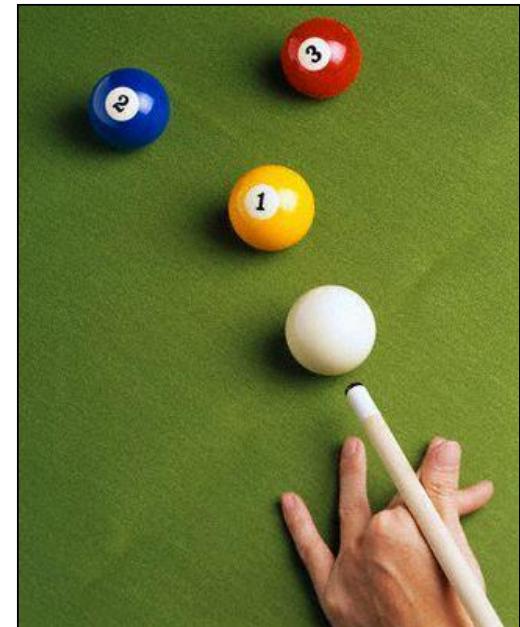
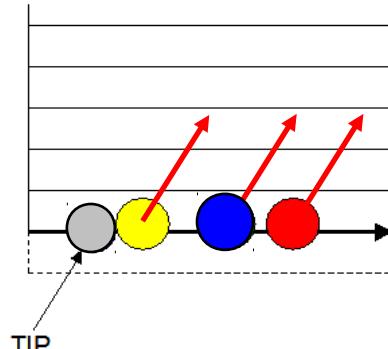
Nanoparticles can be “scattered” by the AFM tip:



Zigzag path:

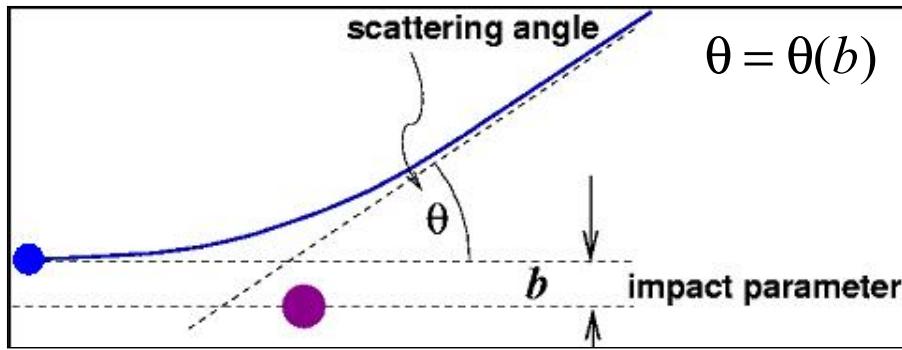


Raster path:

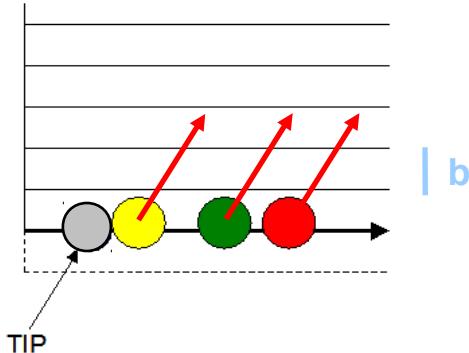


Manipulation of Nanoparticles

Analogy to classical scattering:



Raster path:



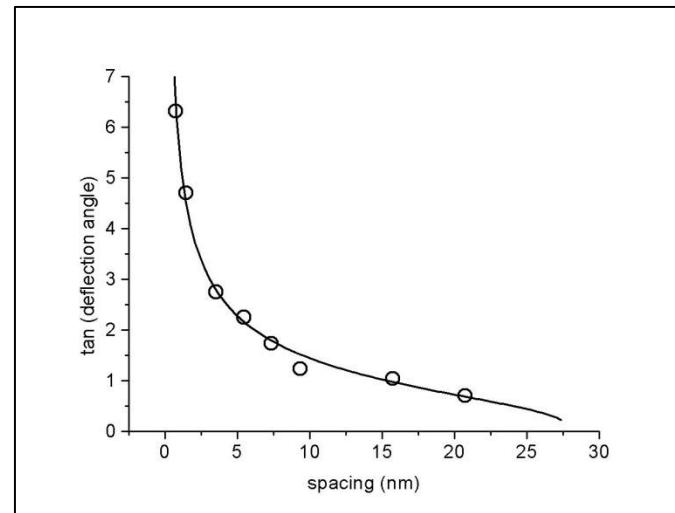
In tapping AFM:

$$\tan \theta = -\frac{b}{2R \left(\cos \alpha_0 + \log \tan \frac{\alpha_0}{2} \right)}$$

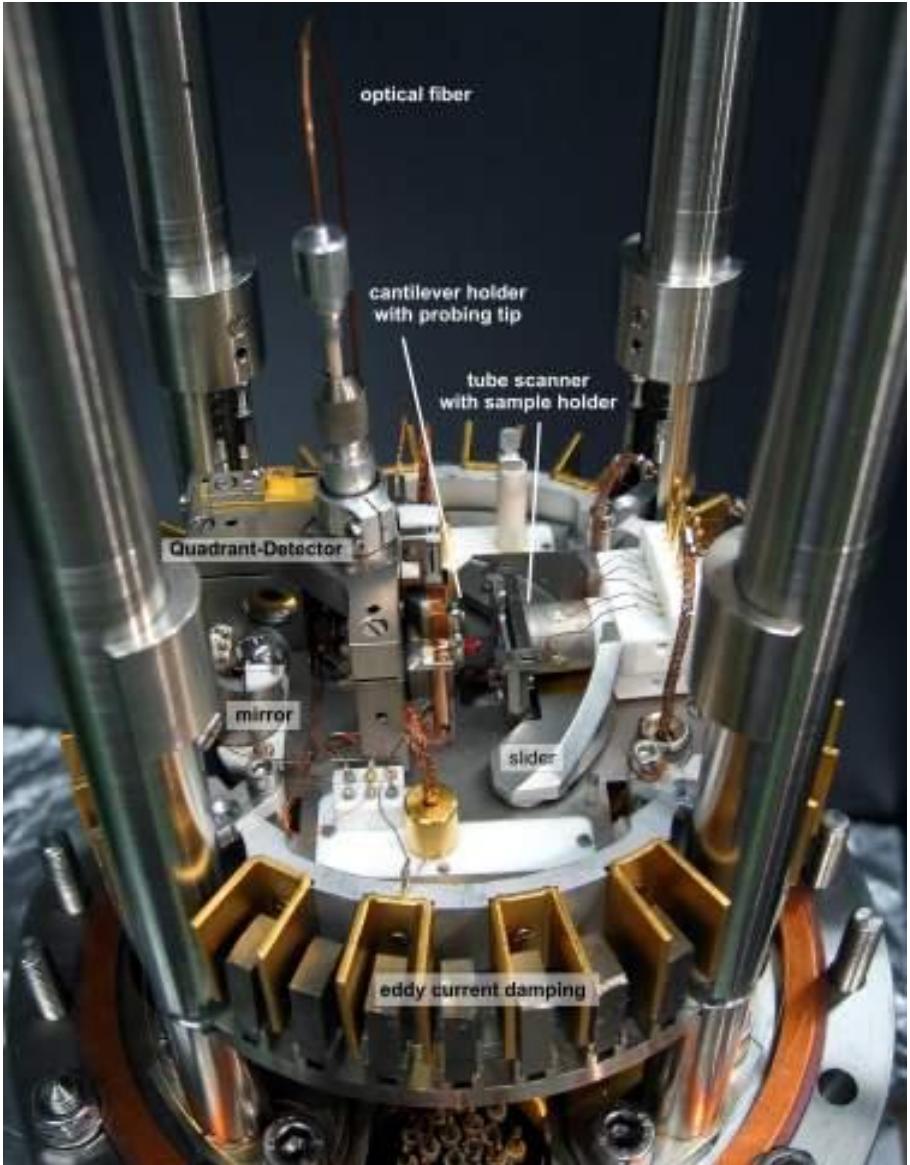
$$\alpha_0 = \arcsin \left(1 - \frac{b}{R} \right)$$

$R \leftarrow$ tip and particle radii

b : spacing between consecutive scan lines



Noncontact-AFM (nc-AFM)

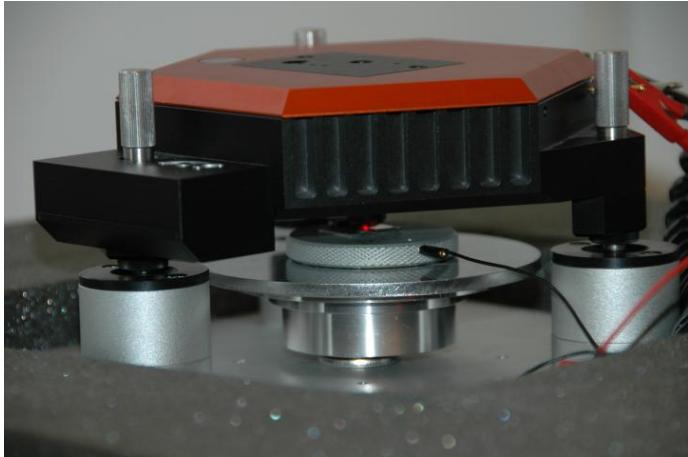


- UHV: Base pressure below 1×10^{-10} mbar
- Operation at room temperature
- Mixed mode: AFM/STM
- Beam deflection method
- Bandwidth of the photodetector: 3MHz
- Evaporation of molecules from a k-cell kept at 165°C or 170°C

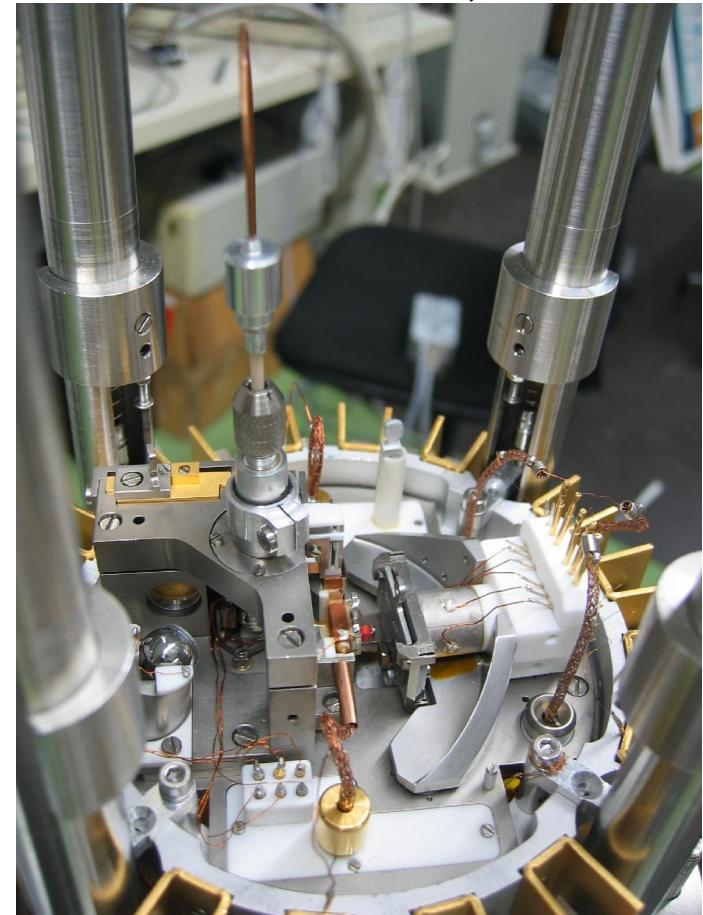
Experimental Setup

AFM/STM

Nanosurf, ambient AFM (Flex-AFM)



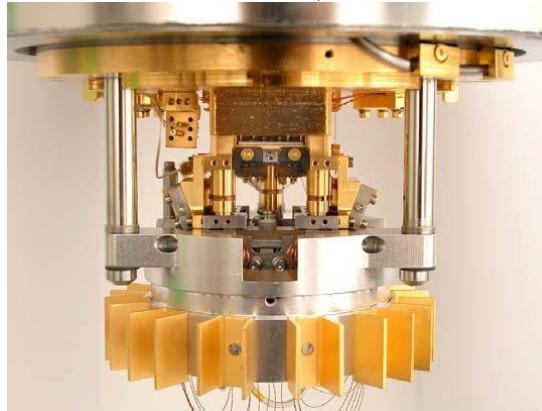
home-build RT-AFM, UHV



Multimode AFM



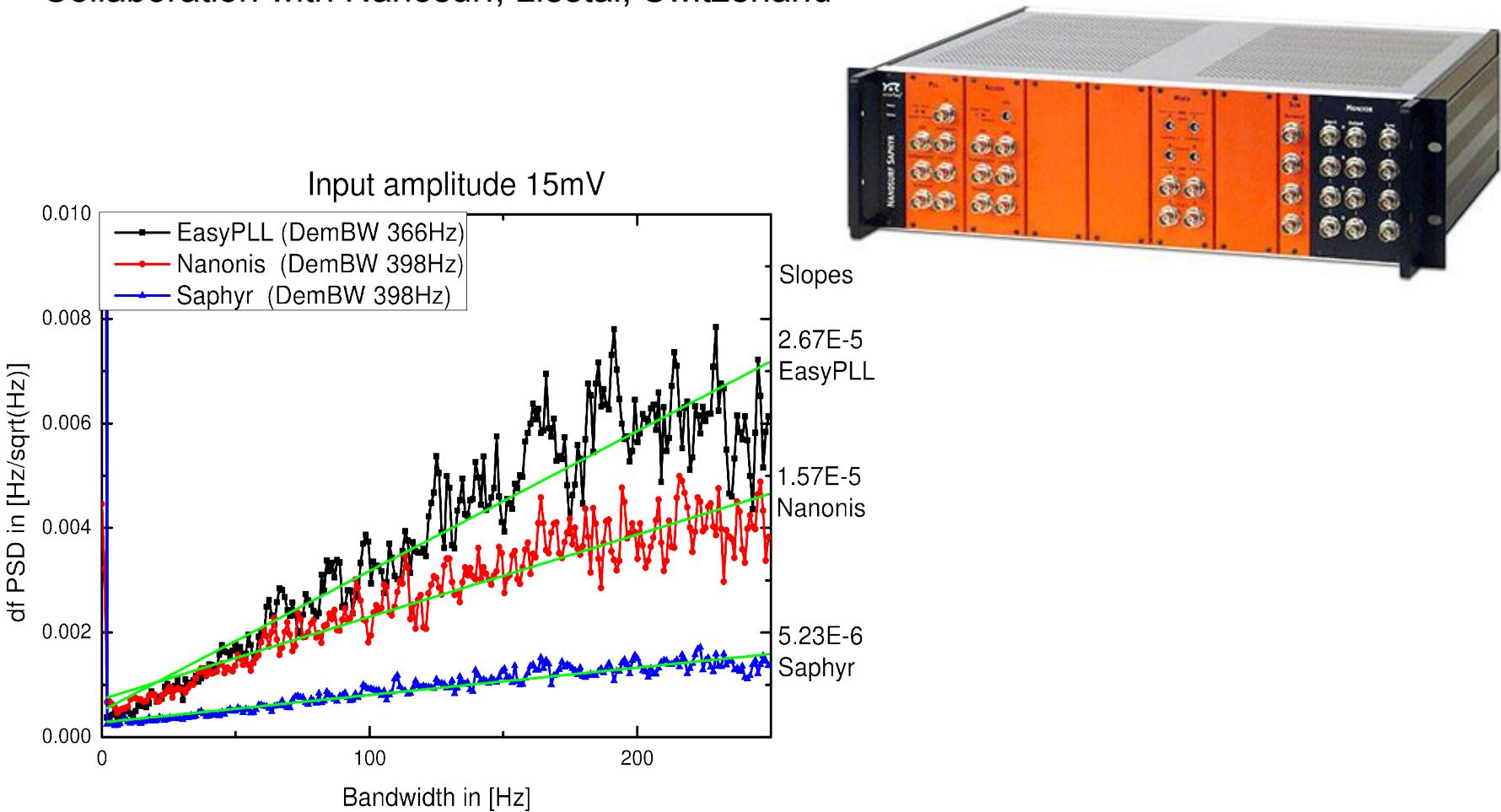
LT-STM/AFM, Omicron



Electronics Development (Saphyr)

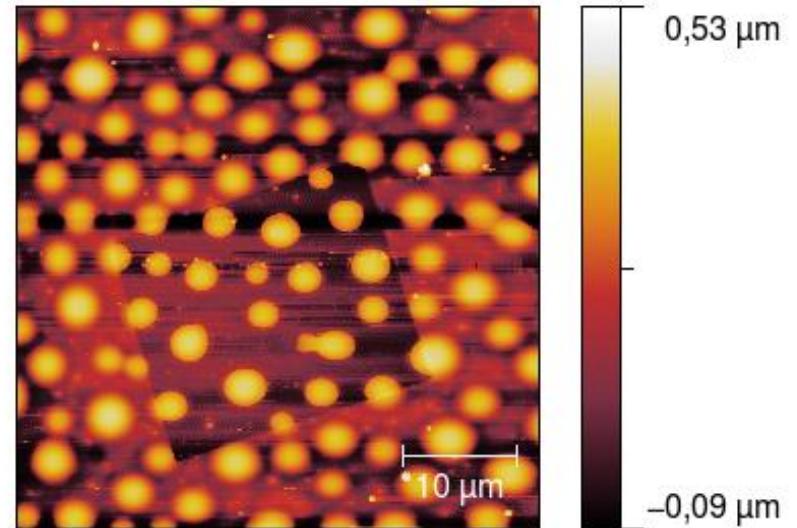
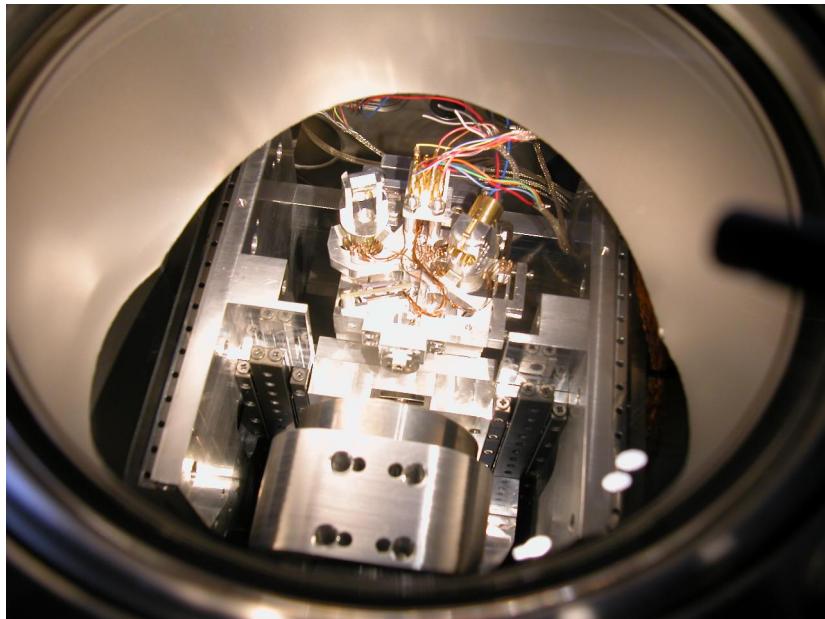
FPGA based multi-PLL system

Collaboration with Nanosurf, Liestal, Switzerland

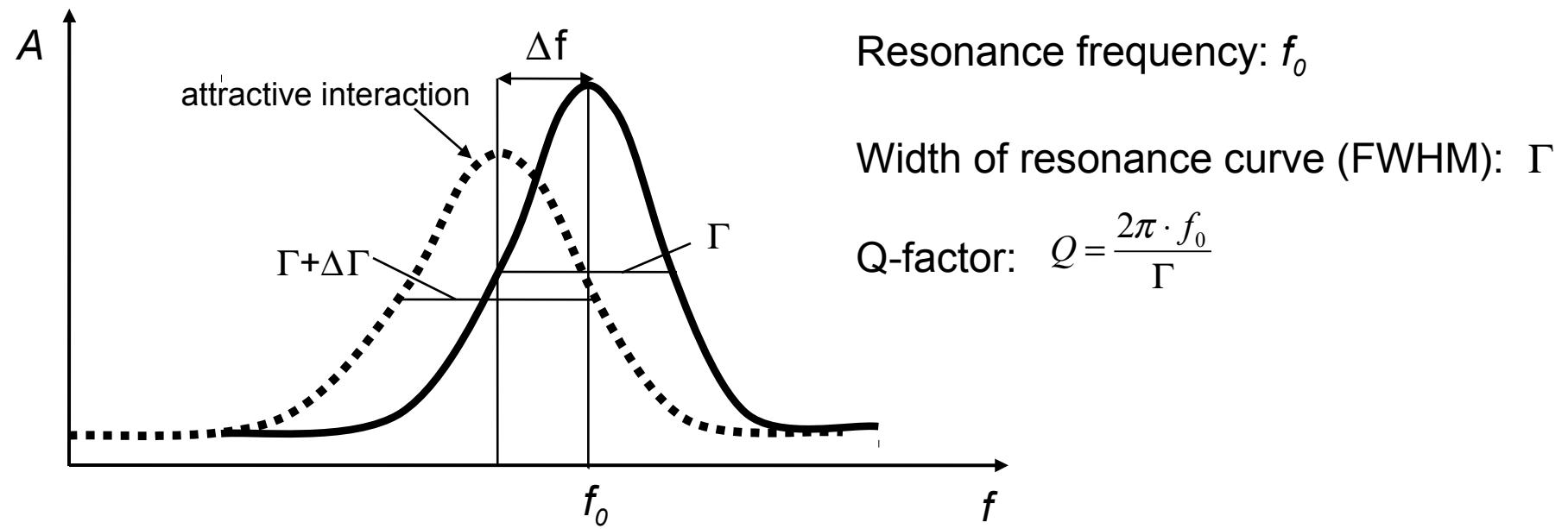


SIMS-SPM Hardware development

Collaboration with Ferrovac, Zurich, Switzerland
and T. Wirtz Lippmann Institute, Luxembourg



Quantitative understanding of nc-AFM



Conservative forces \Rightarrow shift of resonance curve Δf
Dissipative forces \Rightarrow broadening of curve Γ

Forces in nc-AFM

Frequency modulation:

$$f_0 = \frac{1}{2\pi} \sqrt{\frac{k}{m^*}}$$

$$\Delta f = -\frac{f_0}{2k} \frac{\partial F_{tot}}{\partial z}$$

⇒ measured topography = surface of constant $\frac{\partial F}{\partial z}$

$$F_{tot} = F_{chem} + F_{mag} + F_{el} + F_{vdW}$$

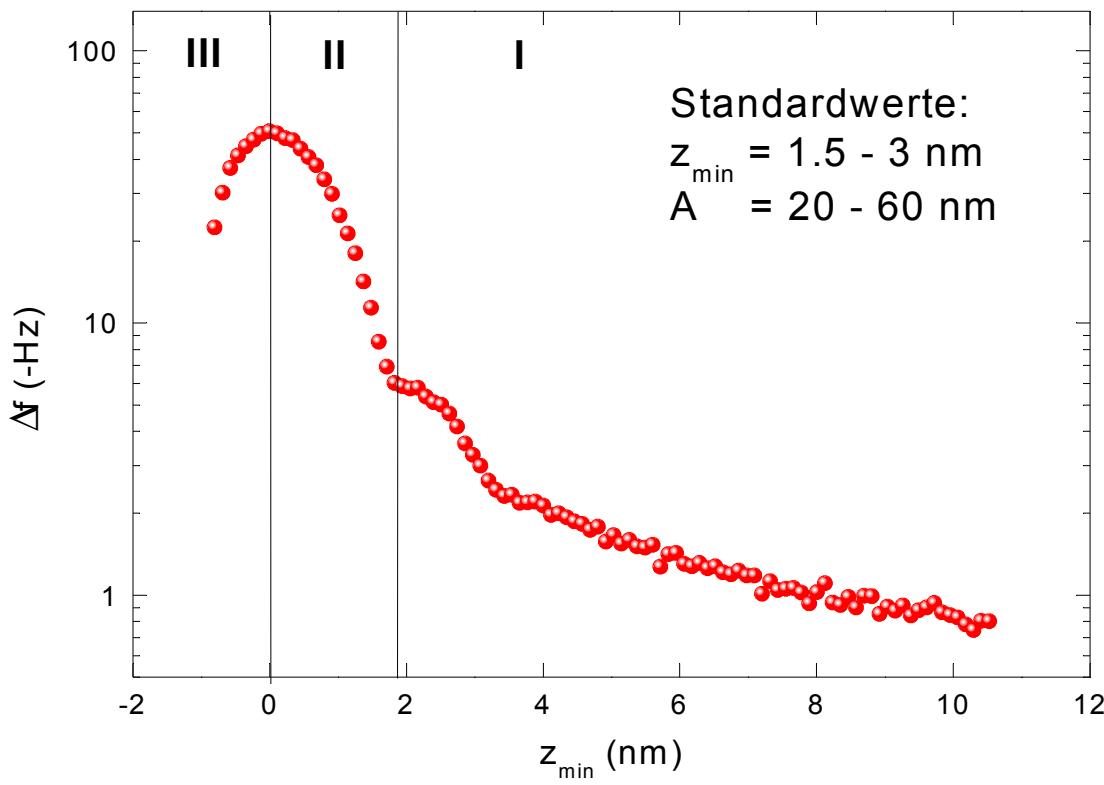
bonding between
tip and sample
atoms
(only for $d < 5 \text{ \AA}$)

only for
magnetically
sensitive tips

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V^2$$

$$F_{vdW} = -\frac{HR}{6d^2}$$

Dynamic Mode, non-contact

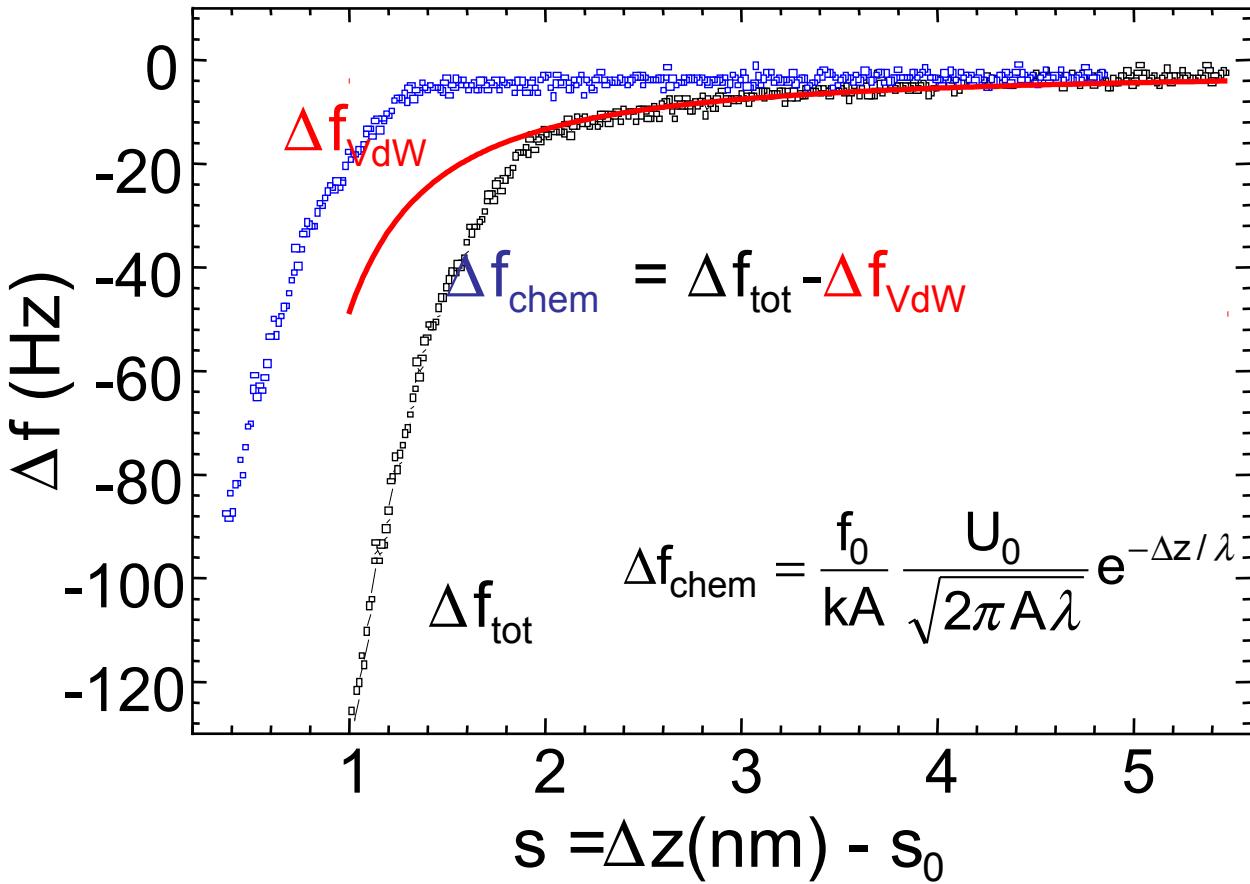


region I:
attractive forces
non-contact mode

region II:
attractive forces
atomic resolution

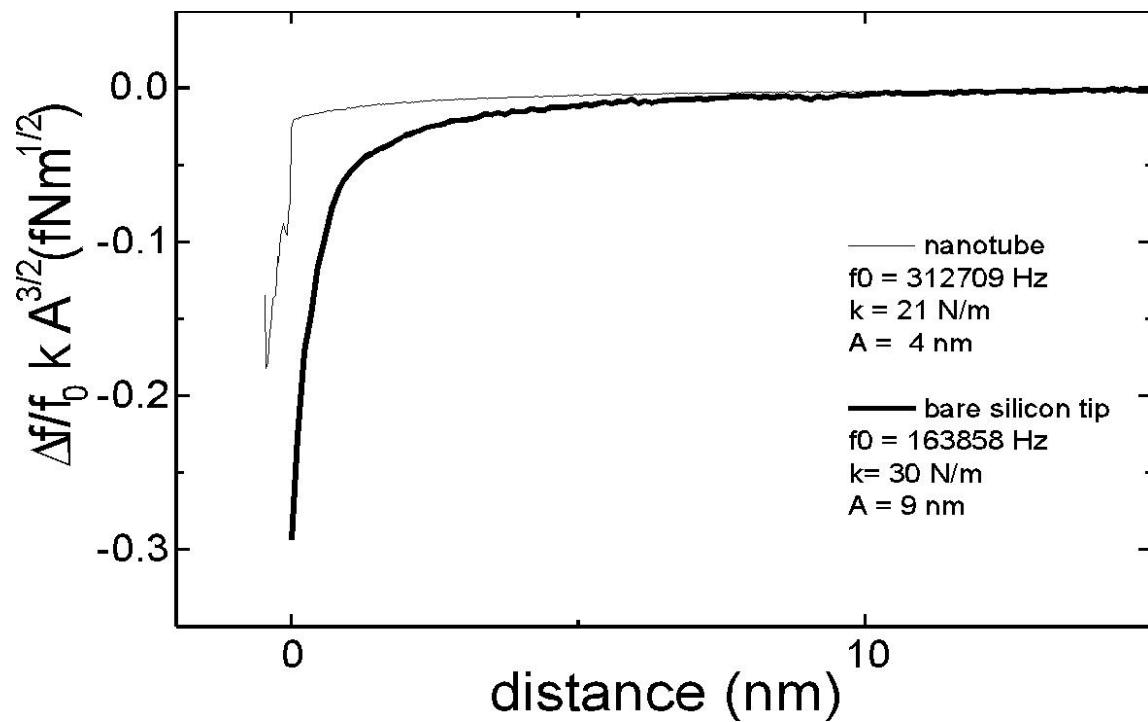
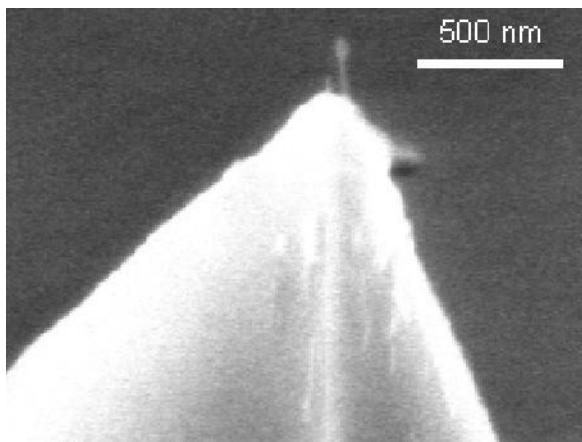
region III:
repulsive forces
tapping mode

Short range interaction



$$\begin{aligned}\lambda &= 0.35 \text{ nm} \\ U_0 &= -4.7 \text{ eV} \\ s_0 &= 0.45 \text{ nm}\end{aligned}$$

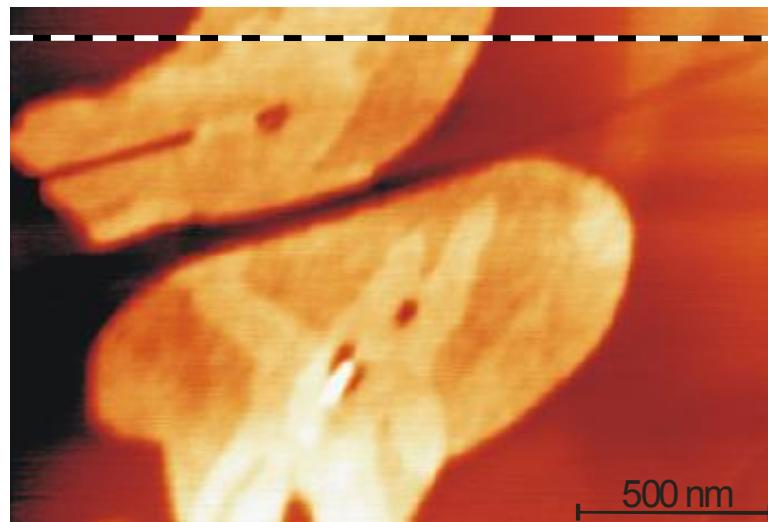
Carbon nanotubes as probing tips for nc-AFM



⇒ Long-range forces are reduced

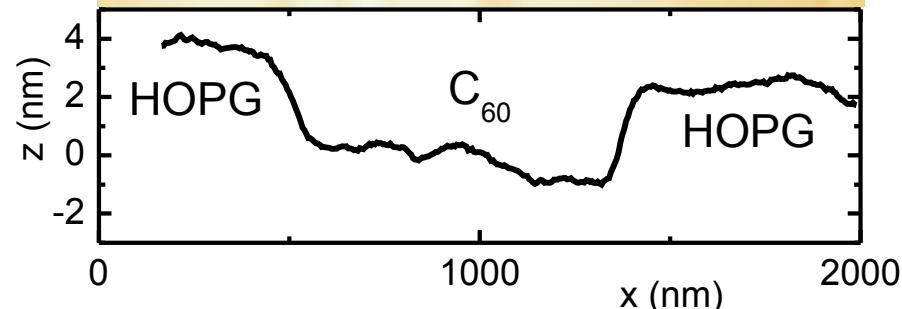
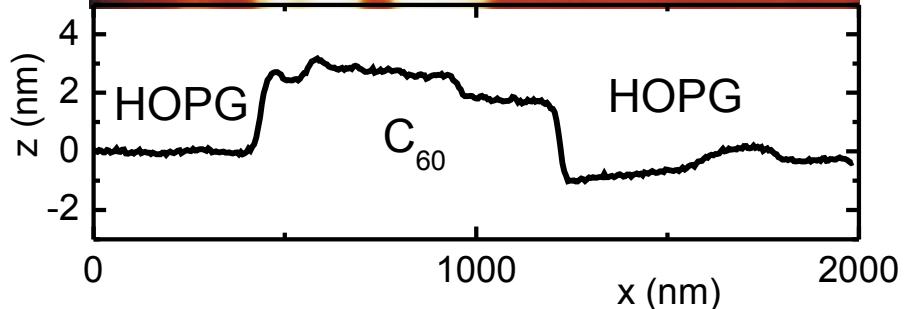
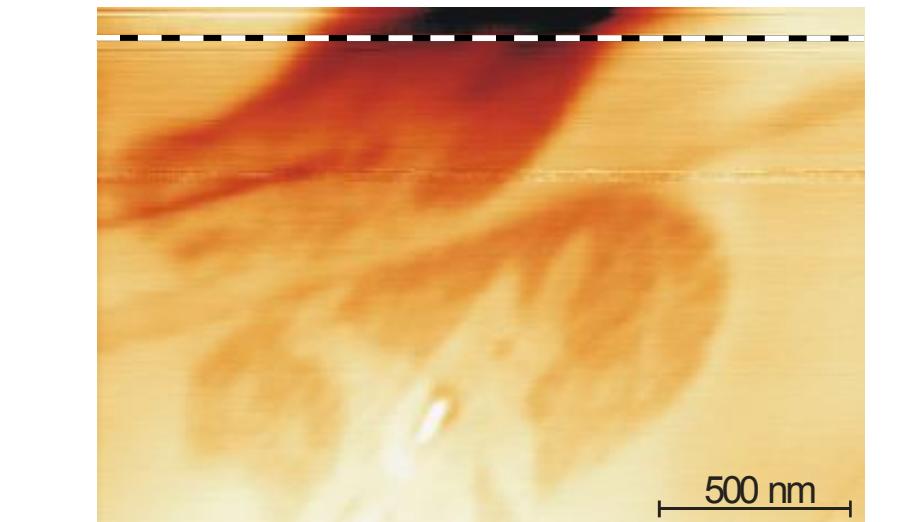
inhomogeneous sample: HOPG + ½ monolayer C₆₀

$V_{\text{bias}} = 0 \text{ V}$



Topography

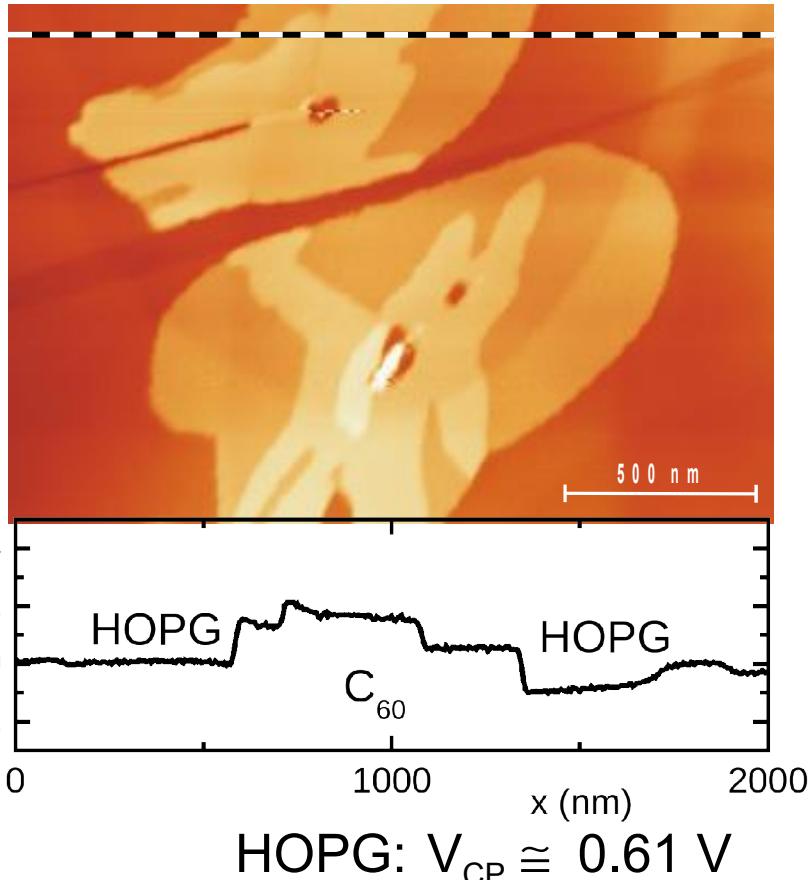
$V_{\text{bias}} = 1.34 \text{ V}$



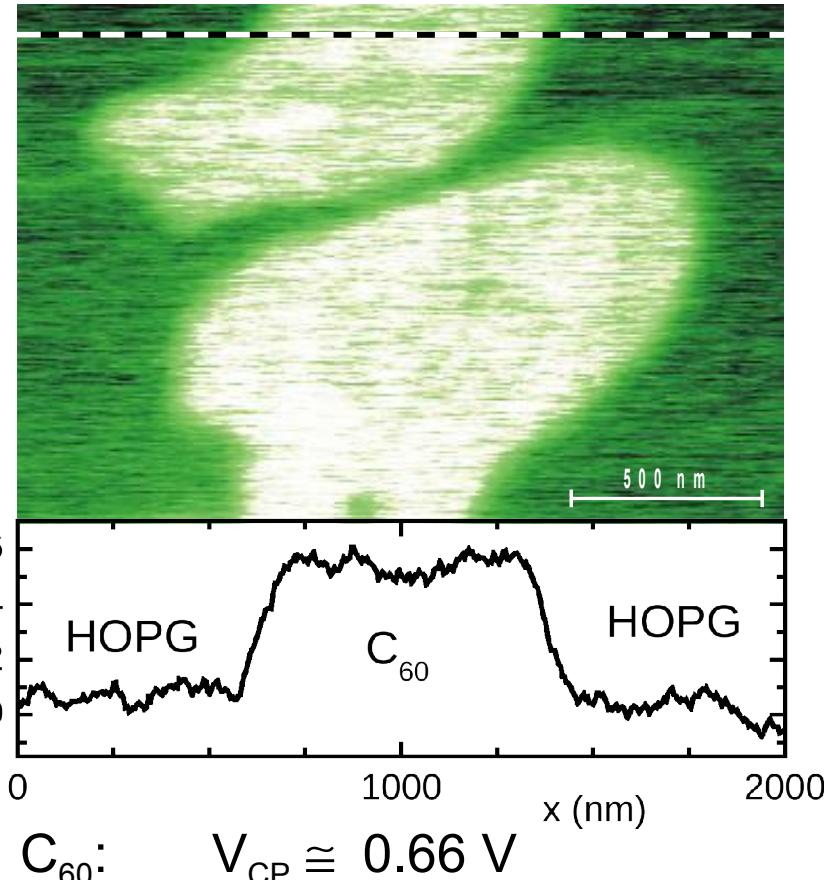
→ contrast reversal: HOPG \leftrightarrow C₆₀

inhomogeneous sample: HOPG + ½ monolayer C₆₀

topography



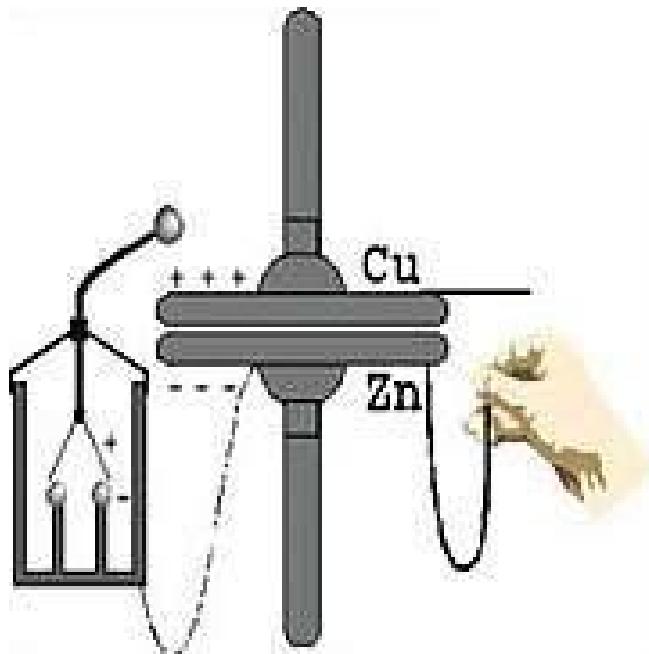
contact potential



⇒ NC-AFM: residual electrostatic force for fixed V_{bias}

Makroskopische Kelvin-Sonde

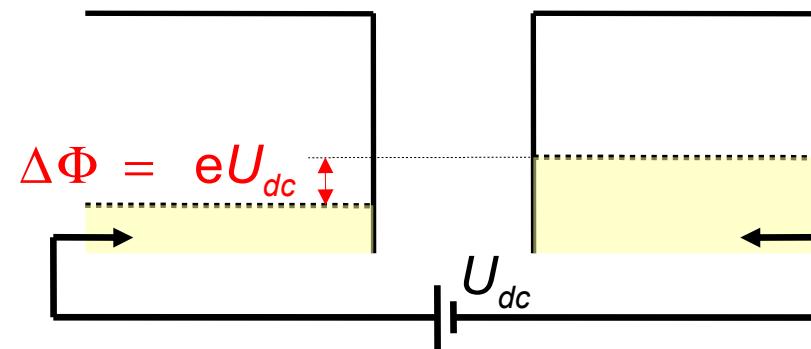
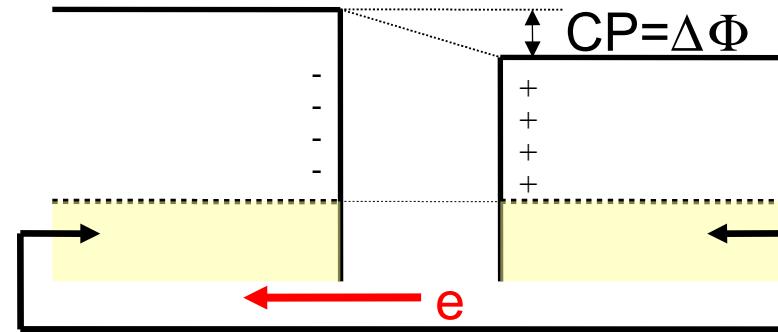
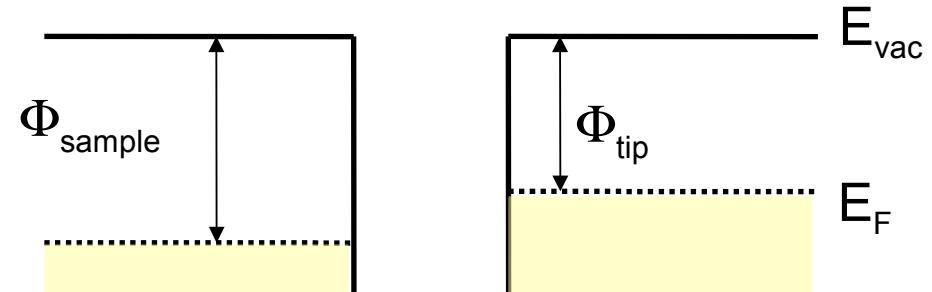
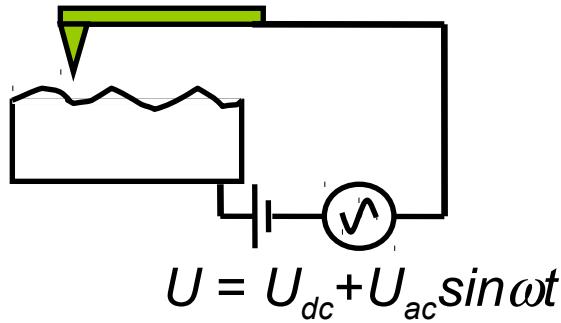
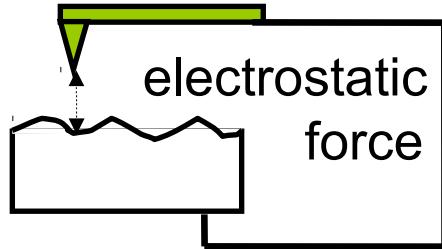
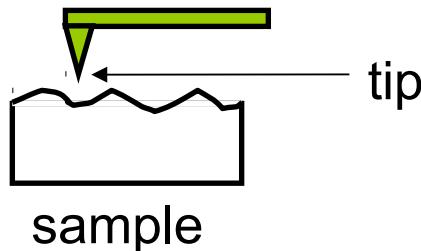
Lord Kelvin 1861



Verschiebestrom

$$I(t) = (U_{dc} - U_{CPD}) f \Delta C \cos \omega t.$$

Kelvin Principle



Electrostatic Forces in nc-AFM

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V_{eff}^2 \quad \Rightarrow \quad F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} (V_{bias} - V_{CP})^2$$

$$V_{CP} = 1/e \cdot (\Phi_{tip} - \Phi_{sample})$$

contact potential

Φ - work function

apply bias:

$$V_{bias} = V_{dc} + V_{ac} \cdot \sin(\omega t)$$

Kelvin Probe Force Microscopy

$$F_{el} = -\frac{1}{2} \frac{\partial C}{\partial z} V_{eff}^2 = F_{dc} + F_{\omega} + F_{2\omega}$$

$$F_{dc} = -\frac{\partial C}{\partial z} \left[\frac{1}{2} (V_{dc} - V_{CP})^2 + \frac{V_{ac}^2}{4} \right]$$

$$F_{\omega} = -\frac{\partial C}{\partial z} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$

$$F_{2\omega} = \frac{\partial C}{\partial z} \frac{V_{ac}^2}{4} \cos(2\omega t)$$

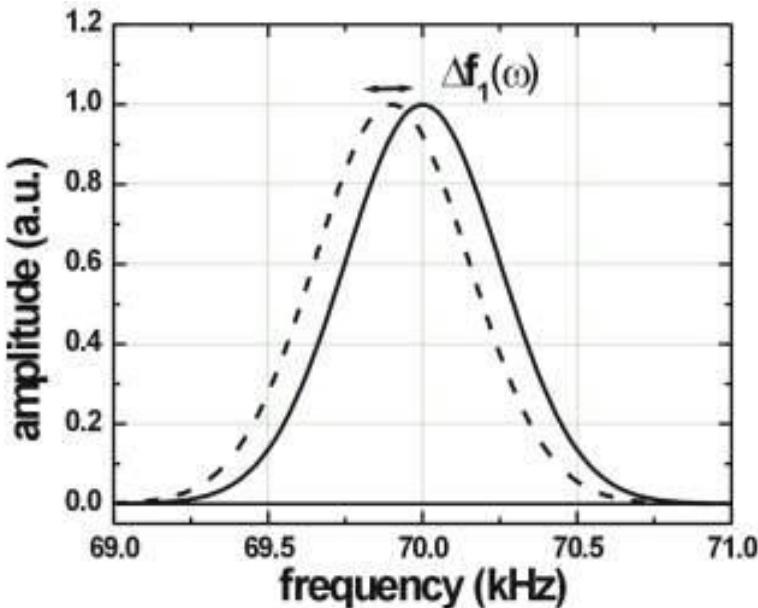
AM-KPFM
Amplitude Modulation

FM-KPFM
Frequency Modulation

FM – KPFM

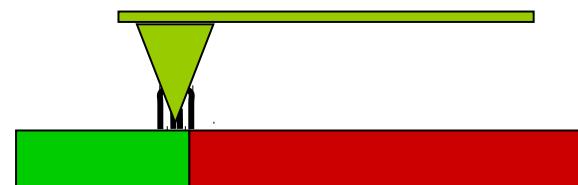
Frequency Modulation Detection

$$\Delta f(\omega) \propto \frac{\partial F_{el}}{\partial z} \propto \frac{\partial^2 C}{\partial z^2} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$



- frequency ω of V_{ac} between 1-3 kHz
- detection of the oscillation of $A(\Delta f_1)$ with a lock-in
- limiting factor: bandwidth of the FM-demodulator / PLL

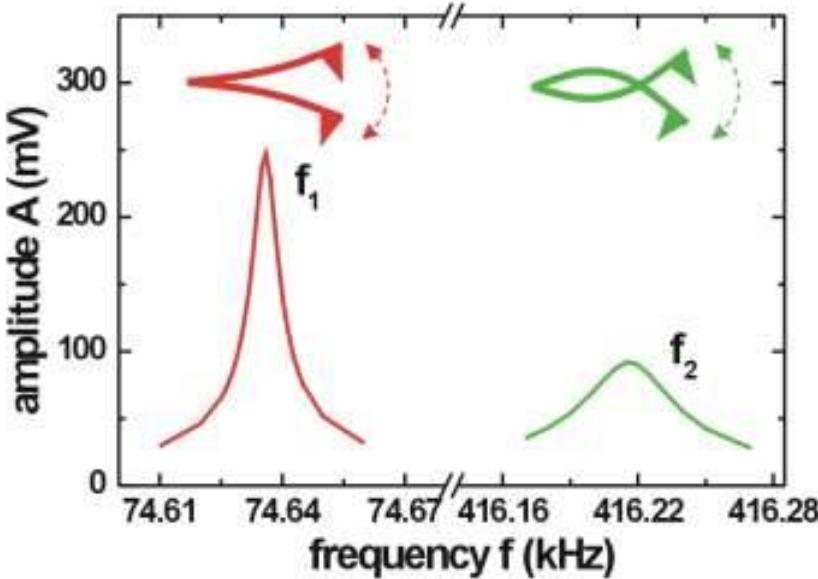
$$A(\Delta f_1) \propto \partial F_{el} / \partial z$$



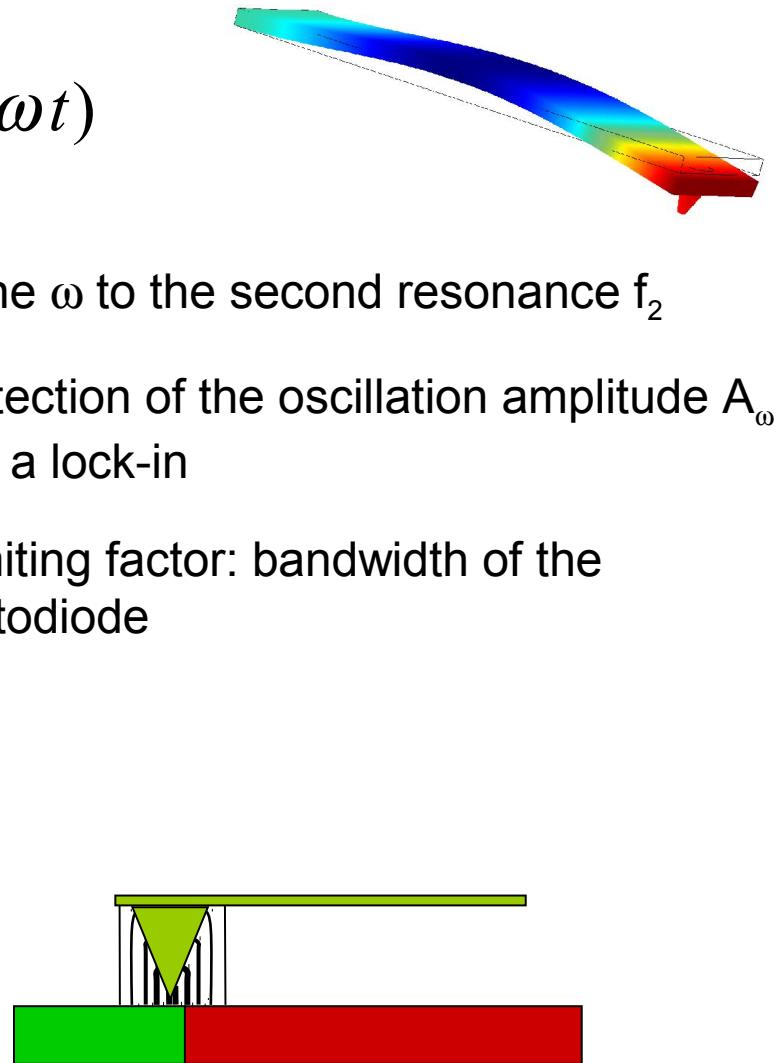
AM – KPFM

Amplitude Modulation Detection

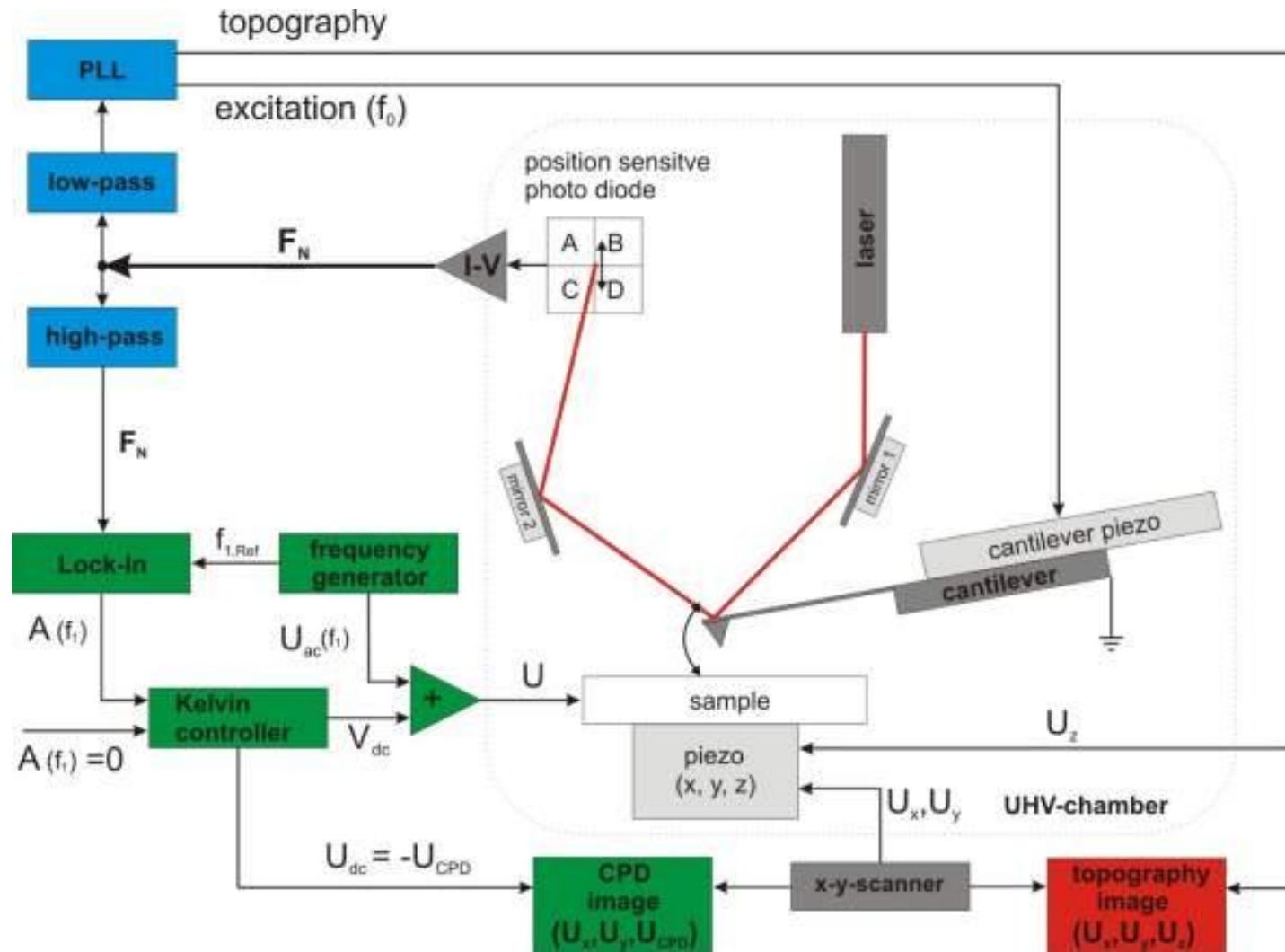
$$F_\omega = -\frac{\partial C}{\partial z} (V_{dc} - V_{CP}) V_{ac} \sin(\omega t)$$



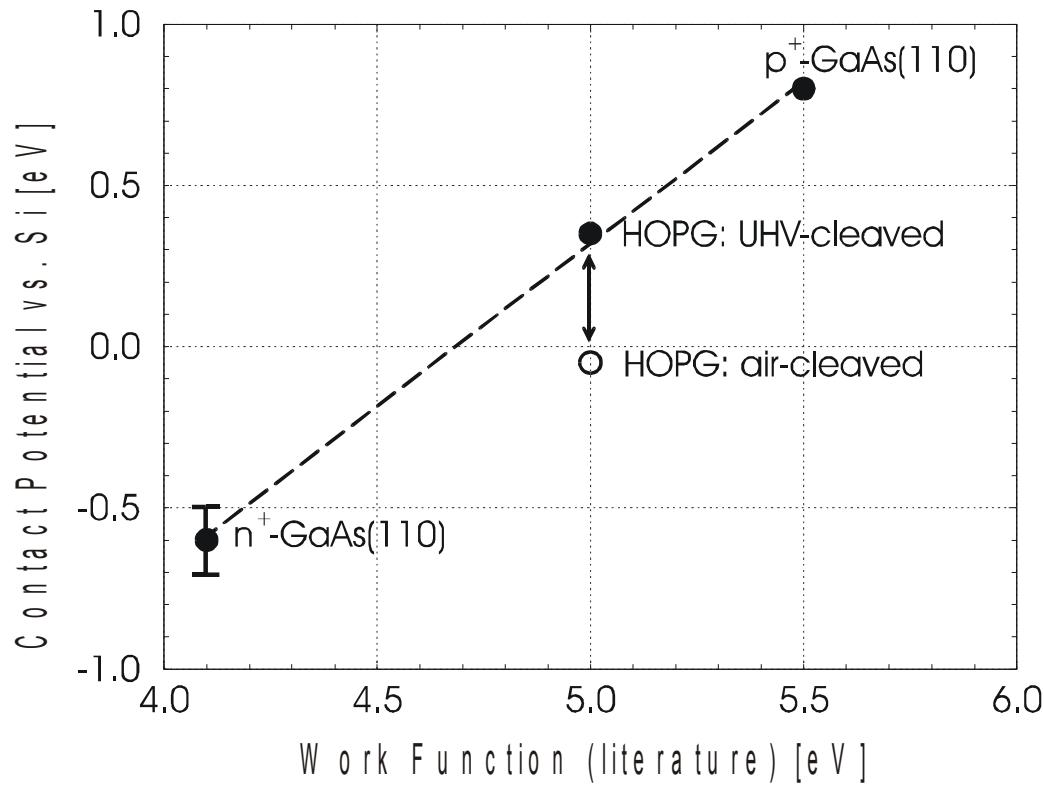
$$A_\omega \propto F_\omega$$



Experimental Setup nc-AFM & AM-KPFM



KPFM calibration and absolute work function

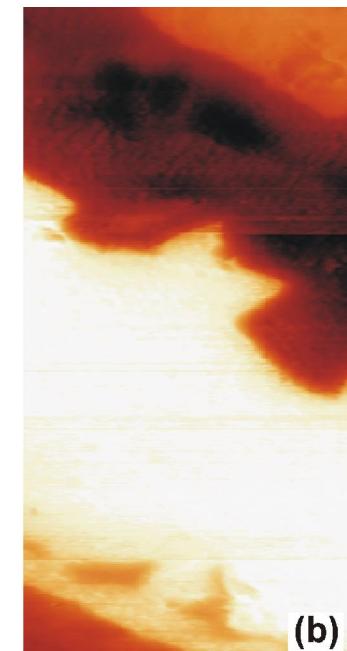
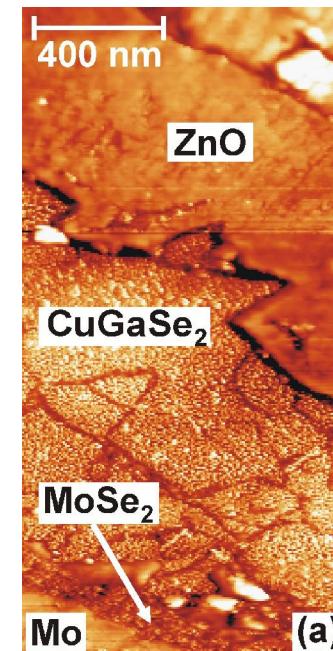
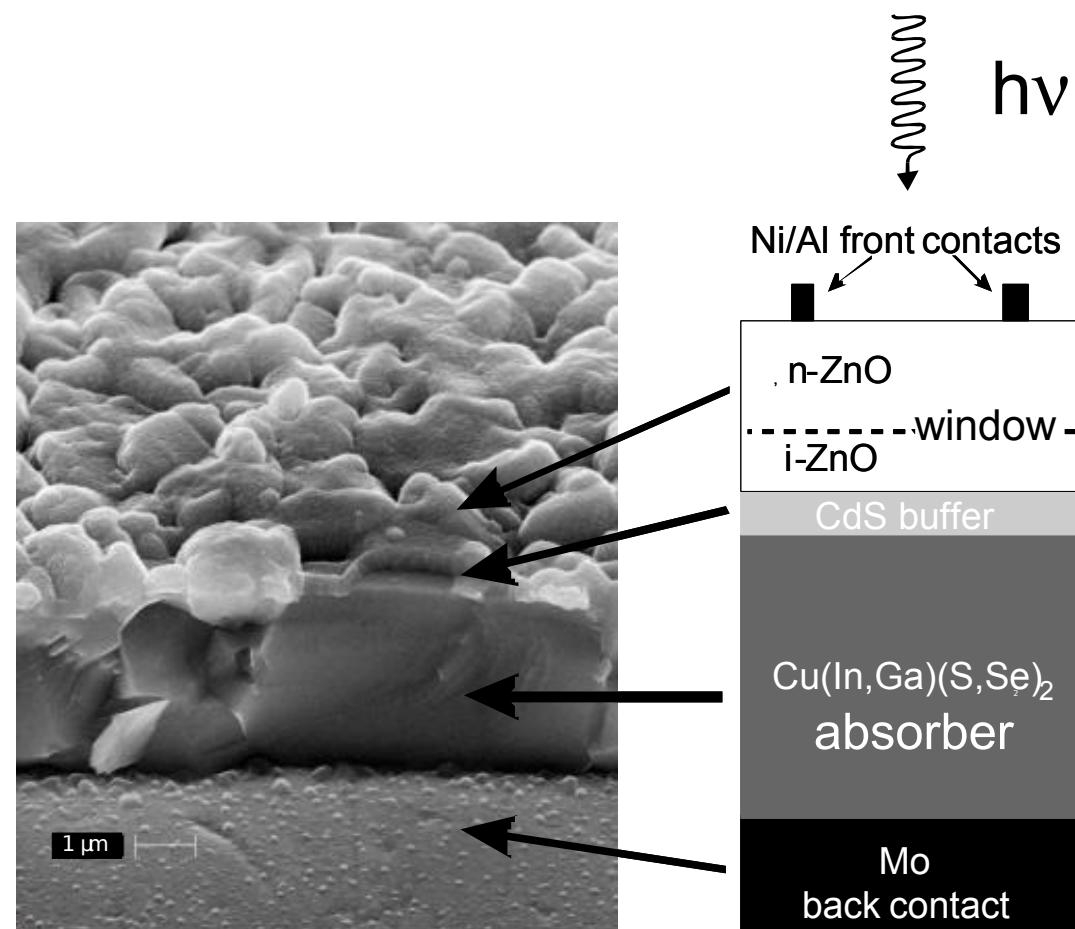


$$\Phi\text{-Si-Cantilever} = 4.70 (\pm 0.1) \text{ eV}$$

$$U_{ac} = 100 \text{ mV}$$

→absolute and quantitative work function determination

Kelvin Probe Force Microscopy (KPFM) chalcopyrite thin film solar cells

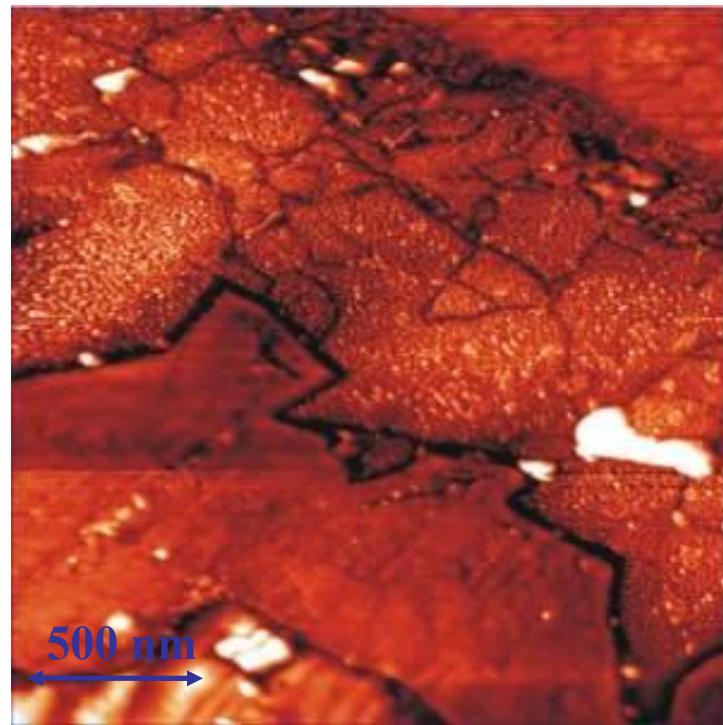


work function
($\Phi = 3.92\text{--}4.86$
eV)

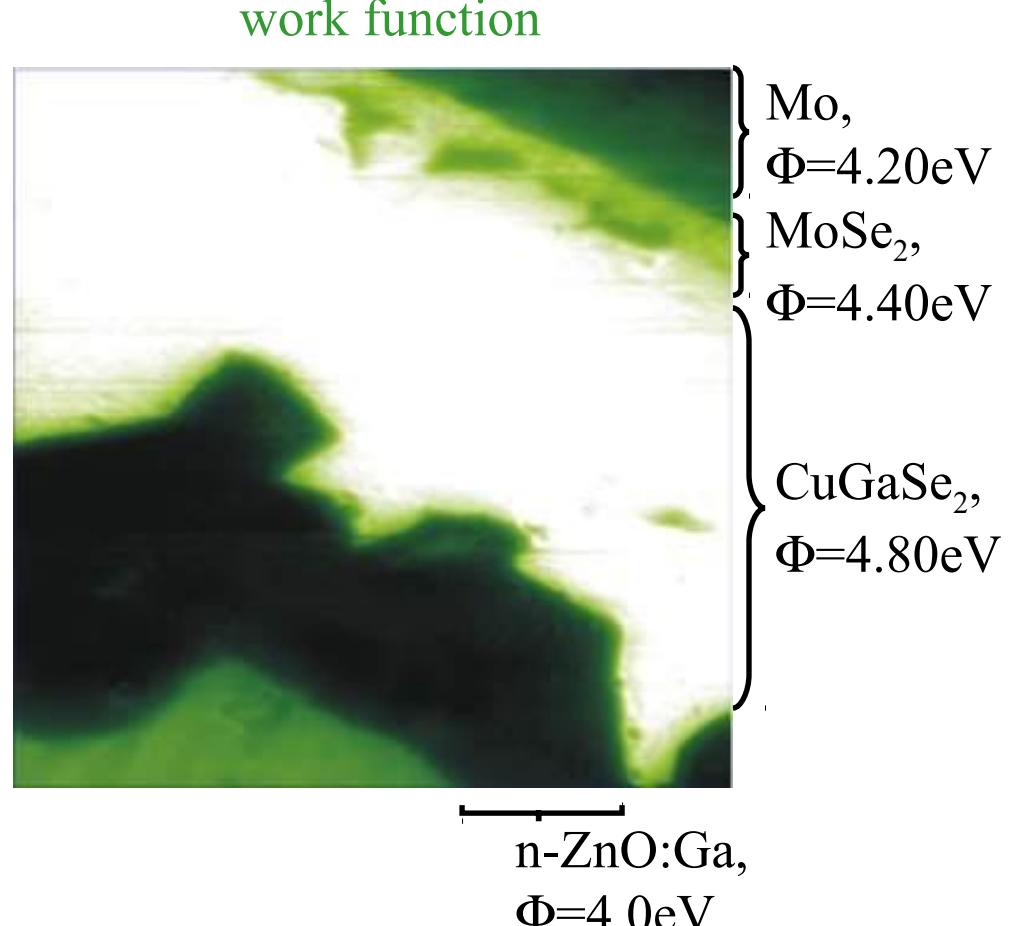
Polished Cross Section of a CuGaSe₂ Solar Cell

CuGaSe₂ solar cell device: $V_{oc} = 820$ mV, $\eta = 4.6\%$

polished and Ar-ion sputtered cross section
topography

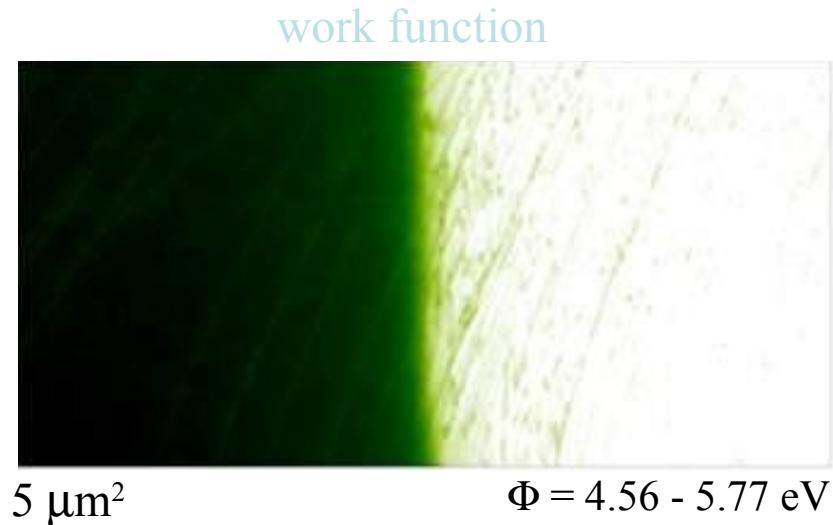


$\Delta z = 65$ nm



AM-KPFM measurement on GaP pn-junction

n-type GaP wafer with p-type GaP layer, $\sim 10^{18} \text{ cm}^{-3}$, UHV cleavage along (110) surface

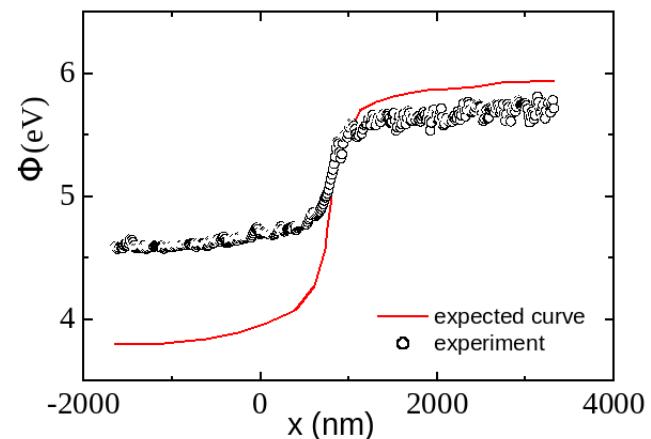


most III-V semiconductors:

no surface states on the (110) surface

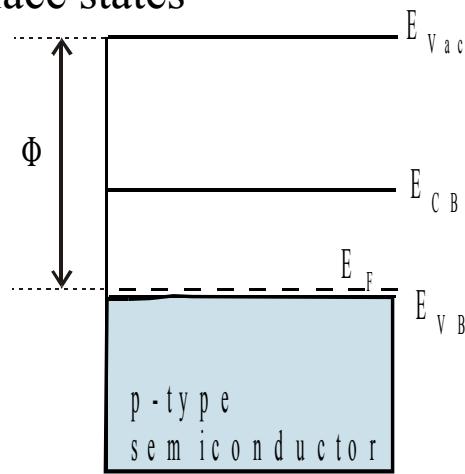
☒ GaP does show surface states

⇒ discrepancy of Φ_{exp} to Φ_{theo} due to surface states!

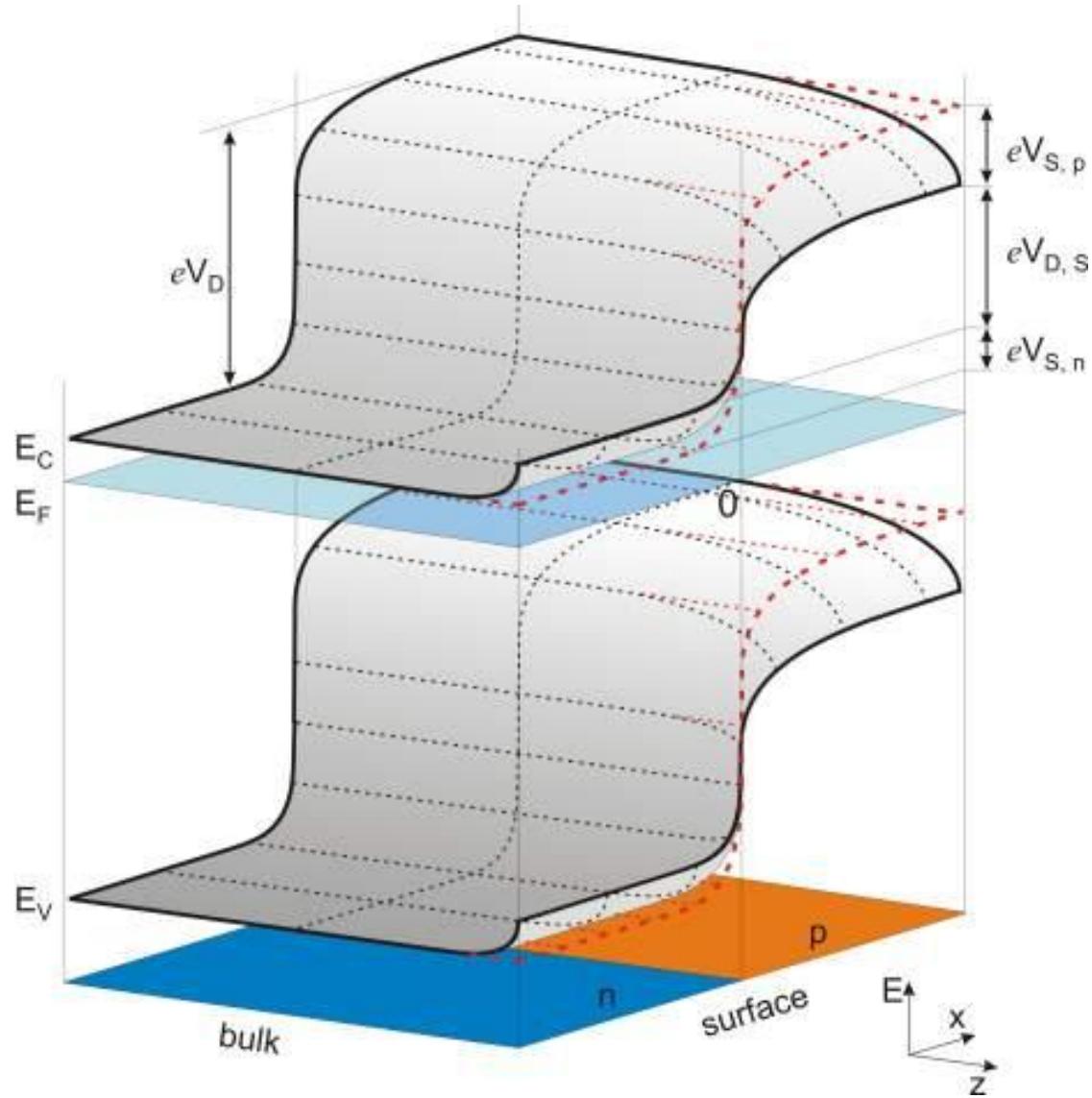
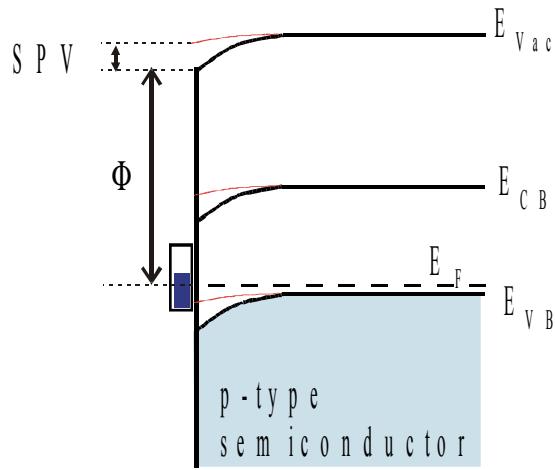


Surface Effects

no surface states



surface states



Surface Photovoltage GaP pn-Interface

n- GaP



$\Delta z = 4.5 \text{ nm}$

p- GaP

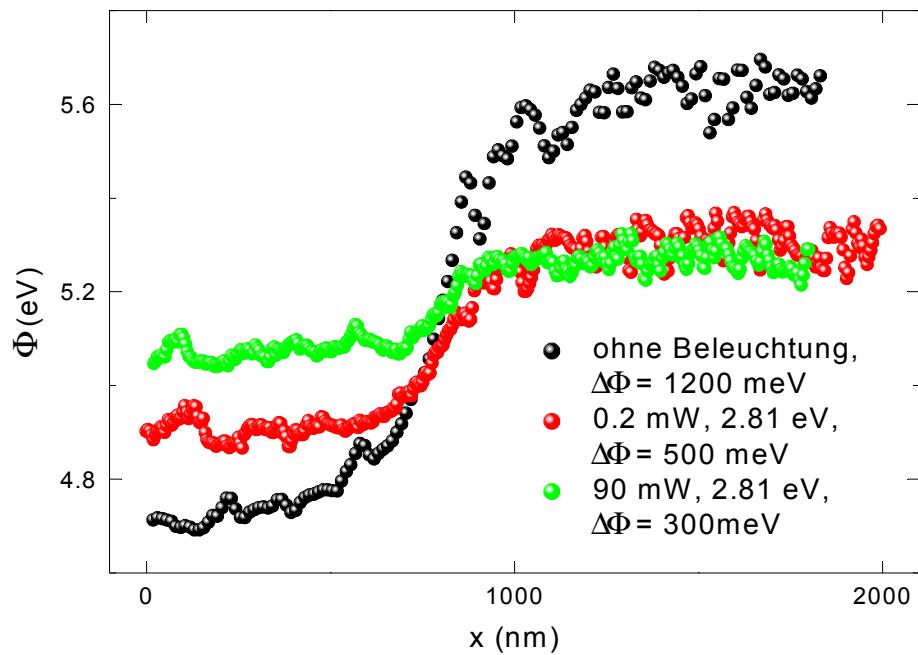
(5000x2500 nm)



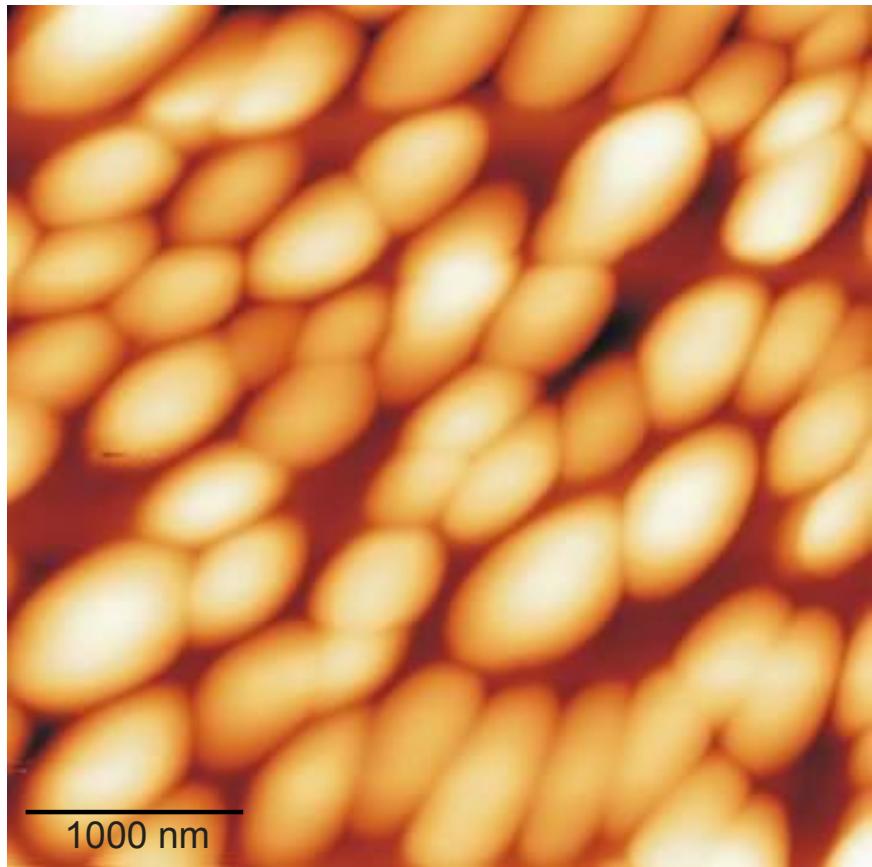
$\Phi = (4.56 - 5.77)\text{eV}$

n- GaP

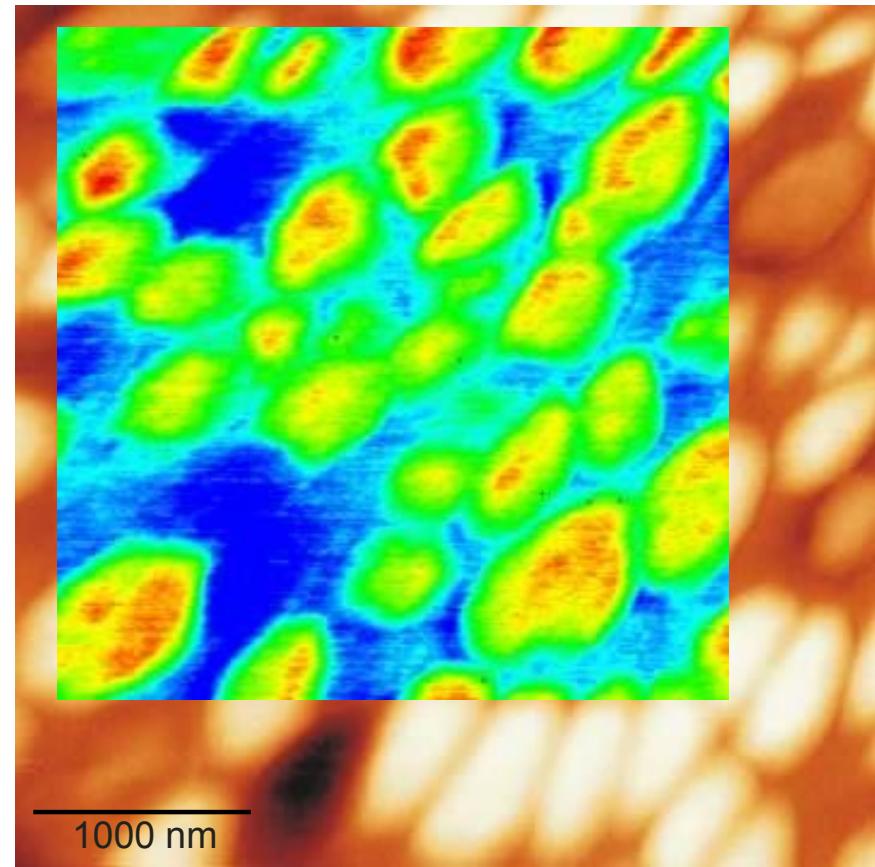
p- GaP



Surface Photovoltage MDMO-PPV/PCBM – 675nm



0 nm 105.6 nm

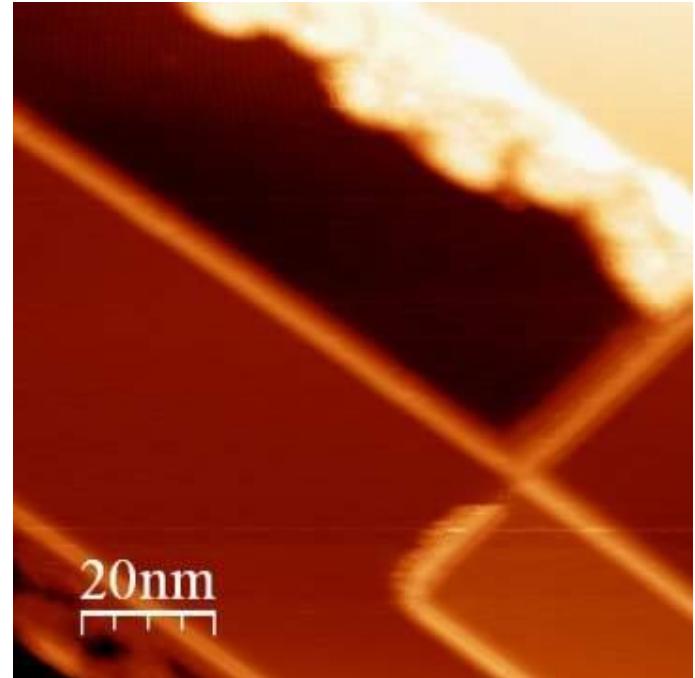
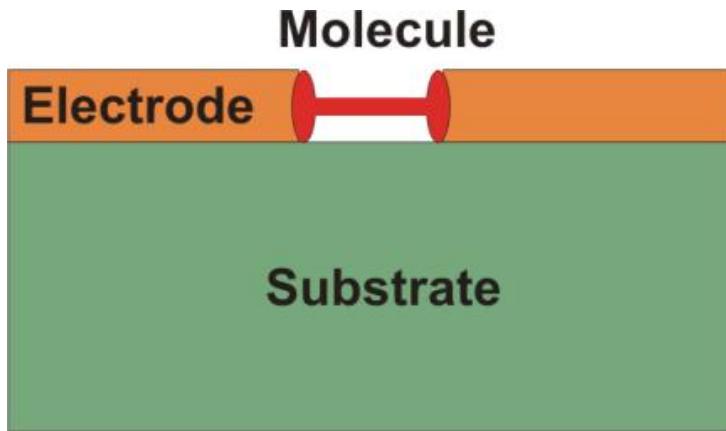


4.19 eV 4.62 eV

-50 mV 220mV

Motivation

Molecular electronics

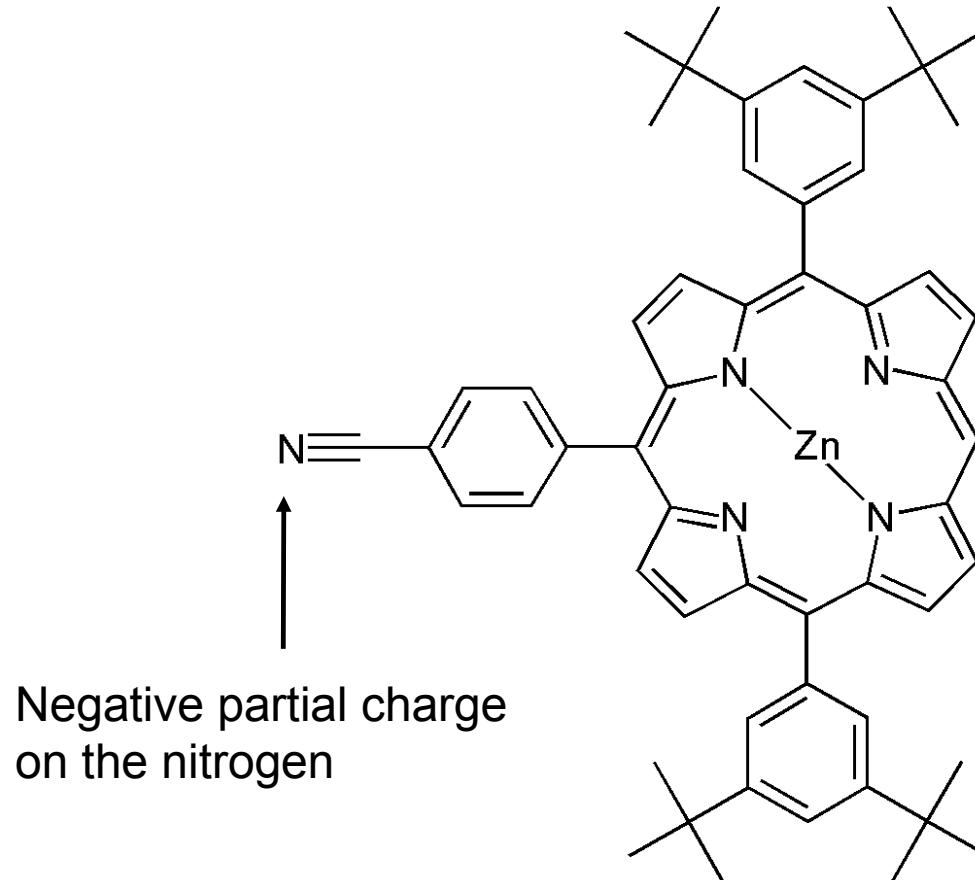
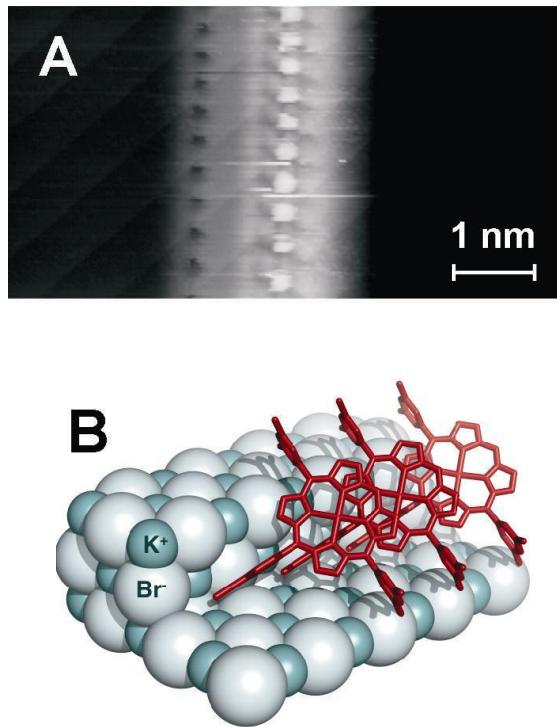


Molecules on Insulators:

- No STM possible – nc-AFM mandatory
- Low diffusion barrier but high intermolecular interaction
- Low temperatures – easier to “fix” molecules but not so easy to find applications

Asymmetric Cyano-Porphyrins

Natural light harvesting complexes

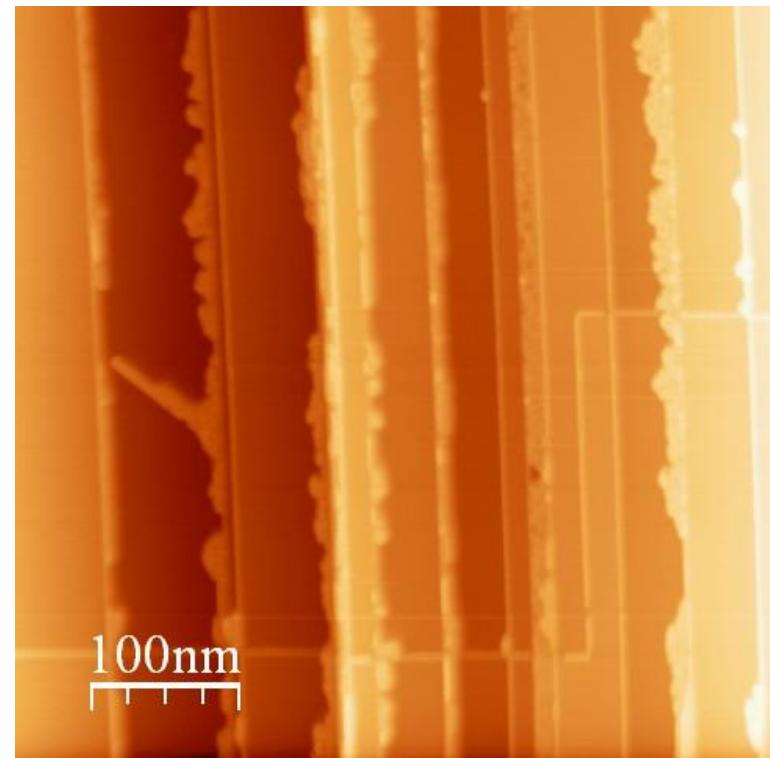


S. Meier et al., Small, 2008, 4, 1115

Wire Formation

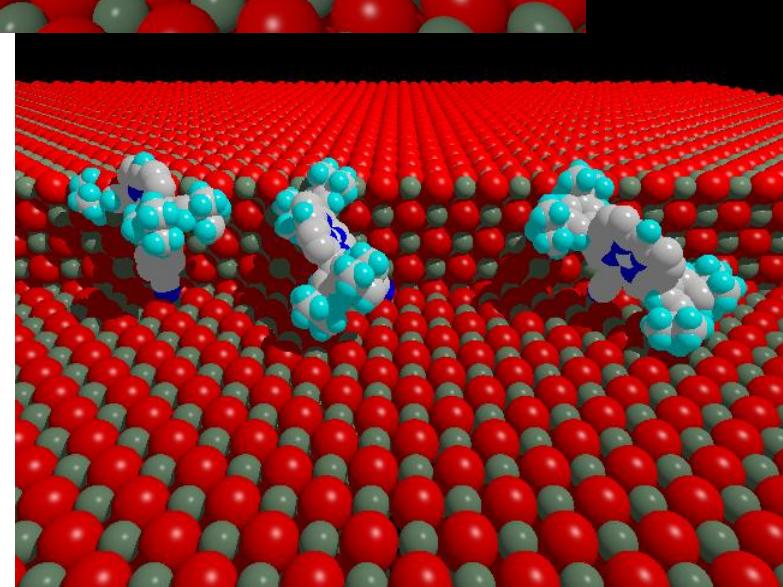
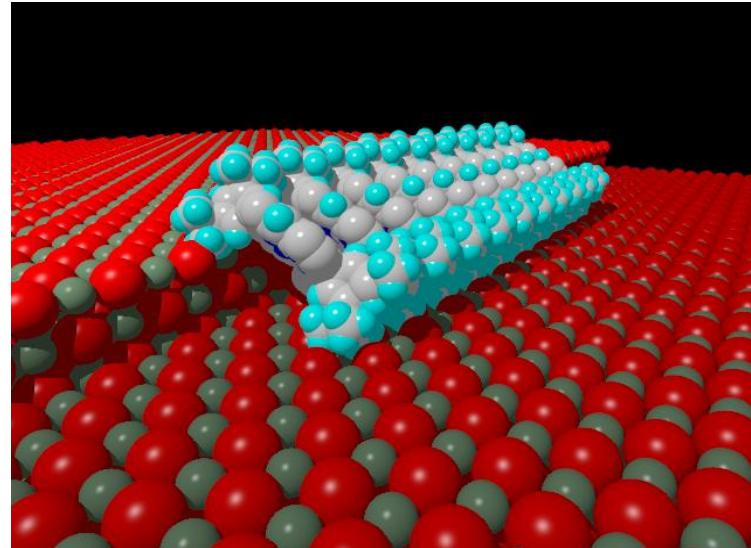
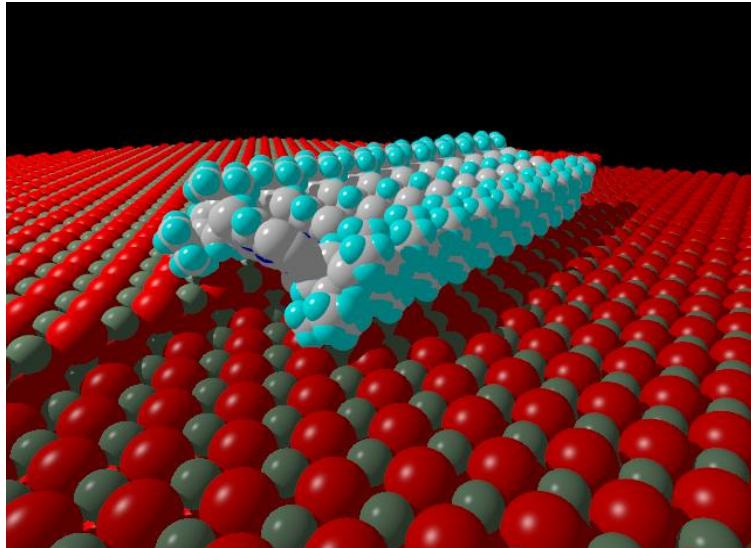
Decoration of step edges on KBr(100)

- In situ cleaved KBr with 0.5 ML of molecules
- Steps (< 1nm) are decorated with monowires
- Higher steps act as nucleation sites for structure growth across terraces



Wire Formation

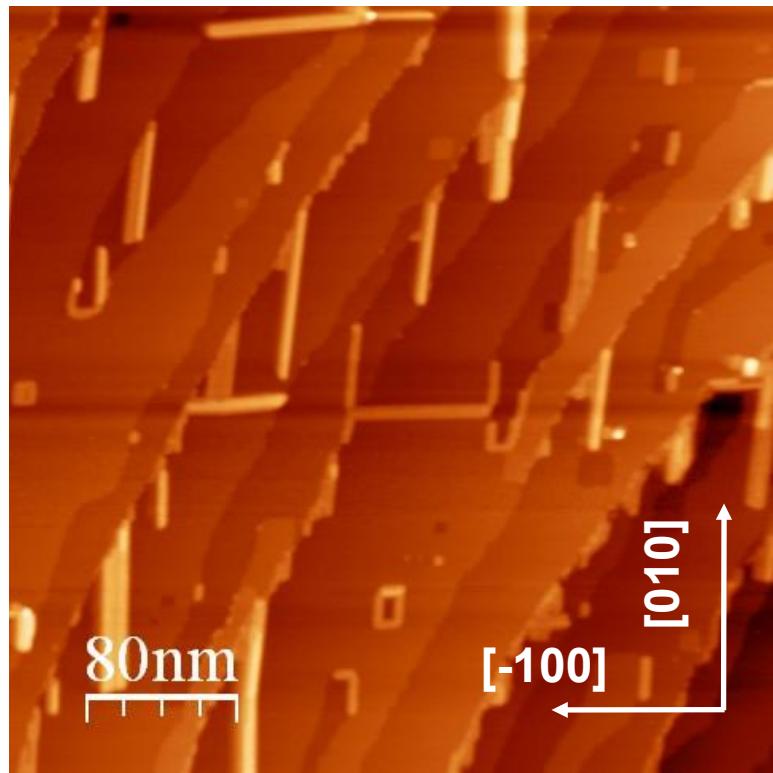
Structural model



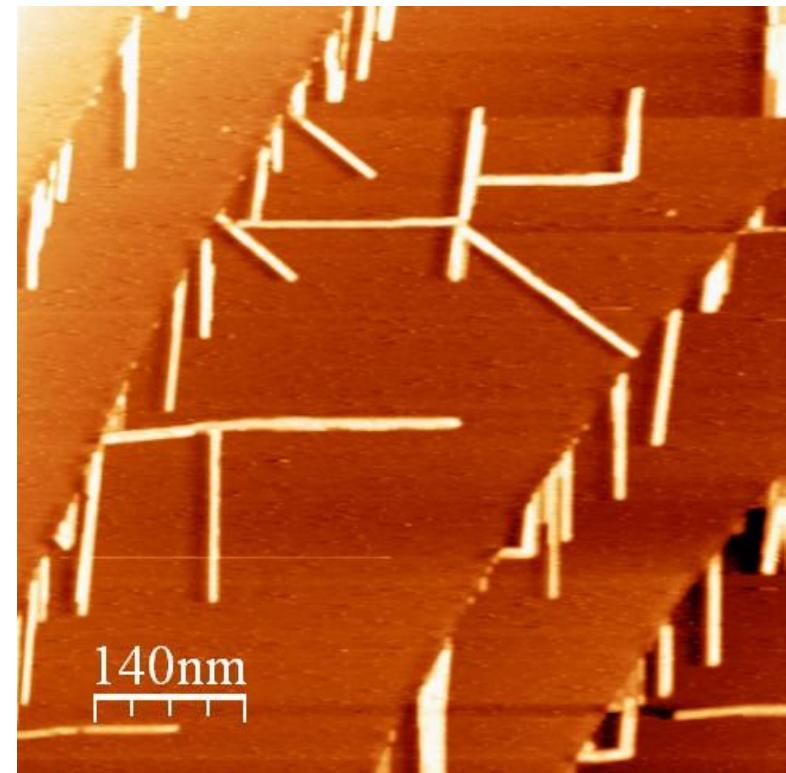
- The tilt angle of the molecules is determined by the side groups, the π - π stacking and the step height.
- Steps higher than 3 ML prevent a π - π stacking.

Molecular Assemblies

Molecular wires on NaCl



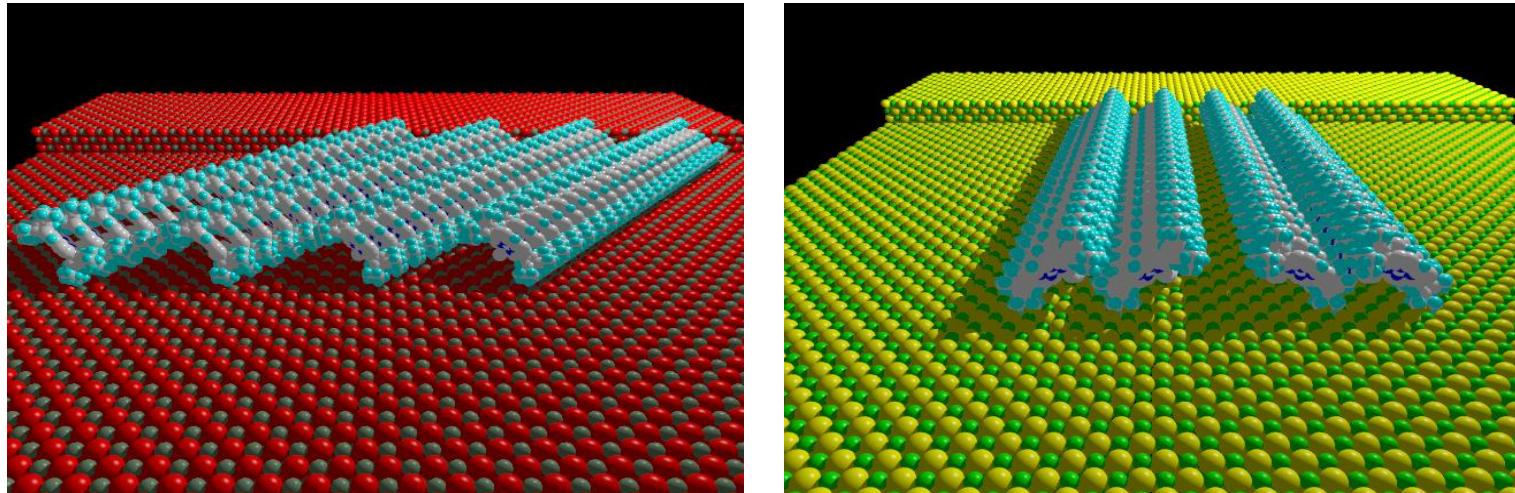
$f_0 \approx 170992\text{Hz}, \Delta f = -9.5\text{Hz}, Q = 15k, A = 40\text{nm}$



$f_0 \approx 170992\text{Hz}, \Delta f = -11\text{Hz}, Q = 15k, A = 40\text{nm}$

Molecular Assemblies

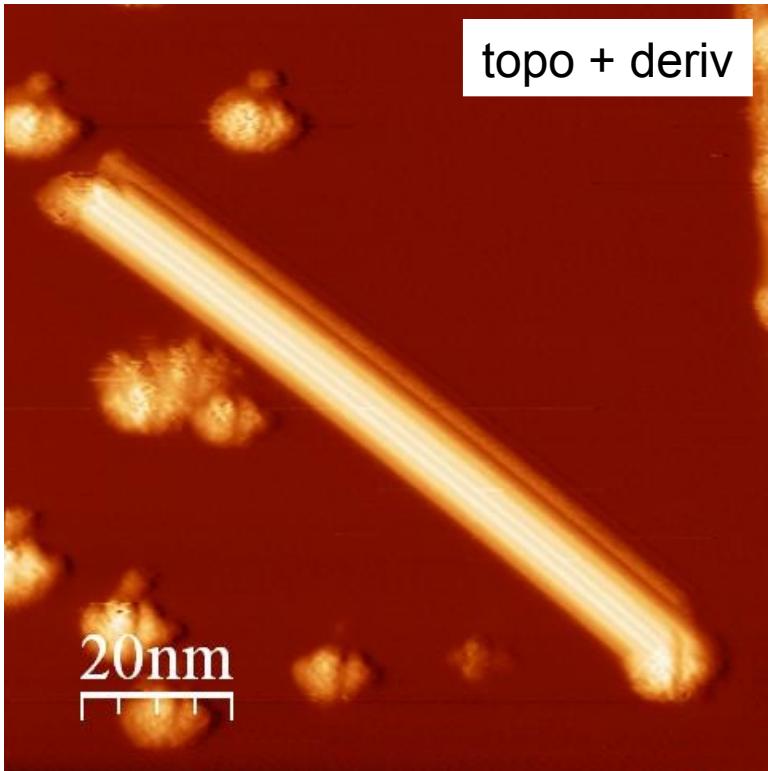
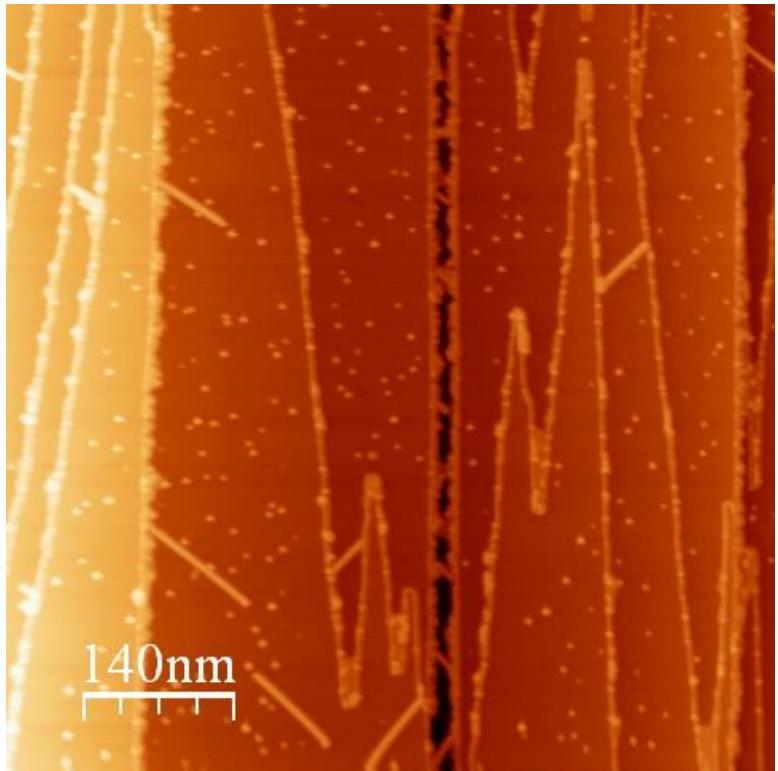
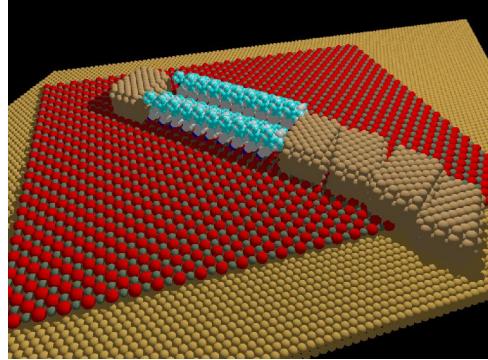
Structural model



- Intermolecular equilibrium separation $\sim 5.7 \text{ \AA}$
- Directed growth by the substrate

Contacting Molecular Assemblies

Au-Molecules-Au

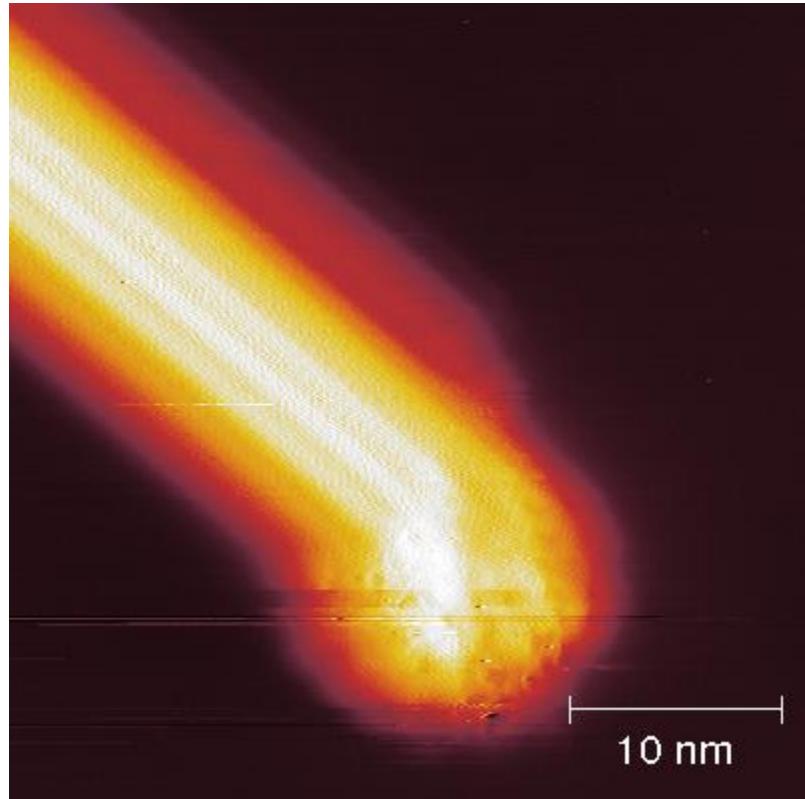


- Molecules arrange at steps and across terraces
- The growth is started/stopped at gold clusters.

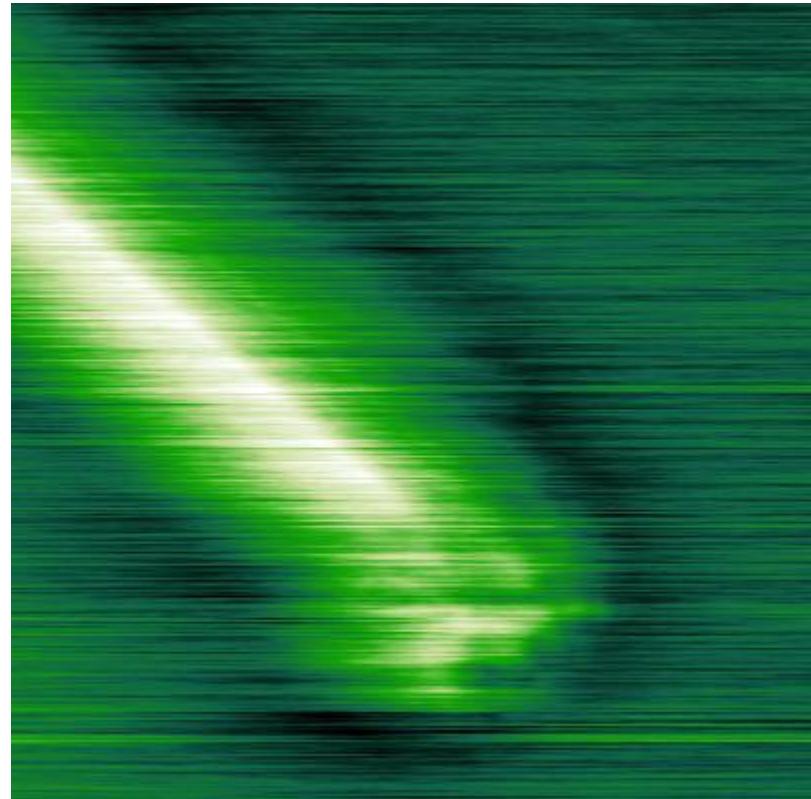
Contacting Molecular Assemblies

KPFM

Topography



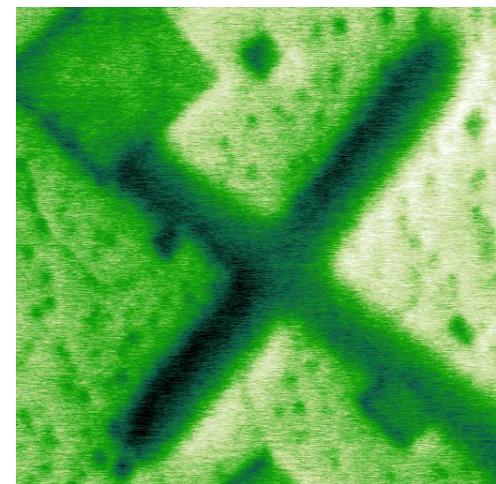
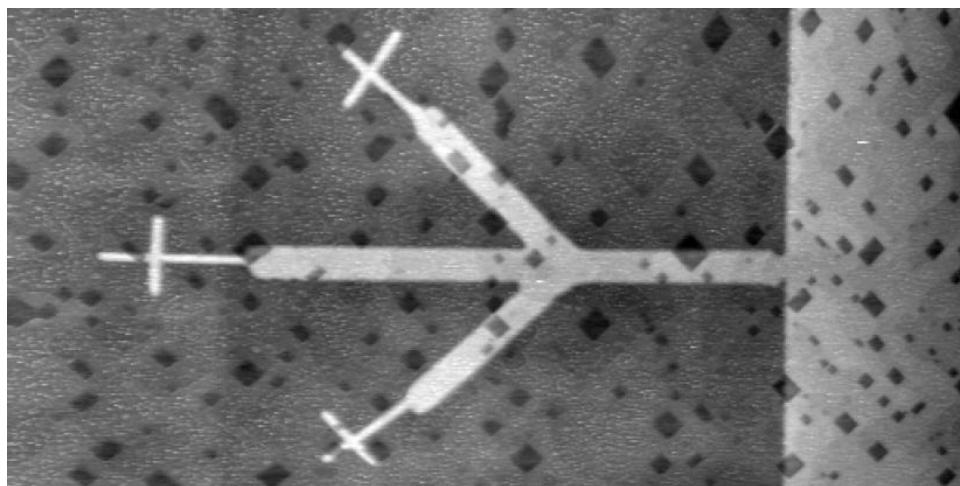
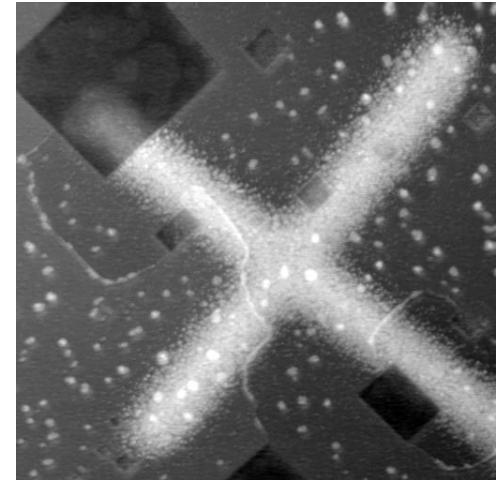
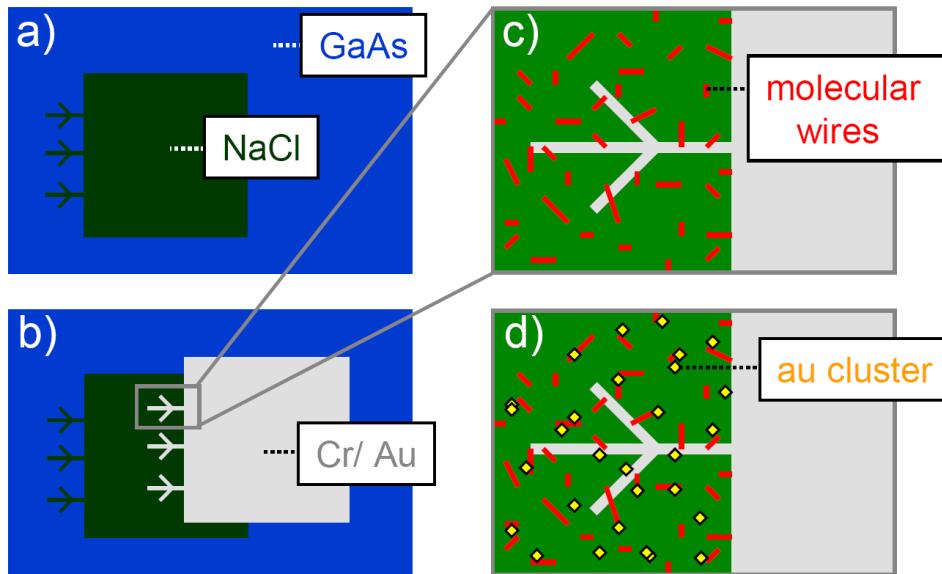
LCPD



$\text{LCPD} = -0.5 \dots 0.2 \text{eV}$

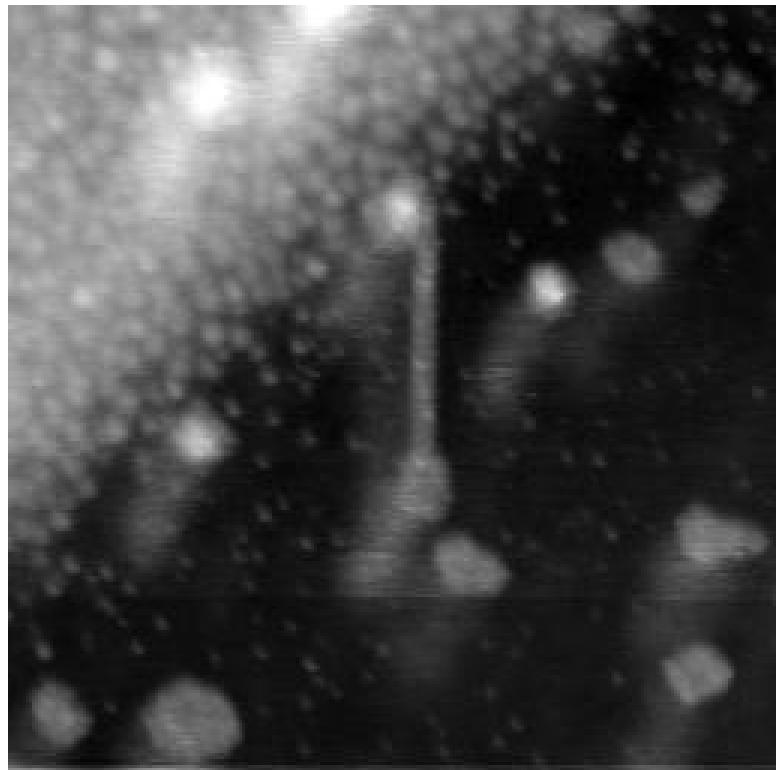
Contacting Molecular Assemblies

Nanostencil (IBM Rüschlikon)

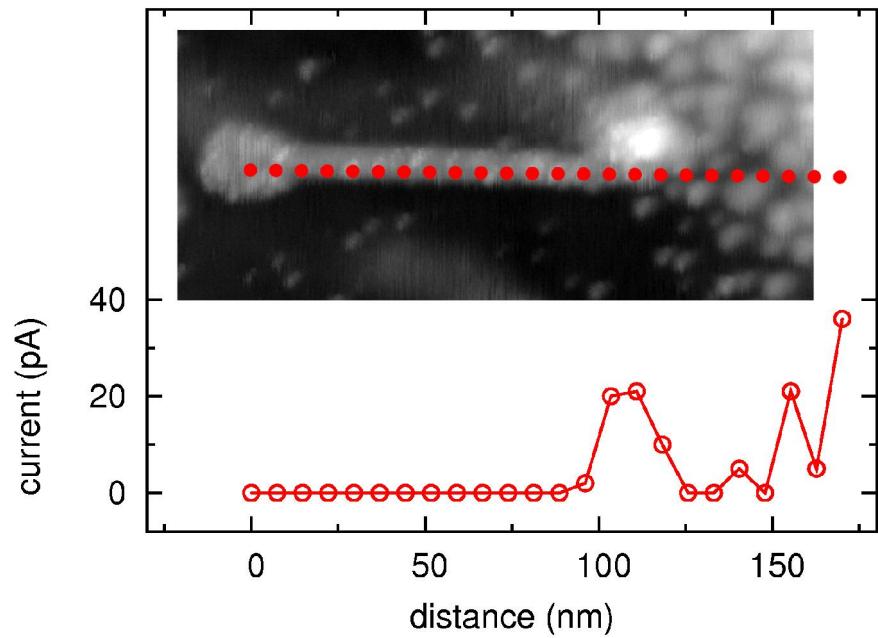


Contacting Molecular Assemblies

Nanostencil (IBM Rüschlikon)

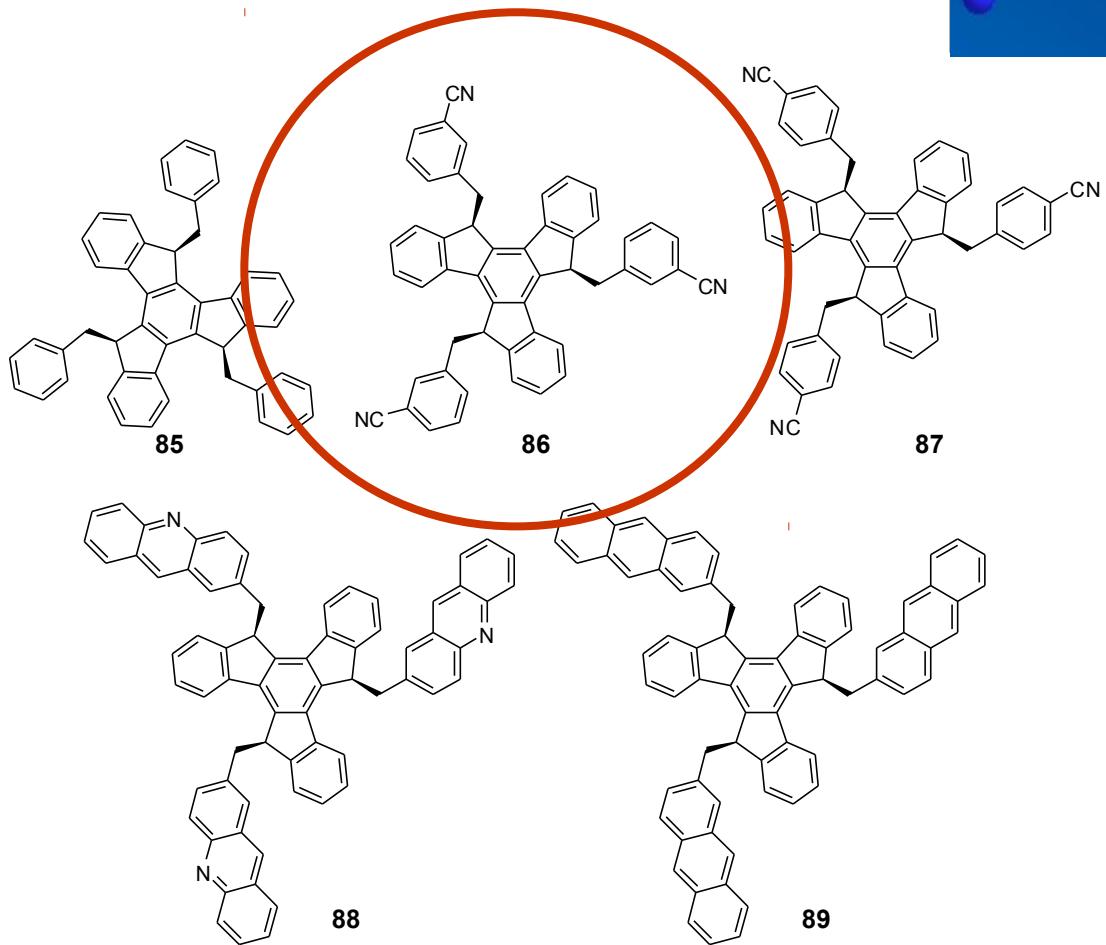
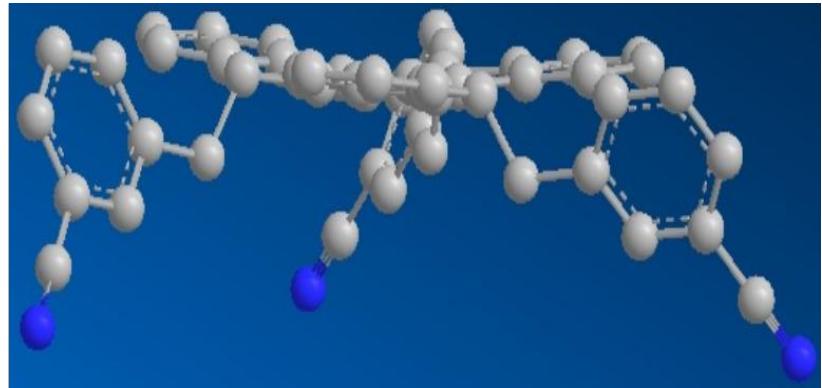


300x300nm²



Functionalized Truxenes

Structure



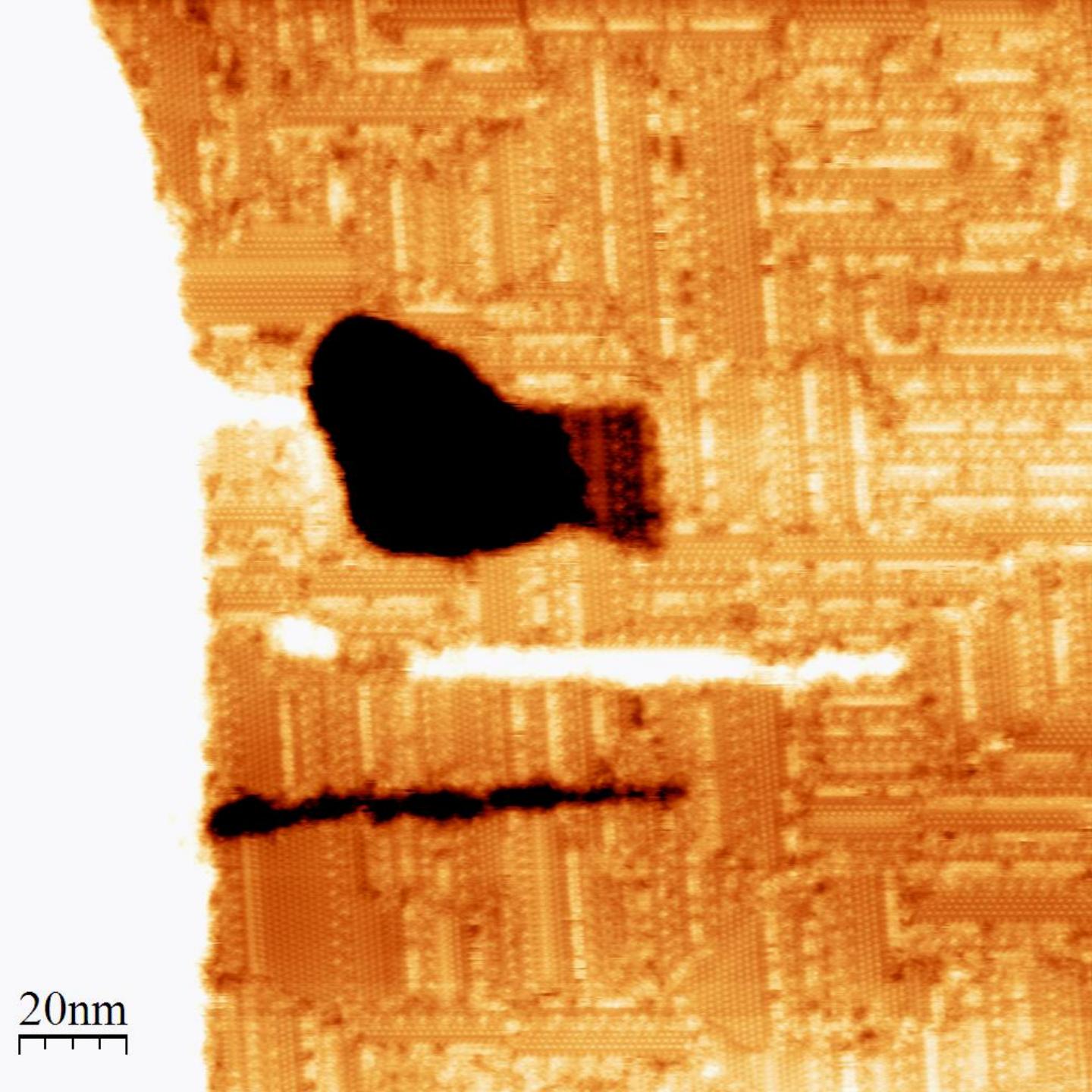
evaporation onto
the sample at RT

40nm

A scale bar consisting of a horizontal line with five vertical tick marks extending downwards from it, representing a length of 40 nanometers.

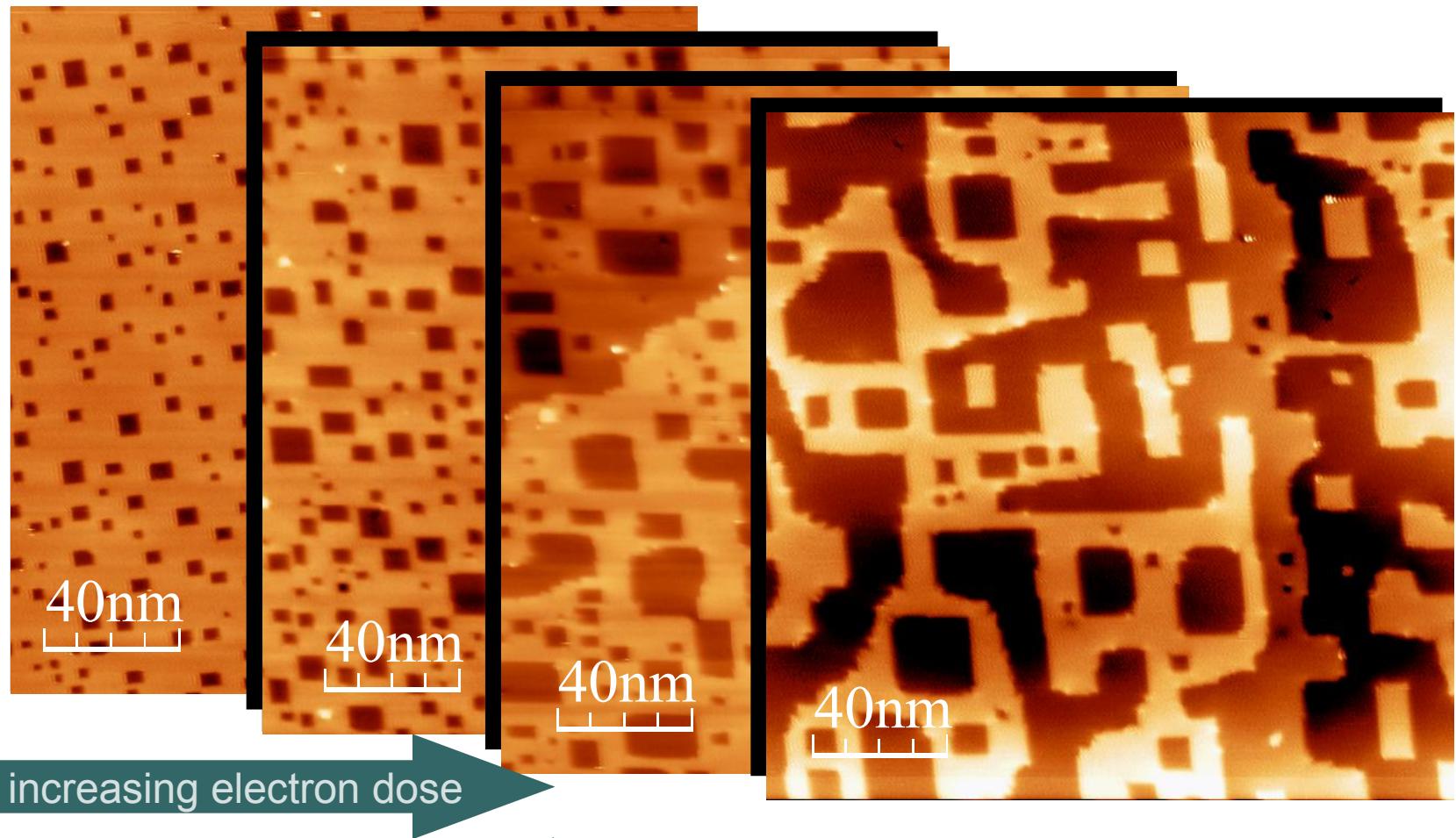
and the result of
post annealing at
155 C for 15 mins

20nm



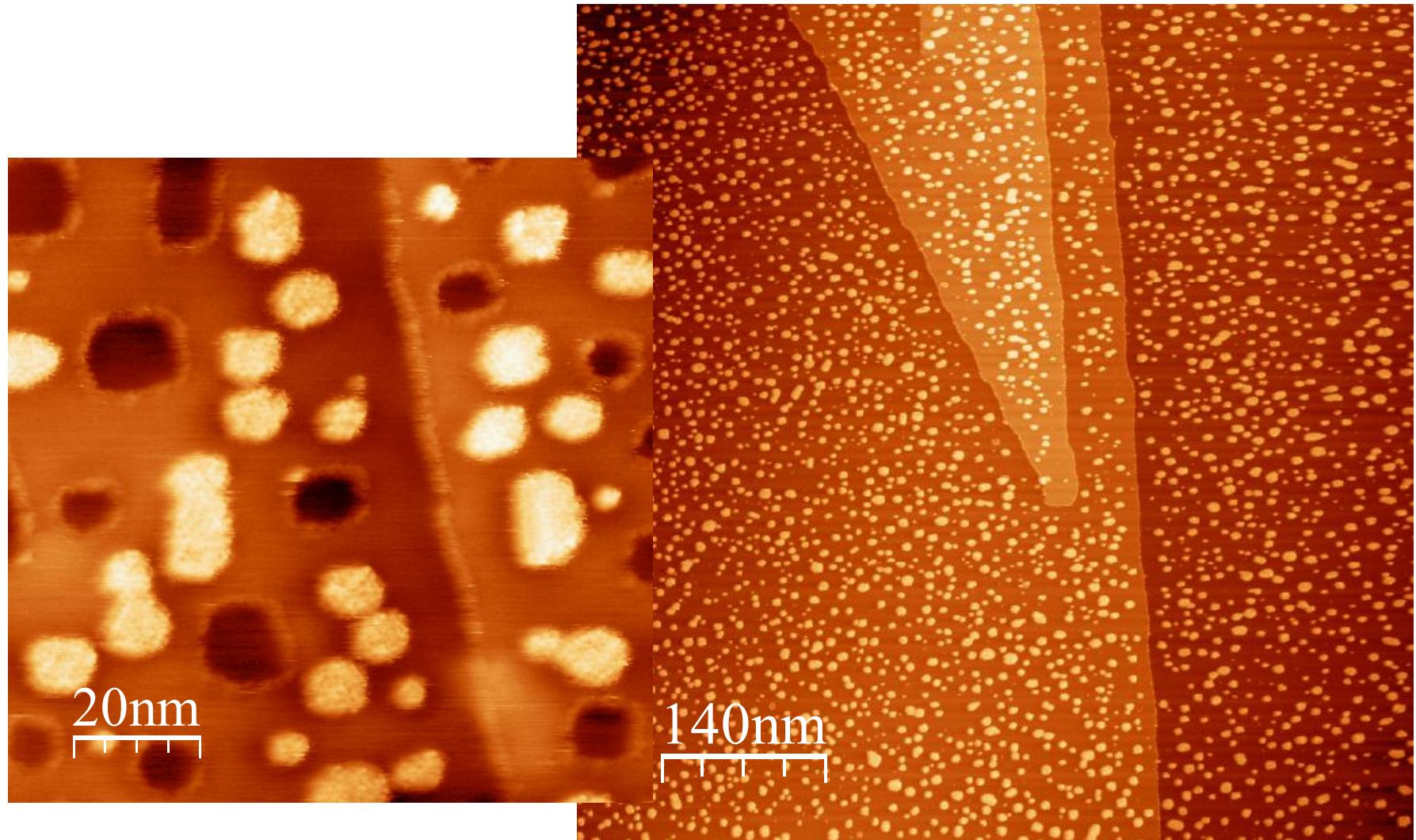
Single Crystal KBr

Substrate patterning by electron irradiation



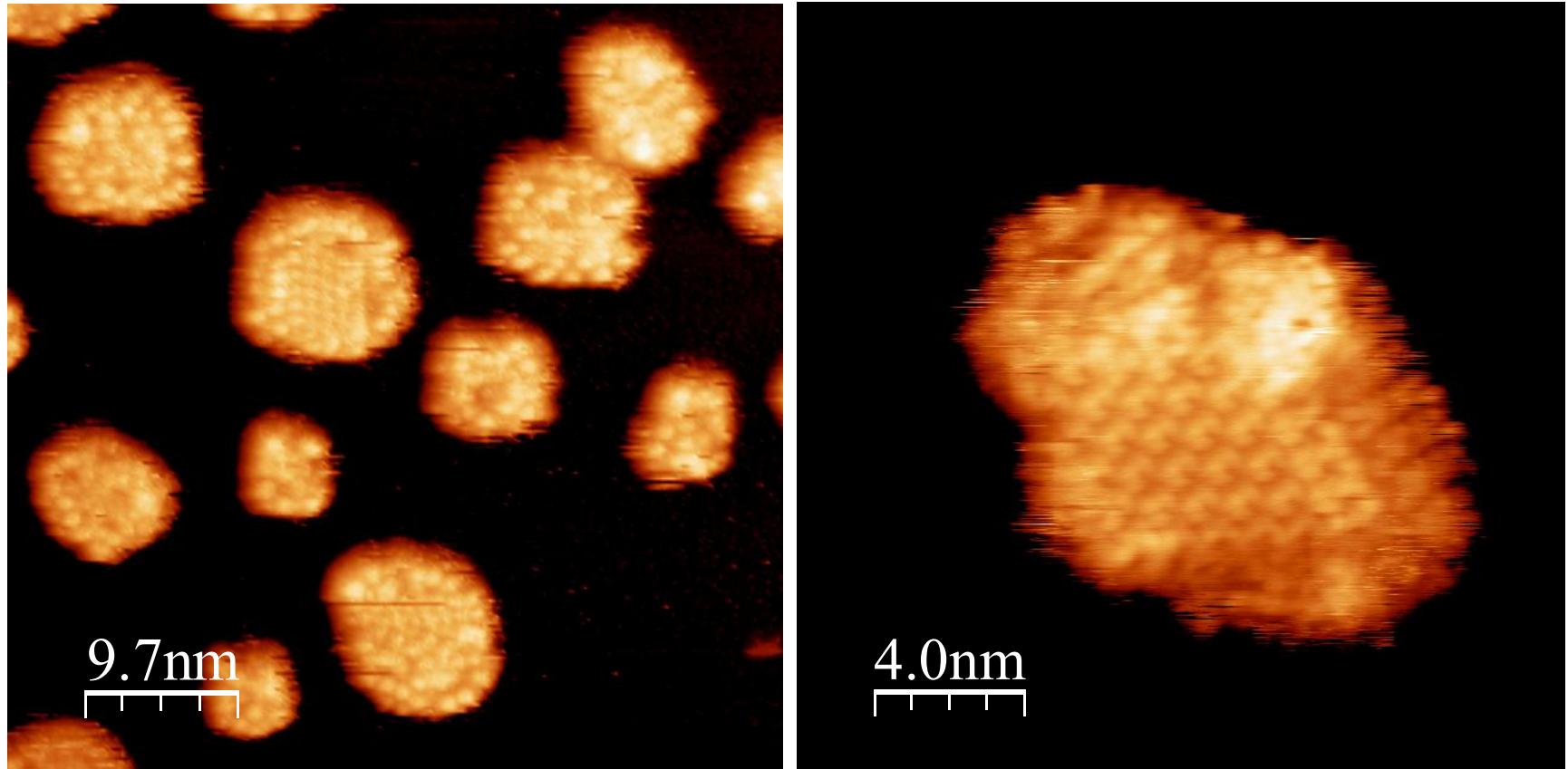
Truxenes on patterned surface

Filled and unfilled pits



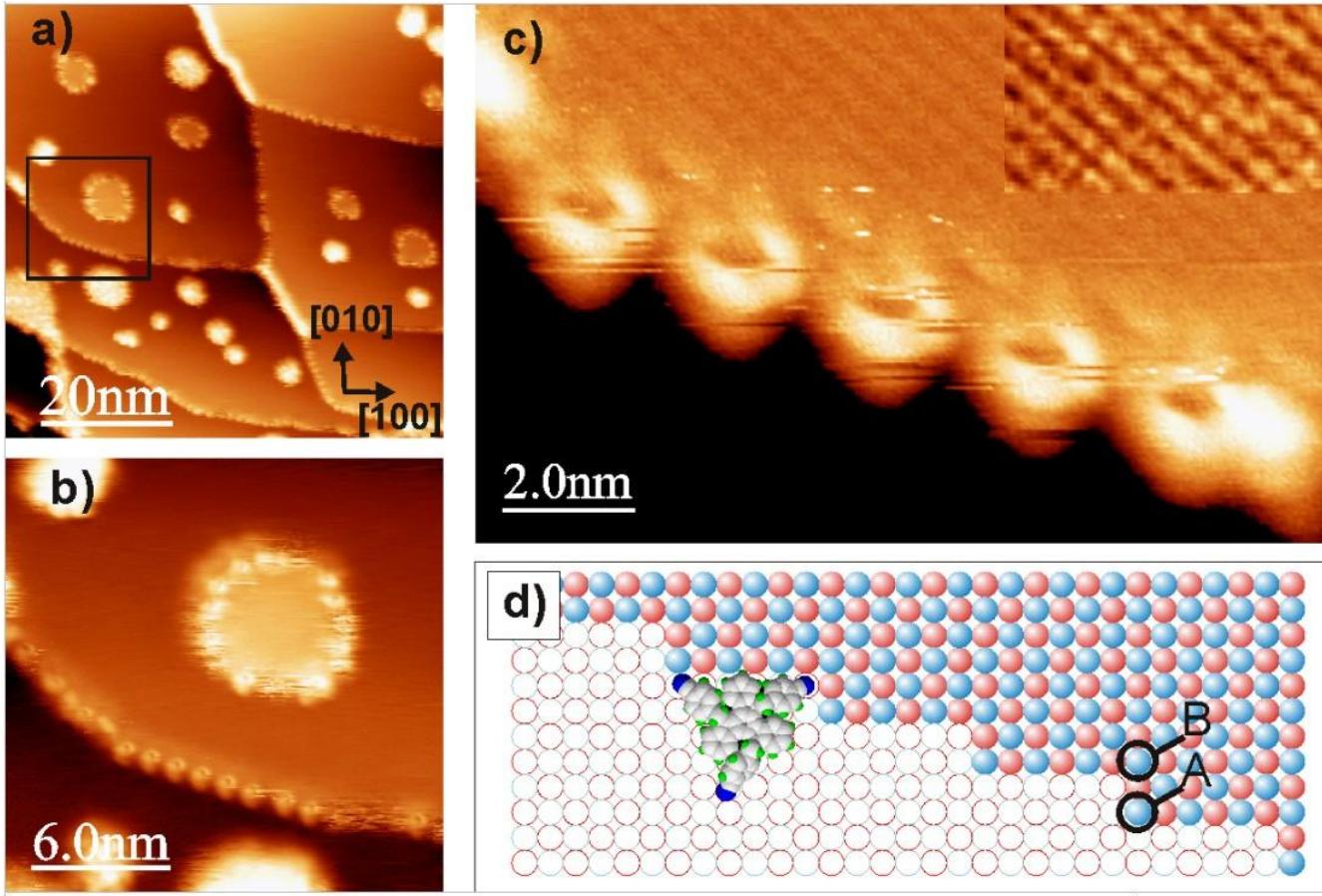
Truxenes on patterned surface

Organization within the pits



Imaging a Single Molecule

Measurements at RT

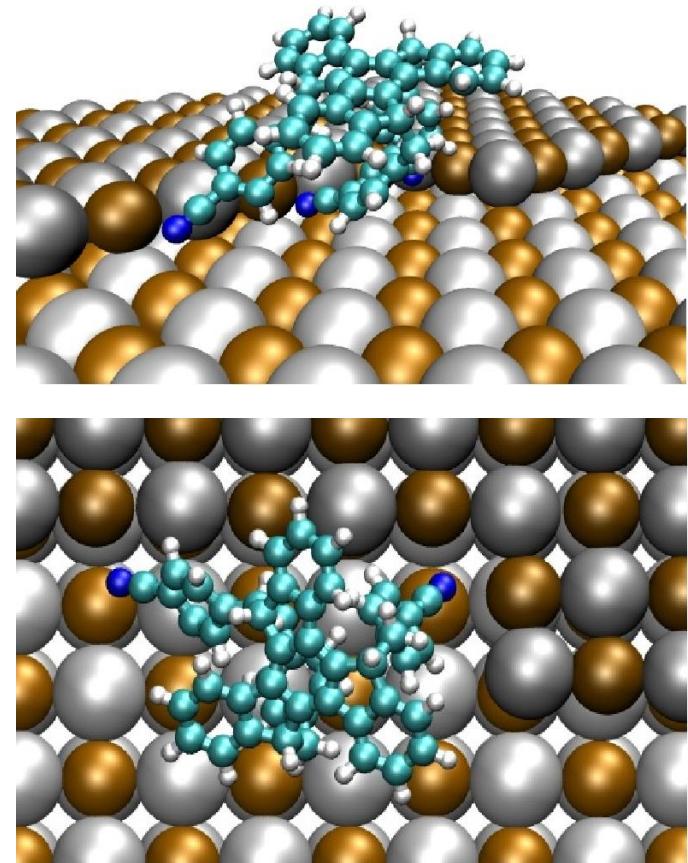
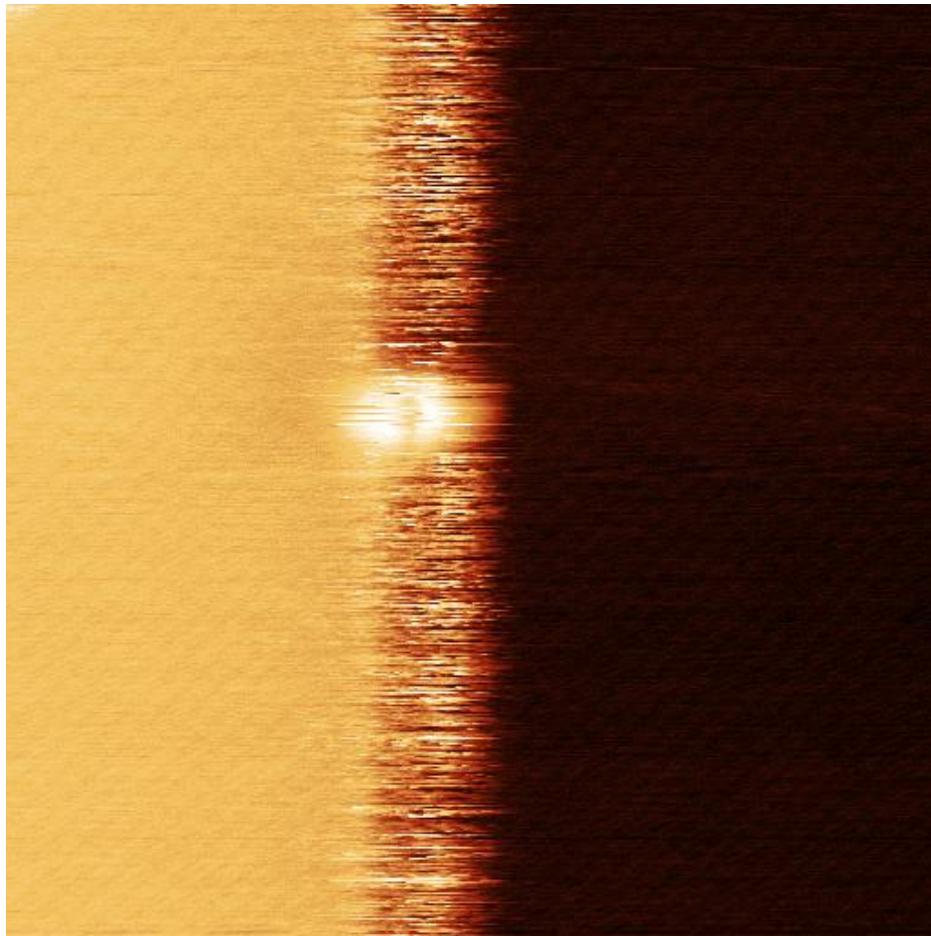


- Re-arrangement of the substrate, edges are running in the [-3 1 0] direction
- no chemical interaction with the surface
- adsorbed on K or Br terminated double atomic kink

Imaging a Single Molecule on KBr

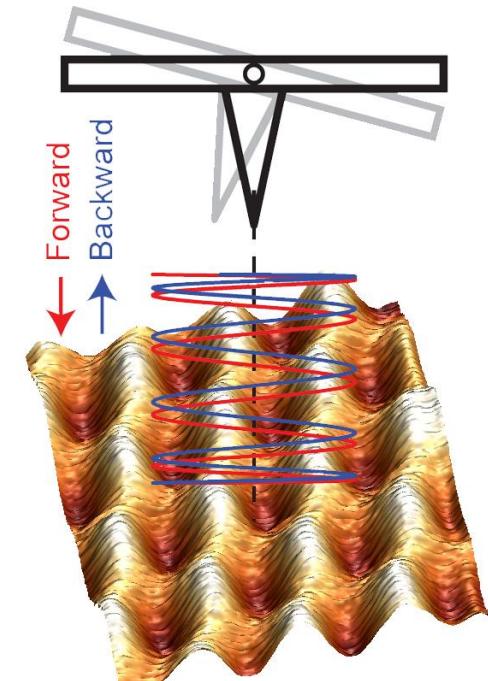
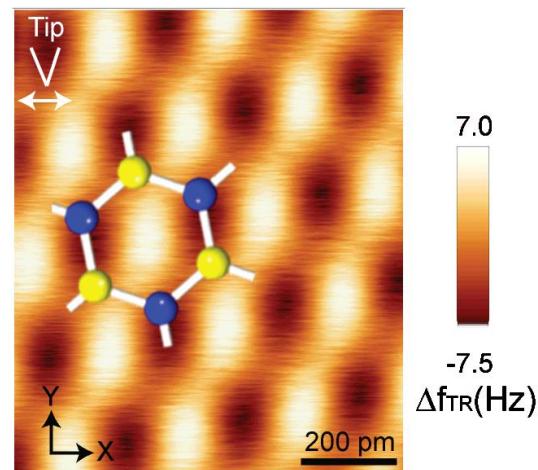
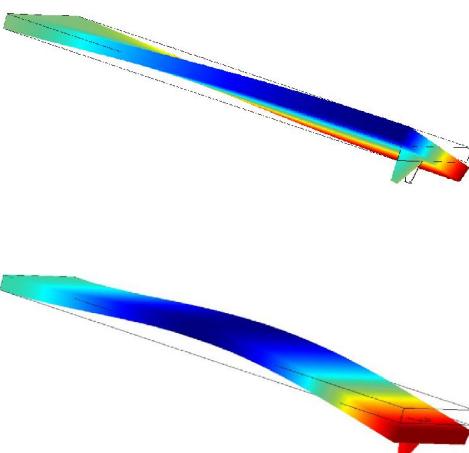
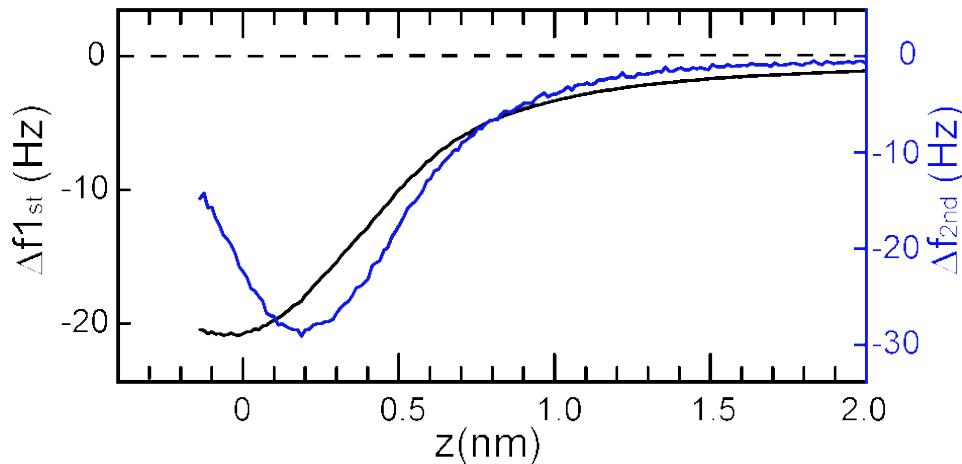
Measurements at RT and Quantum Chemical Calculations

Collaboration with A. Echavarren, Tarragona, Spain
and A. Shluger, UCL, London, UK



Bimodal Dynamic Force Microscopy

increased sensitivity and stability

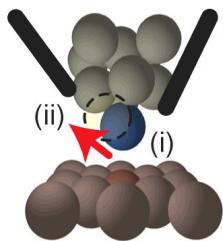


S. Kawai & Th. Glatzel et al.

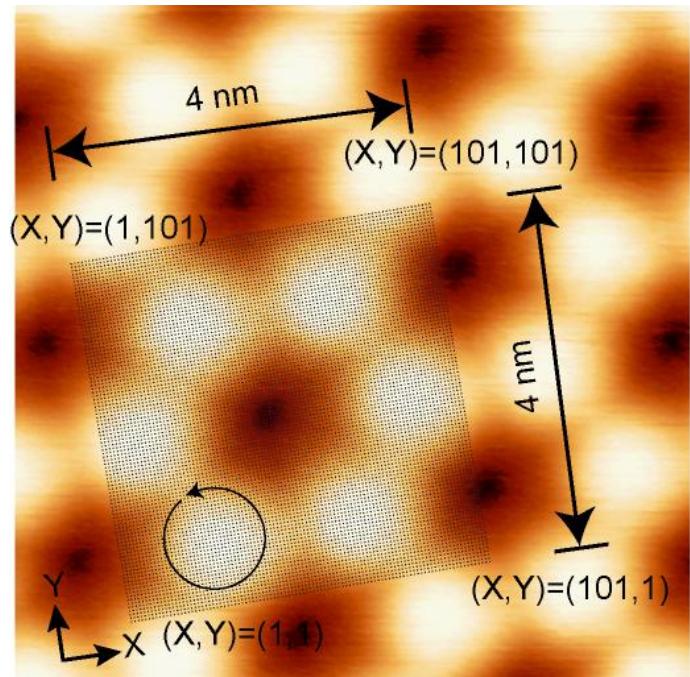
PRB 81, 085420, (2010),
PRB 80, 085422, (2009),
PRL 103, 220801, (2009).

3D Force Fields

atomic and molecular scale



Collaboration with SPECS-Nanonis, Berlin, Germany

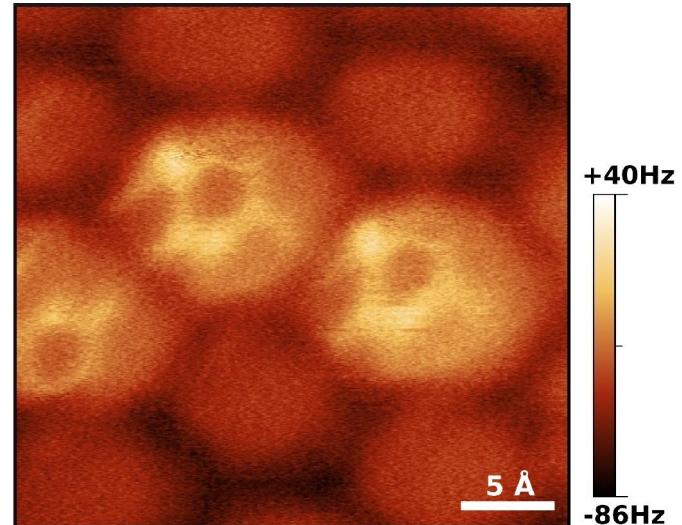
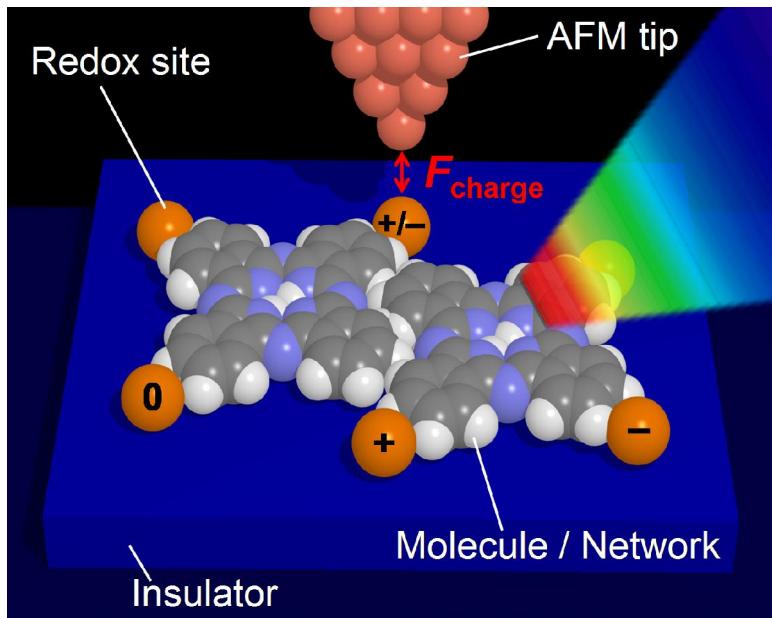


$$A_{\text{TR}} = 50 \text{ pm}$$

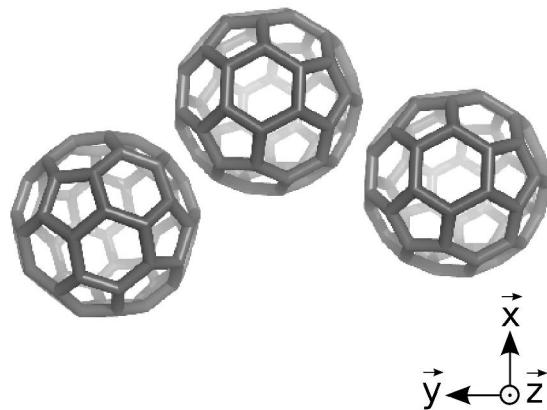


Motivation

Molecular Electronics

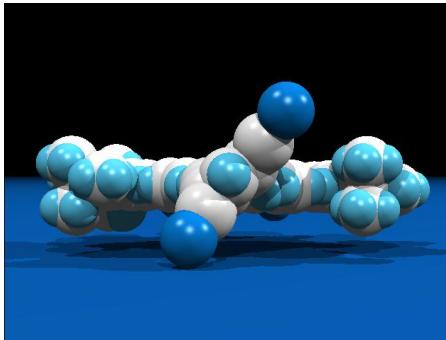
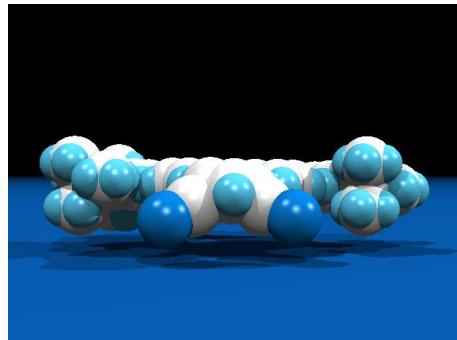
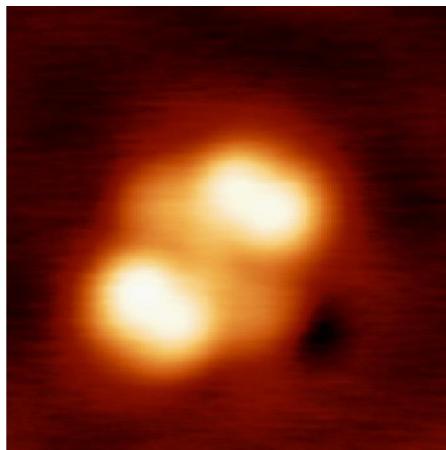
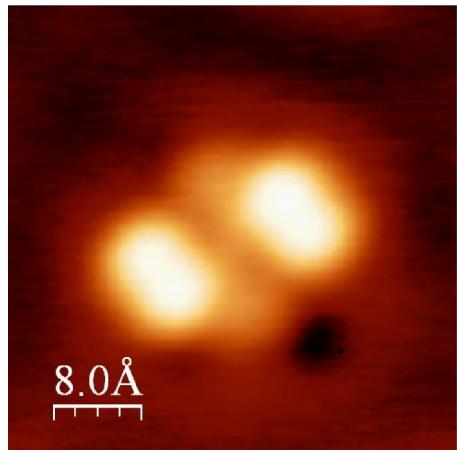


- 2D and 3D spectroscopy
- Combination of optical excitation and high resolution
- Transfer to solar cell devices?

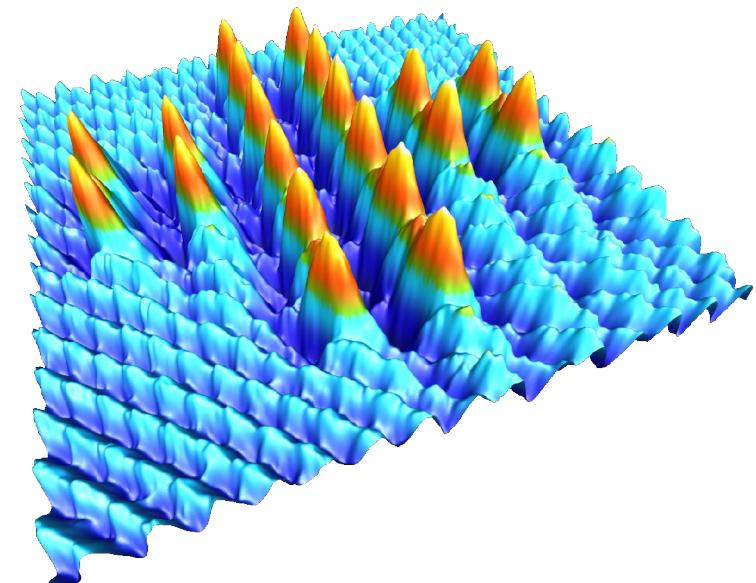


Molecular Manipulation

at room temperature and 5K



Porphyrin on Cu(111) at 5K



Br adsorbates on KBr at room temperature
manipulated by nc-AFM